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Bioinspired, Cholesteric Liquid Crystal Reflectors with Time-Controlled, Coexisting Chiral and Achiral Structures

Cécilia Boyon,¹ Vanessa Soldan,² and Michel Mitov^{*,1}

¹Centre d'Elaboration de Matériaux et d'Etudes Structurales, CEMES, CNRS, Toulouse, France.

²Centre de Biologie Intégrative, CBI, Microscopie Electronique Intégrative, METi, CNRS, University of Toulouse, France.

ABSTRACT:

The twisted structures of the chitin-based cuticle of beetles confer on them specific optical characteristics. Intrigued by the observation of Bragg gratings with a depth-dependent periodicity in the cuticle of *Chrysina* beetles, we determine the experimental conditions leading to their transcription into cholesteric liquid crystal oligomers. We correlate the optical properties of reflectors thus produced with their internal morphology, as observed by transmission electron microscopy. With the use of a single parameter, the thermal annealing time, the reflection color is made time-tunable. Different spectral bands and reflection colors from golden yellow to NIR are available, and irreversibility of the final color is reached at the end. On the basis of the design concept and these properties, these hybrid chiral-achiral materials inspire the fabrication of smart reflective labels. When encapsulated in the package of a product to be kept in cold conditions, the label records the history of the product conservation. Two kinds of information based on color changes are recorded: qualitative information reporting that the product was kept outside of the specified storage temperature and

quantitative information giving an indication of the time elapsed since the temperature exceeded the storage temperature of the product.

KEYWORDS: *cholesteric liquid crystals, photonic crystals, smart labels, time-temperature indicators, structural color, time-responsive reflective color, bioinspiration, structure-gradient materials.*

The morphology of the cuticles of beetles of the genus *Chrysina* inspires the design of reflectors combining twisted and untwisted structures. The resulting materials in turn inspire the fabrication of time temperature indicators that record the thermal history.

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INTRODUCTION

Main features of cholesteric liquid crystals. Cholesteric liquid crystals (CLCs) are chiral materials that occupy a prominent place in the field of photonic crystals and optical materials, generally speaking, because of their helical molecular organization and specific properties¹⁻⁴. First, CLCs exhibit double selectivity in wavelength and polarization. A cholesteric material may selectively reflect light, and the mean reflection wavelength λ_0 is related to the helical pitch p (periodicity, which represents a 360° rotation of the orientation of the rod-like molecules) by Bragg's law: $\lambda_0 = np \cos \theta$, where n is the average refractive index and θ is the angle between the light propagation direction and the helical axis. Second, unlike common mirrors, which reverse the polarization of the light they reflect, the cholesteric planar texture reflects circularly polarized light with the same handedness as the twist of the helix, while light with the other polarization (50% of the ambient unpolarized light at the selected wavelength) is transmitted and possesses the inverse handedness. This rule is known as the polarization-selectivity rule, and up to 50% of the ambient unpolarized light at the selected wavelength can be reflected. Third, an exceptional optical rotation may occur, up to 10^3 to 10^5 °/mm in the visible spectrum, compared with typical values between 0.01 and 100 °/mm for a chiral liquid (sugar water) or crystal (quartz). The wavelength regions for this optical activity are separated by the Bragg band and have opposite signs of rotation. A properly oriented slab of cholesteric material is thus a multifunctional material by acting as a selective filter, a vibrant-color reflector, a polarizer and an optical rotator.

Additionally, the optical properties of CLCs are tunable with control, which gives rise to a wide range of applications in the field of sensors⁵: temperature⁶⁻¹⁰, pressure¹¹⁻¹³, light^{14,15} or humidity¹⁶⁻¹⁸ sensors; electrically tunable bandpass filters^{19,20}; polarizer-free reflective displays²¹⁻²³; and smart window prototypes²⁴⁻²⁶.

Insect cuticles as cholesteric reflectors. Light reflection in the animal kingdom is diverse and ancient²⁷. The earliest known reflectors, diffraction gratings, are 515 million years (Ma) old. Fossil beetle specimens ranging from 15–47 Ma in age still reflect light²⁸ because structural coloration is generally permanent and does not bleach like pigment colors. CLC structures are omnipresent in living matter, and many studies on this topic have focused on the insect cuticle²⁹. The cuticle covers the insect body and consists of three main layers—epicuticle, exocuticle and endocuticle—secreted by a single layer of epidermal cells³⁰. Both the exocuticle and endocuticle involve twisted CLC organization of chitin macromolecules, which causes a spatially varying index of refraction³¹. Chitin macromolecules form fibrils that wrap with proteins, and these fibril-protein associations assemble into fibers. Fibers assemble into bundles, which arrange parallel to each other to form pseudolayers. Finally, the pseudolayers assemble into a cholesteric structure, often referred to as the *twisted plywood model* from Bouligand^{32,33} (Figure 1a). A continuous twist occurs along the axis perpendicular to the fibers, which interpenetrate from one pseudolayer to the next (this is why the term pseudolayer is preferred over layer). A 360° rotation of the fiber orientation defines the helical pitch. When the structure is observed by optical or electron microscopy in a direction perpendicular to the helical axis, a fingerprint texture composed of parallel stripes may appear (as in Figure 1c). The distance between two identical stripes is related to the half-pitch of the twisted organization, not to the full pitch, because a 360° rotation of the molecular orientation defines the pitch (Figure 1a).

In the search for biomimetic or bioinspired materials and devices, the structures and optical properties of insect cuticles are a first-rank source of inspiration for material scientists^{34,35}.

Motivation of the research. The motivation of our research arises from the observation of electron microscopy images of cuticular structures in scarab beetles from the genus *Chrysina*, which show vibrant reflections from bright green to metallic silver–gold, including broadband

Figure 1. Origin of bioinspired reflectors. (a) Transverse cut of the twisted plywood model showing the relationship between the half-pitch and the distance between two stripes with the same orientation of rod-like building blocks. (Adapted from ref. 61. Reprinted with permission from AAAS.) (b) *Chrysina gloriosa* (Image by M. Mitov). (c) TEM view of the cuticle of *C. gloriosa* transverse to the verticality of a green band, which shows a wax layer⁶², the chitin-based part of the cuticle and the epidermis. Pink frames designate the interfacial region between the chitin-based cuticle and epidermal cells. Close-up (d) TEM and (e) SEM transverse views at the interface between the cuticle and the epidermal cells. (f) *Chrysina aurigans*. (Image by O. Boilly, Lille Natural History Museum, France. CC BY-SA 4.0.) (g) SEM transverse view of the cuticle of *C. aurigans*. (Adapted with permission from ref. 39. © 2014 The Optical Society).

reflections³⁶. Here, we refer to the multiband—green and silver—cuticle of *Chrysina gloriosa* (Figure 1b) and the golden yellow cuticle of *Chrysina aurigans* (Figure 1f). Both cuticles of *C.*

gloriosa^{37,38} and *C. aurigans*³⁹⁻⁴² are broadband reflectors from the beginning of the visible spectrum to the NIR spectrum. To our knowledge, the nature of the organization in the border zone (localized by pink frames in Figures 1c, d, e and g) between the end of the twisted region (fingerprint texture) and the epidermis has not been discussed in the literature. Continuity of the variation in the helical pitch in the thickness direction of the cuticle and of the orientation of the helical axis in the case of *C. gloriosa*⁴³ remarkably occurs. It could thus be assumed that the end of the twisted structure does not occur abruptly but instead follows a structural gradient, as suggested by the image in Figure 1e showing the lowest stripes of the endocuticle of *C. gloriosa*: the stripes' contrast gradually fades, and then, the stripes disappear by being overwhelmed in a nonpatterned area. These incomplete lines embedded in a nonpatterned matrix could correspond to a frustrated CLC structure—prevented from complete twisting—coexisting with an unwound helix. Then, these lines are followed by an area without a pattern over a small thickness, which could correspond to a fully unwound helix, an area for which the helical pitch diverges, i.e., an achiral nematic LC (NLC) structure forced by geometric conditions that prevent the free development of a large-pitch helical structure. In this case, it would ensure a soft end to the twist of the chitin molecules. From the top to the bottom, the transverse structure of the chitinous part could thus consist of the coexistence of a pitch-gradient structure followed by a frustrated twisted structure and then a fully unwound helix, with the whole architecture being realized without the requirement of interfaces.

Our goal is not to answer stimulating questions on the morphogenesis of cuticular reflectors but instead to take inspiration from the above observations to design photonic materials with a controlled balance between chiral and achiral structures inside a single material exhibiting no discontinuity or interface—as in nature. We present a simple and effective method to fabricate bandgap-tunable reflective materials whose versatility of the characteristics of the Bragg band does not depend on temperature or external parameters but is underlain by this bioinspired

approach. An optimal use of resources and a reduced number of stages during the fabrication procedure are targeted, in the spirit of eco-design as in nature. We aim to investigate the reflectance and transmittance properties of the bioinspired cholesteric materials in close relation to their transverse structure, as observed by TEM. Finally, a concrete use of materials based on the design of hybrid (chiral–achiral) structures in the field of time-temperature indicators is presented and discussed.

RESULTS

Formulation of LC mixtures. To formulate the CLC and NLC films in relation to our research objectives, we used LC photopolymerizable and crosslinkable oligomers with a siloxane backbone (Figures 2a and b)⁴⁴. They consist of a cyclic siloxane ring to which two types of side chains are attached via aliphatic spacers: an achiral mesogen and a chiral cholesterol-bearing mesogen (Figure 2a) or only achiral mesogens (Figure 2b). When formed into a thin film with a planar texture (the helical axis is perpendicular to the film), the reflection color of the cholesteric oligomer ranges from blue to red in relation to the molar percentage of chiral mesogens in the molecule and the ring size. The helical structure is left-handed. The glassy transition temperature is approximately 0°C, and the clearing temperature is approximately 140°C. After crosslinking, the reflection wavelength becomes nearly independent of the temperature in the range from -196°C (liquid nitrogen) to 140°C. The CLC of the present study was a blend made with 30 wt. % of a blue-reflecting compound, *silicon blue* (SB) and 70 wt. % of a red-reflecting compound, *silicon red* (SR). The composition of the mixture was chosen to fit the golden yellow color of the cuticle of *C. aurigans*. However, the concept underlaid by the present study is independent of the choice of the reflection color associated with the cholesteric mixture. The SB compound contains 65% oligomer **a**, 8% monomer **c** and 27% monomer **d**. The SR compound contains 79% oligomer **a** and 21%

monomer **c**. A nematic compound, *silicon nematic* (SN), containing 70% oligomer **b** and 30% monomer **c**, was used as provided by the supplier and is the NLC of the present study. Monomers **c** and **d** were added by the supplier to the oligomers to reduce their viscosity to facilitate their spreading on a plate in the form of thin films⁴⁴. To the CLC and NLC, 0.6 wt. % photoinitiator **e** and 2.0 wt. % thermal inhibitor **f** were added.

Figure 2. Chemical structures of used chemicals. Photopolymerizable and crosslinkable (a) cholesteric and (b) nematic liquid crystal siloxane-based oligomers. (c) and (d) Functional monomers. (e) Photoinitiator. (f) Thermal inhibitor.

Preparation of experimental cells. The constitution of the cells is shown in Figure 3a. Two open layers of CLC and NLC coated on a glass plate (see Experimental Section) face each other. The thickness of the CLC films ($14.0 \pm 1.0 \mu\text{m}$) was chosen to be almost twice the thickness of the NLC ($7.5 \pm 0.5 \mu\text{m}$). Spacers $12.5 \mu\text{m}$ thick were inserted between the plates. After closing

the cell at room temperature, the open surfaces of the layers were in contact. The cell was then placed on a heating stage at 60°C. To describe the time-dependent optical and structural properties of the system, five scenarios are presented in which the annealing time was varied from 20 s to 50 min. Placing two LC substances between a pair of transparent substrates into contact to generate a concentration gradient allows us to obtain a qualitative overview of some optical textures present in the mixture⁴⁵. Figure 3b displays an optical micrograph showing a concentration gradient between the CLC and NLC compounds in the x - y plane of the substrates. The purpose is to find the conditions for generating a temporal concentration gradient between the layers of chiral and achiral materials in the z -direction perpendicular to the layers; this event has to occur at a thickness several dozen times thinner than the height of the region imaged in Figure 3b.

When the targeted time is reached, the soft material is polymerized and crosslinked with UV light (see the Experimental section). The optical properties of the material are thus fixed into an elastomeric layer, which can be cut in a transverse direction in 150-nm-thick slices to allow TEM imaging of its internal structure.

Figure 3. Representation of the structure gradient to target along with the experimental cell used for this purpose. (a) Experimental cell. (b) Optical micrograph in a planar geometry of

contact preparation between the CLC and NLC of the study showing the different phases and patterns as a consequence of a concentration gradient between the two compounds (transmission mode, crossed polarizers; the black arrows indicate the orientation of the crossed polarizers). The CLC texture is in the upper part and out of the image. The schlieren texture inherent to the NLC is visible in the lower part. The textures follow one another from the top to the bottom as follows: fingerprint texture (the helix is free to develop); cholesteric fingers in a black matrix that corresponds to an unwound helix (frustrated helix); black texture (unwound helix); and NLC texture.

Optical properties of the CLC–NLC cells. The textures and reflection spectra of five cells plus the pristine CLC film after UV polymerization and crosslinking are shown in Figure 4.

Figure 4. Reflection properties of the cells for different annealing times and at normal incidence. (a) Optical textures (reflection mode, crossed polarizers). The white arrows indicate the orientation of the crossed polarizers. Scale bar = 50 μm . (b) Reflection spectra.

The reflectance is displayed between 0 and 50% with regard to the polarization-selectivity rule (see Introduction); the signal level of the spectra is 8-10% due to the reflectance of the cover glass plate (which was not taken into account in the baseline). The mean wavelength of the Bragg band allocated to the golden yellow CLC film is 585 nm. The bandgap limits λ_{MIN} and λ_{MAX} (Figure 5), determined at the half-height of the reflection peak, are 555–620 nm (CLC film), 560–620 nm (20 s), 560–615 nm (1 min) and 560–625 nm (2 min), which means that the diffusion between the CLC and NLC films up to 2 min is not sufficient to significantly modify the position of the Bragg band inherent to the CLC film. A clear-cut bandgap shift is visible at 20 min in Figure 4, with a mean reflection peak at 650 nm (red color). Some light reflection is present above 700 nm (the signal level does not return to 10%): as shown in the following section by TEM, the 20-min sample exhibits a pitch gradient; however, the number of helix turns ascribed to the twist in this spectral region is too low to allow a significant contribution to the Bragg band from this part of the pitch gradient. The reflectance depends on the number of pitches (i.e., the number of helix turns), which means that the twist needs to develop within a minimal film thickness to enable the appearance of Bragg reflections that are detectable under standard spectrophotometry conditions⁴⁶. Annealing the cell for 50 min gives rise to an enlarged Bragg band with λ_{MIN} and λ_{MAX} equal to 690 nm and 915 nm, from dark red to near-IR. The bandgap $\Delta\lambda = \lambda_{\text{MAX}} - \lambda_{\text{MIN}}$ varies from 65 to 220 nm from the pristine golden film to the 50-min red film (inset in Figure 5). The present set of annealing times thus allows the fabrication of different cells with a structural color ranging from golden yellow to the near-IR for a single material with the same chemical composition from one cell to another, the difference only lying

in the time. An NLC is usually considered a CLC with an infinite pitch, and the pitch of a CLC can be augmented by adding a quantity of NLC^{47,48}. Here, spatially directed dissolution of CLC and NLC layers occurs in the transverse direction inside a single material, similar to the result of the design of a gradient-structure material. From 0 s to 50 min, λ_{MAX} undergoes a stronger amplitude increase than does λ_{MIN} : 295 nm vs. 135 nm (Figure 5). It may be expected that this graded stretching of the Bragg band from its *red part* is the consequence of the graded dissolution between a constant-pitch cholesteric LC and an infinite-pitch cholesteric LC.

Transmission spectra and textures are given in Figure S1. Negative peaks occur in spectral bands comparable to those of the positive peaks assigned to the reflection. Transmission investigations mainly provide additional information about the evolution of the transmittance with time (details are given in the caption of Figure S1).

Figure 5. Bandgap limits of reflection spectra as a function of the annealing time. Inset: bandwidth as a function of the annealing time.

TEM investigations of the internal structure of different CLC-NLC films. By taking advantage of the solid nature of the material, TEM allows us to correlate the temporal behavior of reflection properties with the internal structure of CLC-NLC films. TEM imaging of cholesteric polysiloxane-based oligomers reveals the typical fingerprint texture⁴⁹, similar to observations made by optical microscopy (Figure 3b). However, the periodic bright-and-dark contrast in TEM is the result of several processes, including mainly mass loss during irradiation acting as selective etching⁵⁰. Figure 6 shows the TEM textures of cross-sections accompanied by the variation in the half-pitch (distance between two bright stripes) along the depth.

Figure 6. Structural analysis of samples. TEM transverse views accompanied by the half-pitch (distance between two bright stripes) as a function of the normalized depth (depth over the total thickness of the sample) for different annealing times. The half-pitch range differs from the 20 s–1 min–2 min samples (a to c) to the 20 min–50 min (d and e) for representation purposes (the pitch changes in two different ranges). Scale bar = 2 μm .

Different regimes in the thermal diffusion between the chiral and achiral layers are revealed. Let us first remark that the sample thickness may differ from one case of annealing to another (Table S1), in particular because the film laterally spreads out when the annealing time increases. For this reason, we choose for the abscissa a normalized range (depth over the total thickness of the sample) to display the pitch variation for comparison from sample to sample of the pitch variation in the whole volume of the sample. The zero abscissa is chosen as the top of the CLC film. At a short annealing time (20 s), the whole set of bioinspired, target structures is visible (Figure 6a). From the CLC layer (at the left side of the image) to the NLC layer, the following structures successively occur: a fingerprint texture related to a constant pitch of approximately 360 nm, a frustrated fingerprint texture that materializes as deformed and larger bright-and-dark cholesteric fingers, an unwound cholesteric helix that materialize as a bright halo without stripes and the pure NLC (for reference, the TEM image of the NLC material alone is shown in Figure S2a). The term cholesteric fingers is often used in the case of strong frustration of the helical structure when the stripes are separated by homeotropic regions, which correspond to local unwinding of the helix with molecules perpendicular to the surfaces of the CLC film^{51,52}; the helix is unwound when the film thickness is close to the pitch or above it. At 1 min (Figure 6b), the fingerprint texture with a near-constant pitch is followed by a region corresponding to a CLC-NLC gradient in which the helix is frustrated; this event occurs over a thicker thickness compared to the same kind of region at 20 s (this is the result of progressive

diffusion). The textural pitch progressively increases and gives rise to umbilical structures for which the direction of the stripes is no longer parallel to the film surfaces. The NLC layer disappears due to it being consumed in the course of rapid thermal diffusion. The situation at 2 min (Figure 2c) is quite similar; the curve shows that the stripe periodicity is lowered compared to that at 1 min. A third situation is shown at 20 min (Figure 6d): the sample no longer includes frustration of the helix but instead a pitch gradient from 430 nm to 1000 nm, from approximately 32% of the thickness to the end of the film. Finally, a cholesteric structure with a nearly constant 540-nm pitch is obtained at 50 min (Figure 6e). The pitch value is 1.5 times larger than that inherent to the CLC film, similar to the result of full mixing of CLC and NLC layers.

To summarize, the annealing process presents four stages: (a) the coexistence of a constant-pitch helix, a frustrated helix, a (nematic-like) unwound helix and a nematic structure made from the pure NLC compound; (b-c) the coexistence of a constant-pitch helix followed by a frustrated helix; (d) the coexistence of a constant-pitch helix (over approximately one-third of the thickness) followed by a pitch-gradient helix; and (e) a quasi-constant-pitch helix.

DISCUSSION

The association of LC layers was realized in the literature by using cholesteric layers with the goal of broadening the characteristics of the Bragg band³, mainly the wavelength band. Two cases were considered with regard to the discontinuous or continuous (interface-free) nature of the final architecture: stacking of layers leading to a multilayer system^{53,54} and thermal diffusion between a pair of layers with distinct Bragg bands, in which diffusion aims to broaden the bandgap by covering the pristine bandgaps and the wavelengths in between^{55,56}. Our system differs from the latter case by involving a chiral layer along with an achiral layer and a diffusion process leading to different cases of narrow-band Bragg reflectors. This last characteristic is relevant for applications aimed at the field of time-temperature indicators that we discuss below.

We searched for an experimental protocol in which the reflection band was tunable by controlled interpenetration of cholesteric and nematic layers. The first layer defines the left limit of the Bragg band and is not intended to stay fixed in the course of the annealing process. The role of the second layer is to allow different scenarios for the behavior of the Bragg band that are not achievable with two CLC layers: increasing the helical pitch up to an infinite pitch, realizing a variable-pitch twisted structure, and finally realizing a constant-pitch CLC.

Time-temperature indicators record the thermal history⁵⁷. Our design inspired us to realize such smart devices on the basis of the graded, irreversible evolution with temperature and time of the label color (i.e., the pristine bilayer of the present study encapsulated in a plastic matrix or carrier). Usual temperature-variable cholesteric labels (LC thermometers⁵⁸) do not have this dual property with added irreversibility, which is of paramount importance for this kind of application. We may name these indicators *cholesteric cookies* by analogy with the web cookies designed to record a user's browsing activity. When tagged on the package of a food product or medicine—such as vaccines to be stored and transported under ultracold conditions⁵⁹—to be kept in a precise low temperature range, the color of the cholesteric indicator is frozen because the diffusion between layers is forbidden (in our case, the glassy transition temperature of LC oligomers under study is approximately 0°C). When the temperature of the product goes beyond a critical value defined in relation to the product specifications, the label color changes, and this color evolves with time. These color changes reveal to the potential user that the thermal chain has been broken and, in addition, could give an indication of the time since the temperature exceeded the storage temperature range of the product. The user would have at her/his disposal a color chart giving the relationship between color, temperature and time. The fact that the label will never return to its initial color, regardless of what the handler does, is fundamentally based on the physical interdiffusion between chiral and achiral materials. In addition to a temperature-time tunability stage, an irreversibility stage is necessary. Formulation of the

properties of the label layers (chemical composition, thickness) in their native state in relation to the temperature-color variation desired by the manufacturer is required.

CONCLUSIONS

We have designed wavelength-selective cholesteric reflectors by determining the experimental conditions leading to a fine balance between helical, distorted helical and unwound helical structures in the bulk of an interface-free polymer film. Four scenarios were described: (i) constant-pitch helix—frustrated helix—unwound helix (forced nematic structure)—stable nematic structure; (ii) constant-pitch helix—frustrated helix; (iii) pitch-gradient helix; and (iv) monotonic-pitch helix. Obtaining one scenario or another scenario at a constant temperature depends on a single parameter—time. Different spectral bands and reflection colors from golden yellow to NIR are available, and irreversibility of the final color is reached at the end. The combination of these properties precisely inspired us to use the concept of chiral-achiral hybrid structures for the manufacturing of cholesteric labels. When encapsulated in the package of a product to be kept in cold conditions, the smart label records the history of the product conservation. Two kinds of information based on color changes are recorded: qualitative information reporting that the product was kept outside of the specified storage temperature and quantitative information giving an indication of the time elapsed since the temperature exceeded the storage temperature of the product.

EXPERIMENTAL SECTION

Specimen of *Chrysina gloriosa*. A male specimen of *Chrysina gloriosa* was collected at Madera Canyon (Santa Cruz County, Arizona, USA) in July 2014.

Materials. LC photocrosslinkable side-chain oligomers with a siloxane backbone were obtained from Wacker-Chemie GmbH, Munich, Germany. The photoinitiator Irgacure[®] 907 was purchased from Ciba Specialty Chemicals Inc. The thermal polymerization inhibitor BHT was purchased from Sigma-Aldrich (W218405). Kapton sheets for spacers were provided by Micel.

Treatment of glass plates. A polyvinyl alcohol (PVA) solution layer formed on 150- μm -thick glass plates supporting the LC films was used as a promoter of planar alignment of the molecules and as a sacrificial layer to obtain free-standing films after the sample was placed in water. PVA as a powder (P8136, Sigma-Aldrich) was dissolved at 8 wt. % in distilled water. The aqueous solution was stirred and deposited on plates using a spin coater (Ossila Ltd.). A single sequence of spin-coating consisted of the deposition of 0.20 ml of solution during a program consisting of three sequences with the following spin rates and durations: 5000 rpm@20 s—4000 rpm@40 s—3000 rpm@60 s. This operation was repeated two times. The thickness of the dried film was approximately 2 μm . The film was finally rubbed three times with a rayon cloth (YA-20R, Yoshikawa Chemical Company, Ltd.).

Preparation of the contact sample. A 12.0 ± 1.0 - μm -thick sandwich film of CLC was formed at 60°C between untreated glass plates. The NLC was inserted at 80°C by capillarity into the cell. The cell was annealed at 73°C for 15 hours. The image in Figure 3b was taken at 80°C.

Preparation of individual films. Two distinct 585-nm-reflecting CLC and transparent NLC samples were prepared on a heating stage at 80°C. The NLC was used as provided by the supplier. The CLC was a blend made with 30 wt. % of a blue-reflecting compound (SB) and 70 wt. % of a red-reflecting compound (SR). To the CLC blend and NLC compound, 0.6 wt. % of

the photoinitiator Irgacure[®] 907 and 2.0 wt. % of the thermal polymerization inhibitor BHT were added. A 14 ± 1 - μm -thick sandwich film of CLC was formed between two treated glass plates and annealed at 80°C. A 7.5 ± 0.5 - μm -thick sandwich film of SN was formed between two plates and annealed at 100°C. The thickness was measured by the interference method (based on the wave interference due to the internal reflection between the two inner surfaces of the cell substrates⁶⁰). The dispersion in thickness of the individual CLC and NLC films constituting the five samples was kept as low as possible. Special care was taken to ensure that the NLC films had a monodomain texture (Figure S2b) and exhibited a very high transmission (90%—Figure S3). To remove one plate and obtain open films while avoiding the formation of scratches on the surface of the film, the sandwich cells were kept at low temperature in a refrigerator; one of the plates was sharply removed from the film with the aid of a Handi-Vac[®] coverslip pick-up tool (Ted Pella, Inc.). That the free surface of the film remained smooth after the operation was checked by optical microscopy.

Polymerization conditions. The films were placed at 60°C on a heating stage and irradiated with UV light for 10 minutes with a maximum intensity at 365 nm equal to 0.50 mW/cm².

Inclusions of synthetic samples for ultramicrotomy. To obtain free-standing films ready to be embedded in a piece of resin, each sample was placed in a vial of distilled water for 1 day. Due to the presence of water-soluble PVA layers, the films detached as a block from the plates while maintaining clear-cut surfaces, as verified by TEM transverse views. A small piece of material was embedded in EMbed-812 resin (Electron Microscopy Sciences), which was then cured at 62°C for 2 days.

Inclusions of biological samples for ultramicrotomy. The cuticle samples corresponded to the elytra and were removed with a razor blade. A small piece of material was embedded in EMbed-812 resin (Electron Microscopy Sciences), which was then cured at 62°C for 2 days.

Preparation of biological samples for TEM by ultramicrotomy. A diamond knife at ambient temperature was used to cut 80-nm-thick slices (Figure 1c) or 120-nm-thick slices (Figure 1d) with a Leica UCT ultramicrotome. The samples were cut perpendicular to the film surface (cross-sections). Slices were placed on single-slot Formvar-coated copper grids (GS2x1-C3, Gilder Grids Ltd.) and stained with UranylLess (brand mixture of lanthanides from Delta microscopy) for 20 min and with 3% Reynolds lead citrate (from Chromalys) for 5 min.

Preparation of synthetic samples for TEM by ultramicrotomy. A diamond knife at ambient temperature was used to cut 150-nm-thick ultrathin slices with a Leica UCT ultramicrotome. The material was cut perpendicular to the film surface (cross-sections). Slices were placed on single-slot Formvar-coated copper grids (GS2x1-C3, Gilder Grids Ltd.).

TEM conditions. A Jeol JEM-1400 microscope operating at 80 kV equipped with a Gatan Orius SC1000B CCD camera was used.

Periodicity profiles from TEM images. Quantitative image analysis was performed with DigitalMicrograph software from Gatan. The intensities of the gray levels on a given surface of the images (typical surface = $8 \times 18 \mu\text{m}^2$) were analyzed using the Profile option. The distance between two consecutive positive or negative peaks corresponded to the distance between two successive stripes of the same contrast in the fingerprint texture. Positive (*resp.* negative) peaks correspond to bright (*resp.* dark) stripes because the transmitted light is higher (*resp.* lower) in

bright (*resp.* dark) stripes. Peaks were excluded from the analysis when they corresponded to intensity changes related to noisy peaks for which the measurement was questionable (this explains why experimental points are sometimes lacking in Figures 6a to e).

SEM conditions. A Zeiss 1540 XB Crossbeam SEM equipped with a low-energy-capable focused ion beam (FIB) column was used (Figure 1e). WD = 4.9 mm. EHT = 20 kV.

Optical micrographs. Textures were photographed with an Olympus BX51 stereomicroscope equipped with an Olympus DP73 digital camera. A 50x objective with a numerical aperture (NA) of 0.50 was used.

Optical measurements. The reflectance and transmittance spectra of the materials were collected at RT and normal incidence with unpolarized incident light (halogen source HL-2000 from Mikropack GmbH emitting from 360 nm to 1700 nm). An HR2000CG-UV–NIR spectrophotometer from Ocean Optics was used. Baselines were recorded with air (*resp.* a mirror) in the beam path for transmittance (*resp.* reflectance) measurements.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at:

<https://pubs.acs.org/doi/10.1021/acsami.1c08218>

Transmission properties of the cells for different annealing times; TEM and optical textures of the NLC sample; Typical transmission spectrum of an individual NLC film; Thickness of CLC and NLC individual films and samples after polymerization (PDF).

AUTHOR INFORMATION

Corresponding Author

*E-mail: mitov@cemes.fr

ORCID

Michel Mitov: [0000-0003-1831-2152](https://orcid.org/0000-0003-1831-2152)

Author Contributions

C. B. contributed to the development of samples, carried out the manufacture of samples and cells along with their optical characterization, and participated in the collection of data and in the analysis of results. V. S. prepared and cut the inclusions by ultramicrotomy and carried out TEM observations. M. M. led the conceptual development of the research, oversaw data collection, performed the analysis of results, organized the content of the manuscript, prepared the figures and wrote the manuscript. All authors revised the manuscript and gave approval to the final version.

Notes

The authors declare no competing financial interest.

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Supporting Information

Bioinspired, Cholesteric Liquid Crystal Reflectors with Time-Controlled, Coexisting Chiral and Achiral Structures

Cécilia Boyon,¹ Vanessa Soldan,² and Michel Mitov^{*,1}

¹Centre d'Elaboration de Matériaux et d'Etudes Structurales, CEMES, CNRS, Toulouse, France.

²Centre de Biologie Intégrative, CBI, Microscopie Electronique Intégrative, METi, CNRS, University of Toulouse, France.

*Corresponding Author. E-mail: mitov@cemes.fr

Figure S1. Transmission properties of the cells for different annealing times and at normal incidence. (a) Optical textures (transmission mode, crossed polarizers; the white arrows indicate the orientation of the crossed polarizers). Scale bar = 50 μm . (b) Transmission spectra. For the 20 s, 1 min and 2 min samples, the transmittance of the whole signal, inside and outside of the Bragg band, is low with regard to that of the CLC film; the latter exhibits a high transmittance from 85% at 450 nm to 95% at 1000 nm. These low diffusion times do not allow the films to reach a clear-cut planar texture after the two layers are put into contact: in (a), peculiar oily streaks in the textures related to the 20 s, 1 min and 2 min samples are visible, which are absent in the texture related to the CLC film. As a consequence, diffuse reflection occurs instead of specular reflection. This observation is congruent with that of the fingerprint textures of cross-sections shown in Figures 6a to c: the stripes are slightly wavy in the contact region between the two layers corresponding to a frustrated helical structure; consequently, the

direction of light propagating through this region is not kept perpendicular to the film surfaces. A better planar texture is obtained at 20 min: the transmittance level is higher, and in (a), the texture shows oily streaks thinner than those seen at 20 s, 1 min and 2 min; their density is also lower. A planar texture with a transmittance level closer to that of the CLC film is shown in the 50-min film.

Figure S2. Textures of the NLC sample. (a) TEM transverse view of a 10-μm-thick polymerized NLC sample. No patterns or defects are visible. This is probably the consequence of the clear-cut planar alignment in the whole volume of the thin NLC film. (b) Optical micrograph (transmission mode, crossed polarizers; the white arrows indicate the orientation of the crossed polarizers). The black arrow indicates the rubbing direction of the alignment layers.

Figure S3. Typical transmission spectrum of an individual NLC film.

	CLC	NLC	20 s	1 min	2 min	20 min	50 min
Thickness (μm)	14.0 ± 1.0	7.5 ± 0.5	18.5 ± 0.5	17.5 ± 0.5	18.0 ± 0.5	22.0 ± 0.5	13.0 ± 0.5

Table S1. Thickness of individual CLC and NLC films—as calculated from the interference method (see the Experimental section), and samples after polymerization (as calculated from TEM transverse images).