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MAGNETOSTRICTION OF RAPIDLY QUENCHED AMORPHOUS RARE EARTH-Fe ALLOYS

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Abstract. – The magnetostriction of rapidly quenched amorphous $R_{1-x}Fe_x$ alloys ($R = \text{Pr, Nd, Sm, Gd, Tb, Dy, Er}$ and Y ; $0.4 < x < 0.9$) has been measured in the temperature range from 77 K to 300 K. The R dependence of the magnetostriction is discussed in terms of the single-ion model.

1. Introduction

The magnetostriction originates from a strain derivative of a magnetic anisotropy energy and then reflects a local atomic short range order in the amorphous state [1, 2]. In contrast with intensive studies about the magnetostriction of transition metal-metalloid amorphous alloys [1, 2], only a few measurements have been carried out so far for amorphous rare earth (R)-transition metal (TM) alloys [3, 5]. Amorphous R-Fe alloys exhibit several interesting features such as an extraordinary coercive force at low temperatures and aspero magnetic character [6], which supposedly relates to a local magnetic anisotropy in the amorphous state, hence to the magnetostriction. The magnetostriction of amorphous R-Fe alloys is therefore quite interesting from such a physical view-point, as well as from a search for a giant magnetostrictive material.

2. Experimental

Amorphous $R_{1-x}Fe_x$ alloys were prepared by a rapid quenching in an argon atmosphere, where R is Pr ($0.5 \leq x \leq 0.89$), Nd ($0.5 \leq x \leq 0.83$), Gd ($0.5 \leq x \leq 0.8$), Dy ($0.4 \leq x \leq 0.85$), Sm ($x = 0.8$), Tb ($x = 0.8$), Er ($x = 0.8$) and Y ($0.4 \leq x \leq 0.8$). They were formed into ribbon, about 1 mm and 10-20 μm thick [5].

The magnetostriction was measured by a three terminal capacitance method in magnetic fields up to 20 kOe and in the temperature range from 77 K to 400 K. This method enables us to measure the magnetostriction using only a sheet of ribbon with 7 mm length even in brittle R-Fe amorphous ribbons [5]. Although the magnetostriction $\lambda = 2/3(\lambda_{\parallel} - \lambda_{\perp})$ for $Gd_{1-x}Fe_x$ and $Nd_{1-x}Fe_x$ ($0.78 \leq x \leq 0.83$) shows a saturation at $H < 5$ kOe in all the temperature range, λ 's for the others exhibit a sluggish increase up to 20 kOe. The maximum field applied was 20 kOe, so the magnetostriction λ at 20 kOe is reported in this paper instead of saturation magnetostriction λ_s .

3. Experimental results and discussion

The temperature dependence of λ for $R_{0.2}Fe_{0.8}$ alloys is shown in figure 1. The Curie temperatures of

these alloys range from 277 K for $R = \text{Er}$ to 504 K for $R = \text{Gd}$. The values of λ for $R = \text{Pr, Sm}$ and Gd are about 630, -400 and 13×10^{-6} at 77 K, respectively, and decrease monotonically with heating. λ for $R = \text{Nd}$ changes a sign from negative to positive at about 100 K with increasing temperature. A compensation of magnetization takes place at about $T = 210$ K for $R = \text{Dy}$ and at $T < 77$ K for Tb and causes a hysteretic λ - H curve at temperatures below T_{Hc} , resulting in an apparent decrease of λ (see Fig. 1). Here T_{Hc} is defined as the temperature below which λ - H curves become irreversible even above 10 kOe. Hysteretic λ - H curves were also observed in $R = \text{Pr, Nd}$ and Dy alloys with $x \leq 0.7$: these are attributed to an extraordinary coercive force at low temperatures [6].

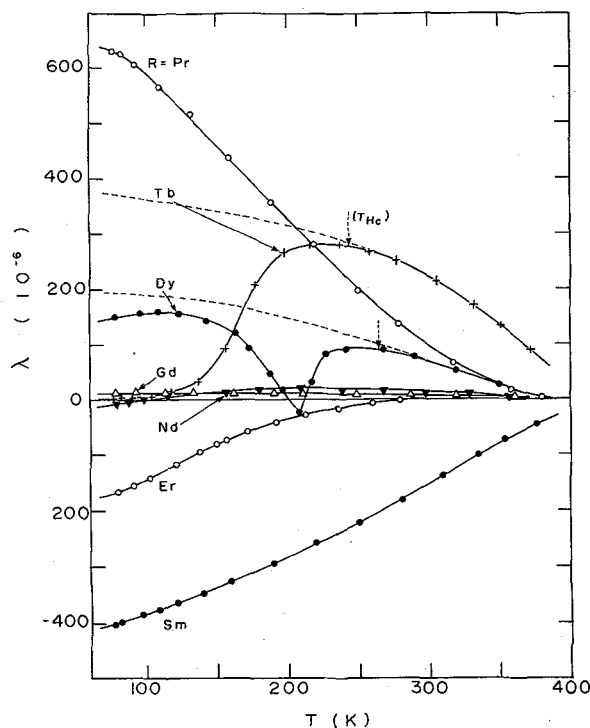


Fig. 1. – Temperature dependence of λ at 20 kOe for amorphous $R_{0.2}Fe_{0.8}$ alloys. For the broken lines and T_{Hc} , see text.

The composition dependence of λ for $R_{1-x}Fe_x$ ($R = Nd, Pr, Gd$ and Dy) at 77 K is plotted in figure 2. The λ -values having error bars in the figure were obtained by an extrapolation from data above T_{Hc} . A few examples of extrapolation are drawn by broken lines in figure 1. Note that λ for $R = Pr, Nd$ and Dy takes a maximum at $x = 0.7-0.8$.

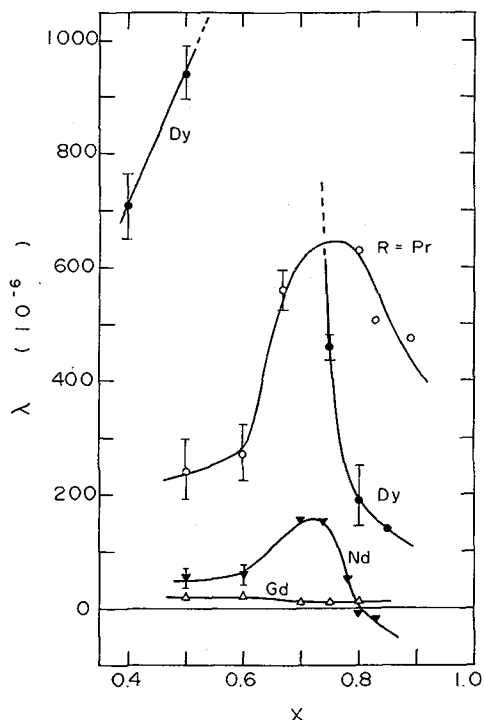


Fig. 2. - Concentration dependence of λ at 77 K for amorphous $R_{1-x}Fe_x$ alloys.

The values of λ at 77 K of $R_{0.2}Fe_{0.8}$ are plotted against R elements in figure 3. The magnetostriction of both R-Fe amorphous alloys and compounds is considered to originate from a uniaxial distortion of an atomic short range order (SRO) surrounding R atoms and was discussed in terms of the single-ion model so far [4, 7]. According to this model, a ratio of the magnetostriction of one lanthanide ion to another is given by the ratio of $\alpha J(J-1/2)\langle r_f^2 \rangle$, where α is the lowest order Stevens' factor, J the ground state angular momentum for 3^+ ions and $\langle r_f^2 \rangle$ is the average radius squared of the 4f electron shell. The value of $\alpha J(J-1/2)\langle r_f^2 \rangle$ is indicated by solid lines in the figure, normalized by λ of $R = Pr$. Note that the experimental values of λ agree well to the theoretical expectation. It is therefore obvious that the magnetostriction of the amorphous

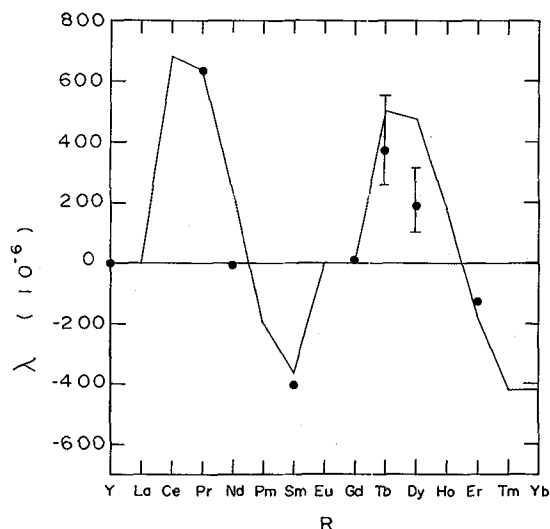


Fig. 3. - λ at 77 K as a function of the lanthanide element R for amorphous $R_{0.2}Fe_{0.8}$ alloys.

R-Fe alloys originates from the single-ion model type coupling between the angular moment of R elements and the crystal field surrounding R elements. The further implication is that SRO around R is almost equal through amorphous R-Fe alloys with $x = 0.8$.

On the other hand, the composition dependence of λ of Pr, Nd, Dy-Fe exhibits the maximum at about $x = 0.7$ as seen in figure 2. This behaviour is quite similar to those of R-Fe compounds [4]. This may indicate a similarity of SRO around R elements between the R-Fe amorphous alloys and their compounds. Hence the change of SRO in the amorphous state supposedly causes the composition dependence of λ in amorphous $R_{1-x}Fe_x$ alloys.

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