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The spray-forming of thin wall shapes

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The works described in this paper are part of a general study concerning new methods for the realization of parts of turbo-engines and turbo-pumps; the results presented concern—the superalloy INCONEL 625. At first are presented the metallographic structure of the deposits and secondly test-pieces developed in order to characterize their mechanical properties (i.e. NOL-Rings, tubes,...). The physical and mechanical properties of the sprayed and heat treated materials are then compared with those of forged and sintered materials. The performances of the sprayformed test-pieces was observed to be similar to the forged or sintered ones.

INTRODUCTION

The progress in the design of turbo engines in order to increase the efficiency, for example of the combustion chambers, and the availability of new materials with higher mechanical properties at high temperature increase drastically the machining costs by traditionnal processes.

The spray-forming appears then to have an important industrial potential in that it could permit to obtain thin complex parts with high mechanical properties at a moderate cost (1,2). This technique consists to spray a more or less thick coating on a substrate of the given final shape and then to remove this substrate in order to obtain the part constitued only by the coating.

The aim of this paper is to present some results about the mechanical properties of such deposits obtained by plasma spraying under low pressure. Several materials were tested; in this paper are presented the results obtained with the INCONEL 625, a well known alloy appreciated for its ductility and commonly used as armour plates.

A similar work dealing with VPS (vacuum plasma spray) forming of INCONEL 718 was recently published by Holmes et al (3). The results obtained showed excellent properties; the fine grained, recrystallized material exhibited better mechanical properties than critical grade cast inconel 718 at cryogenic, room and elevated temperature.

EXPERIMENTAL PROCEDURE

The chemical composition of the IN 625 powder used in this work for the sprayforming is presented in the Table 1.

The powder grain size distribution determined by image analysis is presented in the table 2. The figure 1 illustrates the spherical morphology of the atomized powder.

Ni

bal

Cr

21.1

weight % ppm

Al Ti C Fe Si P Nb Mn S 02 N2

3.52

0.27

18

518

Table 1: Chemical analysis of the INCONEL 625

0.19 0.1

Table 2:	Powder	size	distribution

Mo

8.84

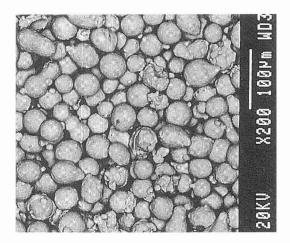
0.1

0.1

0.2

1.72

SIZE	INCONEL 625	
[<i>µ</i> m]	[% by volume]	
S < 16	0.32	
16 < S < 25	38.82	
25 < S < 40	57.12	
40 < S < 63	3.75	



502

Figure 1: The INCONEL 625 powder

EXPERIMENTAL CONDITIONS

The material was sprayed under a low-pressure of argon with a Plasma-Technik F4 gun. The level of residual oxygen into the chamber was kept under 50 ppm during the whole operation. Substrates of a cylindrical form with different compositions and dimensions were set in front of the torch, on a revolving axis. The gun was animated with a longitudinal motion, as shown schematically in the figure 2.

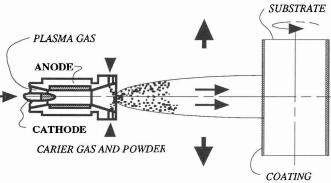


Figure 2: The torch-substrate configuration

After spraying, the substrate was removed either by machining, chemical etching or other means depending upon its nature.

METALLOGRAPHIC STRUCTURES

The figure 3 shows the metallographic structure of the as-sprayed INCONEL 625 revealed by chemical etching. The structure appears very close to the conventionnal structure of the INCONEL 625, unless the presence of some unmolten particles and macroporosities (0.05 to 0.1 vol %).

In order to further homogenize the structure, heat-treatments were performed. The figure 4 presents the metallographic structure of the heat treated material.

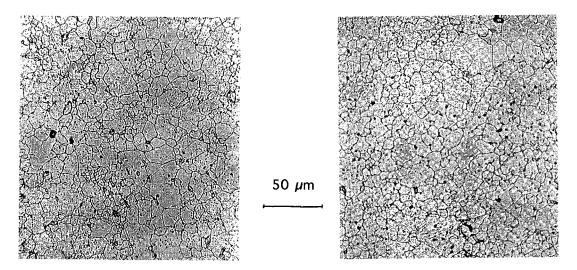


Figure 3: As-sprayed INCONEL 625 structure

Figure 4: Sprayed INCONEL 625 structure after heat-treatment

TENSILE TESTS

Two differents ways were investigated to study the tensile properties of the deposited material. The first one was to use micro-tensile specimens milled in the thickness of the cylinders (figure 5) and the second one was to use NOL-ring specimens (figure 6) directly prepared by spraying. Mechanical properties were measured at different temperatures, from cryogenic (20 K), to room temperature and high temperatures (350 and 500° C).

The figure 7 shows the elongation (e), the yield strength (Y.S) and the ultimate tensile strength (U.T.S) versus temperature. Each value is the average of several tests performed in the same conditions (usually 3 tests). As in the previously mentionned case of the inconel 718 (3) the results appear to be very similar to those obtained with samples issued form a forging or a powder metallurgy route (figure 8) (4).

Furthermore, no significant difference was observed between the results obtained from the NOL-ring and from the micro-tensile specimens, in good correlation with the fine and homogeneous microstructure previously mentionned.

The fracture analysis (figure 9) indicates a ductile fracture, without any preferential propagation zone.

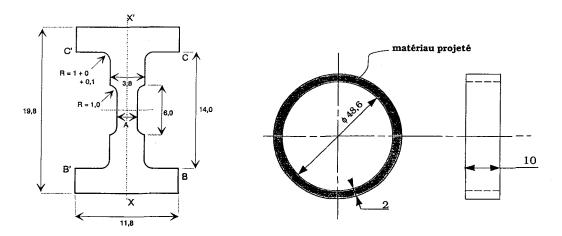


Figure 5: Micro-tensile specimen

Figure 6: The NOL-ring specimen

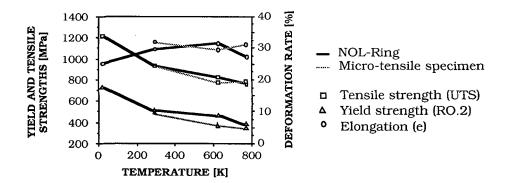


Figure 7: Tensile properties of the sprayed and heat-treated INCONEL 625 versus temperature

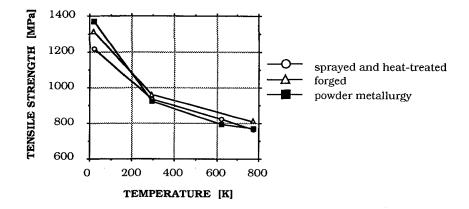
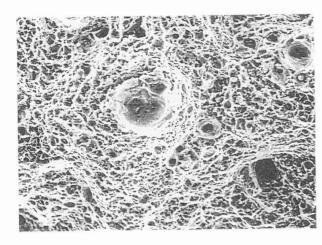


Figure 8: Yield strength of the INCONEL 625 versus temperature, for different fabrication routes



20µm

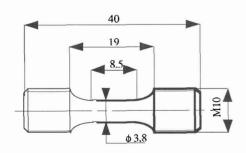
Figure 9: Fracture zone observed by SEM

LOW CYCLE FATIGUE (LCF) TESTS

The LCF life-time was explored in the following conditions: temperature 500° C, stress 600 and 650 MPa, frequency 10 Hz, R 0.05. The figure 10 shows the morphology of the specimens. Under these conditions the average number of cycles before failure is very high (1.200.000 cycles for 600 MPa and 60000 cycles for 650 MPa) and largely exceed those obtained for powder metallurgy elaborated specimens (40.000 cycles). This excellent behaviour appears to be due to the appearance of a plastified zone. Further investigations are underway.

ADHERENCE TESTS

The spray-forming of thin parts may include the necessity to realize assemblies between a bulk material and a sprayed one, as illustrated schematically in the figure 11. In order to assess the mechanical resistance of such assemblies, adherence tests were performed in order to determine the adherence of the sprayed material on a bulk material.



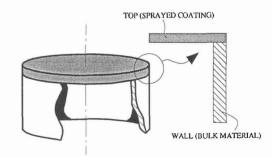


Figure 10: Morphology of the LCF specimens

Figure 11: Schematic view of a composite asembly: bulk material and sprayed one

The configuration of the specimens is illustrated in the figure 12. The base material used was HASTELLOY X. The surface preparation was made by sandblasting and furthermore using a negative transferred arc prior to the spraying.

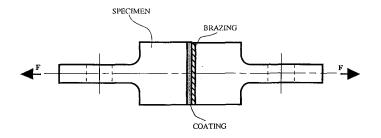


Figure 12: Morphology of the adherence test specimens

After spraying the face of the first half test piece with IN 625 (up to a thickness of 1 mm), the joining was obtained by brazing the two parts in such a way to ensure in the same time the heat-treatment of the materials.

The first results obtained are presented in the table 4. In fact, they appear to have little signification, the rupture occuring the more frequently inside the brazing material. The choice of this brazing material is actually under discussion and further testing will be considered.

Table 4: Results of the adherence tests (rupture inside the brazing material)

T, K	20	300	623	773
Rm, MPa	150	106	28	48

CONCLUSION

Thin and thick parts in INCONEL 625 were realized using plasma spray-forming under low-pressure.

The mechanical testings realized on machined specimens gave the following results:

- i) the mechanical tensile properties of the sprayed material after heat-treatment are very similar to those obtained with the same bulk forced alloy.
- ii) The specimens present excellent and outstanding LCF life-times.

The following step will be the realization of structures containing bulk walls and thin sprayed walls.

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