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J. Fischer, J. Milliken. IONIC LIMIT IN GRAPHITE-FLUOROARSENATE INTERCA-LATION COMPOUNDS. Journal de Physique Colloques, 1983, 44 (C3), pp.C3-723-C3-727. 10.1051/jphyscol:19833142. jpa-00222655

HAL Id: jpa-00222655 https://hal.science/jpa-00222655

Submitted on 4 Feb 2008

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IONIC LIMIT IN GRAPHITE-FLUOROARSENATE INTERCALATION COMPOUNDS

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<u>Résumé</u> - Les mesures de conductivité électrique, de réflectance et <u>de consommation</u> du fluor suggèrent une limite d'ionisation de C⁺₂₀ environ dans les composés de graphite-fluorarsenate. Une oxydation supérieure à cette limite introduit des centres correspondant à des liaisons covalentes C-F provoquant la diffusion des porteurs ; ceci est déduit des mesures de conductivité en fonction de la température et de XPS.

<u>Abstract</u> -Measurements of electrical conductivity, optical reflectance and F_2 -uptake suggest an ionic limit around C_{20} in graphite-fluoroarsenate compounds. Oxidation beyond this limit introduces carrier scattering centers in the form of covalent C-F bonds as deduced from measurements of conductivity versus temperature and XPS.

INTRODUCTION

Exposure of graphite to AsF_s , vapor yields intercalation compounds of stoichiometry $C_{sn}AsF_s$, where n = stage. Complete oxidation of graphite by AsF_s is represented by the following reaction:

$$12C(g) + 3/2AsF_{s} \longrightarrow C_{12}^{+}AsF_{s} \cdot 1/2AsF_{s}$$

However, experimental evidence /1/ indicates that only about 40% of the graphite is oxidized in stage 1 and 50% in stage 2. Based on these results, the compositions of stage 1 and stage 2 are

$$C_{30}^{+}AsF_{6}^{-}$$
 · (0.5AsF₃, 2.25AsF₅) and $C_{45}^{+}AsF_{6}^{-}$ · (0.5AsF₃, 1.35AsF₅),

respectively. A charge density of C_{30}^+ for stage 1 is consistent with those values observed in other acceptor-graphite systems. It has been suggested /2/ that neutral spacers are necessary in these materials to screen the intralayer coulomb repulsion between ions. For example, stage 1 compounds having the following compositions have been prepared /3, 4, 5/:

$$C_{24}^+ NO_3^- \cdot 3HNO_3$$

 $C_{27-21}^+ HSO_4^- \cdot 2.5H_2SO_4$
 $C_{24-20}^+ MF_6^- \cdot 2.5CH_3NO_2$

(Prepared using NO₂MF₆/CH₃NO₂, M=P,Sb,As)

The ionic limit in the graphite-bisulfate system has been studied by a series of <u>in situ</u> chemical and electronic measurements during electrochemical intercalation /6/. Coulommetry and weight uptake measurements show the first

appearance of a stage 1 compound at $C_{27}^+HSO_4^-$ · 2.5H₂SO₄. Further oxidation to to C_{21}^+ results in enhancement of conductivity and metallic reflectance with no further weight uptake. Charging beyond C_{21}^- reduces the conductivity, broadens the reflectivity spectrum and introduces C-O covalent bonds. These results suggest that beyond a limiting concentration of HSO₄, a covalent guest-host interaction is necessary to overcome coulomb repulsion, and that the C-O bonds introduce a new scattering process which limits the conductivity.

To investigate the ionic limit in the AsF_s -graphite system, samples were exposed to F_2 to bring about further oxidation of graphite. Our initial qualitative investigation involved <u>in situ</u> conductivity and reflectance studies yielding results much like those observed in the graphite-bisulfate system. Tensimetry experiments carried out in conjunction with conductivity measurements were used to determine F_2 uptake at the metallic maximum. Measurements of conductivity versus temperature and C(1s) core level XPS helped to identify the origin of the conductivity decrease.

EXPERIMENTAL

Samples of $C_{sn}AsF_s$ were prepared by exposing highly oriented pyrolytic graphite (HOPG) to AsF_s vapor /7/. These samples were exposed to F_2 at pressures ranging from 1.0 to 1.5 atm. Room-temperature conductivity measurements (basal plane) were made <u>in situ</u> using a contactless 1kHz inductive technique /8/. Measurements of conductivity versus temperature were made using a contactless inductively coupled method in which the sample rotates in a magnetic field /9/. Optical reflectance spectra were obtained from an automated single-beam spectrometer /10/. X-ray photoelectron spectra were obtained from a McPherson ESCA 36 photoelectron spectrometer using a MgK_x (1253.6ev) X-ray source.

RESULTS

The increase in conductivity (Figure 1) and blue shift of the plasma edge (Figures 2 and 3) observed during initial exposure of $C_{sn}AsF_s$ (n=1,2) to F_2 , imply an increase in free carrier concentration due to further production of AsF_6 . The subsequent decrease in conductivity and broadening of the edge suggest a reduction in carrier scattering time. This is confirmed by the changes that occur in G_a (T) (Figure 4) upon fluorination and the formation of covalent C-F bonds as determined by XPS (Figure 5). Tensimetry data (Table I) suggest a charge density of C_{20} at the metallic maximum based on the reaction sequence

$$AsF_{3} + F_{2} \longrightarrow AsF_{5}$$
$$AsF_{5} + 2/3e^{-} \longrightarrow 2/3AsF_{6} + 1/3AsF_{3}$$

CONCLUSIONS

The effect of fluorine on the electrical and optical properties of $C_{an}AsF_s$ suggest the existence of an ionic limit at C_{20} . Beyond this limit direct fluorination of the graphite lattice occurs, limiting the conductivity via defect scattering. The similarities between the AsF_s -graphite system and other acceptor-graphite systems, particularly the bisulfate-graphite system, imply that all acceptor intercalants give the same conductivity.

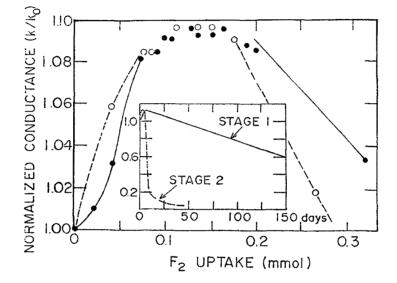


Figure 1: Conductance per C layer versus F_2 uptake (main figure) and versus time of exposure to ~1 atm F_2 (inset). Values normalized to the starting materials $C_{en}AsF_5$. Filled circles and solid curve for stage 1, open circles and dashed curve for stage 2, corresponding to samples C and E in Table I. Note that k/k_0 first increases by 10% then drops to below the pure graphite values (k/k_0 = 0.1 for HOPG).

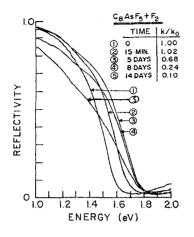
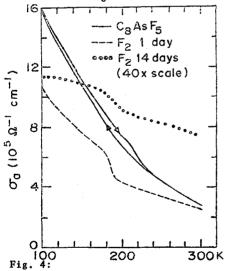
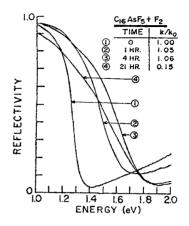


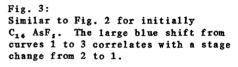
Fig. 2:

Reflectivity spectra recorded after various times of F_2 exposure for initially $C_8 As F_5$. The small initial blue shift of the metallic edge (curves 1-3) and the eventual broadening (curve 5) correlate with the initial increase followed by a decrease of k/k_o in Fig. 1.



Temperature dependent basal plane conductivity of C_8ASF_5 after 0, 1 and 14 days F_2 exposure (solid, dashed and dotted curves respectively). Note the scale change for the highly fluorinated (poorly conducting) sample.





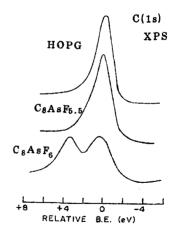


Fig. 5: C(1s) core level XPS spectra for HOPG, lightly and heavily fluorinated C₈AsF₅.

			STAGE I			STAGE 2	
<i></i>			A	B	C	D	E
٦.	carbon		4.01	4,80	5,39	4.13	5.31
2.	AsF ₅ equiv.	L -	0.501	0.600	0.674	0,258	0.332
з.	AsF ₆	BA RELEAT	0.334	0.400	0.449	0.172	0.221
4.	AsF3		0.167	0.200	0.225	0.086	0.111
5.	AsF ₆	MORAN	0.134	0.160	0.180	0.090	0.116
6.	AsF3		0.067	0.080	0.090	0.045	0,058
7.	AsF5		0.301	0.360	0.407	0.122	0.157
8.	F ₂ at k(max)	0.10	0.13	0.14	0.12	0.14	
9.	new AsF ₆		0.067	.087	.093	0.08/ 0.16	0.09/ 0.19
10.	carbon charge		c ₂₀ +	c19	c ₂₀ +	c ₁₆ / c ₂₄ +	c ₁₇ + c ₂₅ +
1.	Approx. fina composition	с _а лsғ _б			C ₈ AsF ₈₋₉		

Table I: Analysis of fluorine tensimetry data. Rows 1-2 characterize the initial compounds assuming $C_{sm}AsF_s$ overall composition (all quantities in mmol.) Rows 3 and 4 give the species present according to Bartlett /11/, while the alternative situation in rows 5-7 results from the conclusions of Moran. Row 8 is the F_2 uptake at the conductance maximum (i.e. Fig. 1) while rows 9 and 10 give the increase in ion concentration and carbon charge based on rows 5-7. The last row gives the final composition after F_2 uptake has essentially stopped.

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