

## Supplementary material:

# Ligand-dependent nano-mechanical properties of CdSe nanoplatelets: calibrating nanobalances for ligands affinity monitoring.

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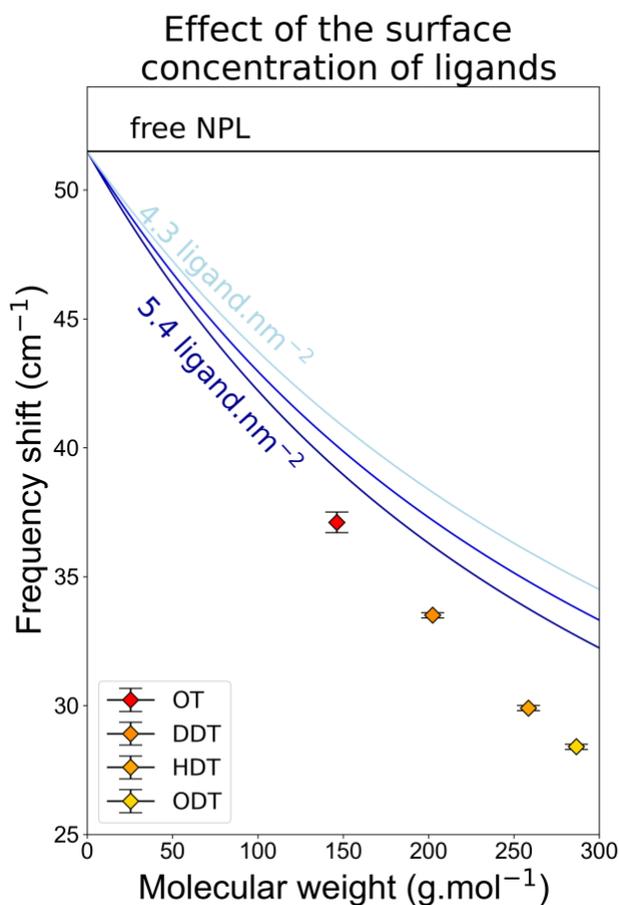
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### **A. Effect of the surface concentration at the surface of a Nanoplatelet**

The surface concentration used in the paper was deduced by considering that a ligand can bind with 2 Cd<sup>+</sup> ions per unit cell of area  $a^2$ , with  $a = 0.608$  nm is the lattice parameter of CdSe zinc blende. Complete coverage (all Cd atoms on the surface are linked by a ligand) therefore corresponds to a surface concentration of 5.4 ligands.nm<sup>-2</sup>. This assumption is probably an overestimation of the reality and figure S1 shows the effect of decreasing the surface concentration at 4.3 ligands.nm<sup>-2</sup> (80% of the Cd atoms are bond with a ligand). As the surface coverage decreases the theoretical frequency of the NPL vibration increases. The difference between experiment and theory therefore increases when the surface concentration is reduced and tends towards a constant frequency (free NPL vibration).

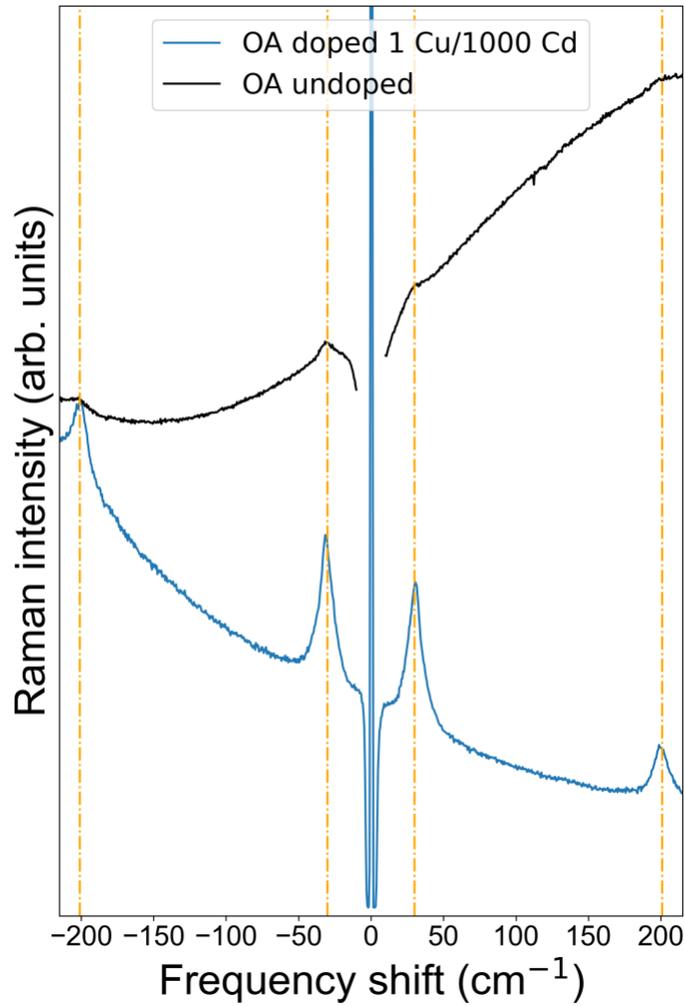


**Figure S1:** Effect of the surface density of ligand. The surface density was varied from 5.4 to 4.3 ligand.nm<sup>-2</sup> and injected into the massload model considering a 4 monolayers thick NPL. The upper horizontal line corresponds to a free NPL, that is to say a surface concentration of 0 ligand.nm<sup>-2</sup>. Experimental data are also represented for comparison.

### B. Effect of Cu doping

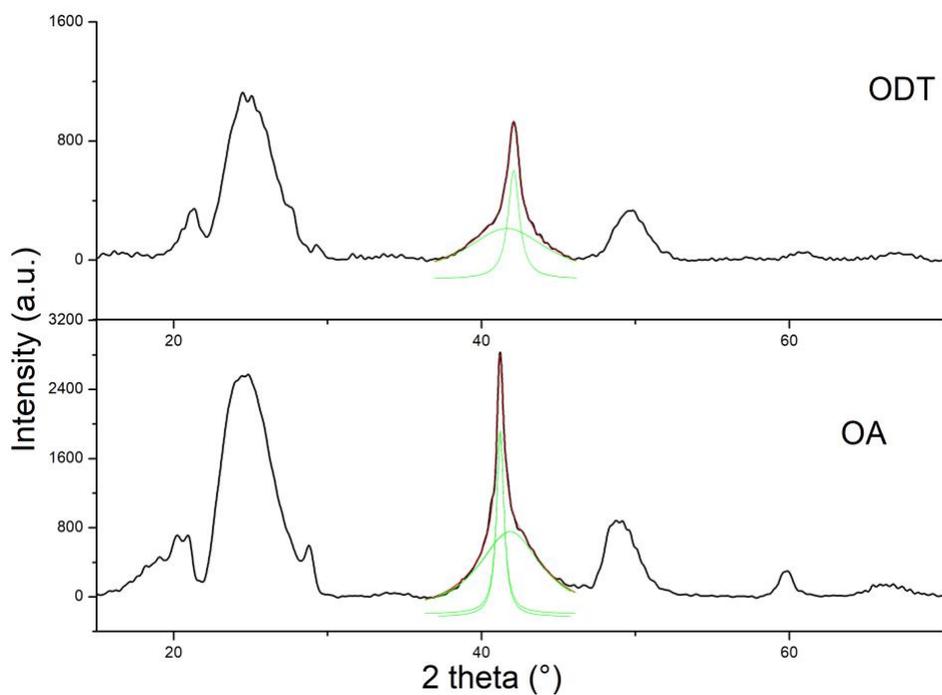
To resolve the Raman peaks, it is necessary to quench the luminescence of NPLs. To achieve this goal, we developed a copper doping technique to significantly reduce the luminescence yield. The raw Raman spectra of the doped and undoped NPLs of 4MLs are shown in figure S2. Both

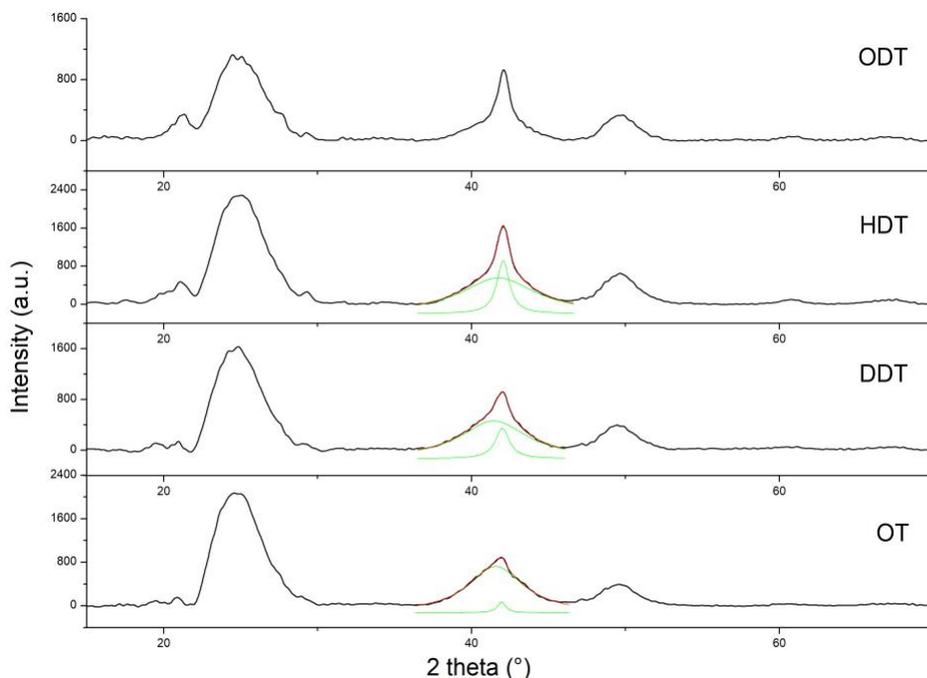
spectra have been normalized to the intensity of the anti-stokes longitudinal optic mode peak for ease of comparison. When the doping ratio increases to 1Cu atom for 1000 Cd atoms, the luminescence background is drastically reduced, allowing Raman peaks to be resolved. On the blue curve, well-defined Raman peaks are observed at the same frequency than for the black curve, showing that the copper doping used (1/1000) has no effect on the vibrational properties of the NPLs.



**Figure S2:** Raman spectra of NPLs of 4 MLs (510 nm exciton) with OA ligands. The black and blue curves correspond respectively to the Raman spectra of the undoped and doped (1Cu/1000Cd) NPLs. Vertical orange dotted lines are guides to the eyes. The spectra have been normalized to the intensity of the anti-Stokes LO phonons for a clear comparison.

### C. Diffractograms





**Figure S3:** diffractograms and the associated fits to extract lattice parameters corresponding to 3ML CdSe NPLs from the same batch and different surface ligands: Oleic Acid (OA), Octadecanethiol (ODT), Hexadecanethiol (HDT) Dodecanethiol (DDT) and Octanethiol (OT).

#### **D. Determining the effect of a lattice parameter variation on the elastic constant of bulk CdSe**

The initial hypotheses are the following:

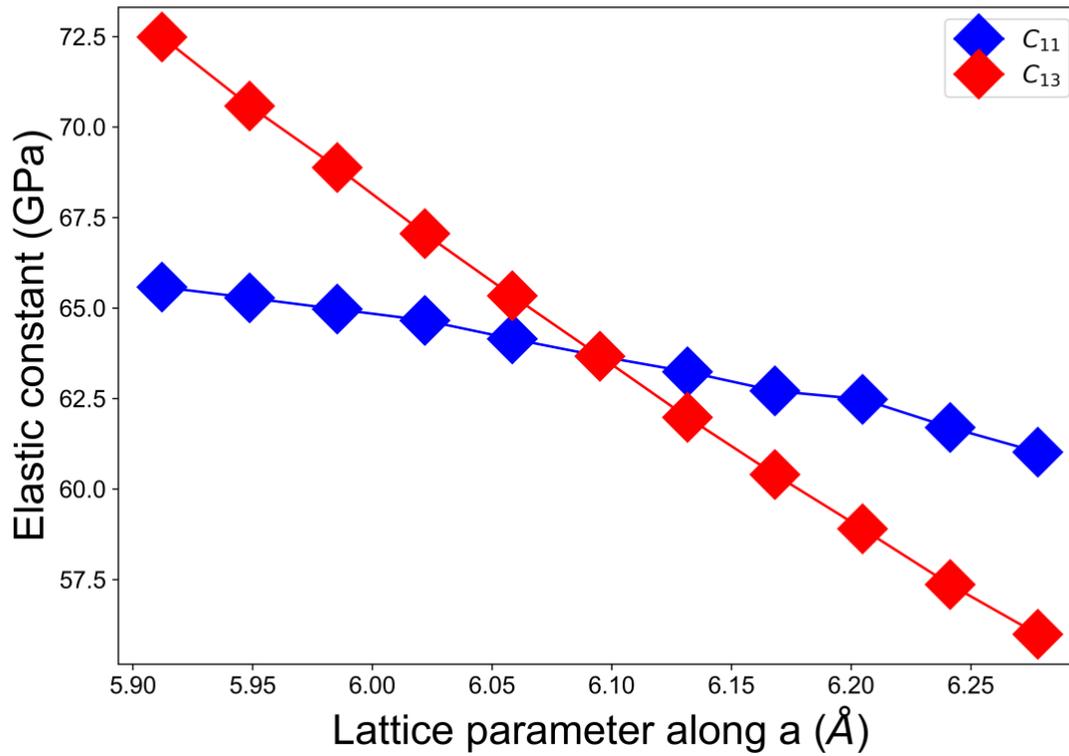
The crystalline structure is Zinc-Blende with  $a_0 = 6.09500$  angstrom (after optimization). There are 8 atoms per cell (4 Cd, 4Se). Using these parameters, the elastic constant at 0K have been calculated by finite differences and are reported in table 1.

	$C_{11}$	$C_{12}$	$C_{44}$
CdSe (Zinc-Blende)	63.67 ( $\pm 0.02$ )	45.15 ( $\pm 0.01$ )	22.31 ( $\pm 0.002$ )

**Table 1.** Elastic constants (GPa) for zinc-blende CdSe at 0K calculated using VASP PBESOL

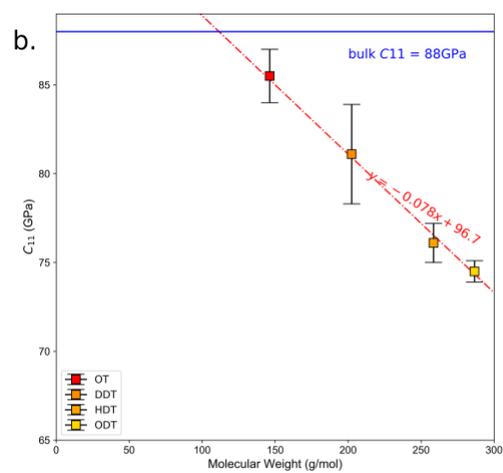
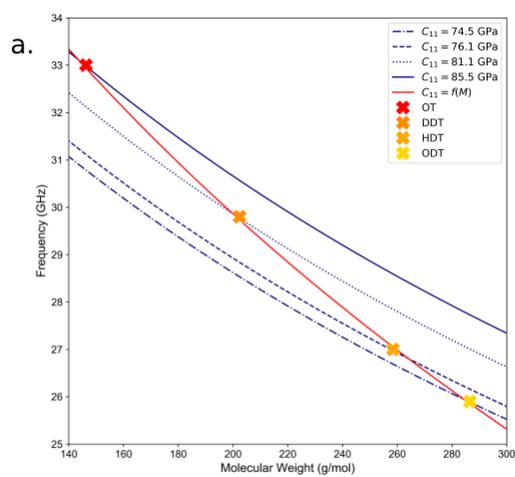
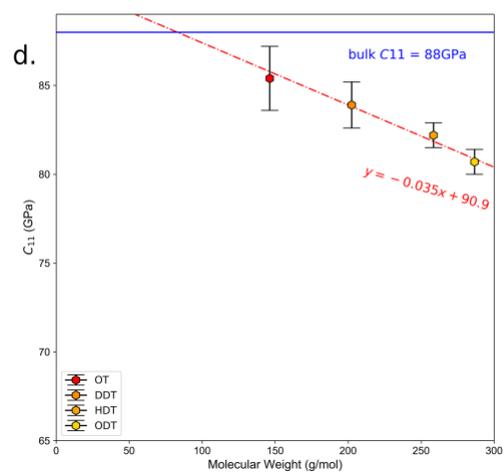
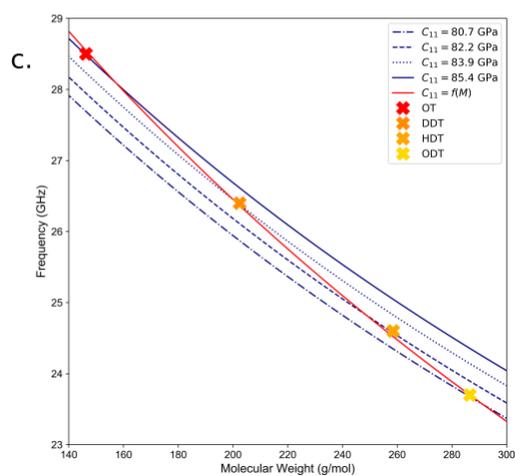
Using these values, we obtained a bulk modulus of 51.32 GPa.

We have then varied the lattice parameter along the crystalline direction  $a$  in the range -3 to +3 %. The lattice parameter along  $b$  and  $c$  are fixed to 6.0950 angstroms. We consider that  $C_{11}$  is the elastic modulus in the direction of the deformation, and  $C_{33}$  is the average along the two other directions (that must be equivalent). The obtained data are represented in figure S1.



**Figure S4.** Dependency of the elastic constant  $C_{11}$  and  $C_{33}$  when the lattice parameter is modified along the crystalline direction  $a$ .

### E. Experimental dependency of $C_{33}$ parameter for 5 (4+1) and 6 (5+1) MLs.



**Figure S5. a. and c.** fitting method: the blue lines correspond to the fit applying the mass-load model with a constant  $C_{11}=C_{33}$  in the case of NPLs containing 5 (4+1) and 6 (5+1) MLs respectively. The crosses represent the measured frequencies. The red curve is the mass-load model considering the dependency of the  $C_{33}$  deduced from **b. and d.** which represents  $C_{33}$  as a function of the molecular weight for NPLs of 5 (4+1) and 6(5+1) MLs, respectively.

Figure S3 shows the fitting method applied to deduce the dependency of  $C_{33}$  as a function of the molecular weight. Typically, the  $C_{33}$  is considered as a free parameter in the mass-load model. Then a least square method is applied to compare the frequency measured and the calculated frequency at the same molecular weight (fixed parameter). The error bars in figure S2 b and d are calculated by considering the error deduced from the measurements. Finally, a linear fit is applied to deduce the elastic constant  $C_{33}$  dependency as a function of the molecular weight.