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Visualization of subsurface damage in woven carbon fiber-reinforced composites using polarization-sensitive terahertz imaging

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ABSTRACT

Polarization-sensitive terahertz imaging is applied to characterize subsurface damage in woven carbon fiber-reinforced composite laminates in this study. Terahertz subsurface spectral imaging based on terahertz deconvolution is tailored and applied to detect, in a nondestructive fashion, the subsurface damage within the first ply of the laminate caused by a four-point bending test. Subsurface damage types, including matrix cracking, fiber distortion/fracture, as well as intra-ply delamination, are successfully characterized. Our results show that, although the conductivity of carbon fibers rapidly attenuates terahertz propagation with depth, the imaging capability of terahertz radiation on woven carbon fiber-reinforced composites can nonetheless be significantly enhanced by taking advantage of the terahertz polarization and terahertz deconvolution. The method demonstrated in this study is capable of extracting and visualizing a number of fine details of the subsurface damage in woven carbon fiber-reinforced composites, and the results achieved are confirmed by comparative studies with X-ray tomography.

1. Introduction

Fiber-reinforced composites, which combine the merits of high strength and low weight, offer an alternative to conventional structural materials, such as metals, and are widely used for various applications including aerospace, automotive, renewable energy and marine industry. As a result, the wide applicability of fiber-reinforced composites has created the correspondingly need for advanced nondestructive evaluation (NDE) techniques for inspection and failure detection during manufacturing and maintenance. Currently, for many materials and applications, there is no adequate NDE technique available to detect important damage types that occur at depth but may not be visually evident at the surface. Given the potential catastrophic nature of failure in some of the aforementioned applications, there has been an intense search for NDE techniques that might provide the desired information.

Terahertz (THz) imaging, which can provide a non-invasive, non-contact, and non-ionizing modality to characterize numerous non-metallic materials, is emerging as a relatively new promising NDE technique for fiber-reinforced composites. The THz portion of the

electromagnetic spectrum, extends from approximately 100 GHz to 10 THz, and lies between the microwave and infrared; the corresponding wavelength range is from 30 μm to 3 mm. THz waves can penetrate various nonmetallic materials that may be opaque in the range of visible and infrared light. Moreover, as nonionizing radiation, THz waves present no known health risks.

THz imaging was first applied to the NDE of fiber-reinforced composites in 2006 [1] and has been explored extensively since [2]. For composites reinforced by glass fibers, THz waves can be utilized to characterize both surface and underlying damage and defects, including mechanical/heat damage [3], voids [4], delamination [5,6], intrusions and moisture contamination [7]. Material parameters [8] and fiber orientation [9] in glass fiber-reinforced composites have also been successfully characterized via THz imaging. Composites reinforced by other non-conductive fibers, such as ultra-high molecular weight polyethylene (UHMWPE) composites, have also been explored with THz imaging for characterization of projectile impact damage [10,11]. However, the studies of THz imaging of carbon fiber-reinforced composites are quite limited, mainly due to the conductivity of carbon

fibers, which hinders the penetration of THz radiation at depth. Thus far, to our knowledge, THz imaging has mainly been applied to detect the damage on the surface of unidirectional carbon fiber-reinforced composites [12,13]. Although carbon fibers hinder the THz penetration, the interaction between the THz polarization and carbon fiber orientation can be utilized to extract detailed information which cannot be obtained by other NDE techniques. In our previous work [14], polarization-resolved THz imaging has been successfully applied to a hybrid fiber-reinforced composite laminate containing glass-fibers/epoxy and carbon-fibers/epoxy plies with a cross-ply stack pattern. Impact-induced inter-ply damage at the interface and intra-ply damage close to the same interface can be differentiated by taking advantage of the sensitivity of THz polarization to the anisotropic conductivity of unidirectional carbon fibers.

In this study, for the first time to our knowledge, we demonstrate how to characterize *subsurface* damage in woven carbon fiber-reinforced composite laminates, which is of great potential importance to the aerospace and automotive industries, via polarization-sensitive THz imaging. THz subsurface spectral imaging based on THz deconvolution is developed and utilized to extract the detailed subsurface damage of several types within the first ply of the laminate caused by a four-point bending test. Subsurface damage types, including matrix cracking, fiber distortion/fracture, and intra-ply delamination, can be clearly distinguished. A comparative study with X-ray tomography is also performed to confirm the results obtained from THz imaging.

2. Theoretical background and method

2.1. THz polarization and carbon-fiber orientation

The electrical conductivity of carbon fibers severely limits the penetration of THz waves into composites. Generally, carbon fiber-reinforced composite laminates can be classified into unidirectional laminates and woven (multidirectional) laminates. Considering one ply of unidirectional carbon fiber-reinforced composite laminate, the electrical conductivity σ is anisotropic and depends on the THz polarization and fiber orientation, which can be expressed as [15].

$$\sigma(\theta) = \sigma_l \cos^2\theta + \sigma_t \sin^2\theta, \quad (1)$$

where θ is the angle between the THz polarization and fiber orientation, and σ_l and σ_t are the longitudinal and transverse conductivities, respectively. Along the fiber direction, the electric current flows through the carbon fibers, so the longitudinal conductivity σ_l depends on the conductivity of carbon fibers σ_f and on the fiber volume fraction ν_f ,

$$\sigma_l \approx \sigma_f \nu_f. \quad (2)$$

For the transverse conductivity σ_t , because the resin material is nonconductive, the flow of electric current only occurs due to random contact between adjacent carbon fibers, which depends on the manufacturing process and the quality of the composites [15]. Therefore, the longitudinal conductivity is much higher than the transverse conductivity. Then we can further conclude that (1) when the THz polarization is parallel to the orientation of carbon fibers, the electric conductivity is maximum and the THz reflectivity also reaches its maximum; (2) when the THz polarization is perpendicular to the carbon fibers, conductivity and reflectivity achieve their minimum. In principle, THz reflective imaging can be utilized as an effective method to characterize impact damage by monitoring the reflection coefficient across the surface of unidirectional carbon fiber-reinforced composites. Suffering from impact, carbon fiber distortion and fracture will occur in the damaged area, which will lead to spatial variations of the reflection coefficient as well as the polarization anisotropy. The reflection coefficient in regions with and without impact damage can be more easily distinguished with polarization parallel to the carbon-fiber orientation [13].

For woven carbon fiber-reinforced composite laminates, the analysis

of the electrical conductivity is more complicated. Besides the electrical properties of the constituent materials and the fiber volume fraction, the electrical conductivity of woven carbon fiber-reinforced composites also depends on the weave pattern, and the contacts between the adjacent tows with different fiber directions in the ply [16,17]. Therefore, the interaction between the THz polarization and woven carbon-fiber structure should be investigated before performing THz imaging. Proper selection of the THz polarization is able to produce enhanced contrasts for the THz imaging results, which we will show in the following.

2.2. THz subsurface spectral imaging

Our focus in this study is on THz subsurface spectral imaging, which is designed to extract *subsurface* damage in woven carbon fiber-reinforced composite laminates. Although THz waves are strongly attenuated while propagating through one ply of woven carbon fiber-reinforced composite, the waves do provide significant subsurface information, especially when the composite suffers damage, THz waves may penetrate deeper and produce echoes when damage is encountered. Therefore, although not superficially evident, the raw reflected THz signals are the superposition of the echo reflected from the surface and temporally overlapping subtle reflections from subsurface features. In order to highlight the subsurface information, it is necessary to separate subsurface echoes and the surface echo using deconvolution techniques.

Our previous work has demonstrated the power of THz deconvolution to resolve overlapping echoes associated with closely-spaced features in depth [18–20]. In short, THz deconvolution aims to reconstruct the impulse-response function, that is intrinsic to the sample and independent of the THz pulse employed, which should ideally consist of a sequence of ideal impulses corresponding to reflections from the interfaces of a stratified system [21]. With THz deconvolution, the depth-resolution of THz imaging can be greatly enhanced. By exploiting the sparse constraint, i.e., the expectation that only a limited number of temporal data points have nonzero values in the impulse-response function, THz sparse deconvolution developed in Ref. [22], is able to provide a clear representation of the impulse-response function with sharp pulses, and increase the depth-resolution to $\sim 45 \mu\text{m}$ in air given the spectral bandwidth of our apparatus.

Exploiting sparse deconvolution, THz subsurface spectral imaging can be performed by using the following procedure: (1) retrieve the deconvolved signals, corresponding to the impulse-response function, by applying THz sparse deconvolution to the received raw signals; in the deconvolved signals, the peaks indicating the location of surface and subsurface echoes are well-resolved; (2) identify the initial peak corresponding to the surface echo, and subtract this peak from the deconvolved signals, then the remaining deconvolved signals only contain the subsurface information; (3) re-convolve the remaining deconvolved signals with the THz reference signal to obtain the THz subsurface time-domain echoes; (4) perform the Fourier transform of the THz subsurface echoes, and integrate the frequency components in the higher frequency range of the available spectral bandwidth to form the imaging contrast; because higher frequency components, corresponding to shorter wavelength, are thus retained, the resulting images enhance both the depth and transverse resolution.

3. Sample and experiment

Two samples were investigated in this study. Both are nominally identical polyphenylene-sulfide-based composite laminates, reinforced with eight plies of woven carbon fibers. The dimensions of the laminates are 160 mm (warp direction, x) $\times$ 25 mm (weft direction, y) $\times$ 2.4 mm (thickness direction, z), shown in Fig. 1 (a). The weave pattern of carbon fibers is 2/1 twill with warp-dominated surface, and the structure of the samples is layered as the sequence

Fig. 1. The typical sample investigated in this study. (a) Optical photograph of the sample; (b) in-plane schematic of the 2/1 twill weave pattern; (c) cross-sectional schematic of the sample ply.

Fig. 2. Photograph of the four-point bending test. Note the carbon fiber-reinforced composite sample placed horizontally near the center of the image. The dimensions of the laminates are 160 mm (warp direction) $\times$ 25 mm (weft direction) $\times$ 2.4 mm (thickness direction).

$[0^\circ/90^\circ, -45^\circ/45^\circ]_{2s}$. The schematic structure of the first ply of this woven carbon fiber-reinforced composites is illustrated in Fig. 1(b) and (c). The volume fraction of carbon fiber in the composite laminate is about 50% and the Young's modulus of the samples is about 56 GPa along the fiber's axes in both warp and weft directions. The damage was introduced by four-point bending tests, shown in Fig. 2, with two different conditions. With the four-point bending test, we are able to introduce various damage types by tension and compression, which are useful for investigating different damage mechanisms. Sample 1 was subjected to the four-point test under one-cycle static loading with the applied force of 1200 N; Sample 2 was subjected to four-point bending test under 500 cycles static loading with the applied force of 800 N. The distance between the rollers on the top surface (compression surface) was about 50 mm, and the distance between the rollers on the bottom surface (tension surface) was about 105 mm.

The commercial THz time-domain spectroscopy (TDS) system (Teraview TPS Spectra 3000), which was employed in this study, is shown schematically in Fig. 3(a). The GaAs photoconductive antenna is excited by an ultrafast (femtosecond) laser to produce roughly single-cycle THz pulses with bandwidth extending from 60 GHz to 3 THz. The ultrafast laser used here is an Er-doped fiber laser that emits 780 nm pulses with sub-100 fs pulse duration at a repetition rate of 100 MHz and has an average output power in excess of 65 mW. Coherent

detection of the THz radiation is performed in a similar photoconductive antenna circuit. By gating the photoconductive gap with a femtosecond pulse synchronized to the THz emission, a current proportional to the THz electric field is measured. By varying the optical path length, the THz time-domain waveform can be sampled, resulting in both amplitude and phase information on a sub-picosecond time-scale.

THz reflective imaging was performed at an angle of incidence of $\sim 10^\circ$. Before scanning the samples, the THz reference signal was recorded by setting a metal plate at the sample position. The time-domain waveform and the frequency spectrum of this reference signal are shown in Fig. 3(b). The THz pulse in the reference signal is composed of roughly one electromagnetic cycle, and its duration is a few picoseconds. The spectrum of the reference signal, obtained by Fourier transforming the temporal pulse, extends from ~ 100 GHz to ~ 3 THz. The ringing after the main THz pulse are due to ambient water-vapor and the pronounced atmospheric water-absorption lines are clearly observed in the spectrum. The focus spot size of the THz beam is frequency-dependent, and is about $300 \mu\text{m}$ at 1 THz. The samples were raster-scanned by a set of motorized stages moving in x (warp) and y (weft) directions with a 0.2 mm spatial step size over the center area of the top surface (subject to compression due to the four-point bending test) of the sample. Each recorded temporal THz waveform contains 2048 data points, and the data sampling period was set to 0.02325 ps. The signal was averaged over 10 shots per pixel to enhance the signal-to-noise ratio. The raster-scan was conducted in a temperature-controlled laboratory at 22°C , and the humidity in the laboratory was held about 38%. In order to consider the interaction between the THz polarization and carbon-fiber orientation, the THz polarization was set first along the warp direction; after completing the scan, the polarization was rotated and set along the weft direction. Therefore, for each sample, two sets of 3D volumetric raw data were acquired.

4. Results and discussion

First, we present the THz spectral images based on the raw THz signals directly obtained from the raster-scanning. Note that the raw THz signals are dominated by the initial reflection from the sample surface, and thus the images so obtained will provide mainly near-surface information. Later, we consider images obtained by windowing-out the initial surface reflection to reveal more clearly subsurface information.

The contrast mechanism M of THz C-scans in the frequency domain

Fig. 3. THz time-domain spectroscopy (TDS) system. (a) Schematic of the THz TDS system in reflection configuration. (b) THz reference signal with its spectrum in the inset.

is obtained by taking the Fourier transforms of the raw reflected THz waveform $E(t)$ at each pixel, and integrating its magnitude in a specified frequency window, which can be expressed as,

$$M = \int_{f_1}^{f_2} |\mathcal{F}(E(t))| df, \quad (3)$$

where $\mathcal{F}$ represents the Fourier transform, f_1 and f_2 are the limits of the frequency window. In order to reveal the small and subtle features, the cutoffs are restricted to relatively high frequencies $f_1 = 1.0$ THz to $f_2 = 3.0$ THz in this study. The contrasts in THz C-scans in this study are all normalized to one (maximum). THz C-scans plotted based on this contrast mechanism should present both the surface and subsurface information since we here include the initial reflected peak from the air-sample interface.

THz C-scans based on raw THz signals for Sample 1 with THz polarization along the warp and weft directions are shown in Fig. 4 (a) and (b), respectively. With THz polarization along the warp direction, the THz C-scan in Fig. 4(a) clearly shows the warp-tow pattern on the sample surface, since warp-tows provide the highest THz reflection and therefore, dominate the contrast. The weft-tow pattern cannot be observed, which indicates the THz reflection from weft-tows is much smaller than that from warp-tows. Besides the warp-tow pattern, two regions of interest (ROI) are identified in Fig. 4(a). In ROI 1, the warp-tows show relatively smaller THz reflections than those outside ROI 1, which indicates less THz reflection, and thus, more THz penetration in these warp-tows. This is because the compression applied, in the four-point bending, leads to the growth of micro-scale cracking of tows and

matrix in these warp-tows [23], which lowers the electrical conductivity and thus decreases the reflectivity there. ROI 2 highlights the most severe damaged area displaying low contrast, in which the pattern of warp-tows is obscured due to the macro-scale damages, including matrix cracking, fiber distortion and fracture.

On the other hand, with THz polarization along the weft direction, in the THz C-scan in Fig. 4(b), the weave pattern, including both the warp- and weft-tows, is observed. Typical THz reflected signals from the warp- and weft-tows with different THz polarization are plotted in Fig. 5 for comparison. It is seen that, with THz polarization along the weft direction (in blue and red), the difference of THz reflections from the warp- and weft-tows is much smaller than that with THz polarization along the warp direction (in green and orange). It is also noticed that, with THz polarization along the weft direction, the amplitude of the THz reflected signal from the warp-tow is even a little higher than that from the weft-tow. This observation can be attributed to the fact that, due to the 2/1 twill weave pattern of the sample, the area of contact between the warp-tow on the surface and the weft-tow underneath is about two times as large as the area of contact between the weft-tow and the warp-tow underneath in the ply (shown in Fig. 1(c)). A larger contact area leads to higher electrical conductivity, and therefore, larger THz reflectivity. One striking point in Fig. 4(b) is that there is a much larger area, in which the warp- and weft-tows provide relatively smaller reflections, can be identified. Again, this feature indicates the growth of micro-scale cracking within the tows. We conclude that, for our 2/1 twill samples, THz polarization along the weft

Fig. 4. THz C-scan images of Sample 1. (a) The THz C-scan based on the raw THz signals with THz polarization parallel to the warp direction. (b) The THz C-scan based on the raw THz signals with THz polarization parallel to the weft direction. (c) The THz C-scan based on the THz subsurface signals with THz polarization parallel to the weft direction.

Fig. 5. Typical THz reflected signals from the warp- and weft tows with different THz polarization. Blue: THz reflection from warp-tows with polarization in the weft direction; red: THz reflection from weft tows with polarization in the weft direction; green: THz reflection from warp-tows with polarization in the warp direction; orange: THz reflection from weft-tows with polarization in the warp direction.

direction provides better imaging contrast, and is more effective in detecting the micro-scale damages within the tows. Macro-scale damages, including matrix cracking, fiber distortion and fracture, can also be detected in Fig. 4(b).

THz C-scans based on raw THz signals for Sample 2 with polarization along the warp and weft directions are shown in Fig. 6 (a) and (b), respectively. As was the case for Sample 1, with THz polarization along the weft direction, the weave pattern, including both the warp- and weft-tows, is observed. Macro-scale damage, including matrix cracking, fiber distortion and fracture can also be characterized with better contrast in ROI 4 of Fig. 6(b). ROI 5 clearly exhibits a typical damage morphology [24] within samples with twill weave under bending, in which the smaller THz reflection from the warp-tow indicates the tow failure, and the lower contrast around the failed tow indicates the growth of cracking. We emphasize that, polarization-sensitive THz imaging is effective in detecting various damages on the surface of woven carbon fiber-reinforced composites. The existence of micro-scale damages will reduce the strength of the THz reflections from the tows, and macro-scale damages will change or obscure the weave pattern.

As mentioned at the beginning of Section 4, THz C-scans based on

the raw THz signals contain both surface and subsurface information; however, since the initial THz reflection from the surface dominates the reflected signals, the subsurface features will tend to be hidden by the surface features. In order to highlight the subsurface information, THz subsurface spectral imaging based on THz sparse deconvolution is utilized in the following. The process by which the subsurface images are obtained involves temporally windowing out the initial surface reflection to emphasize later features associated with regions below the surface in the reflected signal. With THz polarization along the weft direction, typical reflected raw THz signals from Sample 1, corresponding to the signals at points A (64,30), B (383,72), and C (235,15) in Fig. 4(b), are shown in Fig. 7(a). From Point A to C, the amplitude of the first peak of raw THz signals, corresponding to the THz reflection from the surface of the laminate, decreases, which indicates successively increasing micro-scale damage progressing from Point A to B to C. More importantly, the increase of amplitude of the signals after the first peak is observed, which indicates the existence of echoes from subsurface features. The signals boxed in Fig. 7(a) are the superposition of the echoes from the subsurface features and ringing due to ambient water-vapor.

In order to extract the echoes from subsurface features, THz sparse deconvolution based on the shrinkage algorithm was performed. After deconvolution, a sparsity-based impulse-response function is obtained, which entirely depends on material structures. In the deconvolution process, we consider the THz reference signal as the input and the raw THz reflected signal as the output; therefore, the actual impulse-response function associated with reflection coefficients should be obtained by multiplying the deconvolved signal by a factor of -1 for phase correction. The impulse-response function obtained, or the deconvolved signal, consists of a baseline at zero and then a sequence of sharp peaks, which correspond to the interfaces producing the echoes. The details of the algorithm can be found in Ref. [22].

The deconvolved signals achieved by sparse deconvolution are shown in Fig. 7(b). The first positive peaks with the maximum amplitude, corresponding to the echoes from the surface, are clearly identified. From Point A to C, the amplitude of the first positive peaks decreases, which also indicates the increase of the micro-scale damage. There are small negative peaks located before the main peaks, which are attributed to the diffuse reflections (the back scattering of waves from a rough surface, on which the waves are scattered at many angles rather than at just one angle in the case of specular reflection) due to

Fig. 6. THz C-scan images of Sample 2. (a) THz C-scan in the frequency domain based on the raw THz signals with THz polarization parallel to the warp direction. (b) THz C-scan in the frequency domain based on the raw THz signals with THz polarization parallel to the weft direction. (c) THz C-scan in the frequency domain based on the THz subsurface signals with THz polarization parallel to the weft direction.

Fig. 7. Typical THz raw, deconvolved, and subsurface signals from Sample 1 with THz polarization along the weft direction. (a) THz reflected raw signals. (b) THz deconvolved signals based on sparse deconvolution. (c) THz subsurface echoes achieved by re-convolution following temporally windowing out the initial surface echo in the reflected signals. Notice that the horizontal scale in (c) differs that in (a) and (b).

Fig. 8. Surface images of Sample 1 and 2 obtained by X-ray tomography, corresponding to (a) ROI 2 in Fig. 4, (b) ROI 3 in Fig. 4, (c) ROI 4 in Fig. 6, and (d) ROI 5 in Fig. 6, respectively.

the inherent uneven surface of the woven-fiber fabric. The most important features are, at Points B and C, small negative peaks that are successfully resolved after the first positive peaks, confirming the existence of THz echoes originating from features in the subsurface region. In order to eliminate the THz reflection from the surface and recover the time-domain subsurface echoes, the first positive peaks in the deconvolved signals are temporally windowed out, and the time-domain subsurface echoes are obtained by the convolution of the

remaining deconvolved signals with the THz reference signal. The subsurface echoes recovered are shown in Fig. 7(c). It is observed that the amplitude of the subsurface echoes at Point C is larger than that at Point B. There are two physical origins of the subsurface echoes: (1) due to the growth of micro-cracking in tows, more THz power penetrates into the ply, and THz echoes are produced when THz waves encounter the interface between the warp- and weft-tows within the ply; (2) the growth of delamination induced by the four-point bending test will

produce THz echoes with larger amplitude, due to the existence of an air gap. Therefore, the amplitude of the subsurface echoes can be used to evaluate the severity of damage, as higher amplitude indicates the occurrence of delamination. For example, we can conclude that there is delamination at Point C, since the amplitude of subsurface echoes are significantly larger there.

Based on all the subsurface echoes extracted from the 3D volume data, THz subsurface spectral imaging was performed based on Eq. (3), and the subsurface images for Sample 1 and 2 are shown in Figs. 4(c) and Fig. 6(c), respectively. In Fig. 4(c), we observe that areas with high contrast mainly occur under the warp-tows, because the growth of micro-cracking mainly occurs in the warp-tows, and more THz power penetrates there that subsequently bounces back at the interface between warp- and weft-tows. Areas with highest contrast, especially in and around ROI 2, indicate the existence of delamination, corresponding to intra-ply delamination [24]. In Fig. 6(c), areas with delamination are also highlighted with highest contrast. The delamination seen in ROI 5 confirms the damage evolution, as the cracking propagates around the failed tow will inevitably seed the formation of delamination. Another striking point in Fig. 6(c) is that, compared with the damage pattern in Fig. 6(b), larger regions with lower contrast in ROI 4, corresponding to the macro-scale damages in matrix and tows, can be identified. Based on the subsurface images, we conclude that THz subsurface spectral imaging is indeed effective in revealing the fine details of subsurface damage.

Comparison with X-ray micro-computed tomography was carried out to confirm the results from THz imaging. An *EasyTom* system manufactured by *RX solutions* was used. The voxel size of the acquisitions was set to 12 μm for both samples. After the images acquisitions, a 3D reconstruction was done with the *X-Act* software from *RX solutions* for each sample. The obtained volumes were finally investigated using the image processing software *Avizo*. Surface images of typical areas in Sample 1 and 2 obtained by X-ray tomography are depicted in Fig. 8, corresponding to the various ROIs in the THz images. Fig. 8(a) and (c) clearly exhibit the macro-scale damages, including the matrix cracking, fiber distortion and fracture in ROI 2 and ROI 4. Besides the macro-scale damages, X-ray tomography is also able to identify the micro-scale damages. In Fig. 8(b), corresponding to ROI 3 in Fig. 4, the growth of micro-cracking, shown as dim horizontal lines, can be observed in almost every warp-tow. Due to the way the bending was applied in this study, micro-cracking mainly occurs in warp-tows, and propagates along warps-tows. The growth of micro-cracking, considered as the first stage of damage evolution, can also be observed in warp-tows in Fig. 8(a), (c), and (d). Fig. 8(d) shows the same damage morphology in ROI 5 as what is described in Fig. 6. Cross-sectional images obtained by

X-ray tomography are shown in Fig. 9, corresponding to typical cut-slices highlighted in Fig. 8. Fig. 9(a), (c), and (d) mainly show the existence of delaminations in the corresponding ROIs; the formation of delamination mainly occurs at the interface between the warp-tow and weft-tow underneath within the first ply, corresponding to the intra-ply delamination. The growth of transverse micro-cracking is confirmed and identified again in Fig. 9(b). In conclusion, the images obtained by X-ray tomography supports and confirms our analysis of THz images.

5. Conclusion

In this study, THz reflective imaging is utilized to characterize damage in woven carbon fiber-reinforced composite laminates subject to four-point bending tests. We first demonstrate how to take the advantage of THz polarization and THz deconvolution technique to enhance the capability of THz imaging on woven carbon fiber-reinforced composites given the high THz attenuation in these materials, and to extract the fine details of the subsurface damages.

We show that, besides the interaction between THz polarization and carbon-fiber orientation, the weave pattern and the contacts between the tows will also influence the THz reflection across the surface, and further influence the imaging contrasts in THz C-scans. For the samples with 2/1 twill weave pattern in this study, THz polarization along the weft direction provides better contrast, and is more effective in detecting the micro-scale damage compared with THz polarization along the warp direction.

An analysis technique is developed to enhance the ability of THz NDE to reveal subsurface damage in these materials, whereas many previous THz studies have yielded little useful subsurface information due to the low penetration of THz waves within these materials. THz subsurface spectral imaging based on THz sparse deconvolution is utilized to extract the subsurface echoes and highlight the subsurface damages which cannot be identified by the conventional THz imaging. Subsurface damage types, including the matrix cracking, fiber distortion and fracture, as well as intra-ply delamination have been successfully identified. Due to its sensitivity to carbon fibers, THz imaging is able to clearly show the weave pattern of carbon fibers, and the occurrence of both micro-scale and macro-scale damages can be identified. Our results prove that the enhanced THz imaging can be utilized to characterize both the surface and subsurface damages in woven carbon fiber-reinforced composites, and should be considered as a complementary NDE modality to investigate the damage mechanisms of woven carbon fiber-reinforced composites.

Fig. 9. Cross-sectional images of Sample 1 and 2 obtained by X-ray tomography, corresponding to (a) Cut 1, (b) Cut 2, (c) Cut 3, and (d) Cut 4, highlighted in Fig. 8, respectively.

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Appendix A. Supplementary data

Supplementary data related to this article can be found at <http://dx.doi.org/10.1016/j.ndteint.2018.07.001>.

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