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Evidence of hydrogen modulation in amorphous germanium prepared by reactive evaporation

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Reactive evaporation of germanium under a flow of hydrogen atoms on substrates maintained at liquid-nitrogen temperature allows us to obtain α -Ge:H thin films and multilayers. The material contains a considerable amount of polygermane bonds and is porous, as shown by low angle x-ray scattering performed on Ge/Ge:H/... multilayers. The stability of hydrogen is measured by resistivity measurements and by effusion experiments. Finally, low angle neutron scattering shows that it is possible to modulate hydrogen in Ge/Ge:H/Ge/Ge:D/... multilayers with 40 Å thick layers.

Recently, we have shown that the technique of reactive evaporation allows one to obtain amorphous multilayers of hydrogenated silicon¹ and of hydrogenated silicon tin alloys.² In this letter, we show that multilayers of germanium can also be obtained by this preparation method. As the aim of this study is to observe the modulation of hydrogen in very thin layers, it is necessary to suppress diffusion over very short distances. Shinar *et al.*³ have reported the absence of long-range H motion in amorphous silicon if the material was void-rich and contained a lot of SiH₂ bonds. These films were obtained with low substrate temperatures. In order to obtain this microstructure, we prepared hydrogenated amorphous germanium by evaporation under a flow of hydrogen atoms on substrates maintained at liquid-nitrogen temperature. The hydrogen bonding is characterized by infrared spectrometry and the porosity is observed by performing low angle x-ray scattering on Ge/Ge:H/... multilayers, then, the stability of hydrogen is measured by resistivity measurements and effusion experiments. Finally, we show that low angle neutron scattering allows one to observe the modulation of hydrogen and deuterium in Ge/Ge:H/Ge/Ge:D/... multilayers.

Amorphous Ge:H is usually prepared by glow discharge⁴ or by sputtering and only one attempt⁵ has been made to hydrogenate evaporated germanium. In this study, germanium was evaporated from an electron beam gun crucible. The deposition rate, monitored and controlled by a quartz microbalance system, was the equivalent of 1.1 Å/s of pure Ge of bulk density. The substrates were kept at liquid-nitrogen temperature in order to increase the hydrogen concentration and to avoid hydrogen diffusion in the multilayers. Atomic hydrogen was produced by thermal dissociation of molecular hydrogen in a tungsten tube heated to 2400 K. The hydrogen flow was regulated by maintaining the total pressure in the evaporation chamber of 6×10^{-5} Torr.

The first characteristic of these samples is that they contain essentially GeH₂ bonds. Indeed, the infrared transmission spectrum of a freshly prepared film (Fig. 1) shows four strong absorption bands due to the (GeH₂)_n polyger-

mane bonding geometry:⁶ (1) unresolved symmetric and asymmetric stretching modes at 1980 cm⁻¹; (2) the 825 cm⁻¹ scissors mode; (3) the 765 cm⁻¹ wagging mode; and (4) the 570 cm⁻¹ rocking mode. The 1880 cm⁻¹ absorption associated with isolated Ge:H bonds is hardly visible. Moreover, the spectra show an immediate and rapid atmospheric post-oxidation with time. Therefore, in order to avoid post-oxidation, the samples were placed, just after the deposition process, in a liquid-nitrogen container and they were maintained at this temperature until the other characterization experiments (exodiffusion, diffraction) can be carried out.

The second feature is the low density of hydrogenated germanium which was previously⁷ observed from x-ray scattering experiments performed on Ge/Ge:H multilayers prepared by evaporation of germanium alternately with and without hydrogen during 30 s stages. Such multilayers give rise to Bragg peaks in low angle x-ray scattering, the intensity of which is a function of the contrast between the layers. This multilayer is stable since the peaks are always visible after an annealing at 350 °C. As the x-ray scattering factor from hydrogen is very weak, the modulation of hydrogen in the material leads to an insignificant signal and it is not the origin of the x-ray scattering peaks. In fact, the

FIG. 1. Infrared transmission spectrum of evaporated α -Ge:H deposited at 77 K.

FIG. 2. Variation of the dc resistivity of *a*-Ge:H sample vs temperature.

contrast originates in the modulation of the atomic density of germanium since the atomic density is high for germanium evaporated under vacuum and the atomic density is low for noncompact germanium evaporated under a high pressure of hydrogen.

In order to observe the passivation of dangling bonds by hydrogen atoms, the electrical resistivity of a hydrogenated film was measured as a function of temperature. Figure 2 shows that the initial resistivity was too high to be measured ($\rho > 10^{18} \Omega \text{ cm}$) between the deposition temperature and the room temperature. Heating the sample from 300 to 400 K leads to a reversible behavior between these two temperatures (1 $\leftrightarrow$ 2). However, heating the sample to 500 K leads to an irreversible transformation with a sharp decrease of resistivity near 440 K (2 $\rightarrow$ 3). After this transition, the resistivity presents a reversible behavior between 77 and 500 K (3 $\leftrightarrow$ 4), with values comparable to those of an unhydrogenated sample. This transition does not correspond to a crystallization phenomenon since transmission electron microscopy shows that the sample was still amorphous. Therefore, the hydrogenated sample has a dark resistivity value at room temperature which is several orders of magnitude larger than the value obtained for an unhydrogenated sample. The passivation process is well observed. Unfortunately, the range of available resistivity data was too narrow to correctly determine the transport mechanism (thermally activated or hopping). Moreover, with a heating rate of 2 K/min, the resistivity shows a sharp transition at 440 K, as if the hydrogen atoms responsible for the passivation left the sample at this temperature.

Effusion experiments allowed us to directly observe the release of hydrogen from these samples. For these experiments, deuterated films were deposited on float glass substrates. Deuterium was used to suppress background effects due to adsorbed water, then the films were inserted into a quartz tube evacuated by an ionic pump and were heated at a constant rate of 15 °C/min up to 750 °C. The experimental spectrum (Fig. 3) shows only a low-temperature peak, near 230 °C, which is generally attributed to the desorption of molecular hydrogen from GeH₂ bonds in porous materials prepared at low substrate temperature.⁸ This result is in agreement with the preparation conditions, since the substrate is at liquid-nitrogen temperature, and with the

FIG. 3. Effusion spectrum of a deuterated germanium sample. The heating rate is 15 °C/min.

characterization (hydrogen bonding, porosity) described in the first sections. Moreover, taking account of the different heating rates, the maximum temperature of the effusion curve corresponds to the transition temperature observed in the resistivity measurements.

Complementary information on the mobility of hydrogen can be obtained by low angle neutron scattering experiments performed on multilayers with a modulation of hydrogen and deuterium. Indeed, as the neutron coherent scattering amplitudes of H and D are large and of the opposite sign, neutron diffraction experiments are very sensitive to the isotope introduced in the evaporation chamber during the germanium deposition. Low angle neutron scattering experiments were performed on a Ge/Ge:H/Ge/Ge:D/... sample at the D10 instrument of the Institute Laue Langevin. The theoretical thickness of each layer was 33 Å. The spectrum (Fig. 4) shows the peak at 0.078 Å⁻¹ corresponding to that observed with x-ray experiments and attributed to the silicon density modulation. In fact, as the average of b_H and b_D is not zero, this peak is due both to the atomic density modulation and to the hydrogen modulation. In addition, there is a large prepeak at 0.039 Å⁻¹, which is a superstructure peak, due to the fact that hydro-

FIG. 4. Low angle neutron scattering intensity I vs scattering vector q for Ge/Ge:H/Ge/Ge:D/... multilayers.

gen and deuterium are alternating with a double modulation period of 160 Å. The modulation wavelength is larger than the theoretical thickness (132 Å) because of the lower density of the hydrogenated and deuterated films. Moreover, this superstructure peak disappears rapidly at 300 K due to the diffusion of hydrogen and deuterium and it is necessary to keep the samples carefully at 77 K before the experiments in order to maintain the isotopic modulation. As expected, the diffusivity of hydrogen in germanium is then much larger than the diffusivity measured in amorphous silicon⁹ prepared by the same technique, in which neutron experiments can be carried out in the range 250–300 °C.

In conclusion, *a*-Ge:H thin films and multilayers can be obtained by evaporation on a substrate maintained at liquid-nitrogen temperature under a flow of atomic hydrogen. This material is porous, as shown by low angle x-ray scattering, and contains essentially GeH₂ or (GeH₂)_n bonds. Resistivity measurements and exodiffusion experiments show that hydrogen leaves the samples at about 200 °C, depending on the heating rate. Moreover, the low

value of the hydrogen motion allows the observation at 300 K of an isotopic modulation with 40 Å thick layers by performing low angle neutron scattering experiments.

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