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Biomaterials Science

A versatile method for the selective core-crosslinking of hyaluronic acid-based nanogels via ketone-hydrazide chemistry: from chemical characterization to *in vivo* biodistribution

Francielle Pelegrin Garcia^{a,b}, Marlene Rippe^a, Mychelle Vianna^b, Talitha Fernandes Stefanello^{a,b}, Benoit Louage^c, Simon Van Herck^c, Lucie Sancey^d, Jean-Luc Coll^d, Bruno G. De Geest^c, Celso Vataru Nakamura^b, Rachel Auzély-Velty^{*a}

The development of biopolymer-based nanogels has gained particular interest to achieve successful delivery of therapeutics for the treatment of various diseases, such as cancer, infection and diabetes. Herein, we report a new and simple methodology for the covalent stabilization of self-assembled gel nanoparticles based on hyaluronic acid (HA) modified with a thermoresponsive ketone-functional copolymer. This relies on the selective formation of hydrazone crosslinks with bishydrazides within the globular domains of the copolymer chains formed above the cloud point temperature. This approach allows tuning the crosslinking density by varying the dihydrazide crosslinker to ketone molar ratio. The size distributions and morphology of the nanogels were assessed using dynamic light scattering (DLS), cryo-transmission and scanning electron microscopy. *In vitro* cellular uptake in several cancer cells and *in vivo* biodistribution of the nanogels in different mouse tumor models were then explored to assess the effectiveness of this crosslinking strategy. The data from these experiments show prolonged blood circulation, longer than 24 hours, for the crosslinked nanogels and high tumor accumulation.

1 Introduction

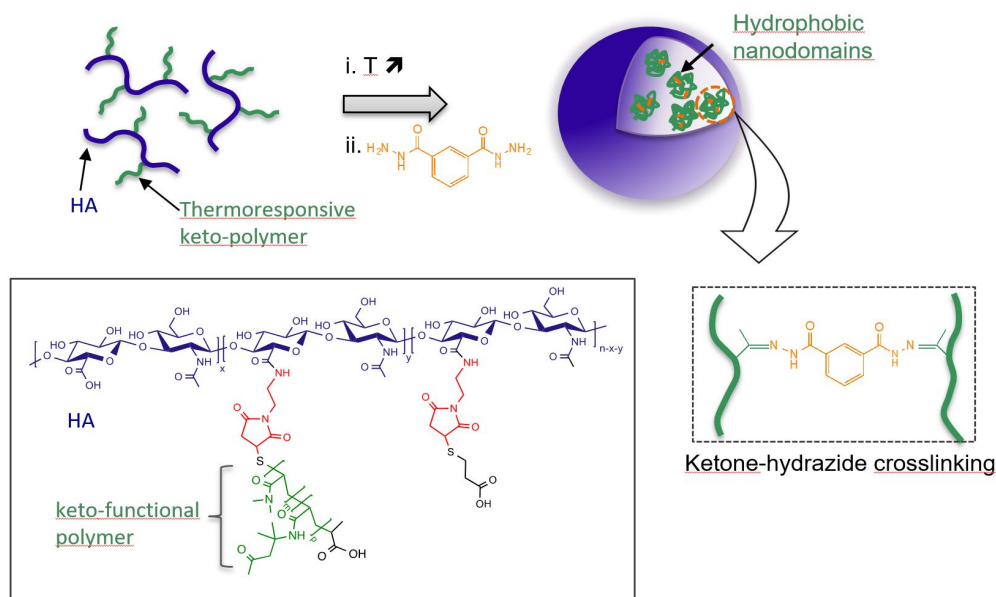
2 Self-assembled gel nanoparticles (also called nanogels) made of hydrophobically modified biopolymers have attracted growing
3 interest for drug delivery to tumors as these systems combine the advantages of hydrogels with nanoscale formulations.¹⁻⁵ Their size
4 can be easily tuned from tens to several hundreds of nanometer, their inner network comprising hydrophobic domains can be used
5 to incorporate poorly water-soluble drugs and their hydrophilic shell can be exploited to control their biological fate and targeting
6 ability. However, one practical challenge with such carrier systems is their tendency to disassemble in the bloodstream because of
7 the large dilution volume and/or interactions with serum proteins.⁶ These negative events reduce the efficiency of tumor-targeted
8 drug delivery *via* the enhanced permeability and retention (EPR) effect^{7,8}. To stabilize the nanocarriers, strategies of shell or core-
9 crosslinking have been employed.⁹⁻¹³ Among these methods, crosslinking the hydrophobic domains in the nanogel core rather than
10 the hydrophilic shell has the advantage of preserving the intrinsic properties of the biopolymer forming the hydrophilic shell.

11 In this study, we aimed to establish a simple and reliable method for the synthesis of core-crosslinked nanogels based on
12 hyaluronic acid (HA) for improved nanogel stability in view of *in vivo* tumor-targeting. HA, a polysaccharide which is a main
13 component of the extracellular matrix, was selected as hydrophilic component due its biocompatibility, biodegradability and the
14 existence of various functional groups that are accessible for chemical modification.¹⁴ Despite intensive research on self-assembled
15 nanoparticles based on hydrophobically modified HA for targeted anti-cancer drug delivery,¹⁴⁻¹⁶ core-crosslinking has only recently
16 been explored to address the issue of nanoparticle stability under diluted serum-rich conditions. In reported examples, the
17 hydrophobic core of HA-based nanoparticles was crosslinked by disulfide- or photo-mediated crosslinking. Here, we present a new
18 approach based on acylhydrazone chemistry allowing to tailor the density of crosslinking.¹⁷⁻²¹ Key to our strategy is the chemical
19 modification of HA by a temperature-responsive keto-functional polymer which can be efficiently crosslinked by bifunctional
20 hydrazides in aqueous medium (Scheme 1). Keto-functionalized polymers have been demonstrated to be useful reactive scaffolds for
21 the design of complex macromolecular architectures and for the conjugation of fragile biomolecules since they can be modified easily
22 and in high yield.²²⁻²⁷ Herein, the keto-polymer plays a dual role: i) in inducing hydrophobically-driven self-assembly of modified HA
23 into nanogels due to its dehydration upon heating, and ii) in stabilizing the nanogel structure through crosslinking by hydrazone
24 bond formation between ketones and hydrazides. We further investigated cellular uptake of the nanogels in several cancer cells and
25 their *in vivo* biodistribution to assess their tumor targeting potential.

26

27 Materials and methods

28 Materials



Scheme 1. Formation of hyaluronic acid-based nanogels by temperature-induced self-assembly and their covalent crosslinking by hydrazone bond formation within the hydrophobic domains of the grafted copolymer chains.

1 Hyaluronic acid ($M_w = 40$ kg/mol) was purchased from Lifecore (USA). Diacetone acrylamide (DAAM), *N,N*-dimethylacrylamide
 2 (DMA), 2,2-azobis(2-methylpropionitrile) (AIBN), phosphate buffer saline (PBS, pH 7.4), Tris-(2-carboxyethyl) phosphine
 3 hydrochloride (TCEP), mercaptopropionic acid (MPA), 3-(4,5-dimethylthiazol-2-yl)-2,5-diphenyltetrazolium bromide (MTT) and 4-
 4 (4,6-dimethoxy-1,3,5-triazin-2-yl)-4-methylmorpholinium chloride (DMTMM), isophthalic acid dihydrazide (IDH), hyaluronidase
 5 (type IV from bovine testes) were purchased from Sigma-Aldrich-Fluka (France). Aminoethylmaleimide hydrochloride was
 6 purchased from Aokchem. *N*-hydroxysulfosuccinimide sodium salt (sulfo-NHS) was obtained from Chemrio. *N*-(3-
 7 dimethylaminopropyl)-*N'*-ethylcarbodiimide hydrochloride (EDC) was purchased from Acros Organics. Dulbecco's Modified Eagle
 8 Medium (DMEM), L-glutamine, sodium pyruvate, fetal bovine serum (FBS), penicillin and streptomycin were provided by Gibco.
 9 Cyanine 5.5-amine (Cy5.5-amine) was purchased from Lumiprobe. All chemicals were used without any further purification. The
 10 positively charged resin, diethylaminoethyl cellulose (DEAE) was purchased from GE Healthcare Life Science. 2-
 11 (Butylthiocarbonothioylthio)propanoic acid (PABTC) was synthesized as previously reported.²⁸ The water used in all experiments
 12 was purified by an Elga Purelab purification system, with a resistivity of 18.2 M Ω cm.

13

14 Analytical techniques

15 ¹H NMR spectra were recorded at 25 °C using a Bruker AVANCE III HD spectrometer operating at 400 MHz. Deuterium oxide (D₂O)
 16 and dimethylsulfoxide-d₆ (DMSO-d₆) were obtained from SDS (Vitry, France). All spectra were recorded by applying a 45° tip angle
 17 for the excitation pulse, and a 10 s recycle delay. Chemical shifts (δ in ppm) are given relative to external tetramethylsilane (TMS = 0
 18 ppm) and calibration was performed using the signal of the residual protons of the solvent as a secondary reference. The number-
 19 average molar mass (M_n), the weight-average molar mass (M_w) and the dispersity (\mathcal{D}) of poly(DAAM-co-DMA) were determined by
 20 size exclusion chromatography (SEC) in dimethylformamide containing 50 mM NaNO₃. Measurements were done on a GPC system
 21 equipped with a Waters model 515 pump, a refractive Index Detector RI 2000 from Schambeck SFD GmbH and a light scattering
 22 detector (MALLS) from Wyatt (USA); Samples were recorded at 30 °C and at a flow rate of 1 mL/min. The cloud point temperature
 23 (T_{cp}) of poly(DAAM-co-DMA) was determined by UV-vis turbidity measurements ($\lambda = 500$ nm) performed on a Varian Cary 50 Scan.
 24 The samples were prepared at room temperature in PBS (1 and 3 g/L), after which the sample was placed in the instrument. The
 25 light transmittance was measured during at least two controlled cooling/heating cycles from 15 °C to 35 °C using a 2 °C interval. T_{cp}
 26 was considered to be the temperature at which the light transmittance was 50 % of that obtained for the same sample at 15 °C. The
 27 critical aggregation temperature (CAT) of HA-m-poly(DAAM-co-DMA) in aqueous solution was assessed by static light scattering
 28 (SLS) using an ALV 5000 device (ALV-Langen, Germany) equipped with a red helium-neon laser at a wavelength of 632.8 nm. A
 29 solution of HA-m-poly(DAAM-co-DMA) in PBS at a concentration of 0.5 g/L was filtered through a 1.2 μ m polycarbonate filter and
 30 heated from 10 to 50 °C using a 5 °C interval. The light scattering intensity (LSI) was measured at 90° angle for 60 s. The CAT, defined
 31 as the temperature at the inflection point of the plotted curve, was found to be 32 °C. The size and size distribution of nanogels were

1 measured by dynamic light scattering (DLS) using a Zetasizer NanoZS Malvern Instruments apparatus operating with a HeNe laser at
2 173°. The hydrodynamic diameters were calculated from diffusion coefficients using the Stokes-Einstein equation. All correlogram
3 analyses were performed with software supplied by the manufacturer. All the measurements were performed in PBS (pH 7.4, [NaCl]
4 = 0.15 M).

5 Synthesis of copolymer poly(DAAM-co-DMA)

6 DMA (0.515 mL, 5 mmol) and DAAM (1.69 g, 10 mmol), the RAFT agent PABTC (35.7 mg, 0.2 mmol) and AIBN (5 mg, 0.03 mmol) in 7
7 mL anhydrous dioxane were placed into a Schlenk tube under nitrogen. After 5 freeze-pump-thaw cycles for the deoxygenation of
8 the solution, the resulting solution was immersed in a thermostated oil bath at 75 °C and left to react for 3 h. The reaction was
9 quenched by cooling and exposure to oxygen. The resulting polymer was precipitated in cold hexane. The precipitate was dissolved
10 in acetone and the copolymer was precipitated again in hexane. This step was repeated 3 times to give 1.85 g of pure poly(DAAM₅₇-
11 co-DMA₃₃). Samples taken before and during the polymerization were analyzed by ¹H NMR to determine monomers conversion. The
12 M_n , M_w and \bar{D} values of the copolymer were determined by SEC in DMF.

13 Synthesis of HA-m-poly(DAAM-co-DMA)

14 Prior to the coupling reaction with HA-maleimide, the poly(DAAM-co-DMA) copolymer was subjected to aminolysis using *n*-
15 butylamine, to convert the RAFT end-group to a thiol. Briefly, the copolymer (0.370 g, 0.225 mmol) was solubilized in
16 tetrahydrofuran (10 mL) and *n*-butylamine (4 mL, 0.04 mol) was added. After 5 min under stirring, the reaction mixture was
17 concentrated under reduced pressure, resulting in a waxy liquid that was solubilized in THF. The copolymer was recovered by
18 precipitation in cyclohexane and dried under vacuum at 45 °C to obtain a white powder. Next, the thiol-capped copolymer (0.360 g,
19 0.0218 mmol) was solubilized in 10 mL of water/ethanol (3:2, v/v) and TCEP (0.0063 g, 0.0218 mmol) was added. After 30 min of
20 stirring at 4 °C under nitrogen atmosphere, the copolymer solution was added to a solution of HA-maleimide (0.124 mmol) in 40 mL
21 of PBS/ethanol (3:2, v/v), to perform the coupling reaction via Michael addition, as previously described²⁹.

22 To prepare the HA-maleimide derivative, HA (0.065 g, 0.162 mmol) was dissolved in 31.5 mL of a water/ethanol mixture
23 (2:1,v/v) and sodium chloride (0.257 g, 7.24 mmol) was added. Next, aminoethylmaleimide hydrochloride (0.068 g, 0.385 mmol),
24 sulfo-NHS (0.070 g, 0.324 mmol) and EDC (0.124 g, 0.65 mmol) were added and the pH of the solution was maintained between 4.5
25 and 5 until no further change of pH was observed (\approx 4 h). The resulting product was purified by diafiltration using a membrane
26 Amicon Ultracel MWCO 10 kDa and recovered in pure water (24 mL). Then, PBS 0.1 M (10-fold concentrated) (2.4 mL) was added,
27 followed by immediately adjusting the pH to 4.5. Finally, the copolymer solution was added to the HA-maleimide solution (0.050 g in
28 18.5 mL, 0.124 mmol) and the reaction mixture was stirred at 4 °C for 4 h. The remaining maleimide units were then converted to
29 carboxylic acid derivatives by addition of excess MPA (42.5 μ L, 0.49 mmol). After stirring for 6 h, the HA-poly(DMA-co-DAAM)
30 derivative was purified via a batch ion exchange process using DEAE Sepharose CL-6B as a weak-anion exchanger. Briefly, DEAE
31 resin (20 mL), stored in a solution of ethanol 30 %, was washed three times with ultrapure water at 4 °C (3 \times 20 mL, contact times of
32 10 min). Excess liquid was removed by centrifugation (10,000 rpm, 10 min) at 4° C. Then, the resin was activated by successive
33 washes with a 0.5 M NaCl aqueous solution (20 mL), a 1 M NaCl aqueous solution (20 mL) and finally, four times with ultrapure
34 water (4 \times 20 mL). The HA-poly(DMA-co-DAAM) derivative was then added to the resin in a conical tube and allowed to interact with
35 the resin overnight at 4°C under stirring with an orbital shaker. Then, the HA-derivative bound to the resin was subjected to four
36 washes with ultrapure water (4 \times 20 mL) to remove non-grafted copolymer. Finally, the HA-derivative was eluted with a 1 M NaCl
37 aqueous solution (4 \times 10 mL). After filtration of the solutions of the recovered HA derivative through a Buchner funnel containing a
38 porous glass filter plate (porosity 4), the solution was dialyzed against deionized water using a dialysis membrane MWCO 100 kDa
39 for 48 h. The product was recovered by freeze-drying as a white powder.

40 Determination of the poly(DAAM-co-DMA) content in HA-m-poly(DAAM-co-DMA) samples by the carbazole reaction

41 The degree of substitution (DS), defined as the average number of copolymer units per repeating disaccharide unit, was indirectly
42 determined by reaction of D-glucuronic acid units of HA with carbazole³⁰. Briefly, 800 μ L of 25 mM sodium tetraborate solution in
43 sulfuric acid was added to an aqueous solution of HA-m-poly(DAAM-co-DMA) (200 μ L) at a concentration of 0.34 g/L. After heating
44 at 100 °C for 10 min, the solution was cooled at room temperature for 15 min, and then a solution of carbazole (200 μ L) in absolute
45 ethanol 0.125 % (m/v) was added. The sample was heated again at 100 °C for 10 min and its absorbance was determined by
46 spectrophotometry at 530 nm. The HA concentration was calculated from a calibration curve (0.050 to 0.250 g/L), which allowed the
47 indirect determination of the copolymer amount in the HA-m-poly(DAAM-co-DMA) sample.

48 Nanogel crosslinking

49 To a solution of HA-m-poly(DMA-co-DAAM) (0.005 g, 0.0026 mmol) with a DS of 0.1 at a concentration of 1 g/L in PBS (pH 7.4) at
50 room temperature, a solution of isophthalic acid dihydrazide (IDH) in DMSO (0.073 g/mL) was added under stirring. The volume of
51 the IDH solution (20, 40 and 60 μ L) was varied to obtain IDH/ketone molar ratios of 0.5, 1.0 and 1.5, respectively. Next, the

1 temperature of the solution was increased to 40 °C. After stirring at 40 °C for 24 h, the nanogels suspension was transferred into a
2 dialysis bag (MWCO = 6-8 kDa) and dialyzed against deionized water for 48 h. The core cross-linked nanogels were recovered by
3 freeze-drying.

4 Labeling of crosslinked nanogels based on HA-m-poly(DAAM-co-DMA) and native HA with Cy5.5

5 Fluorescent nanogels were obtained by grafting the dye Cy5.5-amine on crosslinked nanogels based on HA-m-poly(DAAM-co-DMA)
6 by an amine-acid coupling reaction using DMTMM as a coupling agent³¹. To this end, crosslinked nanogels (0.010 g, 0.0052 mmol)
7 were solubilized in water/DMF (1:1 v/v; 4 mL) and DMTMM (1.467 mg, 0.0052 mmol) was added to the solution, followed by
8 adjusting the pH to 6.5. After 30 min of stirring, Cy5.5-amine (0.099 mg, 0.00013 mmol) solubilized in water/DMF (1:1 v/v) at a
9 concentration of 1 g/L was added to the reaction mixture. After stirring at room temperature for 48 h, the nanogels were purified by
10 dialysis using a membrane MWCO 6-8 kDa against deionized water for 48 h and recovered by freeze-drying. Cy5.5-labeled HA was
11 prepared from native HA and Cy5.5-amine according to the same procedure.

12 Scanning electron microscopy

13 Drops of crosslinked nanogels solutions (0.5 g/L) in pure water at both 5 and 40 °C were deposited onto mica-coated copper stubs
14 (also precooled/heated at 5 or 40 °C, respectively) and allowed to air-drying at 4 or 40 °C. The samples were then coated by
15 approximately 2 nm of sputtered Au-Pd and observed in secondary electron imaging mode with a ZEISS Ultra 55 FEG-SEM (Grenoble
16 INP - CMTC). Images were acquired at low voltage of 3 kV using an in-lens detector.

17 Cryo-transmission electron microscopy

18 Cryo-TEM samples were prepared with an automated vitrification system (Vitrobot Mark IV, FEI, The Netherlands). Crosslinked
19 nanogels solutions (5 g/L) was placed on Lacey carbon grids (TedPella Inc, California, USA) charged previously by glow discharge
20 with 15 mA for 10 seconds in air atmosphere to make them hydrophilic (Pelco easiGlow). Specimens were prepared in a controlled
21 environment with the temperature and humidity set to 22 °C for 5 °C sample and 37 °C and 100 % for humidity, respectively, which
22 prevented sample evaporation sample during the preparation. A 3 µL sample droplet was deposited on a 300 mesh lacey carbon-
23 coated cooper grid (Ted Pella). The excess fluid was blotted off to create an ultra-thin layer, which spans the holes of the support film
24 of the solutions, and prepared with blot force of approximately -5, blot wait of 20 seconds and blot time of 3 seconds of waiting time
25 before blotting. The prepared samples were immediately vitrified by propelling the grids into liquid ethane at its freezing point (-184
26 °C). The vitrified sample grids were transferred under liquid nitrogen by the use of a Gatan (Pleasanton, CA, USA) cryo-holder (Model
27 626) into a JEOL JEM-1400 plus TEM (Jeol, USA) equipped with FEG and operated at 120 kV acceleration voltage. Microscopy was
28 carried out at -175 °C sample temperature with a defocus range of -2 to -4 µm. Images were acquired using an F-416 CMOS camera
29 (TVIPS, Germany). Sample preparation and data acquisition were performed at the Electron Microscopy Laboratory (LME)/Brazilian
30 Nanotechnology National Laboratory (LNNano).

31 Cell culture

32 Cells were obtained from American Type Tissue Culture (ATCC, Maryland), or European Collection of Authenticated Cell Cultures
33 (ECACC). Human cervical cancer cells (HeLa cells) were cultured in DMEM supplemented with 10 % heat-inactivated FBS, 2 mM L-
34 glutamine, and 1 mM sodium pyruvate. Mouse mammary carcinoma cells (TS/A-pc)³² were cultured in RPMI 1640 glutaMAX™
35 medium supplemented with 10 % heat-inactivated FBS. The cells were cultured at 37 °C under a humidified atmosphere containing 5
36 % CO₂. The cells were detached by trypsin and counted under the microscope using a Malassez cell. The cells were then centrifuged
37 (5 min, 200 g) and the pellet was resuspended in appropriate medium or sterile NaCl 0.9% for extemporaneous subcutaneous
38 implant to mice.

39 Analysis of CD44 receptor expression of cells

40 To measure expression of the HA-binding receptor CD44 in TS/A-pc and HeLa cells, TS/A-pc and HeLa cells were seeded into a 6-
41 well plate at a density of 5×10^5 cells per well and incubated for 24 h at 37 °C in a humidified atmosphere of 5% CO₂. Cells were
42 trypsinated, counted and washed before incubation for 1 h in 300 µL of PBS containing 10 µL of fluorescein isothiocyanate (FITC)-
43 conjugated mouse anti-human CD44 antibody (#555478 from BD Pharmingen) at 4 °C. FITC mouse IgG2b κ antibody was used as
44 isotype control (# 555742 from BD Pharmingen). After three washing with PBS, the cells were observed using fluorescence-activated
45 cell sorting (FACS) Accuri 6 (Becton, Dickinson and Company, Pont de Claix, France). Cells were gated to exclude cell debris and
46 10,000 events were recorded in the gate for fluorescence intensity measurement.

47 In vitro enzyme degradation of nanogels

48 In the enzymatic biodegradation experiments, several suspensions of crosslinked HA-m-poly(DAAM-co-DMA) nanogels (1 mg/mL)
49 in PBS (pH = 7.4) containing hyaluronidase at different concentrations (20, 40, 60 and 100 U/mL) were incubated at 37 °C. At a

1 predetermined period, the light scattering intensity, polydispersity index and mean diameter of the samples were measured at 37 °C
2 by DLS.

3 Cellular uptake of nanogels and *In vitro* cytotoxicity

4 HeLa and TSA cells were plated in 4-wells Lab-Tek I Chamber slide (Nunc™, Thermo Scientific) (50,000 cells in 500 µL of culture
5 medium) and incubated overnight in DMEM medium at 37 °C and 5% CO₂. Uptake experiments were performed by incubating cells
6 during 4 h, 8 h, and 24 h with Cy5.5-labeled crosslinked nanogels solution (100 µg/mL) at 37 °C and 5 % CO₂. For confocal imaging,
7 cells were fixed in 4% paraformaldehyde for 10 min, and microscopy images were recorded on a Zeiss LMS510 microscope. Images
8 were processed with ImageJ software. Similar protocol was observed for FACS experiment, with a 4 and a 24 h incubation with PBS
9 containing Cy5.5-labeled crosslinked nanogels solution (50 µg/mL). After three washing with PBS, the cells were observed using LSR
10 II (Becton, Dickinson and Company, Pont de Claix, France). Cells were gated to exclude cell debris and 10,000 events were recorded
11 in the gate for fluorescence intensity measurement. For the cytotoxicity assay, HeLa and TSA were seeded at 5 x 10⁵ cells/mL in
12 DMEM containing FBS, in 96-well microplates at 37 °C, 5% CO₂ atmosphere. After 24 h, the cells were treated with the crosslinked
13 nanogels and initial HA, at different concentrations. After 72 h of incubation, the cultures were evaluated by MTS assay.

14 *In vivo* biodistribution

15 The *in vivo* behavior of crosslinked nanogels was tested in *nude* mice bearing HeLa or TS/A-pc subcutaneous grafts in their right
16 flank (*n* = 3/condition). Female NMRI *nude* mice (6–8 weeks, Janvier, Le Genest-Saint-Isle, France) were injected subcutaneously
17 with 1 × 10⁶ of cells per mouse. After tumor growth (10–12 days for TS/A-pc and 3 weeks for HeLa), anesthetized mice
18 (isoflurane/oxygen 3.5/4 % for induction and 1.5/2 % thereafter; CSP, Cournon, France) were injected in the tail vein with 100 µL of
19 a dispersion of Cy5.5-labeled crosslinked nanogel (3 g/L in PBS). For comparison, Cy5.5-labeled HA (3 g/L in PBS) was also injected.
20 Fluorescence images were acquired by a back-thinned CCD camera (ORCAII-BT-512G, Hamamatsu, Massy, France) after 30 min, 1 h,
21 2 h 30, 5 h, 24 h and 48 h of the injection. After the imaging, at 24 h and 48 h, the organs and the blood were collected and imaged.
22 Image display and analysis were performed using Wasabi software (Hamamatsu, Massy, France). Semi-quantitative data were
23 obtained by drawing regions of interest (ROI) around each organ. All procedures and experimental protocols were approved by the
24 ethical committee of Grenoble (France) for the use of animal research.

25 Results and discussion

26 Synthesis and characterization of core-crosslinked nanogels

27 To prepare core-crosslinked nanogels, our approach was to functionalize HA with a thermosensitive ketone-containing copolymer
28 prepared from diacetone acrylamide and *N,N*-dimethylacrylamide as illustrated in Figure 1. DAAM is not only interesting as a
29 crosslinkable group but also as a hydrophobic co-monomer of DMA, which allows to prepare copolymers exhibiting a lower critical
30 solution temperature (LCST) below the body temperature.²⁶ DAAM and DMA were copolymerized by RAFT polymerization using
31 AIBN as an initiator and 2-(butylthiocarbonothioylthio)propanoic acid as a chain transfer agent (CTA). The molar ratio of
32 monomers:CTA was fixed at 100:1 with a DAAM/DMA feed ratio of 2:1. The kinetic plots proved that, in these reaction conditions,

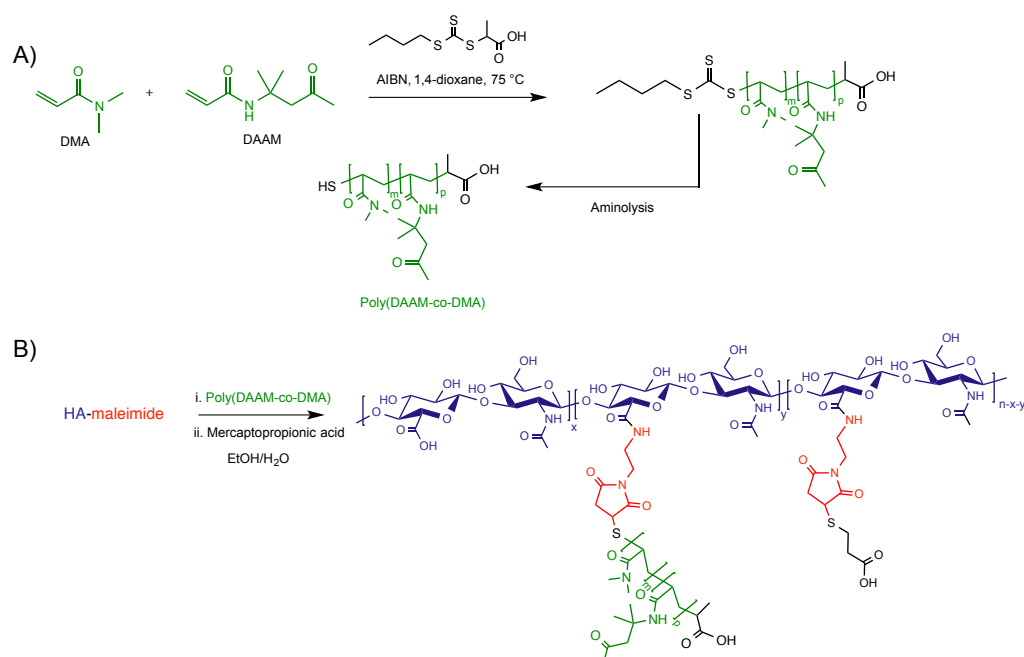
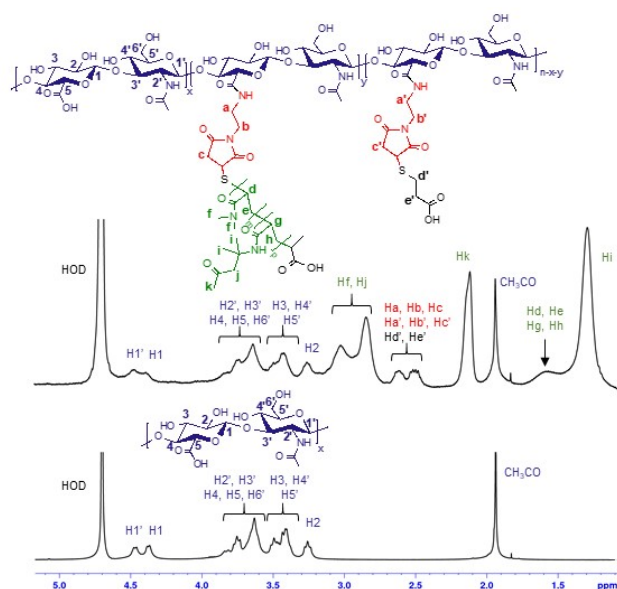


Figure 1. Synthesis of poly(DAAM-co-DMA) and its covalent grafting on hyaluronic acid by a thiol-maleimide coupling reaction.

1 PABTC allowed a good control over the RAFT copolymerization of DAAM and DMA, resulting in a copolymer with a final DAAM/DMA
 2 composition of 57:33 (Figure S1). The resulting copolymer displayed a low dispersity (\mathcal{D}) value (1.11) with a number average molar
 3 mass M_n of 16.45 kg/mol. Its structural integrity was ascertained using $^1\text{H-NMR}$ analysis (Figure S2). The copolymer exhibited a
 4 cloud point temperature of 25.5 °C (Figure S3). Subsequently, poly(DAAM-co-DMA) of which the RAFT trithiocarbonate end-group
 5 was converted to a free thiol by aminolysis using *n*-butylamine was grafted onto HA ($M_w = 40$ kg/mol, namely HA40) containing
 6 maleimide moieties (DS = 0.15, derived from $^1\text{H NMR}$ analysis) by formation of a thioether bond²⁹ (Figure 1).
 7 After conversion of the remaining maleimide units to carboxylic acid derivatives by addition of an excess of mercaptopropionic
 8 acid, the HA-m-poly(DAAM-co-DMA) conjugate was purified via a batch ion exchange process, and finally dialyzed. The structural
 9 integrity of the HA-m-poly(DAAM-co-DMA) was ascertained by $^1\text{H NMR}$ analysis. Indeed, comparison of the $^1\text{H NMR}$ spectra of initial
 10 HA and HA-m-poly(DAAM-co-DMA) shows, for the latter compound, the appearance of new signals in the regions of 3.2-2.1 ppm and
 11 1.7-1.1 ppm, corresponding to the protons of the maleimide-copolymer and maleimide-mercaptopropionic acid adducts, which
 12 confirms the successful grafting of the copolymer on HA (Figure 2). The degree of substitution of the conjugate, determined by
 13 carbazole assay,³⁰ was found to be 0.10.

14



15

16 **Figure 2.** $^1\text{H NMR}$ spectrum (400 MHz, 6 g/L in D_2O , 25 °C) of HA-m-poly(DAAM-co-DMA).

17 HA-m-poly(DAAM-co-DMA) is subjected to a sol-gel transition upon heating in phosphate buffer saline (PBS, pH 7.4) owing to the
 18 temperature-responsive behaviour of the grafted copolymer. Nanogels were thus obtained above 32 °C, corresponding to the critical
 19 aggregation temperature (CAT) of HA-m-poly(DAAM-co-DMA). This was determined by measuring the light scattering intensity of a
 20 solution of the HA derivative in PBS (0.5 g/L) as a function of temperature (Figure S4). The higher value of the CAT compared to the
 21 T_{cp} of the copolymer poly(DAAM-co-DMA) can be related to the presence of the highly hydrophilic and negatively charged HA
 22 backbone. Dynamic light scattering measurements at 40 °C showed that the nanogels had a hydrodynamic diameter around 145 nm
 23 with a low polydispersity index (PDI) of 0.11 (Table 1). The resulting nanogels were subsequently rendered temperature-
 24 irreversible by hydrazone-bond crosslinking the keto-functional copolymer chains with a bishydrazide, isophthalic dihydrazide (IDH),
 25 in aqueous medium as shown in Scheme 1.

26 **Table 1.** Average diameter and PDI of HA-p-poly(DAAM-co-DMA) nanogels.

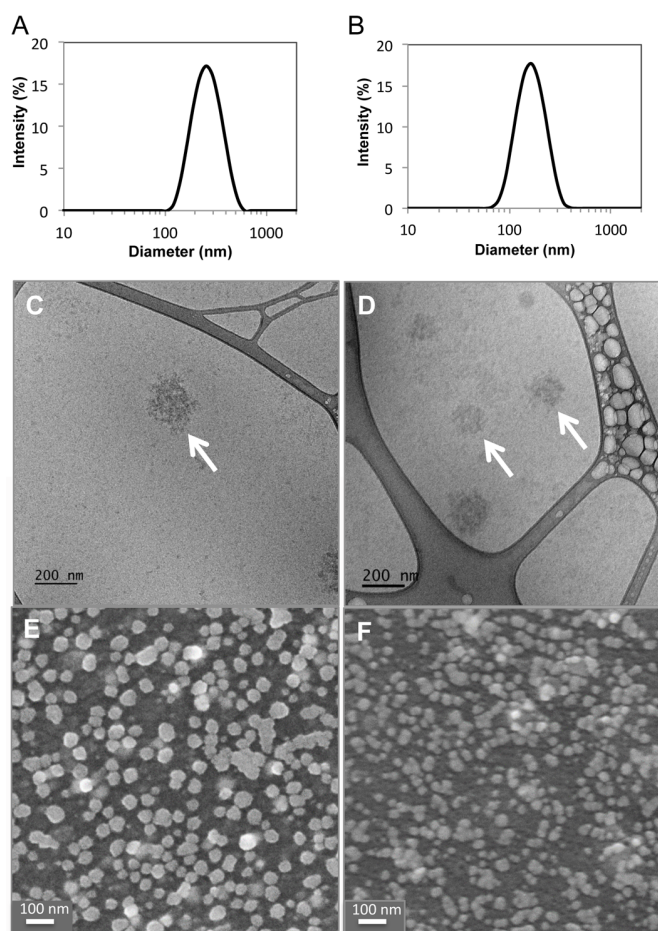
	18 °C		40 °C	
	Diameter ^a (nm)	PDI	Diameter (nm) ^a	PDI
Non-crosslinked nanogels	-- ^b	-- ^b	144.6 ± 1.4	0.11
Crosslinked nanogels (0.5 equiv. IDH)	247.1 ± 1.1	0.15	150.6 ± 2.4	0.10

^a) Z-Average diameter; ^b) no nanogel formation below the CAT

27

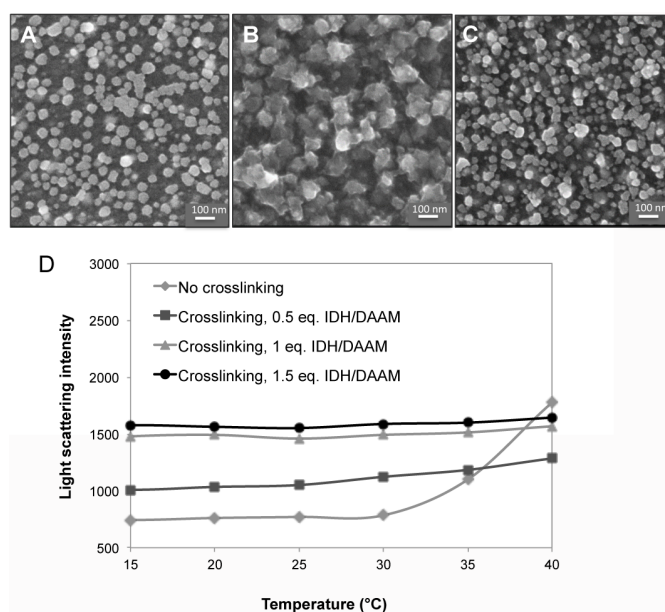
28 The cryo-TEM and SEM images of nanogel suspensions vitrified and dried at 5 °C revealed the presence of nanogels at low
 29 temperature (Figure 3C and 3E). In accordance to the DLS data (Table 1 and Figure 3B), the nanogels at 40 °C observed by cryo-TEM
 30 had a diameter below 200 nm (Figure 3D). The nanogels at 5 °C had a diameter around 200 nm (Figure 3C). This is in line with DLS
 31 analysis which indicated an increase of the average diameter of the nanogels from 150 to 247 nm by lowering the temperature from

1 40 to 18 °C (Table 1). The larger size of the nanogels at 18 and 5 °C can be attributed to increased hydration of the crosslinked
2 copolymer chains, resulting in the nanogel swelling. In the SEM images, the size of the nanogels appeared to be much smaller due to
3 shrinkage during drying.



4
5 **Figure 3.** Size distribution of HA-based nanogels crosslinked with an IDH: ketone molar ratio of 0.5 measured by DLS at 18 °C (A) and 40 °C (B); morphology
6 observed at 5 °C (C, E) and 40 °C (D, F) by cryo-TEM (C, D) and SEM (E, F).

7 In the next step, we varied the IDH to ketone molar ratio from 0.5 to 1.5 in the crosslinking step to ensure that all nanogels were
8 sufficiently crosslinked to be stable at low temperature. The SEM images obtained from crosslinked nanogel suspensions at 5 °C
9 suggested a slightly lower amount of nanogels crosslinked using an IDH to ketone molar ratio of 0.5 (Figure 4A, B and C). This was
10 confirmed by static light scattering experiments performed on the non-crosslinked and crosslinked nanogels (Figure 4D). The non-
11 crosslinked nanogels showed a continuous decrease of the light scattering intensity (LSI) from 40 to 15 °C, indicating gradual
12 disassembly of the nanogels due to hydration of the thermosensitive copolymer chains. The LSI of the nanogels crosslinked with 0.5
13 molar equiv. of IDH slightly decreased upon cooling, suggesting a small extent of nanogel dissociation. In contrast, the LSI of the
14 nanogels crosslinked with 1 and 1.5 molar equiv. of IDH was constant upon cooling, demonstrating efficient nanogel crosslinking
15 under these conditions.



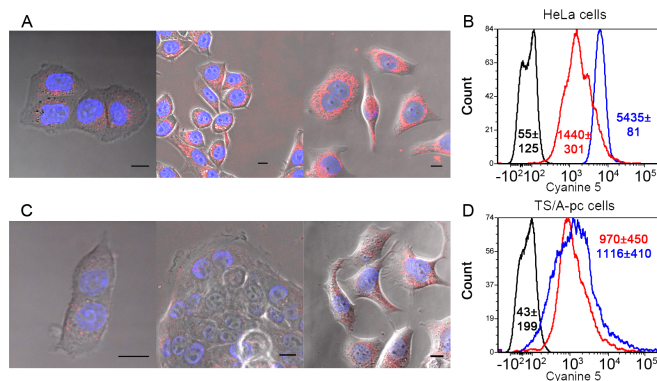
1
 2 **Figure 4.** Influence the IDH to ketone molar ratio on the stability of nanogels at low temperature. SEM observation at 5 °C of HA-based nanogels crosslinked
 3 with IDH to ketone molar ratios of 0.5 (A), 1.0 (B) and 1.5 (C), and light scattering intensity of non-crosslinked HA-based nanogels and nanogels crosslinked
 4 with IDH to ketone molar ratios of 0.5, 1.0 and 1.5 upon cooling from 40 to 15 °C (D).

5 These results also indicate that the crosslinking density can be tuned by varying the degree of dihydrazide to diacetone monomer
 6 molar ratio. Compared to strategies previously used for covalently crosslinking self-assembled nanoparticles based on HA,¹⁶ this
 7 approach based on keto-hydrazide chemistry has the advantage of controlling efficiently the extent of crosslinking. As the
 8 acylhydrazone can be hydrolyzed under acidic conditions, we evaluated the hydrolytic stability of the cross-linked nanogels at the
 9 endosomal pH of 5. We found that at this pH, the nanogels remained stable for prolonged periods of time at 37 °C. We attribute this
 10 to the hydrophobic environment within the collapsed copolymer domains. Moreover, in a previous study on macrogels formed by
 11 crosslinking poly(DAAM-co-DMA) with adipic acid dihydrazide, the networks were shown to remain stable for several days at pH 6.²²
 12 A pH of 2.5 was required to observe degradation in less than a day. Since HA can be readily degraded to lower molecular weight
 13 components by hyaluronidase (HAase) after being taken up by cancer cells,¹⁴ we also investigated the enzymatic degradation of
 14 crosslinked nanogels using different concentration of HAase (20, 40, 60 and 100 U/mL) by DLS at 37 °C. The light scattering intensity
 15 of the samples was found to rapidly drop, leveling off at 1 h post-addition of HAase. At this time, the LSI was reduced by about 75%
 16 and 48 % for HAase concentrations of 20 and 40 U/mL, respectively (Figure S5). Meanwhile, the PDI and the particle size increased
 17 to ~ 0.35 and ~ 200 nm, respectively. Further increase of HAase concentration had no significant effect on the LSI and PDI values and
 18 the mean size did not change (~ 180 nm). These data suggest partial degradation of nanogels due to the ability of HAase to hydrolyze
 19 the outer HA layer of the particles. The evaluation of the biological properties was performed using crosslinked nanogels, which
 20 were stable upon long-term storage, as revealed by the perfect solubilization of the product in aqueous solution after freeze-drying.
 21 By contrast, non-crosslinked HA-m-poly(DAAM-co-DMA) was not stable enough; this may be due to side reactions involving ketone
 22 functional groups though they are more stable than aldehydes. The crosslinking step constitutes therefore an innovative way to
 23 stabilize the nanogels.

24 **In vitro cellular uptake evaluation and in vivo targeting properties of nanogels based on HA-m-poly(DAAM-co-DMA)**

25 The intracellular uptake behaviour was observed by confocal microscopy in two cancer cell lines, TS/A-pc (mouse mammary
 26 adenocarcinoma) and HeLa (human cervical cancer cells), after 4 h, 8 h, and 24 h incubation with the crosslinked nanogels labeled
 27 with Cy5.5. As shown in Figure 5, the time dependent cellular uptake of nanogels was more pronounced in HeLa cells,
 28 overexpressing the CD44 receptor for HA than that in TS/A-pc cells which do not express CD44 (see experimental section and
 29 supporting information, Figure S6). The internalized nanogels were mainly observed into small vesicles, or diffused in the cytoplasm.
 30 This high uptake by HeLa cells was confirmed by FACS experiments, and increased between 4 h and 24 h, as indicated in Figure 5.
 31 Such accumulation may be related to the overexpressed CD44 receptor in HeLa cells³³⁻³⁵, and was safe for both cell lines (see Figure
 32 S7). Indeed, CD44 binds HA and enables its internalization and intracellular degradation.¹⁴ In this regard, several studies
 33 demonstrated that the degree of intracellular uptake of HA nanoparticles depends on the receptor density of the cells; higher CD44
 34 receptor density correlates with higher HA nanoparticle uptake.^{17, 36-38} The *in vivo* biodistribution and tumor accumulation of
 35 crosslinked nanogels (with 0.5 equiv. IDH) labeled with Cy5.5 were assessed by non-invasive full body fluorescence imaging
 36 following their intravenous administration in HeLa and TS/A-pc tumor-bearing mice, and compared to the biodistribution of Cy5.5-
 37 labeled HA. TS/A-pc tumor model was selected in addition to HeLa cell tumor because of its high capacity to accumulate
 38 nanoparticles *via* the EPR effect³⁹.

1

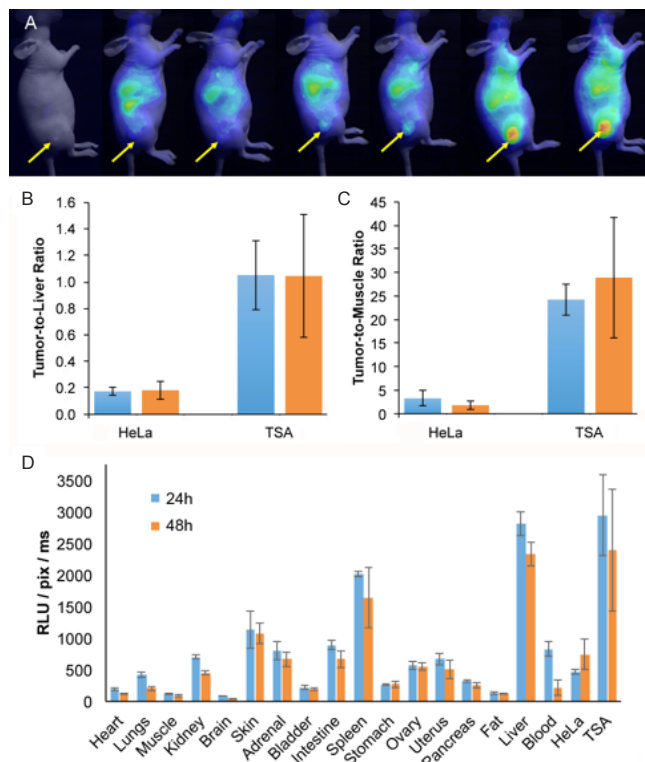


2

3 **Figure 5.** Confocal imaging of HeLa (A) and TS/A-pc (C) cells, 4 h, 8h, and 24 h after incubation with Cy5.5-HA-m-poly(DAAM-co-DMA) crosslinked nanogels.
 4 The Cy5.5-labeled crosslinked nanogels are represented by red color, while the cell nuclei are in blue. Scale bar: 10 μ m. (B; D) FACS analysis were performed
 5 on the same cell lines after 4 h (red line) or 24 h (blue line) and are compared to control condition (Black line).

6 The fluorescence signal of the crosslinked nanogel was recorded as a function of time in animals, and also in organs after sampling
 7 at 24 and 48 h. Fluorescence intensity in the tumor gradually increased up to 24-48 h for both tumor types but to a greater extent in
 8 TS/A-pc tumor-bearing mice (Figure 6A). Indeed, TS/A-pc tumors strongly accumulated the nanogels at 24 h with an *ex-vivo* tumor-
 9 to-liver ratio reaching ≈ 1 and a tumor-to-muscle ratio reaching ≈ 25 (*versus* ≈ 0.2 and < 3.5 , respectively for HeLa tumors, see Figure
 10 6B and C). One should note that these ratios were almost stable between 24 and 48 h. Tumors maintained strong fluorescence
 11 intensities during two days as evidenced by quantitative analysis (Figure 6D). This indicated that nanogels could circulate for a long
 12 period of time in the body and continuously accumulate at the tumor site. The lower level of accumulation of the nanogels in HeLa as
 13 compared to TS/A-pc tumors (Figure 6B and 6C) supports the idea that the capture of nanogels is mostly sensitive to an EPR-driven
 14 passive accumulation rather than a CD44-dependent targeted one. This also reflects that passive accumulation of nanoparticles is
 15 dependent on the tumor type.³⁹⁻⁴¹

16



17 **Figure 6.** (A) In vivo near-infrared fluorescence (NIRF) images of the time dependent biodistribution of Cy5.5-labeled crosslinked nanogels in breast TS/A-pc
 18 and HeLa tumor-bearing mice (n = 3/tumor type). The tumor was engrafted subcutaneously on the right flank of the mouse. The fluorescence was measured
 19 at the following time elapse after administration: 0 min, 30 min, 1 h, 2 h 30, 5 h, 24 h, and 48 h. The tumor locations are indicated by the arrows. (B)
 20 Fluorescence intensity of the tumor-to-liver ratios at 24 h (blue) and 48 h (orange) post-injection and, (C) tumor-to-muscle ratios from excised organs
 21 sampled at 24 h (blue) and 48 h (orange) post-injection. The results are expressed as the mean \pm SD (n=3). (D) Quantification of the *ex vivo* biodistribution of
 22 Cy5.5-labeled crosslinked nanogels in mice, 24 h and 48 h after administration (n = 6/organ). ROIs were defined on the extracted organs to semi-quantify the
 23 amount of photons detected per pixel. The results in each organ are expressed as the mean of relative light unit per pixel per ms \pm SD (n=6).

1 Interestingly, similar results for native HA accumulation were observed as described in Figure S8. Only the tumor-to-muscle ratios
2 were different from the nanogels, reaching 7.18-12.42 for HeLa tumors, and 15.59-8.73 for TS/A-pc tumors, at 24 and 48 h,
3 respectively (Figure S8B and S8C). This highlights the fact that accumulation of nanogels in the tumor is governed by both the nature
4 of their hydrophilic shell and their nanoparticle form. The nanogel fluorescence was also detected in other organs to various extents,
5 with liver and spleen characterized by the highest nanogel accumulations at both 24 and 48 h post injection (Figure 6D). This
6 accumulation may be due to cellular uptake of the nanogels by phagocytic cells of the reticuloendothelial system. It is known that the
7 liver is the major organ that uptakes and accumulates any types of nano/micro particles with high hydrodynamic radius.⁴²
8 Accumulation in liver and spleen may also be related to nanogel clearance from circulation mediated by the hyaluronan receptor for
9 endocytosis (HARE), which is found primarily in the sinusoidal endothelial cells of the liver and spleen.^{14, 43, 44} Regarding HA, its
10 fluorescence signal was detected with high intensity in liver, spleen as well as in intestine, skin and ovary (Figure S8D). Collectively,
11 these results suggest that crosslinked nanogels exhibit high tumor uptake, probably in correlation with the EPR properties of the
12 tumor, and improved biodistribution even after 48 h post-injection.

13 Conclusions

14 In this study, novel biocompatible nanogels were fabricated from hyaluronic acid modified with a ketone-functional copolymer. The
15 nanogels were prepared by a simple method based on the temperature-triggered assembly of the HA conjugate followed by
16 crosslinking the hydrophobic nanodomains of the copolymer chains by hydrazone formation. These crosslinks throughout the
17 hydrophobic domains in the core of the nanogels do not affect the shell-forming polysaccharide structure. In addition, their density
18 can be tuned by varying the dihydrazide crosslinker to ketone group molar ratio. *In vitro* biological studies revealed that the
19 nanogels were internalized into different cell types, with a main vesicle accumulation. Finally, *in vivo* studies demonstrated that
20 following intravenous administration in tumor-bearing mice, the crosslinked nanogels were able to circulate in the bloodstream for a
21 long period of time and accumulated in the tumor tissues, probably in correlation with the EPR properties of the tumor. These
22 results suggest that these nanogels hold great potential in anti-cancer therapy, as they efficiently reach the vascularized tumor. They
23 represent a versatile platform to introduce additional functionalities and/or tune the properties of the hydrophilic shell in order to
24 more optimally exploit the biocompatibility and the beneficial distribution of these novel nanocarriers that might be used for the
25 delivery of chemotherapeutics.

26 Conflicts of interest

27 "There are no conflicts to declare".

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