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INTERACTION OF DEFECTS IN CdTe-CRYSTALS HEAVILY DOPED WITH CHLORINE

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Résumé. — On a constaté que les cristaux de tellure de cadmium dopés au chlore à des concentrations comprises entre 10^{17} et $5 \times 10^{18} \text{ cm}^{-3}$ présentent un degré d'association élevé lorsqu'ils ont été recuits à 850°C puis trempés.

Des mesures électriques et de luminescence, effectuées sur ces échantillons, après des recuits à 600°C pendant des temps croissants, suggèrent qu'en dehors des associations simples et doubles $V_{\text{Cd}}\text{Cl}_{\text{Te}}$ et $V_{\text{Cd}} 2 \text{ Cl}_{\text{Te}}$ des groupements plus larges peuvent se former.

Abstract. — In crystals with chlorine concentrations between

$$1 \times 10^{17} \text{ cm}^{-3} \text{ and } 2 \times 10^{18} \text{ cm}^{-3}$$

a high degree of association is found after vacuum annealing at 850°C and quenching.

Electrical measurements and photoluminescence data along with isothermal anneals for varying times at 600°C are presented which suggest that not only single and double associates $[(V_{\text{Cd}}\text{Cl}_{\text{Te}})$ and $(V_{\text{Cd}} 2 \text{ Cl}_{\text{Te}})]$ form, but larger clusters as well.

Thus, the compensation process in Cl doped CdTe appears to be more complex than thought previously.

In order to clarify the mechanism of compensation in semi-insulating crystals of CdTe doped with chlorine as they are used for γ -ray detectors, we studied the compensation phenomena in crystals with high concentrations of the donor dopant (high N_D). The chlorine concentrations of the five types of crystals studied are presented in table I ⁽¹⁾ along with the

TABLE I

Number of ingots	Chlorine concentration (cm^{-3})	Carrier concentration (cm^{-3})	Fermi levels (eV)
1	1×10^{17}	$p 10^{15}$	0.15
2	2×10^{17}	$p 4 \times 10^{14}$	0.20
3	4×10^{17}	$p 10^{13}$	0.30
4	$(8-10) \times 10^{17}$	$p 10^{11}-10^{12}$	0.4-0.6
5	2×10^{18}	$n (7-10) \times 10^{16}$	0.02

electrical characteristics (carrier concentration at 300 K and position of the Fermi level $\varepsilon(\text{eV})$) obtained after high temperature (850°C) vacuum annealing and a subsequent quench.

These experiments have two characteristic results. Firstly, all crystals except that with the highest chlorine concentration ($\geq 10^{18} \text{ cm}^{-3}$) turn p-type.

⁽¹⁾ The Cl concentrations were determined by annealing of the samples in a Cd atmosphere and a subsequent measurement of the electron concentration [1].

This is in contrast to the results reported in [2] which indicate that crystals even at

$$2 \times 10^{17} \text{ donors.cm}^{-3}$$

cannot be converted to p-type by high temperature annealing. This difference may be reconciled by assuming that the cadmium vacancy concentration (V_{Cd}) is increased in Cl doped samples due to the large degree of association between the donors and V_{Cd} . In [3, table 4] the concentration of electrically active impurities was investigated by measuring the low temperature conductivity. It appeared to be four times less than the amount of chlorine incorporated. Thus, the degree of association can be evaluated as :

$$A = \frac{N_D - N_A}{N_D} = 0.75.$$

This result may be further confirmed by an analysis of the photoluminescence spectra of samples in the region of the *donor-acceptor* (D – A) peak (Fig. 1) in both In and Cl doped samples.

A significant shift of the D – A line in In doped samples toward the long wavelength side relative to the *band-acceptor* (B – A) line at 1.545 eV points toward radiative transitions from a wide spread of energy band tails on the acceptors, as does the considerable (0.01 eV) long wave displacement of the band which occurs when the intensity of the excitation is decreased. Band tails are, of course, a result of high concentrations of randomly distributed donors

FIG. 1. — Photoluminescence spectra of In and Cl doped CdTe. $N_D = 10^{17} \text{ cm}^{-3}$, $T = 4.2 \text{ K}$. Dashed curves-intensity of excitation is reduced 30 times.

and lattice defects. In chlorine doped crystals by contrast, a clear D – A band is observed which drifts only slightly with excitation intensity, i. e. the concentration of charged centers is much lower.

A second characteristic result is also to be observed from table I. With an increase in the chlorine concentration, the Fermi level moves increasingly deeper into the forbidden gap. It takes on successively values of $E_v + 0.15 \text{ eV}$; 0.2 eV ; $0.3 - 0.6 \text{ eV}$ and, finally, for crystal # 5 it is no longer possible to obtain p-type conductivity. Thus, the solubility of V_{Cd} , although increased for these Cl doped crystals, still is somewhat restricted.

Since the Fermi level moves through a virtual continuum of levels, all deeply in the gap, we may conclude that the concentration of the levels which control the conductivity in these samples are comparable to the concentration of the major impurity, chlorine. And since the concentration of chlorine in these samples was definitely larger than the concentration of unwanted electrically active impurities, the levels observed may be assigned to associated centers which contain Cl and V_{Cd} .

We have suggested early on, that the level at $E_v + 0.17$ belongs to the complex $(V_{Cd}Cl_{Te})^-$ [7]. Now we suggest that one more V_{Cd} , locating near that center would lead to a level closer to that of the doubly negative charged $V_{Cd}(E_c - 0.6 \text{ eV})$. Thus, surrounding of the Cl_{Te} by various numbers of V_{Cd} creates the observed spectrum of levels with depths between $E_v + 0.15 - 0.9 \text{ eV}$. That is to say, high temperature annealing leads to clusters of different size, from the smallest possible one ($V_{Cd}Cl$) to multi-component ones, and the great enthalpy of the chlorine compounds makes such clusters probable even at high annealing temperatures [4].

Further evidence for the formation of clusters has been obtained from isothermal annealing experiments at 600°C . Here samples which had been 850°C vacuum annealed were heat treated for varying lengths of time at 600°C . In these experiments we observed anomalous changes of the electron concen-

tration in the materials inasmuch as samples # 2, 3 and 4 became n-type. Detailed results for samples 2, 4 and 5 are shown in figure 2.

FIG. 2. — Change of carrier concentrations in compensated crystals CdTe : Cl depending on time of annealing at $T = 600^\circ\text{C}$. The curves correspond to crystals 5, 4, 2 in table I.

In accordance with [4] annealing at intermediate temperatures increases the size of clusters. Since V_{Cd} or $(V_{Cd}Cl_{Te})$ are the smallest units, we expect that during the 600°C anneal most of the mobile V_{Cd} would separate and join larger clusters, thus creating single uncompensated Cl_{Te} . However, the competing process of incorporating additional V_{Cd} (extraction of Cd from the crystal) somewhat balances the electrical effects here. All this represents itself in the photoluminescence spectra by a gradual disappearance of the band at 1.457 eV which represents the transition into the $(V_{Cd}Cl_{Te})$ complex. The total intensity of the band decreases and the spectra shift to the longer wavelength side [5].

Theoretical calculations of the degree of association for CdTe : Cl crystals have been carried out by Canali *et al.* [6]. There it was suggested, that only two stages of association occur in compensating the crystal, namely $(V_{Cd}^-Cl_{Te}^+)$ and $(V_{Cd}^{2-}Cl_{Te}^+)$ (¹). Using their model and the enthalpy values given there, however, we could not obtain the high degree of association which we found in our experiments.

Thus, the results presented here indicate a far greater complexity in the phenomena of compensation in CdTe : Cl crystals and we attribute the increased solubility of V_{Cd} and the anomalous behavior of the crystals during low temperature annealing to the existence of multicomponent clusters which contain relatively large numbers of V_{Cd} .

Such processes then may also be significant in the crystals with lower dopant content ($N_D \sim 10^{17} \text{ cm}^{-3}$) which are used for radiation detectors.

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