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Jenny Faucheu, Elodie Bourgeat-Lami, Prévot Vanessa. A Review of Vanadium Dioxide as an Actor of Nanothermochromism: Challenges and Perspectives for Polymer Nanocomposites. *Advanced Engineering Materials*, 2019, 21 (2), pp.1800438. 10.1002/adem.201800438 . hal-02131942

**HAL Id: hal-02131942**

**<https://hal.science/hal-02131942>**

Submitted on 13 Nov 2020

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# A review of Vanadium Dioxide as an actor of nanothermochromism: challenges and perspectives for polymer nanocomposites.

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## Abstract:

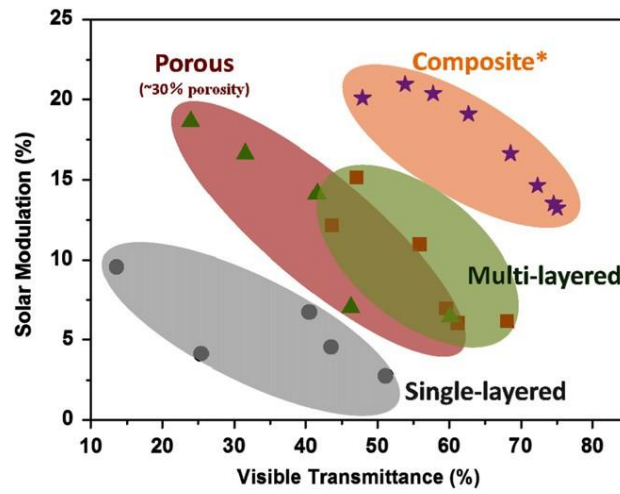
VO<sub>2</sub> has been intensively studied as a model system for Metal-Insulator Transition, ever since its phase switching behavior at relatively low temperature ( $T_c=68^\circ\text{C}$ ) was discovered. Its thermo-active characteristics have been studied for applications in thermal solar collectors to prevent overheating of the fluid absorber, smart windows and smart tiles to optimize thermal comfort in buildings. In the last decades, strong efforts were made to develop protocols to process VO<sub>2</sub>-based material into thin films, coatings, porous layers and glazings in order to provide thermo-active properties to common substrates. A selection of complementary works in both the synthesis of VO<sub>2</sub> nanoparticles and formulation of VO<sub>2</sub>-polymer nanocomposites is proposed in the present review. The perspective of this work is to provide an easy entry to the thermochromic nanocomposite research field as a tutorial to tackle future challenges such as improving the application-oriented properties, in particular transition temperature  $T_c$  and solar transmittance modulation  $\Delta T_{\text{sol}}$ . Synthesis techniques to fabricate VO<sub>2</sub> particles and nanoparticles and strategies to insert VO<sub>2</sub> particles in a polymer matrix are presented.

## 1. Introduction

Among the different VO<sub>2</sub> polymorphs, the monoclinic M or M<sub>1</sub> phase and the tetragonal rutile-type R phase have attracted a lot of attention due to the reversible Metal-Insulator Transition (MIT) that takes place between these two phases.<sup>[1]</sup> The phase transition changes the behavior from semiconducting (VO<sub>2</sub>(M)) to metallic (VO<sub>2</sub>(R)) and thus modifies the electrical conductivity and the optical properties. The nature of the MIT has been much debated, but recent work indicates a Mott transition.<sup>[2][3][4]</sup> The reversible phase transition is a first order transition that results in a change of crystallographic structure at a transition or critical temperature  $T_c$ . VO<sub>2</sub> has been intensively studied as a model system for MIT, ever since its phase switching behavior around  $T_c=68^\circ\text{C}$  was discovered<sup>[2]</sup>. Under heating conditions, the material starts at  $T < T_c$  in its semiconductor state VO<sub>2</sub>(M) and sees the metallic phase VO<sub>2</sub>(R) germinate during the MIT transition. Under cooling conditions, the material starts at  $T > T_c$  in its metallic state VO<sub>2</sub>(R)

and sees the semiconductor phase germinate during the transition. Note that this transition exhibits a hysteresis meaning that the transition under heating conditions occurs at a slightly different temperature than the transition under cooling conditions. The optical properties of VO<sub>2</sub> such as reflectance,<sup>[5]</sup> emittance<sup>[6]</sup> or transmittance<sup>[7]</sup> are modified during the phase transition. Its thermo-active characteristics have been studied for coating applications related to solar-irradiation on substrates that can be either transparent or absorbent materials for instance, VO<sub>2</sub> coatings could be efficient in thermal solar collectors to prevent overheating of the fluid absorber<sup>[8]</sup>, and in smart windows<sup>[7][9][10][11][12][13][14]</sup> and smart tiles<sup>[15]</sup> to optimize thermal comfort in buildings. Solar irradiation onto the atmospheric envelope can be approximated by blackbody-like radiation from the surface of the sun and falls at  $0.2 < \lambda < 3 \mu\text{m}$ . Visible light is governed by the luminous efficiency of the eye, which is given by a bell-shaped curve falling at  $0.4 < \lambda < 0.8 \mu\text{m}$  with a peak around  $\lambda = 0.55 \mu\text{m}$ . Solar transmittance modulation  $\Delta T_{\text{sol}}(\lambda)$  of a thermoactive coating depends on the wavelength and is defined as the difference between the transmittance spectrum at low temperature and the transmittance spectrum at high temperature for  $0.2 < \lambda < 3 \mu\text{m}$  (the transmittance modulation  $\Delta T_{\text{IR}}(\lambda)$  can also be defined for the IR part of the solar spectrum i.e.  $0.8 < \lambda < 3 \mu\text{m}$ ). The solar transmittance modulation  $\Delta T_{\text{sol}}$  is defined as the integrated value over the wavelength range. In particular, VO<sub>2</sub>-glazings have been the subject of intensive research efforts in recent years due to their potential application as smart windows. Numerical calculations<sup>[16][7]</sup> show that 0.1-0.3  $\mu\text{m}$  thick films exhibit solar transmittance modulations  $\Delta T_{\text{sol}}$  around 15-20% (Figure 2a, b, c). These characteristics can induce up to 45% energy consumption improvement with VO<sub>2</sub> coated-glass compared to plain float glass.<sup>[17]</sup> In particular, the MIT involves an abrupt change in near-infrared transmittance that makes VO<sub>2</sub> an attractive thermochromic material to intelligently adjust the near infrared radiation according to the environment temperature.<sup>[18][19][20][21][22]</sup>

Transmittance modulation by VO<sub>2</sub>-based coatings of various morphologies has motivated the bulk of the work in this field with three main challenges.<sup>[13]</sup> The first challenge was to reduce the transition temperature  $T_c$  of VO<sub>2</sub> in order to be more adequate for applications related to Human thermal comfort such as smart windows. The second challenge was to maximize the solar transmittance modulation specifically in the IR region. And finally, the third challenge - specific to smart windows applications in which transparency is mandatory - was to maximize the transmittance in the visible range. These limitations can be alleviated by the elaboration of composite film materials containing VO<sub>2</sub> nanoparticles or nanodomains. Indeed, several experimental results and calculations indicate that nanocomposites containing VO<sub>2</sub> particles embedded in a dielectric matrix such as polymers or metal oxides have superior properties, that combine high luminous transmittance and high solar transmittance modulation, than VO<sub>2</sub>-thin continuous solid films (**Erreur ! Source du renvoi introuvable.**)<sup>[23][24][25][26][27][28]</sup> in particular, due to the triggering of plasmonic absorption by VO<sub>2</sub> nanoparticles in their metallic state. “Nanothermochromism” is a term that was first coined by Li et al.<sup>[7]</sup> to describe the approach consisting in mixing VO<sub>2</sub> nanoparticles with a glass or a polymer matrix. This term is now commonly used.<sup>[29]</sup> Besides these functional constraints, stability and durability in outdoor environments are still open questions that will take advantage of the huge empirical and research works on durability of composites that are available.



**Figure 1. Main performances of VO<sub>2</sub> films prepared by different methods. The results for composite films were made from numerical calculations.<sup>[7]</sup>**

This review intends to extract papers dedicated to thermochromism provided by VO<sub>2</sub> nanoparticles in order to draw the challenges and perspectives of using these nanoparticles into polymer matrices. If such challenges were overcome, a wide range of applications related to adaptive passive control of thermal properties would become industrially realistic (thermal comfort in buildings, thermal stealth in military applications, thermal sensors...). A selection of complementary works in both the synthesis of VO<sub>2</sub> nanoparticles and formulation of VO<sub>2</sub>-polymer nanocomposites is proposed in the present review. The perspective of this work is to provide an easy entry to the thermochromic nanocomposite research field as a tutorial to tackle future challenges. Section 2 describes the existing strategies for improving the application-oriented properties, in particular transition temperature  $T_c$  and solar transmittance modulation  $\Delta T_{sol}$ . Section 3 is dedicated to synthesis techniques adequate for fabricating VO<sub>2</sub> particles and nanoparticles. Section 4 collects the existing approaches to insert VO<sub>2</sub> particles in a polymer matrix. Section 4 will wrap up perspectives and challenges of using VO<sub>2</sub>-polymer coatings as smart architectural coatings for buildings and urban structures and more generally for adaptive passive thermal coatings.

## 2. VO<sub>2</sub> a model material

This work is dedicated to polymer-VO<sub>2</sub> nanoparticles composite paints. A comprehensive review of strategies explored to improve the thermochromic properties of VO<sub>2</sub>-based materials is presented. The VO<sub>2</sub> topic has already drawn interest in several reviews<sup>[30][31][32]</sup> under different scope in the past. In particular, Warwick and Binions<sup>[30]</sup> presented the fundamental physical and chemical principles underlying the electronic structure and properties of VO<sub>2</sub>, and the chronological developments in the theory behind the thermochromic transitions such as, the effects of electron-electron interactions and structural phase changes due to lattice distortions. An extensive discussion and observations on the current understanding of the nature of MIT exhibited by VO<sub>2</sub> investigated via computational, experimental and theoretical studies, were detailed. This previous review also provides a panorama of techniques for thin film deposition such as sol-gel and solution, pulsed laser deposition, physical vapor deposition, atomic layer

deposition, chemical vapor deposition, atmospheric pressure chemical vapor deposition or aerosol assisted chemical vapor deposition. A more complete review on thin film deposition techniques is provided by Wang et al.<sup>[31]</sup> To date, gas-phase deposition methods, such as sputtering and chemical vapor deposition and solution-based deposition are the most common methods for the fabrication of VO<sub>2</sub> films. Gao et al.<sup>[32]</sup> reviewed the state-of-the-art solution processes used to prepare VO<sub>2</sub> films, with a special emphasis on polymer-assisted deposition methods. The resulting VO<sub>2</sub> films are compared in terms of optical properties and transmittance modulation. Gao's review suggests that chemical preparation is inexpensive, easy to scale up, and best suited for the practical applications of the fabricated materials. VO<sub>2</sub> films prepared by these methods show controllable morphology and thickness but complex optical properties compared with those prepared by gas-phase methods. These methods are not best suited to produce nanoparticles to be further used as pigments in a polymer matrix. However, the final properties and the strategies explored to reach these final properties of the deposited coatings are interesting insights to be considered in a nanoparticle/polymer system. In particular, under specific experimental conditions, the final morphology exhibits nanostructured VO<sub>2</sub> films such as nanoporous films or films made of partial sintering of nanoparticles.

## **2.1. Properties of discontinuous VO<sub>2</sub> coatings: towards polymer-VO<sub>2</sub> nanocomposites**

Among the variety of elaboration processes explored to fabricate VO<sub>2</sub> coatings, some were specifically explored in order to form "discontinuous" coatings<sup>[33][34]</sup> such as porous coatings<sup>[35][36][37][38][39]</sup> or etched films<sup>[40][41]</sup>. These works were primarily motivated by the need for increasing luminous transmittance without degrading the infrared modulation. For instance, freeze-drying was employed as an efficient way to avoid aggregation phenomena in the process of synthesizing metal oxide nanoparticles<sup>[42]</sup> and so fabricate porous VO<sub>2</sub> films. Cao et al.<sup>[35]</sup> reported VO<sub>2</sub> nanostructures with porosities from 16% to 45% fabricated using freeze-drying. With a decreasing pre-freezing temperature, grain sizes and pores of VO<sub>2</sub> films tended to be slightly increased. The solar transmittance modulation and the visible transmittance were increased with the increase in pore sizes and porosity. Wang et al.<sup>[36]</sup> reported the preparation of VO<sub>2</sub> films with different nanostructures using sol-gel processes. Compared to a conventional VO<sub>2</sub> sintered films, the nanoporous structure formed exhibited improved thermochromic properties, with a 16% increase in the visible transmittance. Ding et al.<sup>[37]</sup> prepared VO<sub>2</sub> films with a micro/nano hierarchical porous structure through a nanoparticle assembly, with cetyl trimethyl ammonium-vanadate (CTAV) as a precursor. The obtained VO<sub>2</sub> films exhibited a visible transmittance significantly increased (+25 %) compared to continuous films. Wang et al.<sup>[40]</sup> reported the use of surface engineering methods, including acid etching and self-patterning to enhance the thermochromic performance of VO<sub>2</sub> thin films. Two different morphologies, i.e. dense-packed and nanograin-array patterned continuous VO<sub>2</sub> thin films could be obtained by controlling the thickness of the VO<sub>2</sub> thin film. The self-patterning-induced nanograin-array patterned VO<sub>2</sub> thin film exhibited much larger  $T_{lum}$  than the dense-packed counterpart, which should be mainly ascribed to the antireflection effect. With acid etching, the dense-packed surface could be changed to patterned surface, which showed improved  $T_{lum}$  and  $\Delta T_{sol}$ . Li et al.<sup>[41]</sup> compared a nanostructured VO<sub>2</sub> film with a grain size of about 8 nm and a microstructured VO<sub>2</sub> film with grain sizes of about 1-2  $\mu$ m. They observed a transition temperature  $T_c$  of 35°C for the nanostructured film compared to 68°C for the

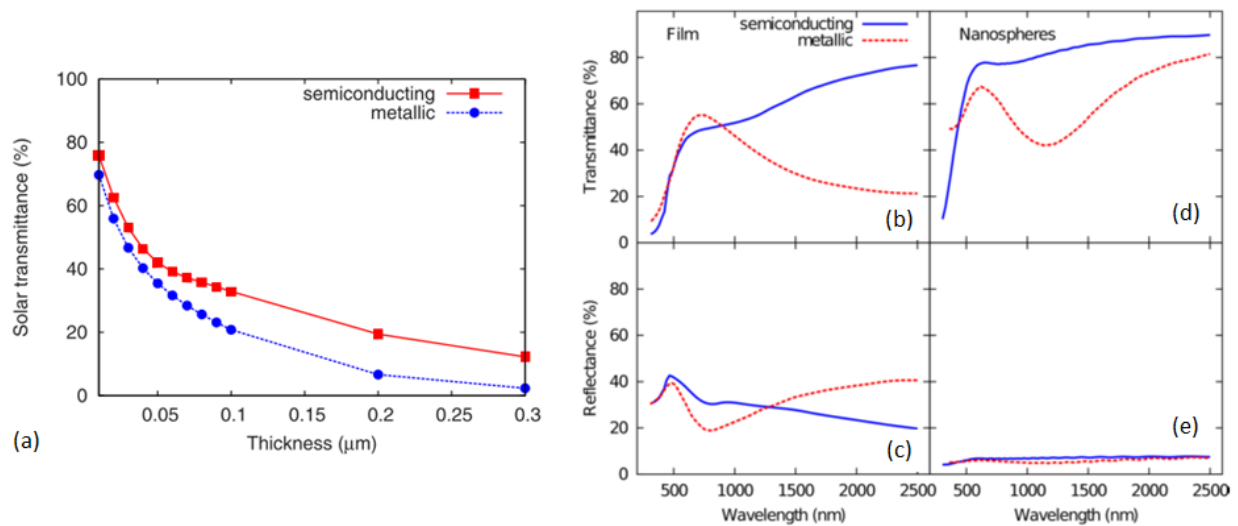
microstructured one. Xu et al.<sup>[38][39]</sup> reported the fabrication of porous nanostructured VO<sub>2</sub> films with sphere-shaped and rod-shaped nanocrystals using a surfactant-assisted sol-gel method. Different surfactants have been used as the nanostructure directing agents including cetyl trimethyl ammonium bromide, polyethylene glycol and sodium dodecyl sulfate. They noted that the T<sub>c</sub> of the nanostructured films was lower than the one of dense VO<sub>2</sub> films. In addition, the hysteresis width was also much wider. However, among the nanostructured films, the films with rod-shaped nanostructures exhibited a narrower hysteresis probably due to improved contacts between particles, which favored phase transition propagation.

DC reactive magnetron sputtering process was investigated to produce granular coatings<sup>[33]</sup>. Zhang et al.<sup>[34]</sup> prepared VO<sub>2</sub> films with spheroidal nanoparticles of different sizes, from 45 to 80 nm diameter, by using DC reactive magnetron sputtering process to investigate the effect of grain size on the hysteresis. Adjusting the annealing time controlled the size of the VO<sub>2</sub> nanoparticles. It was observed that the hysteresis width was inversely proportional to the average grain size.

Besides, several authors have shown that the thermochromic performances of VO<sub>2</sub> films can be significantly increased by controlling the morphology of the nanostructure of the film material using silica templates,<sup>[43][44][45]</sup> latex templates<sup>[46][47][18][48]</sup> or other micro-patterned templates.<sup>[49][50]</sup> For instance, Zhou et al.<sup>[47]</sup> reported a two-dimensional periodic VO<sub>2</sub> structure. The resulting porous thermochromic film exhibited a very high T<sub>lum</sub> of 81% but a moderate ΔT<sub>sol</sub> of 7.9% due to the large pore size (200 nm).

In summary, the variety of processes explored to fabricate discontinuous VO<sub>2</sub> films highlights the advantages and drawbacks compared to continuous films. Indeed, both the luminous transmittance and the solar transmittance modulation are efficiently improved with addition of porosity. This effect is ascribed to interfacial scattering phenomena that limit direct surface reflection. However, discontinuity in the films also tends to lower T<sub>c</sub> but widen the hysteresis. It appears that there is still room for a convenient and large-scale method to produce optimized discontinuous VO<sub>2</sub> films. Consistently, VO<sub>2</sub> nanoparticles have attracted much interest in recent years, because they provide additional optical effects that could reinforce the optical properties provided by the thermochromic characteristics. Besides intrinsic optical properties of bulk VO<sub>2</sub>, the optical properties of VO<sub>2</sub> nanoparticles embedded in a dielectric medium are impacted by scattering and absorption that can both be strongly influenced by the plasmon resonance. The fact that VO<sub>2</sub> nanoparticles in their metallic state can undergo a plasmon resonance has been demonstrated both numerically<sup>[51][52]</sup> and experimentally.<sup>[53][54][55]</sup> Lopez et al.<sup>[54]</sup> showed that the MIT of the nanophase precipitates turns on the classical surface plasmon resonance, with specific features that depend on the size and aspect ratio of the VO<sub>2</sub> particles. Bai et al.<sup>[51]</sup> performed numerical calculations of the optical scattering efficiencies of VO<sub>2</sub> nanoparticles with three different shapes (sphere, cube, porous aggregate) using Mie theory and Discrete Dipole Approximation (DDA). All three model nanoparticles contained the same volume of VO<sub>2</sub>, that of a sphere of 200 nm radius. In all three cases, scattering of the VO<sub>2</sub>-metallic (R) peaked in the near-infrared due to a broad plasmon resonance at about 1.4 μm. Granqvist, et al.<sup>[16][7]</sup> performed a numerical study that focused on a nanocomposite approach considering VO<sub>2</sub> nanoparticles embedded in a dielectric matrix (Figure 2d, e). The calculations used a rather low volume fraction of VO<sub>2</sub> of f = 0.01 and a thickness of 5 μm. The most salient feature is the pronounced absorption band in the near-infrared that is centered at a wavelength of 1.2 μm. This transmittance

minimum, which corresponds to an absorption maximum in the infrared is due to localized plasmon resonance of free electrons in VO<sub>2</sub> nanoparticles.



**Figure 2. (a) Computed data of solar transmittance as a function of thickness of VO<sub>2</sub> films in semiconducting and metallic states, (b, d) Spectral transmittance and (c, e) reflectance for a 50 nm-thick thin film of VO<sub>2</sub> (b,c) and for a layer comprising a dilute dispersion of VO<sub>2</sub> nanospheres, with an equivalent VO<sub>2</sub> thickness of 50 nm, in a medium mimicking transparent glass or polymer (d, e). Adapted with permission from ref.<sup>[16][7]</sup> Copyright 2016 and 2010, MDPI and AIP Publishing.**

The calculations demonstrate that embedding VO<sub>2</sub> nanoparticles in a dielectric second phase is a reasonable strategy combining a high luminous transmittance and a significantly enhanced solar energy transmittance modulation. This strategy overcomes the drawbacks of thin films such as low transmittance that induce an unfavorable yellow color, which is unpleasant to human eyes and low solar transmittance modulation. In addition, it has been shown numerically that the difference in solar transmittance between the semiconducting and metallic phases is 20% for dispersions of spheres while the corresponding number for the film is about 7%. Random distributions of nanoparticles show that the highest transmittance levels are found for the spherical particles, compared to rod-like and platelet-like particles<sup>[7]</sup>. Finally, this numerical study shows that efficient thermochromic polymer-VO<sub>2</sub> nanocomposites, quasi-spherical VO<sub>2</sub>(M) nanoparticles with uniform sizes seem to be ideal to achieve good visible transmittance and high modulation of infrared light. However, the formation of spherical VO<sub>2</sub> particles and the control of the transition characteristics (hysteresis width and T<sub>c</sub>) remain a challenge.

## 2.2. Effect of elemental doping on transition and optical characteristics

The physical origin of the Mott transition<sup>[2][3][4]</sup> is the interplay between the Coulomb repulsion of electrons and their degree of localization (electronic band width). Several studies have focused on strategies to decrease the transition temperature to nearer room temperature. The most common strategy to alter the transition temperature is elemental doping, which involves the intentional incorporation of atoms or ions of suitable elements into the host

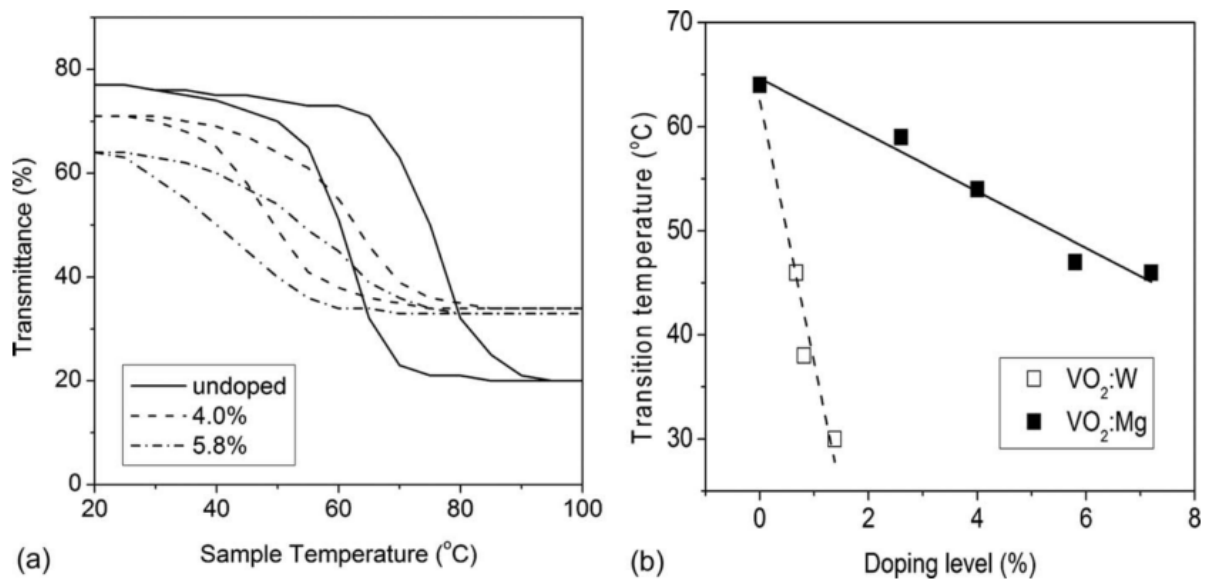
lattice.<sup>[56][57]</sup> Two main mechanisms are considered to explain the altering of  $T_c$  by elemental doping in  $\text{VO}_2$ . If the interatomic distance is large, the electrons remain localized in the energy bands separated by the band gap. If the electron density increases by elemental doping for example, delocalization of the electrons can occur and the material transits to a metallic state. It is also considered that elemental doping modifies the density of defect-nucleation sites, which in turn influences the transition propagation and so the transition temperature. Hu et al.<sup>[58]</sup> studied the band gap widening provided by Mg-dopants into the  $\text{VO}_2$  lattice. They managed to reach a qualitative agreement between experimental and computational data which opens insights to discover alternative dopants to widen band gaps in thermochromic materials. Tan et al.<sup>[59]</sup> unveiled the atomic structure evolutions of W dopant and their role in tailoring the  $T_c$  of  $\text{VO}_2$  transition. They found that the local structure around the W atom is intrinsically symmetric with a tetragonal-like structure. Using synchrotron radiation X-ray absorption spectroscopy combined with first-principles calculations, they gave the experimental evidence that the symmetric W core drives the de-twisting of the nearby asymmetric monoclinic  $\text{VO}_2$  lattice to form rutile-like  $\text{VO}_2$  nuclei, and the propagations of these W-encompassed nuclei through the matrix lower the thermal energy barrier for phase transition. Thus, besides elemental doping, other strategies were explored to alter the transition temperatures based on the modification of the density of defect-nucleation sites. Internal stresses can be introduced by elemental doping<sup>[60]</sup> but also by film preparation processes.<sup>[61]</sup> The transition temperature has also been shown to be affected by film strain<sup>[62]</sup> and it has been demonstrated that strain can be introduced by a careful choice of deposition conditions.<sup>[63]</sup> Therefore, it is interesting to note that doped  $\text{VO}_2$  elaborated by different techniques can lead to contradictory conclusions. For instance, the doping by Al (+III) of  $\text{VO}_2$  films obtained by pulsed laser leads to a decrease in  $T_c$  (40°C),<sup>[64]</sup> whereas the use of sol-gel demonstrates its increase (80°C).<sup>[56]</sup> The effect of internal stresses (more compressive during laser ablation than for sol-gel) could play a role in the transition temperature in addition to the elemental doping.

Wan et al.<sup>[65]</sup> performed a theoretical study of doping processes and investigated the M-doped monoclinic M1 phase of  $\text{VO}_2$  with  $M = \text{B, C, N}$  at O site, and  $M = \text{B, Mg, Al}$  at V site. This study showed that all these doping elements enabled to decrease  $T_c$  and highlighted that the C-doped  $\text{VO}_2$  could be the best effective doping element in M-doped ( $M = \text{B, C, N, Al, Mg}$ )  $\text{VO}_2$ . From an experimental point of view, the most effective route for appreciable reduction of  $T_c$  is doping some donor impurity atoms, which could be easily realized by chemical substitution methods.<sup>[66]</sup> Considering the V site for doping, the doping of  $\text{VO}_2$  films with cations of (+V) and (+VI) valences leads to a reduction of the transition temperature. For instance,  $\text{M}_x\text{V}_{1-x}\text{O}_2$  can have lowered  $T_c$  with M being W(+VI),<sup>[67]</sup> Mo(+VI),<sup>[68]</sup> Ta(+V),<sup>[69]</sup> Nb(+V),<sup>[70]</sup> Sb(+V).<sup>[71][72]</sup> Note that the elemental doping can also impact the optical properties of the films. Hanlon et al.<sup>[68]</sup> reported the doping of  $\text{VO}_2$  films with Mo(+VI) and at a doping level of 7 atomic percentage (at%) the transition temperature was measured at 24°C. They noted that the optical reflectance in the semiconductor state increased slightly with additional dopant, while the reflectance in the metallic state remained constant. It was found that the most effective metal ion dopant was tungsten W(+VI) which lowers the transition temperature by 25°C for every atomic percent of incorporated dopant.<sup>[73]</sup> Zhang et al.<sup>[74]</sup> prepared W-doped  $\text{VO}_2(\text{M})$  nanobelts by hydrothermal treating  $\text{VO}_2(\text{B})$  nanobelts in the presence of tungstic acid. The  $T_c$  of W-doped  $\text{VO}_2(\text{M})$  nanobelts was reduced to 50°C in the heating cycle and to 42°C in the cooling cycle. The transmittance modulation was also



increased compared to undoped VO<sub>2</sub>. Shi et al.<sup>[75]</sup> reported a successful synthesis of W-doped VO<sub>2</sub> particles by hydrolysis of vanadyl sulfate mingled with a small amount of sodium tungstate dihydrate and subsequent calcination. Chen et al.<sup>[76]</sup> prepared thermochromic W-doped VO<sub>2</sub> nanopowders by solution-based method using ammonium tungstate. It was shown that tungsten dopant concentration has almost no effect on the morphology of the nanoparticles and the granular particles were about 20– 60 nm. The reduction of transition temperature was estimated to be about 17°C per 1 at% of W doping with the tungsten extents of ≤ 1 at%, only about 9.5K per 1 at% with the tungsten extents of >1 at%.<sup>[73]</sup> With 3 at% W-doped VO<sub>2</sub>, the phase transition temperature could be reduced to 35°C. Binions et al.<sup>[77]</sup> and Piccirillo et al.<sup>[78]</sup> indicated that tungsten doping could significantly decrease hysteresis loop widths to about 10°C.

Mlyuka et al. compared Mg and W doping in thermochromic films of M<sub>x</sub>V<sub>1-x</sub>O<sub>2</sub> (M = Mg or W) made by reactive DC magnetron sputtering onto heated glass<sup>[79]</sup>. The transition temperature decreased by ~3°C/at% Mg, while the optical transmittance increased concomitantly. Specifically, both the luminous and the solar transmittance were enhanced by ~10% when the Mg content was ~7 at%. They showed that a similar decrease of T<sub>c</sub> needs a higher Mg content than W content (Figure 3).



**Figure 3. (a) Transmittance at  $\lambda = 2300$  nm vs. temperature and (b)  $T_c$  vs. doping level for 50-nm-thick VO<sub>2</sub>, Mg<sub>x</sub>V<sub>1-x</sub>O<sub>2</sub> films and W<sub>x</sub>V<sub>1-x</sub>O<sub>2</sub> films, the lines are solely to guide the eyes. Adapted with permission from ref.<sup>[79]</sup> Copyright 2009, AIP Publishing.**

Zhou et al.<sup>[80]</sup> reported the successful preparation of Mg-doped VO<sub>2</sub> nanoparticles via hydrothermal synthesis. The transition temperature (T<sub>c</sub>) decreased by approximately 2°C per at% Mg. The T<sub>c</sub> decreased to 54°C with 7.0 at% dopant. Li et al.<sup>[27]</sup> reported undoped and Mg-doped thermochromic VO<sub>2</sub> films with atom ratios up to 21 at% deposited by reactive DC magnetron sputtering onto heated glass and carbon substrates. It was found that VO<sub>2</sub> films with less than 6 at% Mg dopant and judiciously chosen thicknesses, can have much improved luminous transmittance and solar energy modulation than pure VO<sub>2</sub> films. The addition of single-layer anti-reflection coatings

on the top of the VO<sub>2</sub>-based films was shown to further enhance the performance of Mg-doped films. Moreover, modeling of nanoparticle composites confirmed that nanoparticle composites are superior to films but suggested that undoped VO<sub>2</sub> nanoparticles are more favorable than Mg-doped ones.

Doping by cations of small sizes and valences less than or equal to that of V leads to higher transition temperatures at 68°C. For instance, an increase of  $T_c$  is observed with M being Ge(+IV),<sup>[70]</sup> Al(+III),<sup>[81]</sup> Sn(+IV),<sup>[82]</sup> Ti(+IV),<sup>[83]</sup> or Sb(+III).<sup>[71]</sup> Various efforts have been conducted to study the effects of Ti doping on VO<sub>2</sub>.<sup>[56][84][85]</sup> It was found that the transition temperature increased as the Ti contents increased. Comparative studies also suggest that Ti doping significantly decreased the hysteresis loop width. Du et al.<sup>[86]</sup> prepared Ti-doped VO<sub>2</sub> films with different Ti/V ratios by polymer assisted deposition. The transition temperature increased and the hysteresis loop width was narrowed from 38.2 to 3.5°C. This was qualitatively analyzed using nucleation theory proposed by Lopez et al.<sup>[64]</sup> that suggested that the driving force for the transition of undoped VO<sub>2</sub> thin films was the extrinsic defects introduced during the film preparation process, contrary to Binions' et al.<sup>[63]</sup> approach that relates the hysteresis loop width to crystallographic orientations. Indeed, Du et al.<sup>[86]</sup> prepared Ti-doped VO<sub>2</sub> films that had a large size distribution without any changes in crystallographic orientations. Ti doping could probably enhance the density of defect-nucleation site. Undoped films had small particle sizes (130 nm). Hence the small volume resulted in a lack of nucleation sites in each particle for undoped VO<sub>2</sub> films, inducing a large hysteresis loop width. For doped films, the particles were even smaller. According to Lopez's nucleation theory, the hysteresis loop width should be widened. However, a comparatively large doping concentrations ( $x > 2\%$ ) would greatly increase the density of defect-induced nucleation sites, which would play a leading role compared to particle size. As a result, in spite of the diminishing particle sizes, a high density of defect-induced nucleation sites would significantly reduce the hysteresis loop width.

### 2.3. Effect of elemental doping on nanoparticle phase and morphology

The monoclinic VO<sub>2</sub>(M) has a preferential orientation along the [110] direction and is prone to grow to one-dimensional nanostructures, such as nanowires, nanorods and nanobelts.<sup>[87][88][89][90][91]</sup> Doping has also been studied as a way to control sphericity of nanoparticles.<sup>[80][90][71]</sup> Doping has the ability to control the size and morphology of nanocrystals using various mechanisms, such as competitive nucleation and dipole polarization of the dopant ion. Zhou et al.<sup>[80]</sup> reported the successful preparation of Mg-doped VO<sub>2</sub> nanoparticles via hydrothermal synthesis. The morphology changed from nanorods for undoped VO<sub>2</sub> to irregular nanoparticles for the Mg-doped VO<sub>2</sub>, revealing the Mg-doping's strong impact on the morphological evolution of the nanoparticles. Baik et al.<sup>[90]</sup> reported that the observed transition temperatures, which ranged between 62 and 70°C for the nanowires measured, trended downward with decreasing nanowire width. Gao et al.<sup>[71]</sup> reported the preparation of crystalline Sb-doped VO<sub>2</sub> nanoparticles with tunable sizes via a one-pot hydrothermal method. This doping strategy enabled the simultaneous control of the size, morphology and phase of the outcome material. Without Sb(+III) doping, VO<sub>2</sub>(A) nanorods grew with a preferential orientation along the [001] direction. For the sample  $x = 0.91$  at% of Sb(+III), VO<sub>2</sub>(A) was still a

major phase with VO<sub>2</sub>(M) being a minor phase. As x increased to 2.68 at%, VO<sub>2</sub>(M) became the main phase. Pure VO<sub>2</sub>(M) was obtained for x = 4.69 at%, 13.60 at% and 33.10 at%. TEM observations provided evidence that the size of the nanocrystals was gradually reduced and that the size distribution tended to be more homogeneous with increasing Sb(+III) contents. They suggested that Sb(+III) doping promoted the formation of VO<sub>2</sub>(M) nanoparticles from the rod-like VO<sub>2</sub> nuclei at about 240–250°C. As the temperature was increased to 260°C and/or the reaction time was prolonged, the VO<sub>2</sub>(A) rods transformed into VO<sub>2</sub>(M) nanoparticles most likely due to dissolution-recrystallization. Gao et al.<sup>[71]</sup> proposed the following mechanism. With Sb(+III) doping, the substitution of two ions of Sb(+III) for V(+IV) in VO<sub>2</sub> requires an extra oxygen vacancy for charge compensation, and higher surface energy of nanoparticles favors oxygen vacancies to aggregate on their surfaces, which may induce transient electric dipoles with their positive poles outward. These transient electric dipoles hinder the diffusion of VO<sup>2+</sup> as well as other charged nuclei to the grain surface, thus retarding the growth of VO<sub>2</sub> particles. The above discussion suggests that an extra repulsion force induced by dopant-resultant defects plays a significant role in the modification of particle sizes. This mechanism was confirmed by using other dopants in VO<sub>2</sub> systems. Using a high-valence ion dopant, such as Sb(+V), to substitute for V(+IV) sites in the VO<sub>2</sub>(M) lattice may induce negatively charged defects for the charge balance. The concentration of these defects on the grain surfaces causes an electric dipole that can accelerate the diffusion of VO<sup>2+</sup> ions in solution to promote the grain growth. In addition, the impacts of Bi(+III) and Ti(+III) doping on the growth of VO<sub>2</sub> nanocrystals under the analogous solution conditions were studied to directly validate the proposed mechanism. Bi(+III) and Ti(+III) ions possess the same valence as Sb(+III) and are speculated to induce similar effects in VO<sub>2</sub> crystal growth. As expected, both dopants benefit the formation of hexahedral VO<sub>2</sub>(M) nanocrystals with significantly smaller sizes. These results confirmed that doping is able to induce the transformation of VO<sub>2</sub> polymorphs.

## 2.4. Effect of “interfacial doping” on transition and optical characteristics

In a larger extent than elemental doping, the integration of a secondary domain of inorganic components into VO<sub>2</sub> films or onto VO<sub>2</sub> particles has been proven effective in modulating the optical properties of the films. Core-shell particles with enhanced thermochromic properties include VO<sub>2</sub>@SiO<sub>2</sub>,<sup>[92][93][94][95][96][97][48]</sup> VO<sub>2</sub>@ZnO,<sup>[98]</sup> VO<sub>2</sub>@Al-O,<sup>[99]</sup> or VO<sub>2</sub>@TiO<sub>2</sub><sup>[100]</sup> as well as inverted core-shell structures such as SiO<sub>2</sub>@VO<sub>2</sub>.<sup>[101][102]</sup> Li et al.<sup>[100]</sup> reported the synthesis of VO<sub>2</sub>/TiO<sub>2</sub> nanorods highlighting an approach to control the optical and transition properties by modifying the surface/interface state and microstructure of VO<sub>2</sub> particles with core/shell nanotechnology. TiO<sub>2</sub> coating can not only enhance the chemical stability of VO<sub>2</sub> but also exhibits the potential to control the transition temperature and hysteresis by designed "interfacial doping" with Ti through annealing. Such "interfacial doping" has the advantage of avoiding the introduction of a large amount of defects as compared with bulk doping. Bulk defect deteriorates the modulation ability of VO<sub>2</sub> to a large degree. Other cationic or anionic ions for example, Mo, W, Cr, F etc, are expected to be able to adjust the transition property by bulk doping and may also be designed to achieve "interfacial doping" through core/shell nanotechnology, thus to modify the phase transition of VO<sub>2</sub> without bringing in excessive defects and retain a comparably sound switching efficiency as in pure VO<sub>2</sub>. Qureshi et al.<sup>[103]</sup> reported composite

TiO<sub>2</sub>–VO<sub>2</sub> films showing a significant reduction in the thermochromic transition temperature (54°C). It was shown that composite films of this kind did not interfere with the functionality of each individual component and that solid solutions were not formed. The films exhibited both photocatalytic properties from TiO<sub>2</sub> and thermochromic properties from VO<sub>2</sub>. Du et al.<sup>[104]</sup> reported the preparation of VO<sub>2</sub>–ZrV<sub>2</sub>O<sub>7</sub> composite films on silica glass substrates by polymer-assisted deposition. The addition of ZrOCl<sub>2</sub> in the precursor solution greatly influenced the film microstructure and transition characteristics. The composite films with similar thickness of about 95 nm showed decreased particle sizes and significantly enhanced luminous transmittances with increasing Zr/V ratios, from 32.3% to 53.4% at Zr/V=0 and 0.12 respectively. The transition temperature T<sub>c</sub> was about 60°C for all samples, which is 8°C lower than that of the single crystal. Miyazaki et al.<sup>[102]</sup> reported W-doped VO<sub>2</sub> coated SiO<sub>2</sub> nanoparticles with mono-dispersion fabricated by the chemical solution deposition method. The resulting particle size was about 40 nm. The transition temperature of non-doped VO<sub>2</sub> coated SiO<sub>2</sub> particles was 68°C, and the transition temperature of 1.1 atomic % W-doped VO<sub>2</sub> coated SiO<sub>2</sub> particles was 52°C. For instance, noble metals<sup>[105][106]</sup> such as Au and Ag can be combined with VO<sub>2</sub> films resulting in high absorption but low reflectance. Kana Kana et al.<sup>[107]</sup> reported on the preparation of highly nanocrystalline Au–VO<sub>2</sub> composite thin films by RF-inverted cylindrical magnetron sputtering. The plasmonic properties and thermochromic behavior of the synthesized nanocomposite films of Au–VO<sub>2</sub> by ICMS are comparable to those seen previously with Au–VO<sub>2</sub> thin films grown by laser ablation<sup>[53]</sup> or chemical vapor deposition.<sup>[108]</sup> Kang et al.<sup>[109]</sup> reported a study that involves the optical properties of Pt/VO<sub>2</sub> composite films, with a Pt top layer and a VO<sub>2</sub> bottom layer. It was shown that the deposition of Pt layers substantially enhanced the infrared reflection of the film while retaining its thermochromic properties. At the same time, the transition temperatures of the heating semi-cycles were reduced by 9.3°C after a 9 nm Pt coating. The visible transmittance, which was diminished by the Pt layers, was effectively boosted by the addition of SiO<sub>2</sub> antireflection coatings. These results suggest that a VO<sub>2</sub> film with additional layers or structure design is a promising candidate for new smart windows. Li et al.<sup>[110]</sup> reported the preparation of nanofibers of VO<sub>2</sub>(A) with 100 nm diameter and 10-20 μm length via a one-step hydrothermal method. The obtained VO<sub>2</sub>(A) was coated by SiO<sub>2</sub> to form VO<sub>2</sub>(A)@SiO<sub>2</sub> core-shell nanocomposites, which were then transformed into VO<sub>2</sub>(M)@SiO<sub>2</sub> by annealing under nitrogen atmosphere. The resulted composites maintained the original fibrous morphology, particularly with a large amount of pores emerging inside the fiber due to the volume shrinkage during the phase transition, which may improve its thermal insulation ability in real applications. The VO<sub>2</sub>(M)@SiO<sub>2</sub> nanofibers were arranged into a self-supporting film by filtration, which showed excellent thermochromic properties. The transmittance modulation at 2500 nm was more than 30%, indicating an excellent thermochromic property of the fibers. Furthermore, the film showed a relatively high luminous transmittance over 50% probably due to the fibrous structure with small fiber radius and inter fiber spaces for light transmission. Interestingly, the film did not show the intrinsic yellow color of VO<sub>2</sub>(M), rather being grayish because of a structure effect, which will be beneficial for visual perception.

These results enrich the fact that doped-nanoparticles or composite-nanoparticles embedded in a polymer matrix can combine improved optical properties (narrow hysteresis, high luminous transmittance and high solar

modulation), convenient process deposition according to polymer coatings technologies and low transition temperature.

### 3. Synthesis techniques for preparing VO<sub>2</sub> (nano) particles

As previously described, the final properties of VO<sub>2</sub>(M) are strongly related to the sample quality, stoichiometry, purity, crystallinity and morphology. Consequently, the development of various synthetic strategies for the synthesis of VO<sub>2</sub>(M) with improved thermochromic properties has attracted a great attention these last two decades. It is worth noticing that the high temperature used during the reported synthesis leads to the direct formation of VO<sub>2</sub>(R) which is spontaneously converted through the fully reversible transition into VO<sub>2</sub>(M) at room temperature. VO<sub>2</sub>(M) is thus mainly formed from the rutile VO<sub>2</sub> thanks to the fully reversible MIT of VO<sub>2</sub>(R). As outlined above, both physical and chemical deposition methods have been reported to prepare VO<sub>2</sub>-based coatings including sputtering, pulsed laser deposition, chemical vapor deposition and sol-gel processes. These deposition methods leading to VO<sub>2</sub> film preparation have been extensively reviewed<sup>[111][31][112][113][30]</sup> and will not be described here. Keeping in mind that the main idea of this paper is to focus on VO<sub>2</sub> nanocomposites for thermochromic applications, we will mainly review in this section the chemical methods allowing the elaboration of VO<sub>2</sub>(M)/VO<sub>2</sub>(R) particles and/or nanoparticles, suitable to be further dispersed within a polymer matrix without significant aggregation. Interestingly, as previously underlined, decreasing the VO<sub>2</sub> particle size can not only facilitate material processing but can also affect in a positive way the phase transition temperature and hysteresis properties.<sup>[114][60]</sup> Controlling the thermochromic properties of VO<sub>2</sub> nanocrystals by playing on the synthetic conditions is clearly a challenging task.

One possibility to decrease the VO<sub>2</sub>(M) particle size and produce particles easy to disperse into a matrix involved mechanical ball-milling.<sup>[115][116][75]</sup> Liu et al.<sup>[115]</sup> described the preparation of VO<sub>2</sub>(M) nanoparticles with an average size of 20 nm simply by milling a micro-sized commercial VO<sub>2</sub>(M) dispersed in an organic solvent containing a dispersing agent for 4h. In parallel, the possibility of obtaining nanoparticles directly during the VO<sub>2</sub>(M) structure formation has also been deeply investigated. Two main synthetic approaches can be distinguished in the literature involving either direct synthesis or multistep processing in which the last step is usually based on annealing of vanadium-based material precursors. Obviously, many experimental parameters including the reaction time, the temperature, the annealing conditions the pH, the filling volumetric percentage, and the nature or morphology of the precursors play a key role on the structure and purity of the vanadium oxide formed as well as on its morphology and optical properties. These different factors are reviewed in the following paragraphs.

#### 3.1. Direct synthesis of VO<sub>2</sub>(M)/VO<sub>2</sub>(R) in solution

Direct synthetic protocols in solution are particularly attractive being simpler, with lower costs and allowing direct functionalization compared to methods involving solid-state transformations and annealing processes. However even if the rutile structure of VO<sub>2</sub> is the most stable phase from the thermodynamic point of view, only a few of one-step strategies in solution have been reported to successfully prepare VO<sub>2</sub>(R) nanomaterials. Clearly, stabilization of

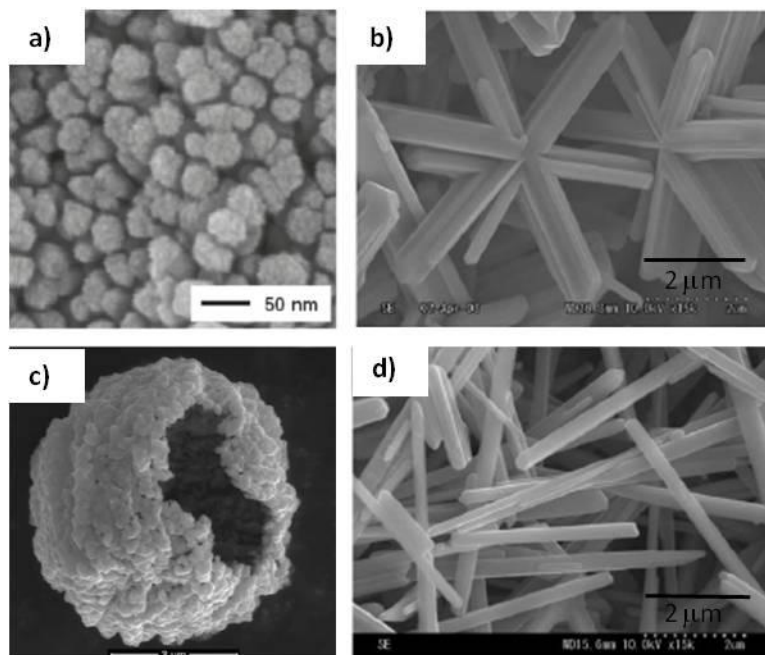
the VO<sub>2</sub>(M) phase directly from molecular precursors or solid phases in solution is challenging. This can be explained by the high valence state of vanadium (V<sup>4+</sup>, V<sup>5+</sup>) and its tendency to form condensed polyvanadates rather than detached VO<sub>6</sub> octahedra critical for rutile structure formation.<sup>[117]</sup> However, it is still possible to directly produce pure nanostructured VO<sub>2</sub>(M) in a cost effective and scalable way thanks to a fine tuning of the synthesis parameters and the nature of the precursors.

Table 1 sums up some of the direct solution protocols described in the literature. Systematically, a careful control of the synthetic parameters including the nature of the precursors, the heating method, the temperature, the reaction time, the type of reductant, the pH and the presence of further additives is required to obtain VO<sub>2</sub>(M) directly in solution. As described previously, the addition of doping element can induce distortion into the structure in favor of the stabilization of the rutile phase playing then a key role in the polymorph transformation facilitating the one-step preparation of VO<sub>2</sub>(M) by solution based approach. Numerous studies are still under progress to define an easy and reproducible way to form the thermochromic VO<sub>2</sub>(M) phase from precursors by solution processing.

The hydrothermal method has been efficiently applied for the direct formation of VO<sub>2</sub>(M) even if most of hydrothermal syntheses involving vanadium materials yield particles of metastable VO<sub>2</sub>(B) or other vanadium oxides phases rather than VO<sub>2</sub>(M).<sup>[118][29]</sup> Vanadium sources such as V<sub>2</sub>O<sub>5</sub>, V<sub>2</sub>O<sub>4</sub>, VOSO<sub>4</sub> and NH<sub>4</sub>VO<sub>3</sub> have been used as precursors.

Starting from commercial V<sub>2</sub>O<sub>4</sub>, VO<sub>2</sub>(M) nanowires were synthesized by hydrothermal treatment in the presence of aliphatic alcohols such as methanol, 2-propanol and 1,3-butanediol through a hydration/cleavage/exfoliation of the bulk precursor. The phase transition temperature could be decreased down to 32°C for the shorter nanowires obtained by the addition of 1,3-butanediol after 7 days at 210°C evidencing the interest of controlling VO<sub>2</sub>(M) particle shape and size.<sup>[119]</sup> VO<sub>2</sub>(M) nanorods were also prepared using V(OH)<sub>2</sub>NH<sub>2</sub> as V<sup>3+</sup> precursor. Such low valence appeared suitable compared to high valence states (V<sup>4+</sup>/V<sup>5+</sup>) to promote the formation of VO<sub>2</sub>(M) at quite low temperatures (200°C) in dilute HNO<sub>3</sub> solution providing suitable acidic conditions in favor of both detached VO<sub>6</sub> formation and weak oxidation conditions.<sup>[120]</sup>

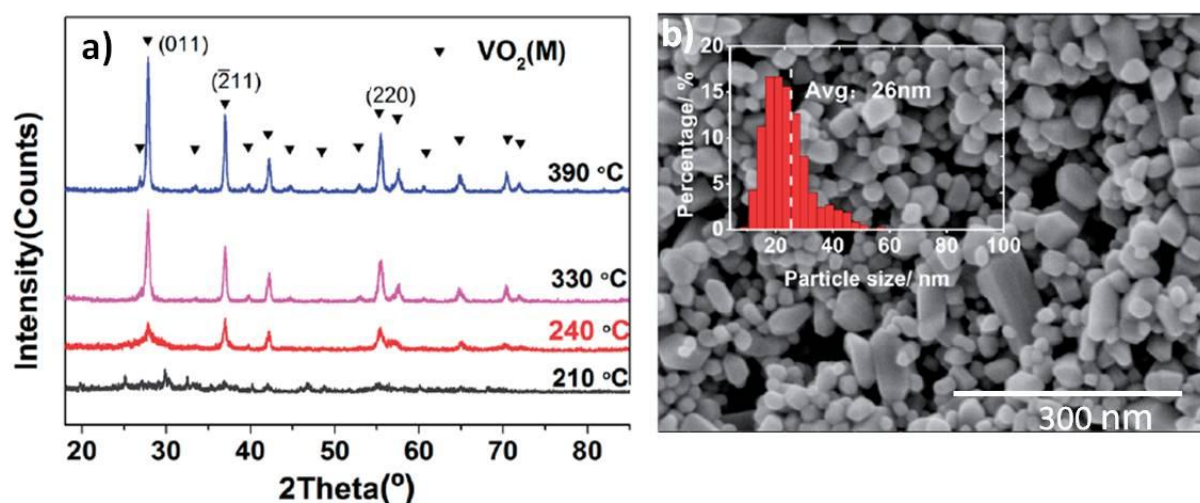
V<sub>2</sub>O<sub>5</sub> was also involved in VO<sub>2</sub>(M) one-step solution synthesis by hydrothermal reaction. In general, V<sub>2</sub>O<sub>5</sub> is reduced in a first step in the presence of oxalic acid or hydrazine, leading to a change of the vanadium oxidation state from V<sup>5+</sup> to V<sup>4+</sup>. Son et al.<sup>[46]</sup> thoroughly studied the influence of precipitation conditions such as pH and reactant concentration on the direct hydrothermal synthesis of VO<sub>2</sub>(M) starting from V<sub>2</sub>O<sub>5</sub> as vanadium precursor and N<sub>2</sub>H<sub>4</sub> as reducing agent. Interestingly, varying the reaction conditions controlled the size, the shape and the purity of the resulting VO<sub>2</sub> particles. For instance an increase of pH tuned the morphology from asterisk-shaped microcrystals to nanocrystals (Figure 4) as did an increase of temperature.<sup>[46]</sup>



**Figure 4. SEM images of various  $\text{VO}_2(\text{M})$  morphologies prepared by hydrothermal synthesis starting from  $\text{V}_2\text{O}_5$ . Adapted with permission from a) ref.,<sup>[46]</sup> b, d) ref.<sup>[121]</sup> and d) ref.<sup>[122]</sup> Copyright 2010, 2014, American Chemical society and Elsevier.**

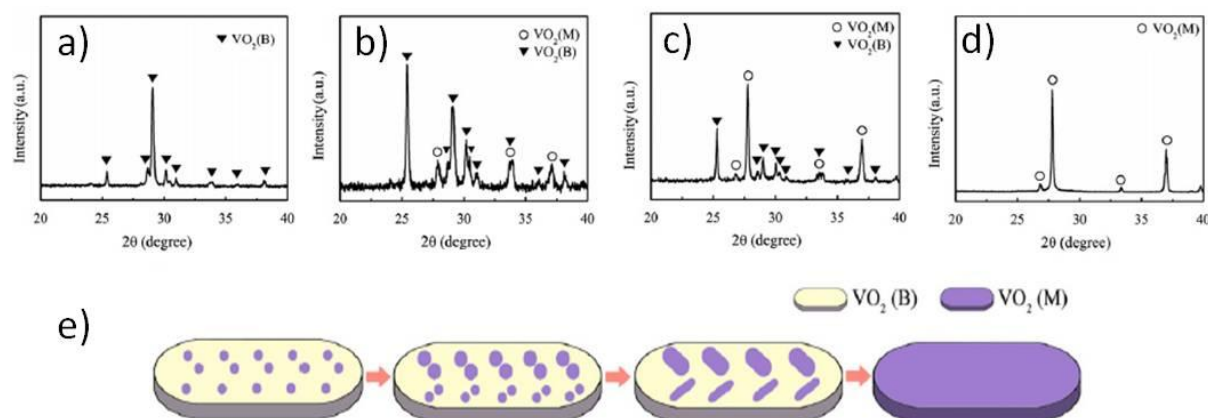
Using a similar synthetic approach that used  $\text{V}_2\text{O}_5$  and  $\text{N}_2\text{H}_4$  as precursors, a clear solution of  $\text{VO}^{2+}$  was prepared in acidic solution, and further submitted to an hydrothermal treatment.<sup>[19]</sup> Such a treatment induced the decomposition of the molecular precursors. As illustrated in Figure 5, X-Ray diffraction (XRD) patterns evidenced crystal structure transformation with the formation of  $\text{VO}_2(\text{M})$  nanoparticles above  $240^\circ\text{C}$  with particle sizes ranging from 25 to 45 nm depending on the hydrothermal temperature applied.<sup>[26]</sup>

Interestingly in the presence of  $\text{H}_2\text{O}_2$ ,  $\text{V}_2\text{O}_5$  is able to form a gel,<sup>[123]</sup> which improved  $\text{V}_2\text{O}_5$  dissolution and its subsequent full reduction.<sup>[124]</sup> Vanadium gels were reported as suitable vanadium precursors to react with  $\text{N}_2\text{H}_4$  acting as a strong reductant in basic conditions and producing pure single phase  $\text{VO}_2(\text{M})$  nanoparticles. In such approach the control of the  $\text{V}_2\text{O}_5/\text{N}_2\text{H}_4$  molar ratio appears essential to prepare  $\text{VO}_2(\text{M})$  nanoparticles.<sup>[125]</sup>



**Figure 5. a) XRD patterns of the particles prepared at different temperature from  $V_2O_5$ ,  $N_2H_4$  and HCl and b) SEM images of  $VO_2(M)$  nanoparticles prepared by hydrothermal synthesis at  $330^\circ C$ . Adapted with permission from ref.<sup>[26]</sup> Copyright 2014, The Royal Society of Chemistry.**

Oxalic acid was also reported to be an efficient reducing agent for the synthesis of  $VO_2(M)$  starting from  $V_2O_5$  through the formation of  $VOC_2O_4$  as an intermediate phase.<sup>[121][122][126][127][128][125][129]</sup> Again,  $VO_2(M)$  formation appeared to be very sensitive to the synthesis parameters. Ji et al.<sup>[121]</sup> reported that for low hydrothermal treatment temperatures, the  $VO_2(B)$  polymorph was preferentially formed, while increasing the temperature to  $260^\circ C$  induced a phase transition from  $VO_2(B)$  to  $VO_2(R)$ . Longer reaction times had a similar effect as increasing the temperature. Such phase transition from  $VO_2(B)$  to  $VO_2(R)$  nanorods under hydrothermal treatment at  $260^\circ C$  was further explored.<sup>[126]</sup> Figure 6 shows the XRD patterns of  $VO_2$  samples prepared for increasing reaction times from 4h to 20h, revealing the progressive transformation of  $VO_2(B)$  to  $VO_2(R)$  with increasing the hydrothermal reaction time. This evidenced a mechanism based on nanobumps nucleation and growth on the surface of  $VO_2(B)$  until complete conversion to  $VO_2(M)$  as the reaction time increases.<sup>[126]</sup>

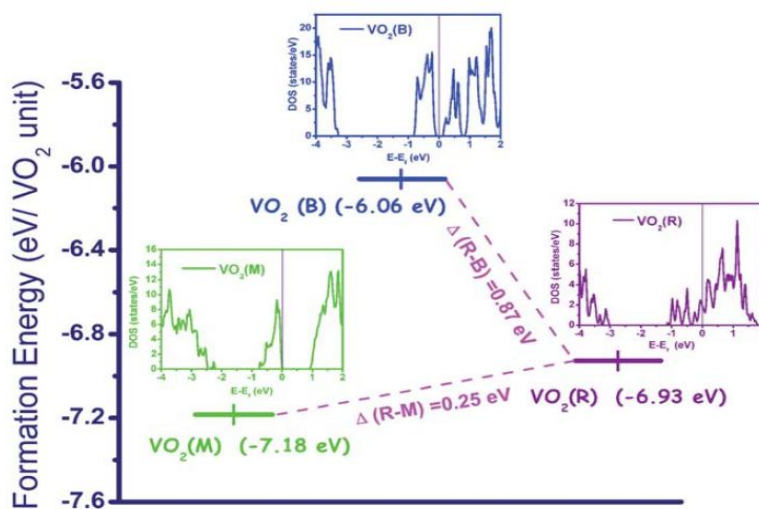


**Figure 6. XRD patterns of  $VO_2$  samples prepared by hydrothermal treatment at  $260^\circ C$  for a) 2h, b) 16h, c) 18h and d) 20h and e) scheme of the mechanism. Adapted with permission from ref.<sup>[126]</sup> Copyright 2011, Elsevier.**

It is also of significance to note that the presence of additives into the reaction medium allowed controlling the structure and morphology of the final particles.<sup>[121][129]</sup>  $H_2SO_4$  addition promotes modification of zeta potential value and crystallization/dissolution equilibrium of  $VO_2(R)$ . While  $VO_2(R)$  asterisk-shaped microcrystal were formed in absence of  $H_2SO_4$  (pH 3.5), a decrease of pH to 1.8 by the addition of  $H_2SO_4$  promoted nanorods formation (Figure 4).<sup>[121]</sup> The importance of controlling the  $NH_3$  addition to favor the formation of the  $VO_2(M)$  polymorph was highlighted by Dong et al.<sup>[130]</sup> who described an hydrothermal method involving an amorphous precursor  $[VO(OH)_2(OH_2)_3]$ . The presence of  $NH_4^+$  influenced the synthesis of pure  $VO_2(M)$  allowing the formation of a  $(NH_4)_2V_4O_9$  intermediate rather than the direct crystallization of  $VO_2(B)$ . Polyvinylpyrrolidone in the reaction medium containing  $V_2O_5$  and oxalate can act as a soft template to promote  $VO_2(R)$  hollow microsphere preparation based on nanorod self-assembly through a single-step assisted hydrothermal process.<sup>[129]</sup>



$\text{VOSO}_4$  was also involved as vanadium precursor in hydrothermal reactions to directly produce  $\text{VO}_2(\text{M})$  in the presence of hydrazine as reducing agent and urea as pH controller. In these conditions, the higher was the vanadium precursor concentration, the lower was the particle sizes.<sup>[131]</sup> By playing on the filling ratio during the hydrothermal treatment, Ji et al.<sup>[128]</sup> confirmed that the phase transition from  $\text{VO}_2(\text{A})$  to  $\text{VO}_2(\text{R})$  was pressure sensitive as suggested by the calculation of the ground state energies for  $\text{VO}_2(\text{A})$ ,  $\text{VO}_2(\text{B})$  and  $\text{VO}_2(\text{R})$  (Figure 7).



**Figure 7. The calculated formation energy of different vanadium oxides according to the hybrid density functional theory. Adapted with permission from ref.<sup>[120]</sup> Copyright 2011, The Royal Society of Chemistry.**

Alternatively,  $\text{VO}_2(\text{M})$  nanocrystals were also prepared using a solvothermal method involving  $\text{V}^{5+}$  diperoxo alkylammonium complexes as vanadium precursors in the presence of aliphatic amine acting as capping agents to limit the crystal growth. During solvothermal treatment at  $180^\circ\text{C}$  in organic medium, the complex precursors decomposed generating vanadium oxide nuclei. Figure 8 shows the TEM images of the obtained nanocrystals with various morphologies synthesized from vanadium diperoxo tetraoctylammonium complexes in oleylamine at  $180^\circ\text{C}$  for 5h with increasing water/toluene volume ratios from 0/40 to 20/40.<sup>[132]</sup> At low water content, monodisperse particles were formed in the dispersed water pools of a water/toluene microemulsion while an increase in water content induced a water/toluene phase separation and nanorods and aligned nanorods formation at the water/toluene interface. The SAED pattern revealed single-crystalline nanorods with a preferential elongated axis along the  $[101]$  direction. Varying the reaction conditions such as water/toluene ratio, stearic ligands of complexes, alkyl chain length of capping amines and nature of solvent could therefore control the shape and size of the nanocrystals from nanospheres to nanoplatelets, nanocubes and nanorods.

Another alternative synthetic strategy to the hydrothermal reaction has consisted in a direct combustion in a confined space of vanadium oxide acetylacetonate precursors  $\text{VO}(\text{acac})_2$  dispersed in ethanol  $\text{VO}_2(\text{M})$  leading to the formation of pure  $\text{VO}_2(\text{M})$ . Such approach provided in a short time suitable thermal energy to induce  $\text{VO}_2(\text{M})$  formation and reducing/inert atmosphere to avoid further  $\text{VO}_2$  oxidation.<sup>[133]</sup> In a similar approach,  $\text{VO}_2(\text{M})$  was prepared in a short time by the combustion of  $\text{NH}_4\text{VO}_3$  dissolved in a mixture of ethylene glycol and EtOH in air. The mechanism of formation involves the formation of a vanadyl ethylene glycolate intermediate whose complete pyrolysis strongly depends on both the volume involved and the curvature of the container.<sup>[134]</sup>

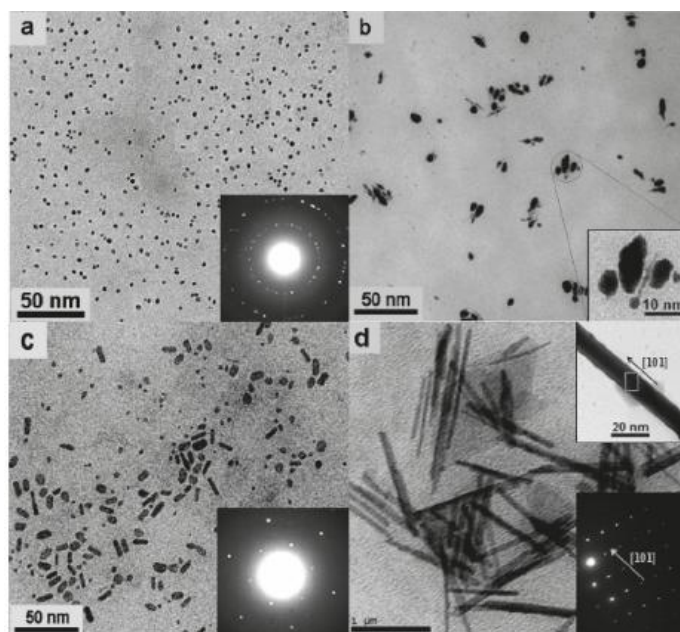


Figure 8. Effect of the increase of water content on the morphology of  $\text{VO}_2(\text{M})$  nanocrystals obtained from  $\text{VO}(\text{O}_2)_2(\text{TOA})$  in oleylamine at  $180^\circ\text{C}$  for 5h. Adapted with permission from ref.<sup>[132]</sup> Copyright 2009, American Chemical Society.

**Table 1. Experimental conditions reported for direct solution-based synthesis of VO<sub>2</sub>(M)**

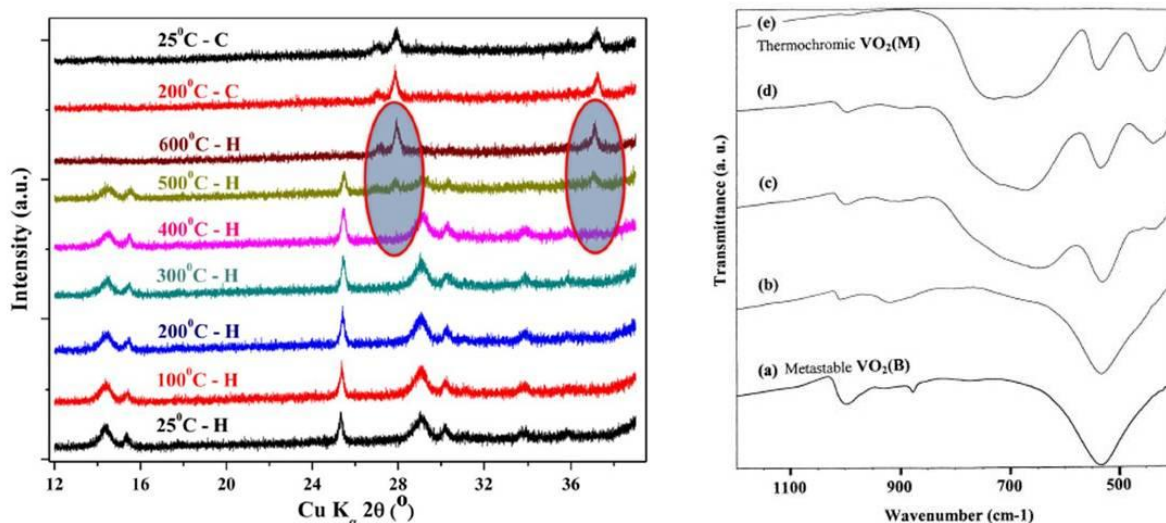
| Ref.       | Vanadium precursor  | Additives and conditions   | Morphology , Size  |
|------------|---|--|--|
| [119]      | V <sub>2</sub> O <sub>4</sub> (commercial)  | Water, 1,3-butanediol, hydrothermal, 210°C, 7 days   | Nanowires (< 10µm in length and 160 nm in diameter)                                    |
| [46]       | V <sub>2</sub> O <sub>5</sub>   | H <sub>2</sub> SO <sub>4</sub> , hydrazine, NaOH, hydrothermal, 220°C, 48h                 | Asterisk-shaped microcrystals (1-20 µm); elongated microrods; nanocrystals (20-100 nm) |
| [121]      | V <sub>2</sub> O <sub>5</sub>   | Oxalic acid, H <sub>2</sub> SO <sub>4</sub> , hydrothermal, 240-260°C, 16h-48h             | Asterisk-shaped microcrystals; nanorods (several µm x 200 nm)                          |
| [130]      | V <sub>2</sub> O <sub>5</sub>   | Oxalic acid, NaOH, NH <sub>3</sub> , hydrothermal, 280°C                                   | Nanorods (150 nm); nanoparticles (50 nm)   |
| [135]      | V <sub>2</sub> O <sub>5</sub>   | Oxalic acid, PVP, hydrothermal, 300°C, 0-6h  | Microspheres   |
| [125]      | V <sub>2</sub> O <sub>5</sub>   | H <sub>2</sub> O <sub>2</sub> , N <sub>2</sub> H <sub>4</sub> , hydrothermal, 260°C, 24h   | Nanoparticles (< 50 nm)  |
| [26]       | V <sub>2</sub> O <sub>5</sub>   | N <sub>2</sub> H <sub>4</sub> , HCl, hydrothermal, 300-390°C, 15h                          | Nanoparticles (24-45 nm)   |
| [136][137] | NH <sub>4</sub> VO <sub>3</sub>   | N <sub>2</sub> H <sub>4</sub> , HCl, hydrothermal, 280°C, 18h                              | Nanoparticles (30-80 nm)   |
| [95]       | VOSO <sub>4</sub>   | N <sub>2</sub> H <sub>4</sub> , NaOH, hydrothermal, 240°C, 36h                             | Nanoparticles (19 nm)  |
| [131]      | VOSO <sub>4</sub>   | N <sub>2</sub> H <sub>4</sub> CO(NH <sub>2</sub> ) <sub>2</sub> , hydrothermal, 260°C, 24h | Nanoparticles (20-50 nm); nanorods (500 x 100 nm)                                      |
| [132]      | VO(O <sub>2</sub> ) <sub>2</sub> (CTA)VO(O <sub>2</sub> ) <sub>2</sub><br>(TPA)VO(O <sub>2</sub> ) <sub>2</sub> (TOA) | Toluene, oleylamine, hydrothermal, 180°C, 5h   | Spherical particles (4 nm); nanorods (300 x 20 nm)                                     |
| [120]      | V(OH) <sub>2</sub> NH <sub>2</sub>  | HNO <sub>3</sub> aqueous solution, hydrothermal 200°C, 36h                                 | Nanorods (400-800 nm in length and 30-120 nm in diameter)                              |
| [138]      | NH <sub>4</sub> VO <sub>3</sub>   | N <sub>2</sub> H <sub>4</sub> , 210°C, hydrothermal, 24-72h                                | Nanoparticles (44 ± 30 nm)   |
| [134]      | NH <sub>4</sub> VO <sub>3</sub>   | Ethylene glycol, EtOH, combustion  | Spherical particles (100-300 nm)   |
| [133]      | VO(acac) <sub>2</sub>   | EtOH, combustion (648-930°C)   | Film   |

PVP : Polyvinylpyrrolidone ; CTA : cetyl trimethylammonium ; TOA : tetraoctylammonium ; TPA : tetrapropylammonium

### 3.2. Solid-state conversion of vanadium oxides to prepare VO<sub>2</sub>(M)/VO<sub>2</sub>(R) particles

Apart from the direct solution-based process, a widely used approach to obtain VO<sub>2</sub>(M) consists in a post-thermal treatment of a precursor phase, typically other vanadium dioxide polymorphs or vanadyl complexes. In a preliminary step, the vanadium precursor phase is prepared using appropriate synthetic methods such as hydrothermal synthesis,<sup>[89]</sup> continuous hydrothermal flow synthesis,<sup>[139]</sup> the sol-gel process<sup>[111], [30]</sup> or the polyol method.<sup>[140]</sup> Then, the vanadium-based precursor is converted into VO<sub>2</sub>(M) by annealing. VO<sub>2</sub>(M) formation is found to be strongly dependent on the annealing parameters that highly impact the phase transition. By heating vanadium oxide precursors, the vanadium atoms displacement within the lattice is promoted leading to the desired VO<sub>2</sub>(M)/VO<sub>2</sub>(R) by taking advantage of the structural similarity. Since vanadium atoms cover different valence states ranging from +2 to +5, VO<sub>2</sub> in a valence state 4+ appears very sensitive to oxydo-reduction reactions. That is why during the thermal treatment or annealing, a special attention should be paid to the nature and the flow rate of the gas atmosphere used. VO<sub>2</sub>(B), VO<sub>2</sub>(A), paramontroseite, nsutite-type VO<sub>2</sub>, VO<sub>2</sub>(D) or VOOH were successfully involved as precursors in solid state conversion upon thermal treatment to obtain VO<sub>2</sub>(M). Some of the annealing conditions for these different precursor phases are reported in Table 2. Depending on the nature of the metastable precursors, the annealing temperature could be varied from 250°C to 550°C<sup>[141]</sup> which influenced the particle morphology.

VO<sub>2</sub>(B) has been widely used as precursor to produce VO<sub>2</sub>(R)/(M). For instance, Popuri et al.<sup>[142]</sup> evidenced by *in situ* high temperature XRD (Figure 9a) the full transformation of VO<sub>2</sub>(B) prepared by hydrothermal synthesis to VO<sub>2</sub>(M) above 550°C under vacuum, preventing the VO<sub>2</sub> unstability under oxidizing conditions.<sup>[143]</sup> Similar results were obtained using FTIR (Figure 9b) and *in situ* transmission electron microscopy<sup>[144]</sup> confirming a complex mechanism of transformation involving an intermediate disordered phase with a reduced symmetry. On the FTIR spectra, the characteristic VO<sub>2</sub>(M) double band located at 680-730 cm<sup>-1</sup> appeared above 540°C. The study of the transition behavior of VO<sub>2</sub>(B) nanorods evidenced that the low MIT transition temperature observed after annealing at a lower temperature and shorter time was related to the presence of VO<sub>2</sub>(M) grains in the strain environment of VO<sub>2</sub>(B) nanorods rather than a size effect.<sup>[145]</sup>



**Figure 9. Left : *In situ*-XRPD from 25°C to 600°C (Heating cycle) and 600°C to 25°C (cooling cycle) and right FTIR spectra with temperature of VO<sub>2</sub>(B) calcined at (a) 380°C for 5 min, (b) 490°C for 5 min, (c) 540°C for 5 min, (d) 600°C for 5 min and (e) 600°C for 30 min. Adapted with permission from ref.<sup>[142][146]</sup> 2012, 1998, American Chemical Society, Elsevier .**

Li et al.<sup>[110]</sup> reported the conversion of VO<sub>2</sub>(A) nanofibers prepared by hydrothermally assisted reduction of NH<sub>4</sub>VO<sub>3</sub> in an acidic solution containing 1, 3- propyleneglycol into VO<sub>2</sub>(M). SiO<sub>2</sub> coating maintained the shape of the fibers during the annealing step limiting the strain and distortion during the structure conversion into VO<sub>2</sub>(M). Amorphous VO<sub>2</sub>@SiO<sub>2</sub> nanorods were also prepared using an elegant strategy based on microemulsion.<sup>[147]</sup> A short-time annealing at 800°C in an inert atmosphere led to well-crystallized VO<sub>2</sub>(M)@SiO<sub>2</sub> core-shell nanorods with lengths of about 200 nm. Interestingly the SiO<sub>2</sub> shell avoided sintering of VO<sub>2</sub> particles during the thermal treatment.

Interestingly, a VO<sub>2</sub>(D) metastable precursor was converted to VO<sub>2</sub>(M) in maintaining the precursor particle size through calcination at low temperature (i.e., 250°C),<sup>[148]</sup> due to the high structural resemblance between the two polymorphs. Utilizing VO<sub>2</sub>(D) nanoparticles thus revealed to be an efficient way to produce VO<sub>2</sub>(M) nanoparticles by annealing compared to the VO<sub>2</sub>(B) and VO<sub>2</sub>(A) polymorphs.<sup>[104][148][149][150]</sup>

An ultrafast transformation from VO<sub>2</sub>(P) to VO<sub>2</sub>(M) in air was also reported by Sun et al.<sup>[151]</sup> VO<sub>2</sub>(P) nanocrystals (10-30 nm) obtained by solvothermal method using VO(OC<sub>3</sub>H<sub>7</sub>) and oleylamine were easily transformed in VO<sub>2</sub>(M) thanks to their structural resemblance. Extended X-ray absorption fine structure spectroscopy (XAFS) measurements conjointly with XRD, XPS and TEM (Figure 10) confirmed that the product obtained after annealing at 400°C was composed of high-quality VO<sub>2</sub>(M) nanocrystals, which have maintained the same morphology and particle size upon annealing. These single domain nanocrystals allowed getting better insight into the size dependent MIT behavior thanks to *in situ* variable temperature IR spectroscopy experiments. Conversion of VO<sub>2</sub>(P) nanocrystal prepared by hydrothermal synthesis in the presence of the surfactant: bis(2-ethylhexyl) sulfosuccinate sodium salt (AOT), could be easily performed by mild annealing treatment producing VO<sub>2</sub>(M) nanoparticles with diameters in the range of 59 to 89 nm with no size and morphological changes.<sup>[152]</sup>

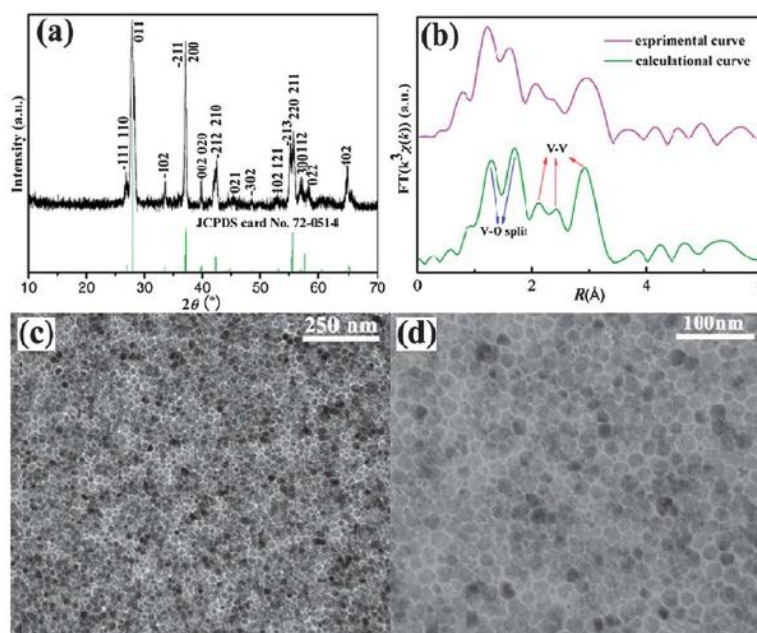


Figure 10. a) XRD pattern, b) vanadium K-edge radial distribution functions, and c) and d) TEM images at low and high magnification of VO<sub>2</sub>(M) nanocrystals. Adapted with permission from ref.<sup>[151]</sup> Copyright 2015, Nature publishing group.

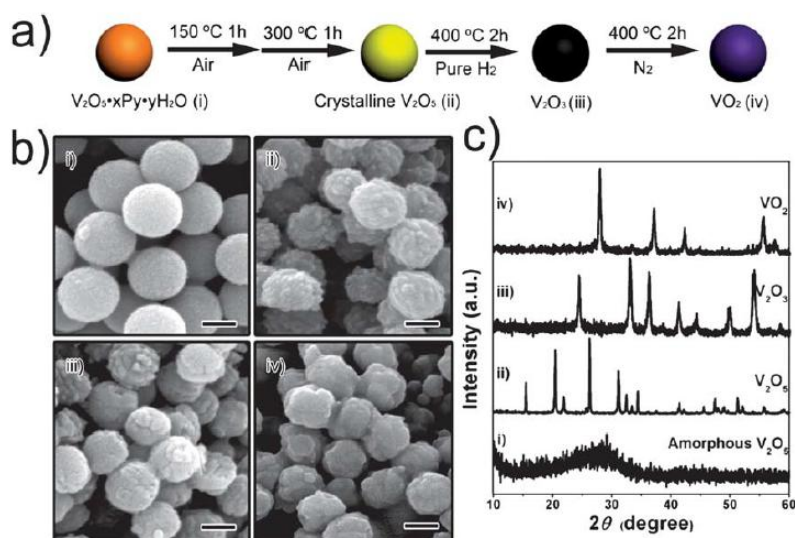
Table 2. Experimental conditions reported to convert vanadium oxide phases into VO<sub>2</sub>(M) by thermal treatment or annealing.

| Ref.  | Vanadium precursor                        | T (°C)    | Time (h) | Atmosphere                            | Size, Morphology  |
|-------|---|-----------|----------|---------------------------------------|---|
| [142] | VO <sub>2</sub> (B)                       | 550       | 1        | Vacuum                                | Platelets   |
| [145] | VO <sub>2</sub> (B)                       | 350-550   | 1        | N <sub>2</sub>                        | Nanorods (several hundreds x 40-90                        |
| [110] | VO <sub>2</sub> (A)@SiO <sub>2</sub>      | 700       | 1        | N <sub>2</sub>                        | Nanofibers  |
| [153] | VO <sub>2</sub> (D)                       | 165       | -        | N <sub>2</sub>                        | Nanoparticles (20-30 nm)                                  |
| [148] | VO <sub>2</sub> (D)                       | 300-450   | 1        | Vacuum                                | Nanoparticles (74-109 nm)                                 |
| [149] | VO <sub>2</sub> (D)                       | 250-550   | 3        | Vacuum                                | Nanoaggregates/ Cucumber-like                             |
| [150] | VO <sub>2</sub> (D)                       | 300-450   | 1        | Vacuum                                | Star shaped particles                                     |
| [154] | VO <sub>2</sub> (D)                       | 320       | 2        | N <sub>2</sub>                        | Hierarchical structures (1-2μm)                           |
| [151] | Paramontroseite VO <sub>2</sub>           | 400       | 0.01     | Air                                   | Nanocrystals (10-30 nm)                                   |
| [155] | Paramontroseite VO <sub>2</sub>           | 400       | 0.0167   | N <sub>2</sub>                        | -   |
| [152] | Paramontroseite VO <sub>2</sub>           | 350       | 1        | Vacuum                                | Nanoparticles (59-89 nm)                                  |
| [156] | Nsutite-type VO <sub>2</sub>              | 350       | 48       | Ar                                    | Hollow spheres (300-400 nm)<br>Shell thickness (40-70 nm) |
| [157] | V <sub>2</sub> O <sub>5</sub> . xPyridine | 150, 300, | 1 or 2   | Air, H <sub>2</sub> or N <sub>2</sub> | Spherical particles (200-600 nm)                          |

Starting from amorphous V<sub>2</sub>O<sub>5</sub> spherical particles obtained through vanadium isopropoxide hydrolysis, Lu et al.<sup>[157]</sup> reported the formation of VO<sub>2</sub>(M) spherical particles dispersible in solution through sequential annealing and reduction steps (Figure 11). After a reduction step in H<sub>2</sub>, an intermediate V<sub>2</sub>O<sub>3</sub> phase was formed and subsequently converted in pure VO<sub>2</sub>(M) during the last annealing step in N<sub>2</sub> through partial re-oxidation by trace of O<sub>2</sub> in nitrogen. The size of the particles could be easily tuned in the range of 200-600 nm by changing the hydrolysis conditions of the vanadium precursors.



Other amorphous ill-defined vanadium based oxides ( $\text{VO}_x \cdot \text{H}_2\text{O}$ ) allowed to prepare  $\text{VO}_2(\text{M})$  by annealing.<sup>[158][159][160]</sup> For instance, amorphous  $\text{VO}_2$  colloids were electrochemically prepared by anodic oxidation of metallic V,<sup>[56]</sup> producing by fast (5s) annealing at 310°C spherical particles of  $\text{VO}_2(\text{M})$  with a size of about 50 nm.



**Figure 11. Synthetic scheme for the  $\text{VO}_2(\text{M})$ , b) SEM images of i) amorphous  $\text{V}_2\text{O}_5 \cdot x\text{Py} \cdot y\text{H}_2\text{O}$ , ii) crystalline  $\text{V}_2\text{O}_5$ , iii)  $\text{V}_2\text{O}_3$  and iv)  $\text{VO}_2(\text{M})$  particles. Adapted with permission from ref. <sup>[157]</sup> Copyright 2011, The Royal Society of Chemistry.**

As an alternative to vanadium oxides, vanadyl ethylene glycolate obtained from ammonium meta-vanadate ( $\text{NH}_4\text{VO}_3$ ) and ethylene glycol mixture appeared as a suitable precursor to produce highly crystalline and pure  $\text{VO}_2(\text{M})$  particles by thermolysis.<sup>[140][161][162][163]</sup> The importance of controlling the annealing treatment under controlled atmosphere and temperature was reported by Mjejri et al.<sup>[112]</sup> Depending on the annealing conditions used, different valence states in vanadium oxides were obtained.  $\text{VO}_2(\text{M})$  particles were formed upon annealing for 2h at 500°C under vacuum while  $\text{V}_2\text{O}_3$  and  $\text{V}_2\text{O}_5$  were obtained under  $\text{Ar}/\text{H}_2$  (95:5) and air, respectively.

Obviously, the structure of the solid vanadium precursors and the annealing conditions are of great importance to form  $\text{VO}_2(\text{M})$ ; but the morphology of the solid precursors appeared to be another key parameter favoring vanadium precursor source conversion. Freeze drying before the annealing step reduced agglomeration of the particles and ensured an homogenous heating promoting the formation of pure  $\text{VO}_2(\text{M})$  nanoparticles of 50 nm.<sup>[124]</sup> Ran et al.<sup>[42]</sup> also illustrated the freeze drying effect on the formation of pure  $\text{VO}_2(\text{M})$  and on the morphology.

In summary, many synthetic strategies in solution involving various reactants were reported to prepare  $\text{VO}_2(\text{M})$  nanoparticles most often involving hydrothermal treatment of vanadium precursors. Obviously, each synthetic parameter plays a key role on the stabilization of the expected  $\text{VO}_2(\text{M})$  polymorph which imposes a strict control of the conditions to stabilize  $\text{VO}_2(\text{M})$  colloidal suspensions.

#### 4. $\text{VO}_2$ -based composite materials

As mentioned above, there are still big challenges associated to the formation of cost-effective VO<sub>2</sub> nanoparticles. In turn, big challenges also remain in incorporating such nanoparticles into coatings on large surfaces. Scientific works provide elaboration routes that can be divided into two classes: fully inorganic materials and hybrid systems composed of VO<sub>2</sub> particles embedded in a polymer matrix. Compared to VO<sub>2</sub>/inorganic composites, examples of elaboration of composite materials made of VO<sub>2</sub> particles incorporated into a polymer matrix are less numerous. According to Madida et al.<sup>[164]</sup>, the elaboration of stable and efficient composite films is challenging for several reasons. One first reason is that vanadium can adopt multiple oxidation states and is thus sensitive to oxygen. Indeed vanadium oxide can be progressively transformed into the more stable V<sub>2</sub>O<sub>5</sub>, which will inevitably affect the overall performances of the coating.<sup>[165][29]</sup> Besides, VO<sub>2</sub> particles are often on the micron scale in size and tend to aggregate after heat treatment. The coarse aggregates need therefore to be broken down. Indeed the presence of large particles can induce significant light scattering and affect the transparency of the coating with the generation of haze. Therefore, the polymer host has to be selected such as: i) to afford an efficient stabilization of the VO<sub>2</sub> particles to prevent their agglomeration during processing, ii) to ensure high environmental stability, and iii) to guarantee a transparent host. In general the polymer is chosen to be as transparent as possible for radiations close to wavelengths of around 1 μm. The VO<sub>2</sub>, on the other hand, can be of various origins (commercial powders or home-made particles). However, as previously mentioned, the thermochromic properties of VO<sub>2</sub> films strongly depend on the particle size and morphology. Producing high quality dispersions with small particles sizes, narrow size distribution and uniform morphology is therefore crucial to obtain highly performant composite films. Furthermore, to satisfy the current requirements in energy saving efficiency, notably in the architectural sector, and allow in particular the formation of VO<sub>2</sub>-based coatings on large surfaces, the deposition technique should be compatible with standard paint technology.

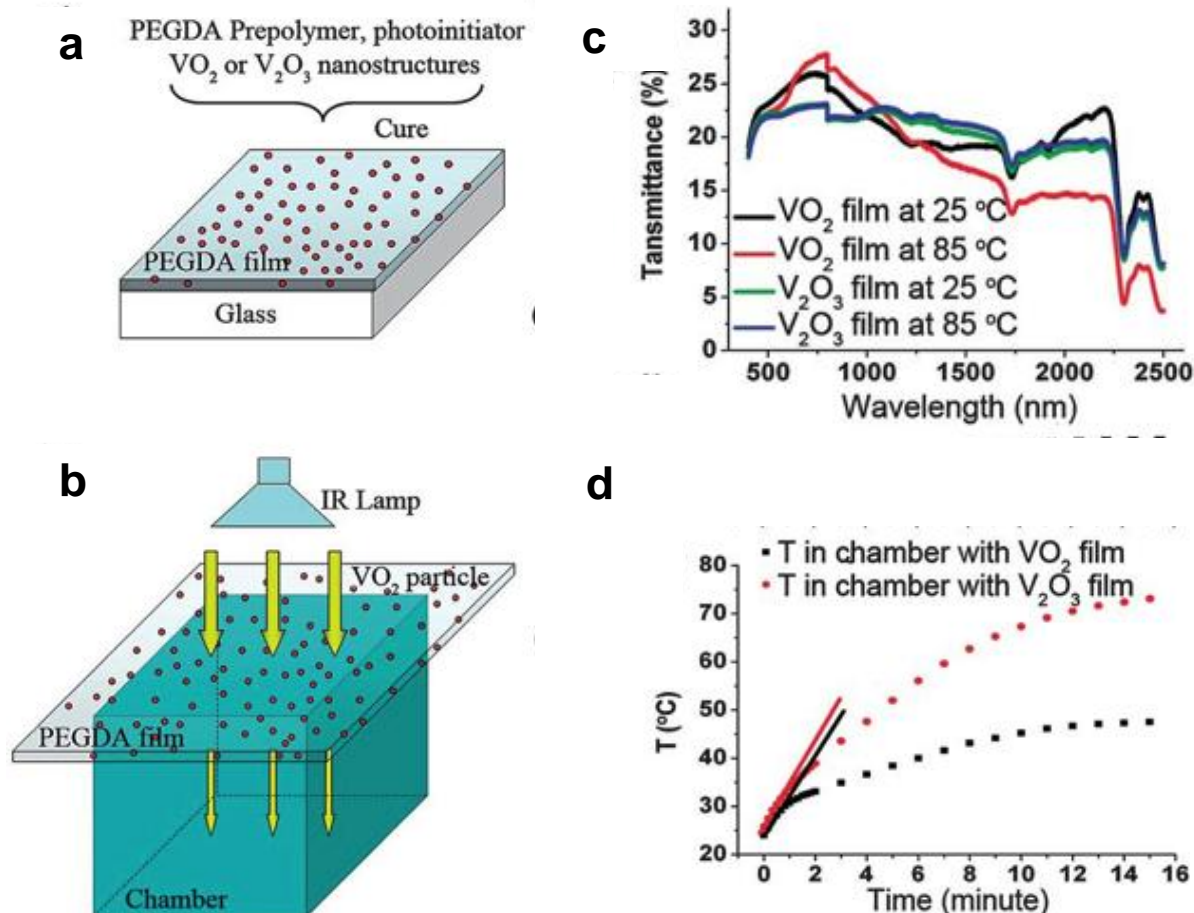
#### 4.1. Acrylic-based coatings

The first reports on the elaboration of VO<sub>2</sub>-based polymer composites were released in 1994 by Alfred-Duplan,<sup>[166]</sup> and Valmalette,<sup>[116]</sup> respectively. Alfred-Duplan et al.<sup>[166]</sup> reported the incorporation of VO<sub>2</sub> particles into low-density polyethylene (LPDE) by melt blending and studied the impact of the thermochromic particles on the electrical properties of the composite material. Complex impedance analyses below and above the insulator-to-metal transition temperature revealed a pronounced influence of the amount of VO<sub>2</sub> on conductivity. A percolation model was proposed to interpret the experimental results. Above the percolation threshold, the metal-insulator transition was effectively reflected in the impedance measurements. Actually, Valmalette and Gavarri were the very first ones to report on the incorporation of VO<sub>2</sub> particles into a polymer matrix for thermochromic applications.<sup>[116]</sup> To reduce the grain size, the VO<sub>2</sub> powder was mechanically grinded using wet ball milling in a mixture of isobutyl methacrylate and a standard volatile solvent with the help of a dispersing agent, and polymerized. However, the nature of the compatibilizing agent was not indicated in this article nor were the polymerization conditions. Instead, the article focused on the modeling of the optical transmittance of the coating below and above the phase transition



temperature as a function of the grain size and spatial distribution for increasing volume fractions of the VO<sub>2</sub> pigment (typically in the range 0.1-5%). It was shown that both the pigment distribution and its mean size played a critical role in the optical properties. Thus it is possible to determine the chemical preparation conditions for manufacturing and optimizing specific pigments for coatings presenting well-defined thermochromic behavior. It is only several years later that the effective performance of a thermochromic coating obtained by embedding W-doped VO<sub>2</sub> particles in an acrylic matrix, was reported by Shi et al.<sup>[75]</sup> The particles were prepared by solution precipitation using VOSO<sub>4</sub> as precursor in the presence of a stoichiometric volume of sodium tungstate. The resulting VO<sub>2</sub> particles, whose T<sub>c</sub> varied from 66°C to -4.1°C upon increasing the W atomic at% from 0 to 2.83%, were next suspended into xylene by grinding in a planetary ball mill, and the resulting dispersion was mixed with the acrylic resin (a styrene-acrylate copolymer with a T<sub>g</sub> of around 54°C) under vigorous stirring. The grinding process not only reduced the particle size, but also decreased the crystal size as attested by XRD. The obtained films displayed good thermochromic properties with a  $\Delta T_{sol}$  of 23% at 2500 nm and a maximum visible transmittance of 60% (see Table 3). A similar approach was reported by Xiao et al.<sup>[167][168]</sup> using tungsten-doped and undoped particles leading to composite films with similar performances. The un-doped films showed an IR transmission modulation of about 37.3% at 2000 nm (which is comparable with CVD films) but a low luminous transmittance ( $T_{lum} = 31.3\%$ ) due to the presence of the VO<sub>2</sub> particles that induced significant scattering and the formation of haze. Tungsten doping decreased both  $T_{lum}$  and  $\Delta T_{sol}$  (at 2000 nm) while the polymer matrix had no influence on the phase transition temperature of the VO<sub>2</sub> particles.

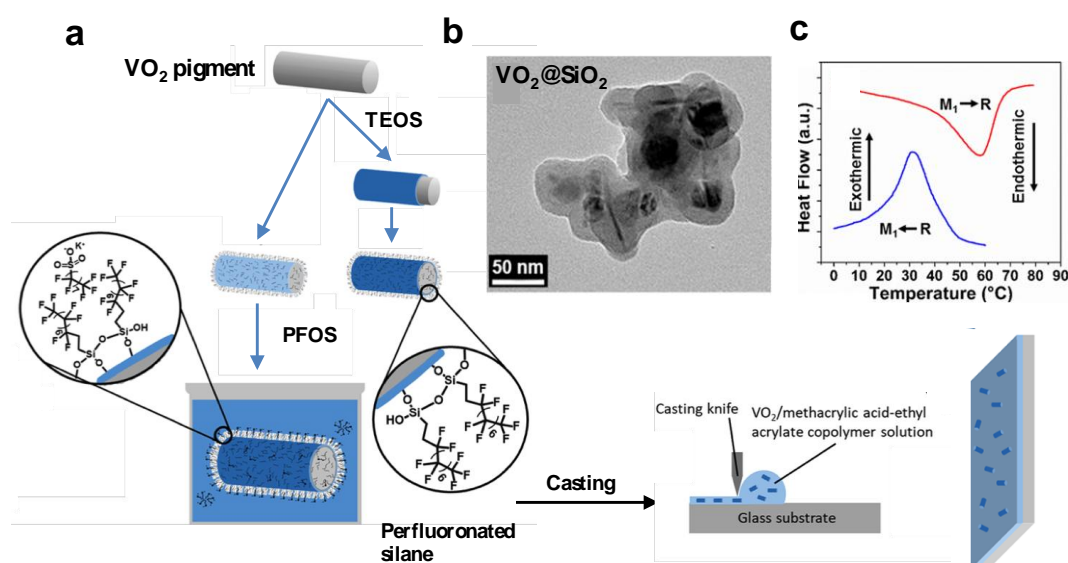
Although the properties reported in these preliminary papers were promising, the performances were not optimal and could be further improved if the grain size could be reduced. Indeed, according to Li et al.,<sup>[23]</sup> reducing the size of the VO<sub>2</sub> particles should lead to a lower transition temperature and to a wider hysteresis loop during the semiconductor-to-metal transition. Following this direction, Lu et al.<sup>[157]</sup> reported the synthesis of colloidal VO<sub>2</sub> particles and their incorporation into a thin plastic film. The particle size could be controlled from several tens to a few hundreds of nanometers, which had a significant influence on the optical properties of VO<sub>2</sub> and enabled the formation of stable suspensions. The trick to obtain such small particles is to use uniform V<sub>2</sub>O<sub>5</sub> colloidal particles as precursors followed by sequential annealing and reduction steps. VO<sub>2</sub>(M) with high purity and different particles sizes was obtained by controlling the size of the initial V<sub>2</sub>O<sub>5</sub> particles. The particles were next introduced in poly(ethylene glycol) diacrylate (PEGDA,  $M_w = 700 \text{ g mol}^{-1}$ ) at the concentration of  $10 \text{ g L}^{-1}$ , and the resulting mixture was deposited between two glass slides as a thin film with a thickness of several hundreds micrometers. The film was finally polymerized by UV-irradiation using dimethoxy-2-phenylacetophenone as photo-initiator (Figure 12a). To test the performance of the composite coating, the glass substrate was placed on top of an adiabatic chamber and irradiated under an IR lamp (Figure 12b). The thermochromic coating enabled to significantly delay the temperature increase of the chamber compared to a reference film containing V<sub>2</sub>O<sub>3</sub> particles: the temperature rose to only 45°C after 15 min compared to 75°C for the V<sub>2</sub>O<sub>3</sub> film (Figure 12d).



**Figure 12.** Illustration of the process used to form: a) a polymer film containing  $\text{VO}_2(\text{M})$  particles by photopolymerization and b) a prototype device to demonstrate the potential of the composite film to regulate the temperature of an adiabatic chamber. c) Transmission of the composite films at different temperatures compared to the same film containing  $\text{V}_2\text{O}_3$  particles and d) temperature inside the chamber for increasing irradiation times. Adapted with permission from ref.<sup>[157]</sup> Copyright 2011, The Royal Society of Chemistry.

Around the same time, He and co-workers<sup>[169]</sup> reported the thermochromic properties of  $\text{VO}_2$ -coated mica sheets embedded in a transparent UV-curing acrylic resin. The  $\text{VO}_2$ /mica pigment was dispersed in the resin and the mixture was spread on a glass substrate to afford a composite film after irradiation with a mercury lamp. The material displayed 35% change in IR transmittance at 2200 nm and a transmittance of around 50% in the visible range. As mentioned above, to minimize light scattering and guarantee a high transparency, it is essential to decrease the size of the inorganic pigment and ensure an adequate dispersion in the polymeric host matrix. Furthermore, the high dielectric constant thermochromic particles need to be refractive index matched with the matrix. These challenges have been addressed by Fleer et al.<sup>[138]</sup> in a recent paper reporting an entirely aqueous processing approach to the elaboration of nanocomposite films with embedded  $\text{VO}_2$  nanocrystals. The as prepared  $\text{VO}_2$  particles were first coated with a silica shell to reduce their degree of agglomeration and facilitate their dispersion in water. It was shown that  $\text{SiO}_2$  deposition did not affect the MIT temperature. The  $\text{VO}_2@\text{SiO}_2$  pigments were further functionalized with heptadecafluoro-1,1,2,2-tetrahydrodecyl - trimethoxysilane, and dispersed in water

with the help of perfluoro-octane sulfonate (PFOS) (Figure 13). The fluorinated silane provided the pigments with effective protection against oxidation when compared to pure  $\text{VO}_2$  or  $\text{VO}_2@\text{SiO}_2$ . The pigments were next mixed with a commercial aqueous solution of methacrylic acid/ethyl acrylate copolymer under basic conditions, and deposited on a glass substrate by casting. A comparison between pigments of varying size distributions showed that the smallest particles displayed the highest magnitude of NIR modulation and the lowest onset wavelength ( $< 800 \text{ nm}$ ) of NIR suppression, indicating an effective reflectance of the considerable solar spectra irradiance in this low wavelength region (in other words, the region of maximum NIR suppression was shifted nearer to the highest solar flux part of the spectrum). The NIR modulation, the visible transmittance and the onset of NIR suppression also depend on the amount of pigment and on their state of dispersion in the matrix. The higher the amount of pigment, the poorer was their dispersion in the matrix, and the lower were consequently the visible transmittance and the NIR modulation efficiency, while the onset wavelength was shifted to higher wavenumbers. The fluorinated core-shell particles exhibited good dispersion properties in the matrix, and led to an optimal combination of performances with a high visible light transmittance of around 62%, a high extent of NIR modulation of 30% and an early onset of NIR suppression of 758 nm (Table 3).



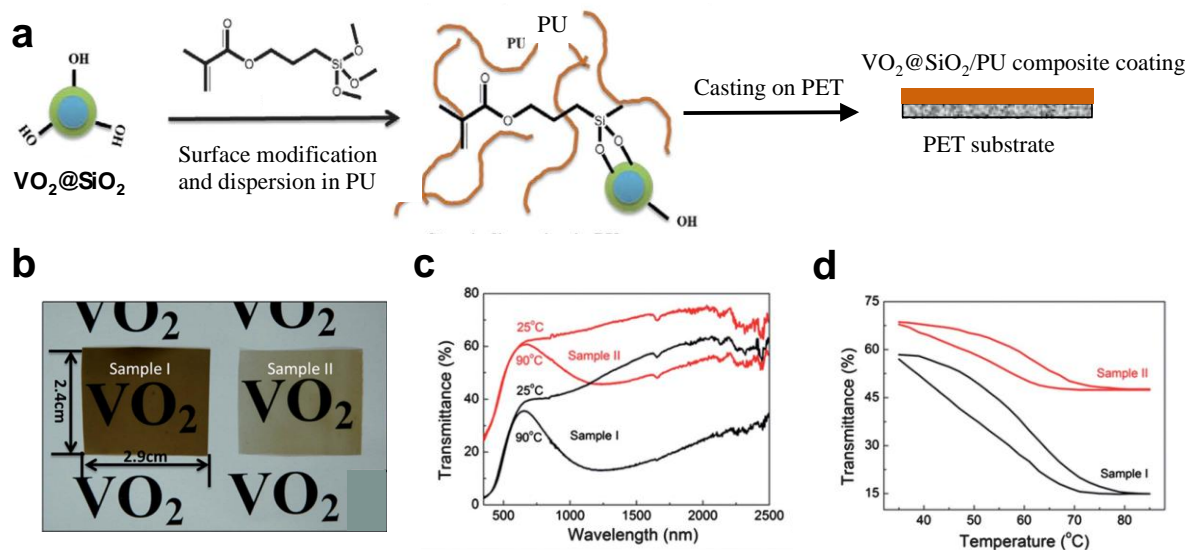
**Figure 13: (a) Scheme illustrating the procedure used to prepare an acrylic thermochromic coating through an all-aqueous route. (b) TEM image of core-shell  $\text{VO}_2@\text{SiO}_2$  nanocrystals and c) DSC traces showing an endothermic pic at  $58^{\circ}\text{C}$  upon heating and an exothermic pic at  $31^{\circ}\text{C}$  upon cooling, characteristics of the MIT transition. Adapted with permission from ref. <sup>[138]</sup> Copyright 2017, American Chemical Society.**

More recently, Liang et al. <sup>[170]</sup> reported the formation of composite film materials with optical modulation abilities in both the visible and NIR ranges (so-called dual-band optical modulation) using an all-solution approach. The films were obtained by UV-curing of a suspension containing a mixture of acrylate monomers, a liquid crystal with a positive dielectric constant (P-LCs), W-doped  $\text{VO}_2$  particles and polystyrene latex particles (to adjust the film thickness). Applying a voltage led to composite films with oriented LC microdomains that were subsequently removed by soaking with cyclohexane, and replaced by liquid crystals with a negative dielectric constant (N-LCs). The

resulting liquid-crystalline polymer network showed a dual-band modulating ability. Indeed, application of an external voltage aligned the LC domains, and actively modulated the visible light transmittance of the film that switched between a fully transparent (for homeotropically aligned LCs) to a strongly scattering state (for disordered LCs) due to refractive index mismatch between the N-LCs domains and the polymer matrix. The phenomenon was fully reversible. Interestingly, the solar modulation efficiency and the luminous transmittance of the coating were strongly dependent on the applied voltage and increased from 7.5% (respectively 11.6%) to 34.6% (respectively 57.8%) with increasing the voltage from 0 to 34 V, which is comparable to the optical performances of VO<sub>2</sub>-based hydrogels previously reported in the literature,<sup>[171]</sup> that also exhibit a dual-band modulation ability (vide infra).

#### 4.2. Polyurethane (PU)-based coatings

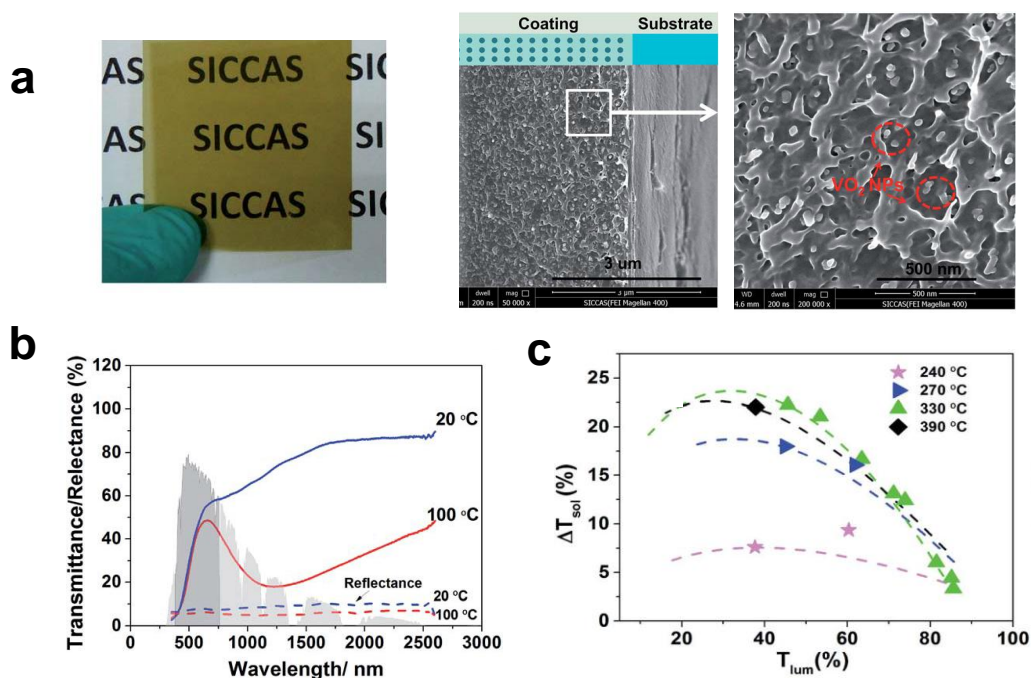
Another family of polymers that is particularly relevant for coating applications is polyurethanes (PUs). PUs-based formulations yield coatings with excellent chemical and solvent resistance and high durability. A few examples are given below. One first illustration is the work of Gao et al.<sup>[92]</sup> who successfully demonstrated the formation of a flexible PU film incorporating VO<sub>2</sub>@SiO<sub>2</sub> thermochromic nanoparticles. The VO<sub>2</sub> particles were synthesized by the hydrothermal method and coated with a silica shell using polyvinyl pyrrolidone (PVP K30) as compatibilizing agent. The shell played a crucial role in the durability properties of the VO<sub>2</sub> particles, providing an effective protection against oxidation and corrosion in acidic media. The core-shell particles were further functionalized with a methacrylate silane to favor their dispersion in the PU resin, mixed with a waterborne high molecular weight PU dispersion stabilized by sulfonate groups (Dispercoll U54, Bayer) and cast on a polyethylene terephthalate (PET) substrate (Figure 14a). The film containing the highest VO<sub>2</sub> content exhibited a brown color (Figure 14b) but displayed nevertheless an acceptable luminous transmittance ( $T_{lum}$ ) of 29%. Besides, the films showed a marked change in IR transmission across the MIT (up to 38% at 2000 nm) and UV-shielding properties (almost no transmittance below 500 nm) (Figure 14c). Their remarkably high solar modulation efficiency  $\Delta T_{sol}$  achieved values up to 13.6% - significantly higher than the modulation ability of pristine VO<sub>2</sub> films or multilayer films so far reported in the academic literature - make these flexible PU composite foils promising candidates for applications in construction and automotive glasses.



**Figure 14.** a) Scheme illustrating the process of formation of flexible PU composite films incorporating  $\text{VO}_2@\text{SiO}_2$  core-shell particles, b) digital photographs of the films with high (sample I) and low (sample II)  $\text{VO}_2$  contents, c) transmission spectra of the sample films at 25°C and 90°C and d) thermal hysteresis of the sample films at a wavelength of 1500 nm. Adapted with permission from ref.<sup>[92]</sup> Copyright 2012, The Royal Society of Chemistry.

In subsequent papers from the same group,<sup>[26][172][173]</sup> the thermochromic properties of the PU composite films were further improved by using a mixture of  $\text{VO}_2$  and antimony-doped tin oxide (ATO) nanoparticles,<sup>[172]</sup> Ti-doped  $\text{VO}_2$ <sup>[173]</sup> or highly crystalline  $\text{VO}_2$  particles,<sup>[26]</sup> and confronted to simulations. The incorporation of ATO particles decreased the solar modulation efficiency but also decreased the IR transmittance at high temperature, which is beneficial for energy saving as it decreases the solar heat gain during summer and thus reduces the cooling energy demand, while the visible transmittance was not affected by the presence of the particles.<sup>[172]</sup> Ti doping (1 wt%) led to a further enhancement of the thermochromic properties of the composite foils that exhibited a  $T_{\text{lum}}$  of 53% and a  $\Delta T_{\text{sol}}$  of 17.2%.<sup>[173]</sup> The best results were obtained for highly crystalline  $\text{VO}_2$  particles, whose composite films exhibited a  $T_{\text{lum}}$  of 45.6% and a  $\Delta T_{\text{sol}}$  of 22.3% (Table 3).<sup>[26]</sup>

Figure 15 shows the microstructure and the optical performance of a  $\text{VO}_2$ /PU composite coating containing 25-45 nm diameter  $\text{VO}_2$  particles synthesized by the hydrothermal method at 330°C.



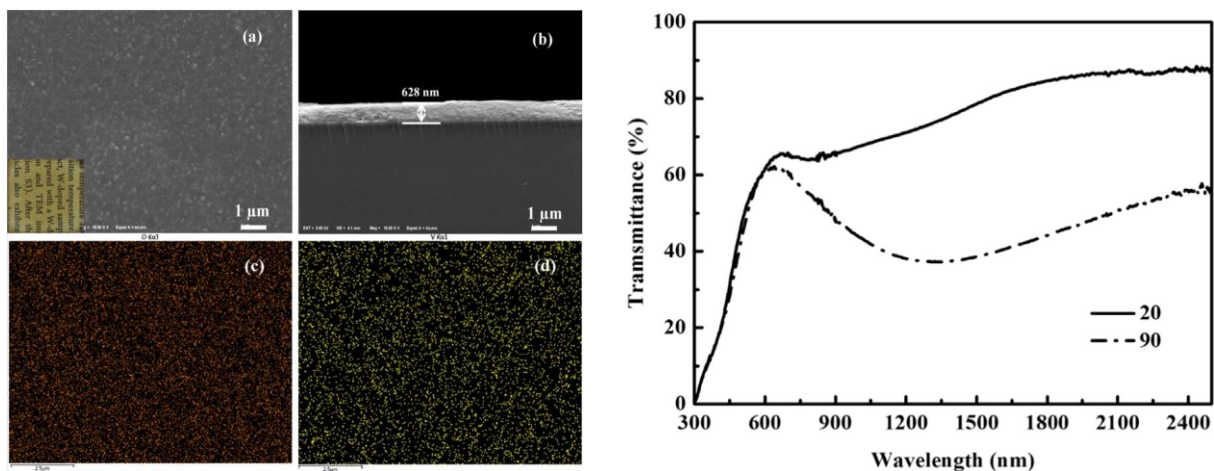
**Figure 15. a) Photograph and cross-sectional SEM images (small and larger magnification) of a PU composite film containing highly crystalline VO<sub>2</sub> particles prepared by the hydrothermal method at 330 °C showing that the nanoparticles are well dispersed within the matrix, b) transmission spectra of the composite film below and above the transition temperature and c) comparison between experimental (symbols) and calculated (dot lines) solar modulation efficiencies ( $\Delta T_{sol}$ ) and luminous transmittance ( $T_{lum}$ ) of composite films with different thicknesses and containing VO<sub>2</sub> particles heated at different temperatures. Adapted with permission from ref.<sup>[26]</sup> Copyright 2014, The Royal Society of Chemistry.**

This optimized temperature led to highly pure particles with a crystallinity close to that of bulk VO<sub>2</sub>. The particles were uniformly distributed within the film and had a significant influence on the thermochromic behavior of the coating, which displayed a huge transmittance difference below and above the transition temperature (Figure 15b). Calculations performed on composite films of varying thicknesses and containing VO<sub>2</sub> particles heated at different temperature showed a very good agreement with the experimental results in particular for the VO<sub>2</sub> particles synthesized at high temperature (Figure 15c). Under these optimized conditions, the composite coating was highly performant with a  $\Delta T_{sol}$  of 22.3% and a  $T_{lum}$  of 45.6%. In another example along the same line, Mo-W-co-doped VO<sub>2</sub> particles with a flower-like morphology, were synthesized by a microwave-assisted hydrothermal method, and incorporated into PU through *in situ* polymerization.<sup>[174]</sup> The PU dispersion was prepared from isophorone diisocyanate, 2-bis(hydroxymethyl) propionic acid, poly( $\epsilon$ -caprolactone) diol, glycerol, and triethylamine by means of phase inversion through the acetone process in the presence of the nanosized co-doped VO<sub>2</sub> particles. The resulting coatings showed a high visible transmittance and good optical properties in the IR range.



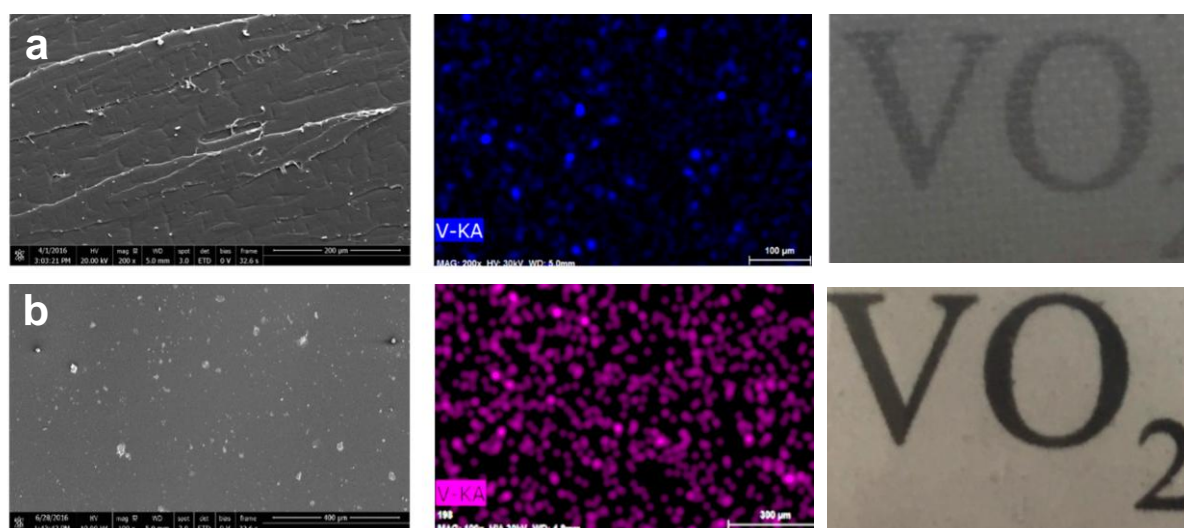
### 4.3. Other polymeric host materials

Several other publications have reported the use of a variety of polymeric host materials to improve the processability and final properties of VO<sub>2</sub>-based coatings. <sup>[101][175][164][162][127][134][176][98][177]</sup> Examples of hosting polymeric materials include poly(lactic acid) (PLA), <sup>[101]</sup> polyvinyl phenol, <sup>[175][164]</sup> ethylene-vinyl acetate (EVA) copolymers, <sup>[176]</sup> polyvinyl butyral (PVB), <sup>[98]</sup> and PVP. <sup>[177]47</sup> For instance, Suzuki et al. <sup>[101]19</sup> reported the synthesis of VO<sub>2</sub>-coated SiO<sub>2</sub> particles and their incorporation into PLA by solution mixing using N,N-dimethyl formamide as solvent. Madiba et al. <sup>[175][164]</sup> reported the elaboration of large surface area and chemically stable thermochromic VO<sub>2</sub>/polyvinyl phenol composite films by spin and dip coating techniques. Coarse VO<sub>2</sub> particles (of commercial grade) were dispersed in ethanol containing the polymer and the resulting dispersion was cast on freshly cleaved sodium chloride or glass substrates. The solutions were subjected to an intensive ultrasonic treatment to reduce the grain size and improve homogeneity. Removal of the solvent led to 850 nm thick VO<sub>2</sub>-based composite films with a granular surface. However, for too high VO<sub>2</sub> contents, poor dispersion of the submicronic crystallites within the matrix compromised the thermochromic performance of the coating. Optimal performances were obtained for a VO<sub>2</sub>-to-polymer mass ratio of 0.8. A similar approach was recently undertaken by Wu et al. <sup>[177]</sup> using porous VO<sub>2</sub> particles dispersed in a PVP matrix. Indeed, optical calculations predict that the incorporation of air voids should enhance the thermochromic properties of VO<sub>2</sub> films. <sup>[175]</sup> The VO<sub>2</sub> particles were obtained by the hydrothermal method using cotton as a template. The authors studied the effect of the hydrothermal temperature (from 120°C to 180°C) on the properties of the VO<sub>2</sub> particles, and showed that the crystallinity and the particle size increased with increasing temperature while the mean pore size first increased and then decreased. The nanoporous VO<sub>2</sub> crystallites were uniformly distributed within the film as attested by energy-dispersive X-ray spectroscopy (EDS) mapping of O and V elements (Figure 16, left). Furthermore, the composite film containing the VO<sub>2</sub> particles heated at 180°C exhibited excellent optical properties with a solar modulation ability ( $\Delta T_{\text{sol}}$ ) of 12.4% while retaining a high visible transmission ( $T_{\text{lum}}$ ) of 56% (Figure 16, right).



**Figure 16. Left : a, b) SEM images of the surface and the cross-section of the VO<sub>2</sub>/PVP composite film containing nanoporous VO<sub>2</sub> particles heated at 180°C, and c, d) EDS mapping of O and V elements showing uniform distribution of the VO<sub>2</sub> crystallites within the film. Right : transmission spectra of the film below and above the transition temperature. Reproduced with permission from ref.<sup>[177]</sup> Copyright 2017, Elsevier.**

Using a polyelectrolyte as the matrix, Miao et al.<sup>[127]</sup> also reported the formation of composite films by the spin-coating technique. The VO<sub>2</sub> particles were dispersed in butyl acetate containing an appropriate amount of polyelectrolyte, followed by grinding in a planetary ball mill. The polyelectrolyte enabled to effectively prevent the particles from re-aggregating during the dispersion process, and particles sizes as small as 40 nm in diameter were obtained after 12 hours, although the crystallinity also decreased with increasing grinding time in agreement with previous literature.<sup>[75]</sup> The films prepared under these optimized milling conditions displayed a high luminous transmittance ( $T_{lum} = 38.5\%$ ) and depressed haze as a consequence of the small particle size. The concomitant loss of crystallinity led however to a lower transmittance difference at 2000 nm below and above the phase transition compared to the film materials obtained after grinding for shorter times. The presence of defects also resulted in a shift of the transition temperature towards a higher temperature on heating and a lower temperature on cooling (i.e., the hysteresis loops broadened). The 12 hours grinding composite film exhibited nonetheless the highest solar modulation efficiency of the series ( $\Delta T_{sol} = 16.9\%$ ), illustrating the tremendous influence of the VO<sub>2</sub> particle size on the optical properties of the thermochromic coatings. The importance of controlling the size and state of dispersion of the VO<sub>2</sub> particles was recently illustrated by Srirodpai et al.<sup>[176]</sup> who investigated the effect of processing conditions on the thermochromic properties of VO<sub>2</sub>/EVA laminate films. As shown in Figure 17, the film obtained by solution mixing displayed a greater visible transmittance as a consequence of a more uniform distribution of the VO<sub>2</sub> pigment. A demo system was further developed by exposing the VO<sub>2</sub>/EVA laminate film (previously coated on a window) to an IR lamp, and measuring the temperature behind the window as a function of irradiation time. The temperature was lowered by around 10°C when compared to a reference film that did not contain VO<sub>2</sub> particles.

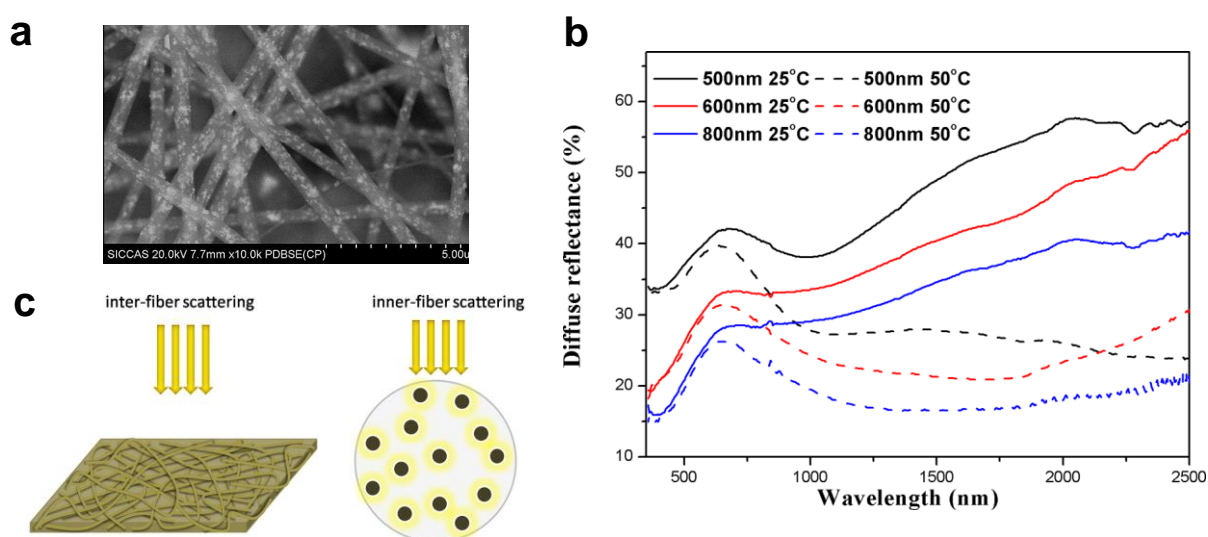




**Figure 17. Influence of processing conditions: melt blending (a) versus solution mixing (b) on the microstructure and transparency of VO<sub>2</sub>/EVA composite films. From left to right: SEM images, X-ray dot mapping and photographs of self-standing films. Reproduced with permission from ref.<sup>[176]</sup> Copyright 2017, MDPI.**

#### 4.4. Electrospun VO<sub>2</sub> / polymer composite fibers

In an attempt to produce porous discontinuous films, the incorporation of inorganic materials into electrospun fibers has retained considerable attention in the recent literature. The resulting materials can find applications in a variety of domains as for instance filtration, tissue engineering and coatings. Some examples of composite fibers incorporating VO<sub>2</sub> particles are given below.<sup>[178][179][180][181]</sup> The first article along this line was reported by Li et al.<sup>[178]</sup> only a few years ago. Thermochromic VO<sub>2</sub> nanoparticles prepared by the hydrothermal method (pure VO<sub>2</sub> or W-doped VO<sub>2</sub>), were ultrasonically dispersed in an ethanol solution containing PVP, and further processed into 500 - 800 nm diameter fibers by electrospinning (Figure 18a). Contrary to pure VO<sub>2</sub>, the fiber mat revealed a decreased reflectance (and greater transmittance) when heated above the phase transition temperature. The larger the fiber diameter, the lower was the diffuse reflectance of the fiber mat both under and above the transition temperature (Figure 18b). This was interpreted by the authors as being the result of multiple inner and inter-fiber scatterings as depicted in Figure 18c. The process was next extended to VO<sub>2</sub> nanorods,<sup>[179]</sup> leading to similar observations.



**Figure 18. a) SEM image of the electrospun VO<sub>2</sub>/PVP composite fibers with a diameter of 500 nm, b) diffuse reflection spectra of composite fiber mats containing W-doped VO<sub>2</sub> particles for different fiber diameters and c) illustration of the multiple scattering-absorbing mechanism of the fiber mats at the origin of the peculiar optical behavior of the mat. Reproduced with permission from ref.<sup>[178]</sup> Copyright 2013, American Chemical Society.**

More recently, Lu et al.<sup>[180]</sup> described the incorporation of VO<sub>2</sub> into poly(methyl methacrylate) fibers still using the electrospinning approach, with the aim of affording thermochromic coatings with enhanced antioxidation properties and higher hydrophobicity. The thermochromic performances of the composite fiber mats coating were however

poor with a luminous transmittance of only 21% and a  $\Delta T_{\text{sol}}$  lower than 7%. The same authors also published the fabrication of thermochromic films with superhydrophobic properties using a similar approach that combines electrospinning and hot pressing.<sup>[171]</sup> However again, the properties of the resulting films were quite disappointing.

#### 4.5. Hydrogel-based coatings

As mentioned above, to reinforce the solar transmittance modulation, VO<sub>2</sub> particles can be combined with thermotropic materials (i.e., materials that exhibit a temperature-dependent change of the light scattering properties) leading to the so-called dual-band optical modulation. A typical example of thermotropic materials is hydrogels. Hydrogels are three-dimensional crosslinked polymer networks capable of swelling and de-swelling reversibly in water under different conditions (pH, temperature or light), which modifies their optical characteristics. Thermotropic effects are caused in this case by a phase separation process above the lower critical solution temperature (LCST) resulting in the formation of scattering domains. Poly(N-isopropylacrylamide) (PNIPAM), one of the most investigated thermoresponsive hydrogel polymers, undergoes an hydrophilic-to-hydrophobic transition at around 32°C which makes it an ideal candidate for thermochromic applications. The crosslinked polymer swells extensively in water at room temperature and exhibits a pronounced collapse on heating which is accompanied by a change of the optical transparency of the coating. Although the thermotropic behavior of polymer gels has been studied for many years, their application in smart windows was hampered by their strong volume reduction and their lack of transparency above the phase transition.<sup>[182]</sup> However, by designing a suitable glass panel set-up, Zhou et al.<sup>[183]</sup> showed that a PNIPAM hydrogel film sandwiched between two glass sides, could be a potential candidate for smart windows applications. After carefully adjusting the film thickness to 52 µm, the PNIPAM hydrogel displayed a high  $T_{\text{lum}}$  of 87.9% at room temperature and an acceptable opacity at higher temperature ( $T_{\text{lum}}(40^{\circ}\text{C}) = 59.9\%$ ). Overall, the hydrogel coating enabled to achieve a solar modulation efficiency ( $\Delta T_{\text{sol}}(20-40^{\circ}\text{C}) = 20.4\%$ ) comparable to that of the most efficient VO<sub>2</sub> thermochromic coatings. The PNIPAM hydrogels were further combined with VO<sub>2</sub> particles to make a hybrid thermochromic material.<sup>[171]</sup> The VO<sub>2</sub>/hydrogel hybrid was synthesized by *in situ* polymerization starting from an homogeneous dispersion of VO<sub>2</sub> particles into an aqueous solution containing the monomer (NIPAM), polyvinyl alcohol and *N,N'*-methylene bis acrylamide) used as crosslinker. Dipropylene glycol methyl ether (DMP) and Disperbyk 180 were used to assist pigment dispersion in water. The approach allowed increasing the  $\Delta T_{\text{sol}}$  to up to 35% while maintaining a high  $T_{\text{lum}}$  of around 82%. Thermochromic materials based on hydroxypropyl cellulose (HPC) have also been reported.<sup>[184]</sup> Compared to PNIPAM, HPC presents the advantage of being less toxic while its LCST can be tuned from 42 to 30°C by the addition of sodium chloride (NaCl). Thermochromic coatings with optimum performances were obtained for 0.5wt% of NaCl and a film thickness of 350 µm. The hydrogel was totally transparent at room temperature ( $T_{\text{lum}} = 86.1\%$ ) and translucent at 45 °C ( $T_{\text{lum}}(45^{\circ}\text{C}) = 48.7\%$ ), which led overall to an optimized  $\Delta T_{\text{sol}}(20-45^{\circ}\text{C})$  of 25.7%. The incorporation of tungsten-doped VO<sub>2</sub> particles into the HPC microgel led to a further enhancement of the thermochromic properties with a high  $T_{\text{lum}}$  of 80% at room temperature and an acceptable  $T_{\text{lum}}$  of 33% at 60°C.<sup>[67]</sup> The luminous contrast of the hydrogel combined

with the IR transmittance contrast of VO<sub>2</sub> gave a high  $\Delta T_{\text{sol}}$  of 36% which is the highest ever reported among VO<sub>2</sub>-based thermochromic composite coatings (Table 3).

**Table 3. Summary of the VO<sub>2</sub>-based composite thermochromic coatings reported in the literature: nature of the polymer matrix and of the precursors used to synthesize the VO<sub>2</sub> particles, transition temperature of VO<sub>2</sub> and optical performances of the coatings.**

| Ref.       | Polymer matrix                     | VO <sub>2</sub>  |   | T <sub>c</sub> (°C) | ΔT <sub>sol</sub> (%) at λ (nm) |         | T <sub>lum</sub> (%) | ΔT <sub>sol</sub> (%) |
|------------|------------------------------------|--|---|---------------------|---------------------------------|---------|----------------------|-----------------------|
|            |                                    | Precursors   | Type  |                     | 2000 nm                         | 2500 nm |                      |                       |
| [98]       | PVB/ethanol                        | V <sub>2</sub> O <sub>5</sub> /OA                                  | VO <sub>2</sub> @ZnO                            | 63.6                | /                               | /       | 51                   | 19.1                  |
| [101]      | PLA/solvent                        | VO(OiPr) <sub>3</sub>  | SiO <sub>2</sub> @VO <sub>2</sub>               | /                   | /                               | 10      | /                    | /                     |
| [164][175] | Poly vinyl phenol /EtOH            | /  | VO <sub>2</sub> powder (commercial)             | 67.8                | 25                              | 24      | /                    | /                     |
| [127]      | Polyelectrolyte/butyl acetate      | NH <sub>4</sub> VO <sub>3</sub> /OA                                | VO <sub>2</sub> nanobelts                       | 64 to 68.5          | 48.7                            | 43      | 38.5                 | 16.9                  |
| [177]      | PVP/ethanol                        | V <sub>2</sub> O <sub>5</sub> /OA/cotton                           | Mesoporous VO <sub>2</sub> nanocrystals         | /                   | 40                              | 25      | 56.0                 | 12.9                  |
| [75]       | Acrylic/solvent                    | VOSO <sub>4</sub> , xH <sub>2</sub> O                              | W-doped and undoped VO <sub>2</sub>             | -4 to 66            | /                               | 23      | /                    | /                     |
| [167]      | Acrylic                            | V <sub>2</sub> O <sub>5</sub> /OA                                  | W-doped and undoped VO <sub>2</sub>             | 58 to 68            | 31.5                            | /       | /                    | /                     |
| [168]      | Acrylic                            | VOSO <sub>4</sub> , xH <sub>2</sub> O                              | W-doped and undoped VO <sub>2</sub>             | 47                  | 37.3                            | /       | 26                   | 3.4                   |
| [157]      | Acrylic/photo polymerization       | VO(OiPr) <sub>3</sub>  | Colloidal VO <sub>2</sub>                       | 65 to 78            | 5                               | /       | /                    | /                     |
| [169]      | Acrylic/photo polymerization       | V <sub>2</sub> O <sub>5</sub>                                      | VO <sub>2</sub> on mica                         | 71                  | 35 <sub>λ=2200</sub>            | /       | /                    | /                     |
| [138]      | Acrylic/waterborne                 | NH <sub>4</sub> VO <sub>3</sub> (aq)                               | VO <sub>2</sub> @SiO <sub>2</sub> nanoparticles | /                   | /                               | /       | /                    | /                     |
| [170]      | Acrylic + LCs/photo polymerization | V <sub>2</sub> O <sub>5</sub> /N <sub>2</sub> H <sub>4</sub> , HCl | W-doped VO <sub>2</sub> nanocrystals            | 43.2                | 22.5                            | 15      | 57.8                 | 34.6                  |
| [92]       | PU/waterborne                      | V <sub>2</sub> O <sub>5</sub> /OA                                  | VO <sub>2</sub> @SiO <sub>2</sub>               | 51.5                | 38                              | 26.5    | 29                   | 13.6                  |
| [172]      | PU/waterborne                      | V <sub>2</sub> O <sub>5</sub> /OA                                  | VO <sub>2</sub> -Sb:SnO <sub>2</sub>            | /                   | 17                              | /       | 53                   | 11.7                  |
| [173]      | PU/waterborne                      | V <sub>2</sub> O <sub>5</sub> /OA                                  | Ti-doped VO <sub>2</sub>                        | 66.8                | 30                              | 20      | 53                   | 17.2                  |
| [174]      | PU/waterborne                      | V <sub>2</sub> O <sub>5</sub> /OA                                  | Mo-W-doped VO <sub>2</sub>                      | 17.6 to 38.5        | 23 - 72                         | 25 - 76 | /                    | /                     |
| [26]       | PU/waterborne                      | V <sub>2</sub> O <sub>5</sub> /N <sub>2</sub> H <sub>4</sub> , HCl | Highly crystalline VO <sub>2</sub> + W          | 21 to 68            | 55                              | 41      | 45.6                 | 22.3                  |
| [178]      | PVP/electrospinning                | V <sub>2</sub> O <sub>5</sub> /H <sub>2</sub> O <sub>2</sub>       | W-doped VO <sub>2</sub> nanoparticles           | 40 to 68            | /                               | /       | /                    | /                     |
| [179]      | PVP/electrospinning                | V <sub>2</sub> O <sub>5</sub> /OA                                  | VO <sub>2</sub> nanorods                        | 60.5                | 17.5                            | 15      | /                    | /                     |
| [180]      | PMMA/electrospinning               | NH <sub>4</sub> VO <sub>3</sub> /EG                                | VO <sub>2</sub> nanoparticles                   | /                   | /                               | /       | 21                   | 6.8                   |
| [181]      | PMMA/electrospinning               | VOSO <sub>4</sub> , xH <sub>2</sub> O                              | VO <sub>2</sub> nanoparticles                   | /                   | 22                              | 18      | /                    | 7.4                   |
| [183]      | PNIPAM/hydrogel                    | /  | No VO <sub>2</sub>                              | /                   | /                               | /       | 87.9                 | 20.4                  |
| [171]      | PNIPAM/hydrogel                    | /  | VO <sub>2</sub> powder (commercial)             | 68                  | /                               | /       | 82.1                 | 34.7                  |
| [184]      | HPC/hydrogel                       | /  | No VO <sub>2</sub>                              | /                   | /                               | /       | 86.1                 | 25.7                  |
| [67]       | HPC/hydrogel                       | /  | W-doped VO <sub>2</sub> (commercial)            | 50                  | /                               | /       | 80                   | 36                    |

## Conclusions

Large-scale production of VO<sub>2</sub> smart coatings is necessary to turn this technology from the laboratory into the industrial and commercial applications. For instance, for uses in buildings envelopes, the final properties to be achieved are high solar transmittance modulation ( $\Delta T_{\text{sol}}$ ) to reach high passive adaptive thermal comfort in buildings with a control of the transition characteristics such as a narrow hysteresis width and a low transition temperature,  $T_c$ . If transparent applications are aimed, such as smart windows, an additional high luminous transmittance should be achieved. For practical applications, VO<sub>2</sub> smart coatings should have 50% luminous transmittance and 15% solar transmittance modulation ability for sufficient energy-saving effect. We have reviewed in this paper complementary building blocks that enable to envision the use of VO<sub>2</sub> nanoparticles as fillers in efficient thermochromic polymer-VO<sub>2</sub> nanocomposites. It was shown that the nanoparticle approach enables to improve the optical characteristics of the coatings ( $T_{\text{lum}}$  and  $\Delta T_{\text{sol}}$ ). However, nanoparticles exhibit a wider hysteresis, which can be in turn reduced by elemental or interfacial doping of the VO<sub>2</sub>. In addition, doping lowers down the transition temperature closer to room temperature, which is more adequate for human comfort thermal applications and also impacts the morphology and phase of the nanoparticles.

It appeared that direct synthetic protocols in solution are particularly attractive being simpler, with lower costs and allowing direct functionalization compared to methods involving solid-state transformation and annealing process. However, only a few of one-step strategies in solution have been reported to successfully prepare VO<sub>2</sub>(R) nanomaterials. Clearly the stabilization of the VO<sub>2</sub>(M) phase directly from molecular precursors or solid phases in solution is challenging and is strongly affected by many parameters especially the presence of doping element. It was found that post-thermal treatment of a precursor phase, typically other vanadium dioxide polymorphs and vanadyl complexes is also a widely used approach to obtain VO<sub>2</sub>(M) nanoparticles. The fact that vanadium can adopt multiple oxidation states makes it very sensitive to oxygen, thus elaboration of composite materials made of VO<sub>2</sub> particles incorporated into a polymer matrix is still a challenge. However, various polymer matrices have been explored such as acrylics, polyurethanes and hydrogels. However, doped-nanoparticles or composite-nanoparticles embedded in polymer matrices can combine improved optical properties (narrow hysteresis, high luminous transmittance and high solar modulation), convenient process deposition according to polymer coatings technologies and low transition temperature. By combining the properties of VO<sub>2</sub> nanoparticles as fillers within the polymer matrix, one will take advantage of the long-run experience in coatings technology to tackle industrial scale-up steps such as coatings formulation for shelf-life and environmental outdoor stability. Future work can be carried out by combining other functional materials such as protective, antireflection and self-cleaning within the same coating or within the same nanoparticles (for instance into core-shell nanostructures). If so, approaches to VO<sub>2</sub> surface functionalization and compatibilisation to favor tuning of nanostructures and improve dispersion into the polymer matrix might be next challenges to tackle.

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