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Colloidal and Supported TiO₂: Towards Non-Extractable and Recyclable Photocatalysts for Radical Polymerizations in Aqueous Dispersed Media

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Abstract

Two high surface area titania forms, dispersed as stable nanoparticles or coated on fused silica microfiber, are used as non-leachable photocatalysts for the radical photopolymerization of methyl methacrylate miniemulsion in water. At low loading (0.17 wt% / wt_{monomer}), both nanoscale TiO₂ yield ca. 50% conversion after 10 min UV irradiation, compared to 63% with a conventional type I photoinitiator (hydroxyacetophenone). High-molecular-weight values (> 180 kDa) and a polydispersity index of about 1.5 are achieved, indicating that undesirable degradation is negligible. In the proposed mechanism, a surface initiation takes place through the generation of hydroxyl radicals from H₂O oxidation. We show that irradiance and TiO₂ content are two key parameters for controlling molecular weight and conversion values. The supported TiO₂ form can be easily recovered and reused up to 4 times, despite a gradual reduction in conversion. Deposited on the wall of the annular photochemical reactor, it enables reaction scaling-up.

FIGURE FOR ToC_ABSTRACT

1. Introduction

Issues including initiator odor, toxicity, migration and contamination, which may affect the properties or uses of polymer products, are currently at the centre of public debate, especially in the context of consumer and worker protection. The problem is particularly acute in UV-curing technology, which is used for the large-scale production of industrial coatings, adhesives, inks and other photosensitive materials.^[1,2] Free radical UV-curing is characterized by sub-second cycle times and the need of high photoinitiator (PI) concentrations to overcome oxygen inhibition, especially in fields such as the printing industry. This leads to a significant amount of non-reacted PI remaining in cross-linked films. Due to their low-molecular-weight, PIs are able to migrate within the film network, and may extract from the surface. In the field of food packaging, this mechanism may result in contamination, and spawned the important food product scandals of recent years which have permanently affected the consciousness and driven more stringent regulations.^[3]

At present, the main response of the industry concerns the development of polymer initiating systems based on oligomeric type II PIs, hence avoiding the release of low-molecular-weight photoproducts. These initiating systems typically involve a couple of non-migratory oligomeric photosensitizers (based on thioxantone or benzophenone^[4]) and a hydrogen donor (polymeric amine) generating initiating species of high-molecular-weight.^[5] However, these “polymeric” PIs involve additional cost, lead to higher viscosities of the UV-curable formulations, and decrease polymerization reactivity when used at similar weight ratios as low-molecular-weight analogues.

To overcome these limitations, semiconductor nanoparticles have been recently presented as potential non-migrating radical photoinitiators.^[6] However, the capacity of semiconductors such as TiO₂, ZnO or CdS to act as initiator of radical photopolymerization was already demonstrated several decades ago.^[7–12] Today, nanoscale semiconductor particles, especially nanoTiO₂, are best known as photocatalysts for advanced oxidation processes (AOP).^[13,14]

TiO₂-photocatalysis is induced by the generation of excitons (electron-hole pairs) upon absorption of UV-radiation by the semiconductor leading to redox reactions of compounds adsorbed at the particles surfaces, *e.g.* molecular oxygen (O₂) and/or water. The resulting reactive oxygen species (ROS), *i.e.*, hydroxyl radical (HO•) cause oxidation and subsequent mineralization of organic contaminants present in water or air.^[15] Superoxide radical anion (O₂^{•-}), generated by the reduction of O₂, might act as a reductant but is not known to induce subsequent radical reactions. Based on the premise that some ROS, such as HO•, might initiate radical polymerizations, Kraeutler et al. demonstrated as early as 1979 that fumed TiO₂ could replace a PI when dispersed in an organic solvent containing methyl methacrylate (MMA).^[16] Mechanistically, it might be assumed that the holes of the electronically excited TiO₂ react by electron transfer with an alcohol (ROH) or a carboxylate (RCOO⁻) added as co-initiator, thus generating C-centered radicals (R•) either by loss of a proton (primary or secondary ROH) or by loss of CO₂ from the intermediary RCOO•. Such initiating systems based on a semiconductor nanoparticle/coinitiator tandem have attracted considerable interest in radical photopolymerization.^[17–25] Nevertheless, non-reacted co-initiators may be prone to migration in contrast to nanometer scale semiconductor particles that are not supposed to diffuse in the polymer network. Alternative pathways involving semiconductor nanoparticles proceeded by photoinduced electron transfer to transition metal ions to trigger atom-transfer radical polymerization (ATRP)^[25,26] or bulk radical polymerization using surface Pt(II), Cu(II) or Fe(III) modified ZnO nanoparticles.^[6]

Yet, the simplest method to avoid the formation of residual migrating species is probably to rely on hole oxidation of water to create initiating hydroxyl radicals (HO•) that are known to be very effective for the oxidation of organic compounds in aqueous solution. The first TiO₂-photocatalyzed polymerization studies were performed in aqueous solution using dispersed titania powder. This restricted the scope of this approach to a short list of water-soluble

monomers (acrylamide,^[8,24,27] acrylic acid^[27] or methacrylic acid^[28]) or partially water-soluble monomers (MMA).^[22,29,30]

Given that most utility monomers are insoluble in water, we suggest that polymerization in aqueous dispersed media may be a more convenient and versatile system to perform nanoscale TiO₂-photoinitiated polymerizations without generation of migrating photoproducts. Currently, polymerization in dispersed media accounts for nearly 20% of the worldwide polymer production, and encompasses a number of distinct processes. Mainly suspension and emulsion processes have been used industrially to produce some of the most important commodity thermoplastics, including poly(vinyl chloride) or styrene-acrylic polymers. Recently, Song et al. showed that solid TiO₂ nanoparticles can act both as photocatalyst and Pickering stabilizer (by adsorption on droplet surface) in emulsion polymerization of styrene.^[31,32]

Keeping this study in mind, we investigated in a first step a model MMA miniemulsion to confirm the feasibility of a TiO₂-photoassisted polymerization in an aqueous dispersed medium. But in all likelihood, this approach may be applied to a range of aqueous monomer emulsions. By a careful control of the amount of added TiO₂ and the conditions of irradiation, we could show that the production of high-molecular-weight acrylic latex can be achieved without substantial chain degradation. In this work, two nanoscale TiO₂ forms have been investigated, as depicted in **Figure 1**: *colloidal dispersion* and *nanosized layer coated on fused quartz fiber*. Both systems have specific advantages, and their initiating efficiencies have been compared to that of a conventional water-soluble type I PI.

1. *Colloidal TiO₂* offers significant process advantages: direct addition to any aqueous monomer dispersion without emulsification or dissolution stage, high surface-to-volume ratio to ensure optimal reactivity, and small average particle diameter (≈ 30 nm) to limit scattering and attenuation of UV radiation. However, the catalyst cannot be extracted after the latex

synthesis stage. This raises concern on the UV photostability of the final product (waterborne coating or adhesive), in particular for outdoor applications.

2. *TiO₂-coated quartz fiber* can act as a “supported” catalyst.^[33] Quartz fibers offer an inert and UV transparent support, while the thin TiO₂ layer deposited on the micrometric fiber may provide a specific surface area for an optimal contact with the monomer emulsion and high reactivity. Supported TiO₂ may be seen as a promising avenue for recoverable and reusable PIs, lessening the environmental footprint of photopolymerization reactions. The main challenge is the risk of fouling of the fiber surface during photopolymerization, suppressing the photocatalytic activity.

Figure 1. Supported and colloidal TiO₂ as photocatalysts for photoinitiation of radical polymerization of monomer aqueous emulsions

2. Materials and methods

2.1. Materials

Technical grade methyl methacrylate (MMA) was purchased from Aldrich and used without further purification. Octadecyl acrylate (OA, Aldrich) was employed as co-stabilizer of monomer miniemulsions and sodium dodecyl sulfate (SDS, Aldrich) was used as a surfactant for preparing the MMA miniemulsions. Distilled water was used in the preparation of the miniemulsions. Used for comparison, the conventional type I water-soluble PI, Irgacure 2959 (2-hydroxy-1-[4-(2-hydroxyethoxy)phenyl]-2-methyl-1-propanone), was provided by BASF

Specialty Chemicals. Nanoscale titanium dioxide was used in two different forms, (i) as a dispersion in water (33-37 wt%, specific surface area of 49 m²/g) produced by Buehler and sold by Aldrich. According to manufacturer information, this stable colloidal dispersion has an average diameter lower than 30 nm, and 90 % of the size distribution lies below the threshold value of 150 nm. (ii) as a TiO₂-coated fused silica fiber provided by Saint Gobain (Quartzel® PCO). Currently used for air or water treatment, this product consists of a felt composed of entangled fiberglass with a nominal TiO₂ thickness of 15-20 nm, and a specific surface area of ca. 120 m²/g.

2.2. Miniemulsion photopolymerisation

The formulation of the 20 wt% solids content monomer miniemulsion is described in **Table 1**. In a typical procedure, an organic phase was first prepared by dissolving the co-stabilizer (OA, 4 wt% / wt_{monomer}) into the monomer. The latter was then added to an aqueous SDS solution (2 wt% / wt_{monomer}). Both phases were then stirred during 10 min at 700 rpm. The resulting coarse macroemulsion was then sonicated during 5 min using a Branson Sonifier 450 (450 W/L) (9 output control, 80 % duty cycle) while maintaining the stirring. The PI was introduced differently depending on its structure. The water-soluble type I initiator or colloidal TiO₂ were simply added to the aqueous phase before the emulsification stage. As regards to the TiO₂-coated fiber glass, the fabric was introduced in the reactor vessel containing the monomer miniemulsion just before irradiation. In the three cases, the weight concentration of the photochemically reactive substance (molecular PI or TiO₂) was kept constant with respect to the monomer content in order to enable comparison. Regarding *c*-TiO₂, the concentration in TiO₂ can be easily deduced from the solids content value. For *f*-TiO₂, the mass content was estimated based on the TiO₂ weight content compared to the total weight obtained from Energy-dispersive X-ray spectroscopy (EDX) data.

Table 1. Miniemulsion formulation

	Compound	Weight
Continuous phase	Water	20 g
PI	<ul style="list-style-type: none"> Type I water-soluble PI 	0.17 % wt% / W_{monomer}^a

	<ul style="list-style-type: none"> TiO₂ dispersion in water (<i>c</i>-TiO₂) TiO₂-coated quartz fiberglass (<i>f</i>-TiO₂) 	
Surfactant	SDS	0.075 g
Monomer	MMA	5 g
Co-stabilizer	OA	0.2 g

^a The concentration is that of the active substance: type I PI or TiO₂ in the case *c*-TiO₂ and *f*-TiO₂

Figure 2 depicts the photopolymerization set-up. The reaction was performed in a 1 mm-thick quartz rectangular cell (340 μL) without stirring and without degassing except when explicitly mentioned. Polychromatic irradiation was applied vertically using a focused beam generated by a medium-pressure Hg-Xe arc lamp (Hamamatsu L8252, 365 nm reflector, 200 W) coupled to a flexible light-guide. Adverse effects from heat (IR radiation) were removed by a 365 nm elliptical cold reflector. Irradiation below 310 nm was removed using a glass filter to avoid self-initiation of acrylate monomers. An irradiance of 430 mW cm^{-2} for the spectral range 310–600 nm was measured by a radiometer (Ocean Optics HR4000 spectrometer) at the surface of the sample at full opening of the diaphragm. The light was tuned by varying the opening of the diaphragm.

Figure 2. Experimental set-up for miniemulsion photopolymerization **1.** Medium pressure Xe-Hg arc lamp, **2.** Long-wave elliptical reflector, **3.** Light guide, **4.** Glass filter, **5.** Spectroscopic cuvette with an optical path of 1 mm containing the miniemulsion

A scale-up of the f -TiO₂-induced photopolymerization was performed in batch using an annular photochemical reactor (Peschl Ultraviolet®). A standard medium pressure Hg arc lamp (Heraeus Noblelight TQ 150, arc length: 4.4 cm) emitting a series of rays from 250 nm to 600 nm is used for irradiation. The lamp is housed in a fused borosilicate sleeve to avoid irradiation at wavelengths below 310 nm. Because of the important IR-emission of the lamp, an external cooling jacket keeps the photoreactor contents at a temperature between 20 and 25 °C throughout the polymerization reaction. The combination of lamp, sleeve and cooling device is introduced into the cylindrical reactor vessel made of borosilicate accommodating up to 120 mL of monomer miniemulsion. The distance between the two annular sections (external sleeve wall and inner reactor wall) is 9 mm and defines the optical path length. Magnetic stirring of the miniemulsion is maintained throughout the photochemically initiated polymerization. Samples were drawn during polymerization to determine conversion.

2.3 Characterization

The average diameter of the monomer droplets (D_d), latex particles (D_p) and colloidal TiO₂ particles were measured by dynamic light scattering (DLS) using a Zetasizer nano ZS (Malvern Instruments). Conversion was assessed by gravimetric analysis. Molecular weights of the different latexes prepared were determined by size exclusion chromatography by using a PL-gel mixed D (5 μ m, 300 mm $\times$ 7.5 mm) Shimadzu column calibrated with polystyrene. The molecular weights provided are then given in polystyrene equivalent. Both types of TiO₂ were characterized by X-ray diffraction (XRD, Philips X'Pert Pro PAnalytical). Transmission electron microscopy (TEM) was carried out on a Philips CM200 microscope operating at 30 kV. Scanning electron microscopy (SEM) was carried out on a Jeol JCM 6000 microscope operating at 15 kV. Quartzel® fibers were analyzed before and after photopolymerization and the TiO₂ content was measured by EDX.

3. Results and discussion

3.1. Characterization of colloidal and supported nanoscale TiO₂

Colloidal TiO₂ (*c*-TiO₂) and TiO₂-coated fiberglass (*f*-TiO₂) were characterized by a range of techniques prior to their use. All data are summarized in **Table 2**. Concerning the morphology of *f*-TiO₂, the SEM analysis showed an entangled, curly continuous fiber structure displaying a mean fiber diameter of 10.5 μ m. Fiber surface appeared smooth and defect-free, *i.e.* without any area non-covered by the titanium dioxide layer (15-20 nm thickness according to the manufacturer). EDX analysis revealed a TiO₂ concentration of 1.4 wt% (see **Figure S1** in Supporting Information). TEM analysis of *c*-TiO₂ showed an ensemble of non-spherical particles that are roughly 24 nm in diameter. In contrast, DLS showed a much greater (hydrodynamic) average diameter (90 nm). TEM provides information about particles which are dehydrated and immobilized on a solid support. This can lead to structural distortions compared to the solvent swollen state of DLS analysis. Discrepancy between DLS and TEM size data is not necessarily contradictory; it may emphasize that inorganic nanoparticles like

TiO₂ are prone to aggregation due to strong Van der Waals interactions. Additionally, TEM provides information about the high crystalline structure of the individual nanoparticles. The two main physico-chemically distinct polymorphs of TiO₂ are anatase and rutile. The precise crystalline phase was determined by XRD analysis (see **Figure S2** of SI): *c*-TiO₂ has a pure anatase structure that is known to have a higher photocatalytic activity than rutile but is thermodynamically less stable.^[34,35] By contrast, the XRD pattern of *f*-TiO₂ revealed a coexistence of the two different crystallographic forms rutile/anatase in the ratio: ca. 80:20 %.

Table 2. Main characteristics of the two nanoscale TiO₂, either coated on amorphous silica fiber (*f*-TiO₂) or dispersed in water as nanoparticles (*c*-TiO₂)

	<i>f</i> -TiO ₂	<i>c</i> -TiO ₂
MORPHOLOGY (SEM, TEM)
CRYSTAL PHASE (XRD)	Pure anatase	anatase/rutile: ca. 80:20
SIZE DATA	Fiber diameter: 10.5 μm (SEM) TiO ₂ layer thickness: 15-20 nm Specific surface area: 120 m ² /g	Hydrodynamic particle diameter (<i>D_d</i>): 90 nm (DLS) Dry particle diameter: 24 nm (TEM) Specific surface area: 49 m ² /g ^[17]
TiO ₂ CONTENT	1.4 wt%	35 wt%. in water

3.2 Nanoscale TiO₂-assisted photopolymerization of a MMA miniemulsion

Table 3 compares conversions, molecular weights, and particle sizes of the products obtained by the photopolymerization of a MMA miniemulsion ($D_d = 50$ nm) irradiated during 10 min in presence of *f*-TiO₂, *c*-TiO₂ or of a conventional type I PI (hydroxyacetophenone). All photopolymerizations were performed in a 1-mm thick spectroscopic cell to optimize the irradiated volume fraction.

Table 3. Comparison between different initiation modes for the miniemulsion photopolymerization of MMA. Irradiation conditions: Hg-Xe lamp, 430 mW/cm², 310-600 nm, 10 min.

PI	Conv. %	M_n kDa	M_n/M_w PDI	Particle size ^a D_p , nm
Type I PI	63	176	2.8	63
<i>c</i> -TiO ₂	54	381	1.6	65
<i>f</i> -TiO ₂	43	181	1.4	56

^a Determined by DLS

Both nanoscale TiO₂ proved to be efficient PI, yielding ca. 50% conversions after 10 min of irradiation compared to 63% for a conventional PI. Control experiments without PI confirmed the absence of monomer consumption upon irradiation. Despite a greater surface area (0.84 m² vs 0.42 m²), *f*-TiO₂ yields lower conversion (43 %) compared to *c*-TiO₂ (54 %). To account for this result, we postulate that the immobilization of the catalyst may cause a less accessible surface area, and reduces the number of active catalyst sites compared to a finely dispersed *c*-TiO₂ system. A second most likely explanation is that the surface of *f*-TiO₂ absorbing the incident radiation may be smaller than the corresponding surface of *c*-TiO₂ dispersed over the whole reactor volume. The M_n values of the precipitated polymers are of the same order or even larger in the case of the photocatalyzed polymerization. The high-molecular-weight values rule out that significant oxidative degradation may occur, and may reflect a segregation (compartmentalization) effect. As two radicals species located in two separate particles are unable to react, the molecular weight can increase due to a decreased termination rate. In addition, the formation of low-molecular-weight products may be limited

by slower oxidation reactions compared to propagation reactions, or by a rather small concentration of ROS. The latter interpretation seems to be the most likely reason considering the much narrower polydispersity index (ca. 1.5) for the TiO₂-initiated samples. In contrast, the reference sample has a PDI value of 2.8, more consistent with a conventional chain reaction (all the SEC traces are provided in **Figure S3** in SI). Finally, the partially converted particles show size values very similar to those of the starting droplets, thus indicating a nucleation preferentially localized in the droplets. It must be emphasized that the number of monomer droplets in *c*-TiO₂ is about 290 times higher than the number of semiconductor nanoparticles. Consequently, the proportion of polymer/inorganic nanoparticles is assumed to remain much lower than pure polymer particles. With regard to this final point, a significantly lower concentration in TiO₂ has been used compared to other studies in aqueous solution.^[22,28] One can expect that if photocatalyst particles were coated with polymer layer, the photocatalytic reactivity would be strongly reduced.

3.3 A tentative photoinitiation mechanism

While these first experiments illustrate the photoinitiation ability of nanoscale TiO₂ in dispersed media, a precise mechanism needs to be established. In principle, superoxide radicals (O₂^{•-}) generated by a transfer of electrons from the conduction band to O₂ (reductive pathway), or hydroxyl radicals (OH[•]) resulting from electron transfer from H₂O to “holes” in the valence band (oxidative pathway ①) may both initiate radical polymerization (Figure 2), provided that O₂^{•-} and its protonated form would not dismutate to H₂O₂ (**Figure 3**). A final possibility would imply the oxidation of MMA (oxidative pathway ②) by “holes of the valence band and deprotonation to yield initiating C-centered radicals of the monomer substrate.

Figure 3. Photocatalytic activity of TiO₂

To shed light into the initiation pathway, O₂ or water were removed from the formulation respectively by nitrogen purging or by substituting water with acetonitrile. Removal of dissolved O₂ led to an increase of conversions from 54 to 60% (*c*-TiO₂) and from 43 to 55% (*f*-TiO₂), thus confirming the negligible role played by the superoxide anions (O₂^{•-}) in polymerization initiation. However, oxygen is not the only species able to be reduced by the electron in the conduction band. Among the other compounds present, it seems that only the carboxylate of MMA and the sulfate group of SDS are electron acceptors. The transfer of an electron to MMA would produce an anion radical that is protonated, the resulting allylic radical could be a weak initiator. The transfer of an electron to SDS would produce the dodecyl alcoholate and the radical of sulfurous acid anion that would instantly dimerize to dithionic acid, itself a weak oxidizing and reducing agent. Unless we would find sulfate by ion chromatography, we would have to assume that MMA is the electron acceptor. If this is the case, a slight acceleration of the polymerization could be only observed, meaning MMA reduction is not the main initiation reaction. The most likely explanation for the slight increase in conversion in the O₂-free experiment is related to the suppression of the inhibition by oxygen. This means that the formation of less reactive ROO[•] radicals unable to add on new monomer molecules is prevented.

In addition, a solution of MMA in anhydric acetonitrile failed to photopolymerize (0 % conversion) in presence of *f*-TiO₂, revealing the minor contribution of oxidation of acrylate molecules by “holes” of the valence band to photopolymerization (pathway ②). This set of experiments suggests that the presence of water is a prerequisite for the formation of initiating species, which are in all likelihood highly reactive hydroxyl radicals (pathway ①). A tentative mechanism would therefore involve OH[•] primarily generated at the surface of the semiconductor. Despite their limited lifetime and diffusion ability, a rather large fraction of OH[•] may react with MMA at the aqueous/organic interface of the dispersed medium. Subsequently, the C-centered MMA-radicals add to other MMA molecules present at high local concentration within the monomer droplets.

3.4 Primary process parameters affecting the TiO₂ photocatalyzed polymerization of MMA

Irradiance. As summarized in **Table 4**, a decrease of irradiance from 430 to 115 mW/cm² reduces the final conversion from 82% to 58% in 10 minutes, and slightly affects the molecular weight values (ca. 500 kDa). For reaction systems implying *c*-TiO₂, we observe a strong discrepancy between droplet size (50 nm) and particle size (85-155 nm), but the presence of TiO₂ particles may distort the final particle size value because of its propensity to form aggregates. The effect of irradiance is much more pronounced in the case of *f*-TiO₂. At 115 mW/cm², the conversion is only 20% versus 43% at 430 mW/cm². This result translates a much lower concentration of initiating radicals causing a significant increase of M_n. Given the fact that the efficiency of electronic excitation is much higher in nanoparticulate suspension than at immobile surfaces, the strong dependence of irradiance on the conversion of *f*-TiO₂ was expected.

Table 4. Effect of irradiance on MMA miniemulsion photopolymerization initiated by nanoTiO₂ ($D_d = 50$ nm)

Nanoscale TiO ₂ wt% / wt _{monomer}	Irradiance mW cm ⁻²	Conv %	M _n kDa	PDI	D _p nm
<i>c</i> -TiO ₂ 3.88%	430	82	499	1.7	95
	335	76	548	1.6	85
	295	67	404	2.1	135
	180	66	586	1.5	95
	135	62	594	1.5	115
	115	58	535	1.7	95
<i>f</i> -TiO ₂ 0.17%	430	43	181	1.4	56
	335	75	135	4.8	67
	180	20	504	1.5	59
	115	20	548	1.5	52

TiO₂ concentration. Regardless of the TiO₂ morphology, variation of the catalyst concentration has a marked effect on conversion and molecular weight (**Table 5**). A higher TiO₂ content enables the generation of a higher concentration of initiating species, yielding higher conversions and lower molecular weight values. However, it seems that *c*-TiO₂ conversion reached a threshold value. Above this value, the average molecular weight decreases. However, at the highest *f*-TiO₂ concentration (0.2 wt%), much shorter chains were obtained as well as a broader molecular weight distribution (PDI = 5.5), and these results may translate an increasing importance of degradation reactions. Particle sizes remain at a level close to those of droplets.

Table 5. Effect of the TiO₂ content on the MMA miniemulsion photopolymerization

	Nanoscale TiO ₂ wt% / wt _{monomer}	Conv %	M _n kDa	PDI	D _p nm
<i>f</i> -TiO ₂	0.07	15	641	1.4	60
	0.08	35	877	1.3	61
	0.16	43	181	1.4	56
	0.20	49	108	5.5	60
<i>c</i> -TiO ₂	0.17	54	381	1.6	65
	0.39	66	600	1.5	66
	0.79	77	615	1.5	68
	1.7	74	534	1.6	73
	3.88	82	499	1.7	95

3.5 New potentialities of nanoscale *f*-TiO₂ initiated photopolymerization

***f*-TiO₂ as recyclable PI.** In contrast to colloidal titania, immobilized *f*-TiO₂ system can be utilized to overcome the problem of post-process separation. A key feature of supported TiO₂ is to offer facile recovery and separation. This section investigates the possibility of reuse of the photocatalyst for the photopolymerization in dispersed media, opening the way for a simple, low-cost, recyclable PI. Preliminary observations of the fibers before and after photopolymerization (**Figure 4**) reveal that a minor part of the synthesized PMMA can adhere to the fibers, which agrees with a surface-initiated reaction. This highlights the necessity for cleaning the microfibers before triggering a new photopolymerization. In the following recycling experiments, the fibers were systematically washed with THF, water, and dried overnight at 45°C.

Figure 4. SEM pictures of the *f*-TiO₂ before (left) and after (right) MMA miniemulsion photopolymerization

Figure 5 depicts the evolution of monomer conversion following 4 consecutive polymerizations. While the final conversion is relatively preserved after 2 cycles, a significant decrease is observed from 40% to 26% in the last two cases. The reduction in conversion reflects a decreased photocatalytic activity due to a diminished surface area (PMMA coating), but also to a loss of fiber materials after the washing process. A weight loss of approximately of 20 wt% was estimated between the first and the fourth experiments. Additionally, one must note that particle size remains constant (58 - 63 nm) regardless of the number of cycles. In

contrast, the molecular weight increases from 181 kDa in the first experiment to 437 kDa for the last one, with a PDI value of about 1.5.

Figure 5. MMA conversion obtained during the photopolymerization of MMA miniemulsion when recycling f -TiO₂

Photopolymerization in annular batch photoreactor.

To show scaling-up capacity of TiO₂-assisted polymerization, MMA miniemulsion photopolymerization was also performed in a 120 mL photochemical reactor (see experimental section for details). The f -TiO₂ felt was placed on the internal wall of the photoreactor to optimize light absorption. **Figure 6** describes the evolution of the conversion with irradiation time. The conversion increases gradually and reached 84% after 100 min of irradiation. The molecular weight reached 425 kDa with a PDI of 1.6. It is noteworthy that no destabilization of the particles was noticed.

Figure 6. Conversion versus time plot of a *f*-TiO₂ initiated MMA miniemulsion photopolymerization using a batch photoreactor:

Conclusions

Due to their inherently large ratio of surface area to volume and the high efficiency of absorption of UV light, colloidal titania and titania deposited on fused silica fiber have been intensively used in the domain of photocatalysis. The idea behind this study was to prove that these beneficial properties can also be used in photopolymerizations of aqueous emulsions containing acrylate monomers. High molecular weight PMMA chains with a relatively narrow polydispersity index of about 1.5 were achieved using two different TiO₂ nanoscale forms (colloidal and supported titania). This new initiation mode combines the advantages of miniemulsion photopolymerization, such as the improvement of light absorption and fast kinetics, with the benefits of TiO₂ semiconductors: simple and low-cost process, no byproducts, and non-extractable PI residues in the polymer products. An emphasis on the *f*-TiO₂ system allowed to go beyond the proof of concept. First, the scale-up of the process was demonstrated and then, experiments were carried out to prove the recyclability. In the future, we foresee that such TiO₂ photoinitiating systems could be used for a broader range of water-insoluble monomer dispersed into aqueous emulsions. In addition, new syntheses could also

be envisaged such as polythiophene, poly(3,4-ethylenedioxythiophene), or polypyrrole via oxidative photopolymerization, proceeding through different mechanisms.

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KEYWORDS: miniemulsion, photopolymerization, photoinitiator, photocatalysis, titanium dioxide

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Intensively investigated as photocatalysts for the photodegradation of persistent organic contaminants, TiO₂ nanomaterials can be tamed to enable miniemulsion photopolymerization. In particular, colloidal and supported titania offer many advantages compared to conventional molecular photoinitiator: facile addition to the monomer dispersion, high surface-to-volume ratio, non migrating, no byproducts, simple and low cost process, and possible recovery/reuse.

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Supporting Information

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Colloidal and Supported TiO_2 : Towards Non-Extractable and Recyclable Photocatalysts for Radical Polymerizations in Aqueous Dispersed Media

*Emeline Lobry, Abdoul Salam BT Bah, Loïc Vidal, Esther Oliveros, André M. Braun, Adrien Criqui, Abraham Chemtob**

Figure S1. SEM picture and quantitative analysis of *f*- TiO_2

Figure S2. XRD pattern of *f*-TiO₂ (trace A) and *c*-TiO₂ nanoparticles (trace B)

Figure S3. SEC traces of PMMA obtained by miniemulsion photopolymerization catalyzed by *c*-TiO₂, *f*-TiO₂ and a conventional type I PI (hydroxyacetophenone). Irradiation conditions: Hg-Xe lamp, 430 mW/cm², 310-600 nm, 10 min UV exposure.