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Low Temperature Sintered Ni-Zn-Cu Ferrite

T. Nakamura and Y. Okano

R&D Division, Toda Kogyo Corporation, 4-1-2 Funairi-minami, Naka-ku, Hiroshima 730, Japan

Abstract The low temperature sintering of Ni-Zn-Cu ferrite has been investigated by usual ceramic technique. The sintering density and the permeability of the sintered ferrite were strongly affected by the size of the starting oxide powders and the pre-sintering temperature. It was most effective to utilize fine particles of iron oxide and to calcine at about 800°C for the preparation of high permeability ferrite. The sintered ferrite with a density greater than 4.5g/cc and the permeability at 10MHz greater than 200 was obtained at relative low sintering temperature (about 900°C). This condition is suitable for multilayer chip inductor application.

1. INTRODUCTION

Recently, the surface mounting devices have been rapidly developed using a multilayer chip inductor[1]. This chip inductor is produced by coating ferrite and electrical pastes alternately and by co-firing. Since Ag is usually used as an electrode material, high temperature co-firing causes unexpected lowering of the inductance due to the interfacial diffusion between the ferrite and the Ag electrode. The interfacial diffusion can be suppressed by co-firing at a temperature lower than the melting point of Ag (960°C). Therefore, low-temperature sintering of ferrite is strongly demanded.

In the present work, we have studied the low-temperature sintering of Ni-Zn-Cu ferrite and clarified the relations between preparation conditions (calcination temperature and particle size of starting materials) and some properties of the sintered ferrite (sintering density and permeability). We have also discussed the permeability variation with the sintering density from the view point of the spin and domain wall contributions.

2. EXPERIMENTS

Different size of Fe_2O_3 , NiO and ZnO particles were used for the preparation of Ni-Zn-Cu ferrite sintered ceramics : calcination and sintering temperature were changed to find the optimum condition for low-temperature ferrite sintering. $\alpha\text{-Fe}_2\text{O}_3$ coarse particles (the average diameter of 0.15mm) were synthesized by the oxidation of Fe_3O_4 particles at 800°C. The Fe_3O_4 particles were precipitated by aerial oxidation of alkaline $\text{Fe}(\text{OH})_2$ suspension at 80°C. $\gamma\text{-Fe}_2\text{O}_3$ fine particles (the average diameter of 0.03mm) were prepared by thermal decomposition of ferrous oxalate at 400°C in air. The ferrous oxalate particles were precipitated by mixing of ferrous sulfate and oxalic acid aqueous solutions at 50°C. Industrial grade particles of Ni oxide, Zn oxide and Cu oxide were used as the coarse particles (their average diameters were 0.5, 0.18 and 0.4mm, respectively). And fine particles of Ni oxide and Zn oxide were also synthesized by thermal decomposition of Ni - and Zn-oxalate at 400°C in air, the average diameter of which were 0.04 and 0.02mm, respectively.

Ni-Zn-Cu ferrite ceramics were synthesized by usual ceramic technique (double sintering method). Starting powders Fe_2O_3 , NiO, ZnO and CuO were mixed together with wet attrition milling. Their molar ratio was adjusted to the composition of $\text{Ni}_{0.26}\text{Zn}_{0.52}\text{Cu}_{0.25}\text{Fe}_{1.97}\text{O}_4$. The mixtures were calcined in air at several temperature ranging from 750 to 850°C for four hours, and re-ground by wet ball milling. The obtained ferrite powders were mixed with an appropriated amount of poly-vinyl-alcohol as a binder, subjected to the press compacting (plate with the diameter of 50mm and the thickness of about 5mm) and sintered in air at temperature range from 850 to 920°C for four hours.

3. RESULTS AND DISCUSSION

First, the calcination temperature was fixed at 800 °C in order to examine the effects of the starting materials. Both the sintering density and the permeability increased with a increase of the sintering temperature for each set of the starting materials. It was found that using the fine particles as the starting materials led to the greater sintering density and the greater permeability at the same sintering temperature. Especially, using fine iron oxide particles was most effective to obtain the sintered ferrite with high sintering density and high permeability. Small starting oxide particles are transformed into smaller ferrite particles during the calcination process. These smaller ferrite particles have larger surface area and larger free energy. Since the surface free energy is considered to be a driving force in the sintering process, the smaller ferrite particles tend to be well sintered. Therefore, the use of fine oxide particles as the starting materials is effective in preparing high density sintered ferrite through low-temperature sintering.

Next, the starting materials and the sintering temperature were fixed and then we examined the effect of the calcination temperature. The sintering density remained almost constant at the calcination temperature below 800 °C, and it decreased when the calcination temperature exceeded 800 °C. The particle size of the calcined ferrite increases when the calcination temperature is raised. Accordingly, large thermal energy is required to sinter the particles : low-temperature sintering is considered to be difficult. On the other hand, the permeability had a maximum value at about 800 °C of the calcination temperature. These results indicate there exists an optimum calcination temperature to obtain high permeability ferrite ceramics.

From these results, both the utilization of the fine iron oxide particles as the starting materials and the appropriate temperature calcination produces the sintered ferrite with high density and high permeability. The sintered ferrite which was synthesized by use of γ -Fe₂O₃, calcination at 800 °C and sintering at 900 °C had the sintering density of about 4.6g/cc and the real permeability about 200 at 10MHz and approximately 30 at 100MHz.

The preparation conditions have a large effect on the permeability, since these ferrites are in polycrystalline form. In order to clarify the factors affecting the permeability, the relationship between the sintering density and the permeability was examined. The permeability value at 100MHz increased with a increase of the sintering density and saturated at about 4.5g/cc. In general, the permeability is related to two different magnetizing mechanisms : the spin rotational magnetizing and the domain wall motion. It is known that the complex permeability of sintered ferrite in 100MHz region is almost determined by the spin rotation magnetizing mechanism[2] and that the spin rotational component of the permeability depends only on the ferrite volume loading, i.e. the sintering density[3,4]. Therefore, the permeability in 100MHz region can be controlled by the sintering density.

Roughly speaking, the permeability at 10MHz also had the same trend and increased with the sintering density. However, the permeability values at 10MHz were spread over a wide range owing to the difference of some conditions : there are highly sintered ferrite ceramics with low permeability. While, in 100MHz region, the permeability can be described only with the spin rotational component, the contribution of the domain wall motion cannot be negligible in 10MHz region. The SEM observation for the fracture surface of the sintered ferrites which had almost the same sintering density indicated that the growth of the ferrite grain resulted in the greater permeability value at 10MHz. Globus suggested that the domain wall motion was affected by the grain size and enhanced with the increase of the grain size[5]. Accordingly, it is necessary to increase the sintering density and the ferrite grain size for high permeability in 10MHz region. Although low-temperature calcination produces small ferrite particles which has high sintering ability even at relative low temperature, resulting in the ferrite ceramics with high sintering density, the grain growth of the ferrite particles is not sufficient. This is a reason why there exists the optimum calcination temperature for high permeability ferrite ceramics.

4. CONCLUSIONS

The sintering density and the complex permeability of Ni-Zn-Cu ferrite ceramics can be controlled by the particle size of the starting oxide materials and the calcination temperature. By using the starting oxide materials with fine particle size and the calcination at an appropriate temperature, the ferrite particles with easy sintering ability can be obtained and they provide the preparation of the sintered ferrite with high density and high permeability in the relative low-temperature sintering.

References

- [1] T.Nomura and M.Takaya : HYBRIDS 3,15(1987).
- [2] Y.Naito : Proc.Int.Conf.Ferrite (1970)558.
- [3] T.Nakamura, S. Yamamoto, T.Tsutaoka, S.Shimizu, E.Hankui and K.Hatakeyama : Proc.Int.Conf.Ferrite (1992)1298.
- [4] T.Nakamura, T.Tsutaoka and K.Hatakeyama : J. Magn. Magn. Mater. **138**, 319(1994).
- [5] A.Globus : J.Phys.Suppl.C1(Proc.ICF-3)(1977)1.