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Magnetostriction of an amorphous \((\text{Fe}_{82}\text{B}_{18})_{0.9}\text{La}_{5}\text{Tb}_{5}\) alloy

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Résumé. — On donne les variations thermiques des magnétostrictions spontanée et forcée d'un alliage amorphe \((\text{Fe}_{82}\text{B}_{18})_{0.9}\text{La}_{5}\text{Tb}_{5}\). À la température ambiante, la magnétostriction est un ordre de grandeur plus faible que la valeur prouvée, ce qui ôte à cet alliage tout intérêt en tant que transducteur magnétostrictif. La variation thermique de \(\lambda_s\) s'interprète assez bien dans le modèle à un ion des ferrimagnétiques.

Abstract. — The thermal variations of the spontaneous and forced magnetostriction of an amorphous \((\text{Fe}_{82}\text{B}_{18})_{0.9}\text{La}_{5}\text{Tb}_{5}\) alloy are given. The room temperature magnetostriction is one order of magnitude smaller than the predicted value, which disproves the potential interest of this alloy as a magnetostrictive transducer. The thermal variation of \(\lambda_s\) is well described by the one-ion model for ferrimagnets.

The room temperature magnetostriction of amorphous \(\text{Tb}_{x}\text{Fe}_{1-x}\) films is known to be large, namely \(\lambda \sim 3 \times 10^{-4}\) near \(x = 0.4\) [1, 2], thus indicating the terbium magnetoelastic coupling to be strong in these amorphous alloys as previously found in the crystalline compounds.

Assuming this large magnetoelastic coupling to hold true also in the dilute terbium alloys, Koon and Das have predicted a magnetostriction as large as \(\lambda \sim 190-240 \times 10^{-6}\) for the amorphous \((\text{Fe}_{82}\text{B}_{18})_{0.9}\text{Tb}_{5}\text{La}_{5}\) alloy which they have found to exhibit soft magnetic properties [3]. Such performances would have designated this alloy as an interesting magnetoelastic transducer, but it remained to experimentally confirm this large magnetostriction.

1. Experiment.

The samples were prepared by the standard melt spinning techniques. The ribbons were approximately 1.8 mm in width, 40 \(\mu\)m thick and only a few centimeters long due to their relative brittleness. X-ray diffraction data are given in figure 1 and the alloy appears to be amorphous, if one neglects a tiny signal at \(2\theta = 34^\circ\) on the upper face of the ribbon. The magnetic properties we observed agree with the data previously given in [3, 4]: the magnetization has been measured up to 20 kOe below room temperature by the extraction method, and the low coercivity \((H_c < 3\ Oe\ at\ 300\ K)\) confirms the amorphousness of the sample. At low temperatures, the coercive
force increases up to \( H_c = 50 \) Oe at 4.2 K. Finally, the Curie temperature \( T_c = 495 \) K that we derived from the forced magnetostriction in figure 2 is slightly higher than 486 K, the value given in [4], and this reveals a small difference in the chemical composition, the amorphous Fe\(_{82}B_{18}\) alloy without any rare earth having a much higher Curie temperature, \( T_c = 636 \) K [5].

The magnetostriction could not be measured by the cylinder method [6] due to the poor mechanical properties, so we piled up 22 pieces about 10 \( \times \) 1.8 mm\(^2\) in area, 40 \( \mu \)m in thickness that we glued together with M bond 600 adhesive from micromeasurements in order to get a bulk sample according to Tsuya's method [7]. The strains were measured along the larger dimension with a capacitance dilatometer, the magnetic field being applied parallel and then perpendicular to the measurement direction, always in the plane of the ribbons.

The thermal variations of both the spontaneous and the forced magnetostrictions are given in figure 2 for the As-quenched alloy.

2. Discussion.

The room temperature magnetostriction we observe is one order of magnitude smaller than the one predicted by previous authors [3], and this can explain the soft magnetic properties of this alloy which no longer offer outstanding technical interest as a magnetostrictive transducer.

The analysis of the thermal variation of the magnetostriction can bring further information concerning the magnetoelastic coupling of the terbium in this alloy.

O'Handley [8] first recognized the magnetostriction of amorphous Fe\(_{80}B_{20}\) alloy to follow a one-ion law, and this behaviour appeared to be common to most amorphous alloys, see e.g. figure 5 in [9]. In the present case, the alloy is a ferrimagnet and the one-ion law no longer involves the macroscopic magnetization \( M_{\text{tot}} = M_{\text{Fe}} - M_{\text{Tb}} \), but is a linear combination (Eq. 1) involving the reduced magnetizations of both sublattices, as already pointed out by Callen et al. [10]:

\[
\lambda_s(T) = \lambda_{\text{Fe}} \tilde{I}_{5/2}\left(\frac{1}{\mu} m_{\text{Fe}}\right) + \lambda_{\text{Tb}} \tilde{I}_{5/2}\left(\frac{1}{\mu} m_{\text{Tb}}\right).
\]

In this equation, we have neglected the thermal variation of the Lamé coefficient \( \mu \), since this approximation has already proved to be valid for a lot of metallic glasses.

We have assumed that the iron magnetization varies as the magnetization of the amorphous (Fe\(_{82}B_{18}\)La\(_{10}\) alloy, the thermal variation of which has been given in [4]. The terbium magnetization has been derived by subtracting the iron magnetization from the magnetization of the amorphous (Fe\(_{82}B_{18}\)\(_{0.5}\)Tb\(_{5}\)La\(_{5}\) alloy, the thermal variation of which has also been given in [4] and confirmed in this work up to 300 K. The best fit of equation 1 and figure 2 is obtained with \( \lambda_{\text{Fe}} = +40 \times 10^{-6} \) and \( \lambda_{\text{Tb}} = +16 \times 10^{-6} \).
The iron contribution must be compared with the data obtained with the amorphous Fe$_{82}$B$_{18}$ alloy [5], namely, $\lambda_s = 46 \times 10^{-6}$ at 0 K. $\lambda_s$ is known to vary roughly as the square of the magnetization for metallic glasses and, taking into account the reduction by 10% of the number of iron atoms in (Fe$_{82}$B$_{18}$)$_{0.9}$La$_{10}$, we predict $\lambda_{Fe} \approx 25 \times 10^{-6}$ which is a satisfactory value if one considers the above approximations.

On the other hand, the terbium contribution appears very small with respect of the magnetoelastic coupling of Tb in amorphous TbFe$_2$ [1]. At 300 K, $\lambda_s \approx 300 \times 10^{-6}$ for this alloy while the Tb contribution in our alloy is only $4 \times 10^{-6}$ at the same temperature, where $I_{5/2} \left\langle \xi^{-1}(m_{Tb}) \right\rangle \approx 0.25$, which gives one order of magnitude smaller when referring to one Tb ion.

But, we must notice that the magnetoelastic behaviour of the amorphous series Tb$_x$Fe$_{2-x}$ is not linear. For $x = 0.05$, the magnetostriction is only $20 \times 10^{-6}$ [1], leading to a negligible contribution of Tb if one subtracts the iron contribution, and so it is not surprising to find a rather small magnetostriction in the present work.

As a conclusion, the magnetostriction of the amorphous (Fe$_{82}$B$_{18}$)$_{0.9}$Tb$_5$La$_5$ is mainly due to the iron contribution. The magnetoelastic coupling of terbium seems to be smaller in dilute alloys than it is in the composition TbFe$_2$. The thermal variation of $\lambda_s$ is well described by the one-ion ferrimagnetic model of Callen.

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