Coexistence of site- and bond-centered electron localization in the high-pressure phase of LuFe_2O_4

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Magnetic-electronic hyperfine interaction parameters of spectral components are obtained from *in situ* ^{57}Fe Mössbauer spectroscopy pressure studies of the mixed-valence LuFe_2O_4 multiferroic, up to ~ 30 GPa and on recovered high-pressure phase samples. Temperature-dependent Mössbauer spectra of the low-pressure phase show that Fe^{2+} and Fe^{3+} sites are discernible, consistent with known site-centered charge order in the triangular (frustrated) Fe sublattice network. Magnetic spectra of the high-pressure phase, stabilized in a rectangular Fe sublattice network at $P > 8$ GPa, exhibit fingerprints of iron in an intermediate valence state only. Temperature-dependent resistivity pressure studies evidence thermally activated small polaron motion in the high-pressure phase. These experimental signatures, complemented by *ab initio* calculations of electronic structure, are considered evidence of asymmetric dimer formation $\text{Fe}^{(2+\Delta)+} \leftrightarrow \text{Fe}^{(3-\Delta)+}$, where the minority-spin electron deconfinement coefficient is $\Delta = 0.3\text{--}0.4$. Bragg satellites discerned in electron diffraction patterns of the metastable high-pressure phase possibly stem from this admixture of site- and bond-centered localization (intermediate-state charge order) in a magnetic background. This breaks inversion symmetry and potentially renders LuFe_2O_4 in its high-pressure phase as a new charge order instigated (electronic) ferroelectric.

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I. INTRODUCTION

Charge order (CO) has often been considered as a site-centered superstructure involving transition metal (TM) sites with different discrete valences. Such TM-centered CO is pictured as mobile carriers confined to a TM sublattice (e.g., t_{2g} minority-spin electron at Fe^{2+}) [1]. Examples are alternation of $\text{Fe}^{2+}/\text{Fe}^{3+}$ in Fe_2OBO_3 [2], or less definitively $\text{Mn}^{3+}/\text{Mn}^{4+}$ in $\text{La}_{0.5}\text{Ca}_{0.5}\text{MnO}_3$ [3,4]. The latter is less definitive because calculations show other energetically favorable CO states occur, notably (i) bond-centered CO and (ii) so-called intermediate-state CO, see Refs. [3,5] and the review [6]. In case (i), in TM oxides, oxygen ions on TM-TM bonds (e.g., perovskite) facilitate electron transfer processes, and charge carriers are confined evenly over a TM-TM symmetric dimer [6–8]. In intermediate-state CO, case (ii), an admixture of site- and bond-centered configurations is energetically favored [3,6]. Carrier delocalization involves preferential confinement to a neighboring TM atom and formation of electronically asymmetric TM-TM dimers.

These alternatives to site-centered CO have generated considerable excitement in: parent or precursor phases of high- T_C cuprates [9,10], colossal magnetoresistance doped manganites [11–13], and so-called type-I multiferroics. The latter include manganites $\text{Pr}_{1-x}\text{Ca}_x\text{MnO}_3$ [13], the famous case of magnetite (Fe_3O_4) [6], and the spin/charge frustrated

case LuFe_2O_4 [14]. This has spawned a hot topical area in multiferroics where CO is predicted to render electronically driven ferroelectricity [6,15]. Intermediate-state CO yields requisite inversion symmetry breaking for electronically driven ferroelectric polarization [3,6].

Our focus is CO in the pressure-temperature phase diagram of LuFe_2O_4 . This has been intensely scrutinized in the low-pressure (LP) phase, and claims for CO-instigated ferroelectricity at ambient pressure are mired in controversy; see the review [16] and Ref. [17]. Rouquette *et al.* [18] showed that a high-pressure (HP) polymorph stabilizes above ~ 8 GPa (hereafter referred to as $\text{LuFe}_2\text{O}_4\text{-HP}$) and is locked-in upon decompression to ambient conditions. It is purported to involve CO, based on: Bragg satellites seen in electron diffraction studies of recovered specimens, resistivity-pressure behavior at room temperature, and detailed structural comparisons of LP and HP polymorphs [18,19]. Damay *et al.* [19] focused on more detailed crystal-chemical, structural, and magnetic elucidations. Synchrotron x-ray data analysis revealed an LP monoclinic $C2/m$ transformation to a monoclinic Pm , distorted rock-salt type, HP polymorph. Frustration in the LP phase is relieved as the original triangular Fe network changes to rectangular; see Figs. 1(a) and 1(b) [20]. Electron diffraction micrographs of recovered $\text{LuFe}_2\text{O}_4\text{-HP}$ showed weak satellites with respect to the Pm parent cell. These are indicative of either charge modulation as proposed earlier by Rouquette *et al.* [18], or a structural modulation involving the misfit-related structure of the monoclinic HP phase. Recovered $\text{LuFe}_2\text{O}_4\text{-HP}$, $\sim 9\%$ denser than the LP phase, exhibited antiferromagnetic ordering up to $T_N \sim 380$ K.

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FIG. 1. Bilayers at 300 K involving (a) triangular networked Fe in the LP phase and (b) rectangular networked Fe in the HP structure where oxygen atoms (not shown) are at vertices of connecting bonds. Vectors on Fe show magnetic moments, and dashed lines represent shortest Fe-Fe distances. ⁵⁷Fe Mössbauer spectra at 300 K of (c) paramagnetic LP phase and (d) magnetic HP phase. Solid lines through data points are overall fits to spectra. Deconvolution into subcomponents representing various charge states is discussed in the text.

The nature of any charge segregation (CO) in LuFe₂O₄-HP is deemed worthy of deeper scrutiny, to potentially unveil a new HP stabilized type-I multiferroic [6]. To this end, our temperature-dependent ⁵⁷Fe Mössbauer-effect spectroscopy (MS) pressure studies and electronic structure calculations, complemented by resistivity-pressure studies, enable direct probing of the Fe magnetic-electronic state.

II. EXPERIMENTAL METHODS AND RESULTS

Experimental methodologies are similar to our previous ⁵⁷Fe MS and resistivity investigations in diamond anvil cells (DACs) of the pressure response of site-centered CO in Fe₂OBO₃ [21,22]. The highly stoichiometric LuFe₂O₄ used here is from previous studies involving an extensive characterization at ambient pressure [23,24]. The recovered HP phase sample from LuFe₂O₄, taken to 12 GPa in previous neutron diffraction studies [19], was subjected to ⁵⁷Fe MS investigations as well.

The Fe magnetic-electronic state is deduced from fitted hyperfine interaction parameters of deconvoluted subcomponents of complex spectral profiles [25]. The most reliably

determined parameters are the isomer shift *I**S* (subspectrum centroid) quoted relative to the α -Fe metal standard, the quadrupole doublet-splitting (*Q**S*), and the internal magnetic field *B*_{hf} (sextet hyperfine structure) [26,27]. Further details on experimental methodologies and analyses are provided in the Supplemental Material [28].

The Mössbauer spectral pressure response of LP and HP phases at 300 K are radically different; see Figs. 1(c) and 1(d), respectively. Paramagnetic spectra of the LP phase are initially a superposition of equally abundant Fe²⁺ and Fe³⁺ doublets representing site-centered CO. There is also an electron-hopping component, Fe²⁺ $\leftrightarrow$ Fe³⁺, initiated at CO domain boundaries at ambient pressure [28,29] (Table S1), the abundance of which increases upon compression; see Fig. 1(c). At $\sim$ 5.2 GPa, beyond the HP-phase onset, the pronounced shoulder (at $\sim$ 1.5 mm s⁻¹), signifying discrete valences, merges into the total spectral envelope. This spectrum is fitted with a single electron-hopping component, already hinting at the disruption of site-centered CO in the HP majority phase.

Our main interest is the spectral attributes of LuFe₂O₄-HP fully stabilized at *P* > 8 GPa; see Fig. 1(d). The magnetic signature resonances emerge at $\sim$ 11 GPa in wings of the

FIG. 2. ^{57}Fe Mössbauer spectra of (a) magnetic LP phase at 15 K and (b) recovered HP phase. Subcomponents representing $\text{Fe}^{(3-\Delta)+}$ and $\text{Fe}^{(2+\Delta)+}$ sites are discussed in the text. Solid lines through data points represent overall fit to a spectrum. Spectrum at 4 K was measured with a synchrotron Mössbauer source. (c) Valence separation representation, by comparison of isomer shifts (relative to α -Fe metal standard) of spectral subcomponents of LuFe_2O_4 and Fe_2OBO_3 reference. Bottom cartoon depicts minority-spin electron distribution/deconfinement at neighboring Fe-Fe sites. (d) Dimers forming Zener polarons along chains of ferromagnetic atomic spins in the a direction. (e) Plot of $\ln(\rho/T)$ versus inverse temperature. Solid gray line is a linear fit to HP data at ~ 25 GPa in the range 180–320 K, representing thermally activated small polarons.

spectrum at 300 K and become more conspicuous in spectra of LuFe_2O_4 -HP at $P > 11$ GPa on an extended velocity scale, as seen in Fig. 1(d). There is appreciable intensity and asymmetry in the central region near zero velocity. Relatively broad spectral line shapes indicate a multiplicity of Fe sites.

The magnetic spectrum of LuFe_2O_4 -HP is first compared with that of the LP phase at ambient pressure, depicted in Fig. 2(a), measured well below the ferrimagnetic ordering onset $T_M \sim 240$ K. In the LP phase, three large- B_{hf} sextets (46–50 T, hatched yellow shading) with similar IS values (0.45 mm s $^{-1}$) represent Fe^{3+} sites. Three comparatively small- B_{hf} sextets (7–31 T, shaded gray) with similar IS values (0.95 mm s $^{-1}$) represent Fe^{2+} [27,30], including an additional diffuse “spin-frustrated” Fe^{2+} component [28,31] (Table S2). This is guided by previous studies [32–35] and compatibility with the known ferrimagnetic structure and

$\text{Fe}^{2+}/\text{Fe}^{3+}$ superstructure (site-centered CO) occurring in each sheet of a bilayer [35–37].

The relatively complex magnetic spectra of both LuFe_2O_4 -HP and its recovered metastable form in Figs. 1(d) and 2(b), respectively, have been analyzed by using similar fitting protocols described above for the magnetic spectrum of the LP phase. This is to reduce the number of fitting parameters [28,30] (Tables S3 and S4). Minimally, three components with some degree of line broadening are employed for both large- B_{hf} and small- B_{hf} suites, yielding acceptably small residuals in the overall fit. This model accounts for additional sites not explicitly included in the deconvolution [38].

A representative fitting of LuFe_2O_4 -HP is shown for the magnetic spectrum at 29 GPa in Fig. 1(d). Three subcomponents with similar (linked) $IS = 0.56$ mm s $^{-1}$ and relatively large- B_{hf} values (31–36 T, hatched yellow shading) account for resonances at high (x -scale) velocities. Additionally three

subspectra with small- B_{hf} values (2–25 T, shaded gray) and $IS = 0.44 \text{ mm s}^{-1}$ describe the asymmetric central region. A similar analysis is implemented for spectra in Fig. 2(b) of LuFe_2O_4 -HP at ambient pressure, recovered from starting material compressed to $P \geq 12 \text{ GPa}$ [19]. Large- B_{hf} components yield $IS = 0.57 \text{ mm s}^{-1}$, and small- B_{hf} components have $IS = 0.68 \text{ mm s}^{-1}$. The spectrum at 4 K in Fig. 2(b) was measured at the European Synchrotron Radiation Facility (ESRF, Grenoble, France) using synchrotron Mössbauer source methodology [39,40]. Yellow-shaded hatched and gray-shaded magnetic components evolve from large- and small- B_{hf} subspectra at 300 K, respectively, based on fitted IS values.

The valence separation plot in Fig. 2(c) compares IS values for fitted static subcomponents of the LP and HP phases of LuFe_2O_4 to those in Fe_2OBO_3 deemed to have mixed-integer valences [2,21]. The large valence separation in Fe_2OBO_3 with site-centered CO has $IS:\text{Fe}^{2+} \geq 1 \text{ mm s}^{-1}$ and $IS:\text{Fe}^{3+} \leq 0.4 \text{ mm s}^{-1}$ at ambient conditions [41]. For magnetic spectra of LuFe_2O_4 -HP in Figs. 1(d) and 2(b), IS values in Fig. 2(c) correspond to suites of large- B_{hf} (closed symbols) and small- B_{hf} subspectra (open symbols). The occurrence of both large- and small- B_{hf} suites is additional evidence of some valence separation [27]. Figure 2(c) is analogous to bond-valence considerations in structural studies of mixed-valence compounds, including Fe_2OBO_3 and LuFe_2O_4 at ambient pressure [2,36].

At 300 K, large- B_{hf} and small- B_{hf} components of LuFe_2O_4 -HP do not have widely disparate IS values anticipated for Fe^{2+} and Fe^{3+} ; see Fig. 2(c). The associated valences are more appropriately designated $\text{Fe}^{(2+\Delta)+}$ and $\text{Fe}^{(3-\Delta)+}$. The deconfinement coefficient Δ quantifies the time-averaged amount of minority-spin charge ($\Delta \times e^-$) delocalized from Fe^{2+} to a neighboring Fe^{3+} site [42]. Its discernment in molecular complexes is known as valence detrapping, $0 < \Delta < 0.5$, whereas $\Delta = 0$ constitutes the valence trapped situation [43,44]. The consequential effect on measured IS values is that:

$$(IS : \text{Fe}^{(2+\Delta)+}) = (1 - \Delta)(IS : \text{Fe}^{2+}) + \Delta(IS : \text{Fe}^{3+}). \quad (1)$$

A similar relation holds for $(IS:\text{Fe}^{(3-\Delta)+})$. Extrinsic shift contributions to IS parameters are eliminated by taking differences of fitted values in Fig. 2(c), from which the electron deconfinement coefficient Δ may be obtained [45]:

$$\begin{aligned} (IS : \text{Fe}^{(2+\Delta)+}) - (IS : \text{Fe}^{(3-\Delta)+}) \\ = \{(IS : \text{Fe}^{2+}) - (IS : \text{Fe}^{3+})\}[1 - 2\Delta]. \end{aligned} \quad (2)$$

The reference integer valence separation on the right-hand side of Eq. (2) stems from typical ionic $IS:\text{Fe}^{2+}$ and $IS:\text{Fe}^{3+}$ values of sixfold coordinated oxides, for example, Fe_2OBO_3 at ambient pressure in Fig. 2(c) [2,41].

Site-centered charge localization is when $\Delta = 0$. Fully deconfined minority-spin electrons occurring at neighboring Fe^{2+} and Fe^{3+} sites with equal probability represent symmetric dimers $\Delta = 0.5$. Partially delocalized minority-spin electrons, $0 < \Delta < 0.5$, have a higher probability of being at one of the neighboring sites in an asymmetric dimer due to a barrier

to excursion, for example, from electron-lattice coupling [46,47]. If these mixed-valence cases manifest as long-range charge modulation, then these are site-centered CO ($\Delta = 0$), bond-centered CO ($\Delta = 0.5$), and an admixture (coexistence) of site- and bond-centered CO ($0 < \Delta < 0.5$). For symmetric dimers, statically fitted Mössbauer subspectra are indistinguishable and have the same $IS:\text{Fe}^{2.5+}$ of $\sim 0.7 \text{ mm s}^{-1}$ at 300 K [43,48]. The Fe_2OBO_3 reference at pressure represents this case [21]; see Fig. 2(c). For asymmetric dimers, fitted IS and B_{hf} parameters of subspectra representing $\text{Fe}^{(2+\Delta)+}$ and $\text{Fe}^{(3-\Delta)+}$ are distinguishable [44,47], as in LuFe_2O_4 -HP in Figs. 2(b) and 2(c).

Note that the frequency of electron hopping or tunneling, $\text{Fe}^{2+} \Leftrightarrow \text{Fe}^{3+}$, must be considered with respect to the time-sensing window of the probing technique. For MS, this “shutter speed” is $\sim 140 \text{ ns}$, but it is much shorter for x-ray probes and determines what valences are discerned [49,50].

III. DISCUSSION

Recovered LuFe_2O_4 -HP at 300 K in Fig. 2(b) has $IS:\text{Fe}^{(3-\Delta)+} = 0.57 \text{ mm s}^{-1}$ and $IS:\text{Fe}^{(2+\Delta)+} = 0.68 \text{ mm s}^{-1}$ for large- B_{hf} (27–32 T) and small- B_{hf} (4–22 T) subspectra suites, respectively. Comparison with Fe_2OBO_3 LP phase reference and using Eq. (2) yields $\Delta \sim 0.4$. Time-averaged valences for Fe constituents manifest as asymmetric dimers, denoted $[\text{Fe}^{(2.4)+} : \text{Fe}^{(2.6)+}]$. This emanates from comparatively fast minority-spin electron exchange, exceeding several megahertz [50]. An oxygen intermediary between neighboring Fe atoms facilitates electron exchange, $\text{Fe}^{2+} \Leftrightarrow \text{O} \Leftrightarrow \text{Fe}^{3+}$ in Fig. 1(b), involving σ bonding orbitals [42,51].

The increased valence (isomer shift) separation at 4 K ($\Delta \rightarrow 0.35$), Fig. 2(c), suggests “freezing” of electron dynamics. Minority-spin carriers tend to dwell more at one of the Fe sites in a dimer as temperature is lowered. All B_{hf} values increase as anticipated in an approach to saturation magnetization.

Figure 2(c) shows that, at *in situ* high pressures, IS values of large- and small- B_{hf} components are less separated and are reversed in relative magnitudes compared to recovered LuFe_2O_4 -HP. Nonetheless, large- and small- B_{hf} spectral components are discerned, suggestive of valence separation similar to recovered LuFe_2O_4 -HP. This pressure evolution is therefore attributed to different pressure dependencies of ferrous- and ferric-type isomer shifts [21], analogous to their different temperature dependencies in Fig. 2(c). Integer valences Fe^{2+} and Fe^{3+} and site-centered CO can definitely be discounted, although it is less clear how asymmetric the dimers are at high pressures.

We also performed electronic structure calculations of LuFe_2O_4 -HP by a full-potential linearized augmented plane wave method (WIEN2k) [52]. Cell parameters and two Lu and four Fe positions were taken from structural investigations of Damay *et al.* [19,38]. Oxygen positions were determined by structure relaxation and agreed closely with the HP recovered phase elucidated by Damay *et al.* [19]. The number of k points in the Brillouin zone was 2000 (440 points in its irreducible part). The hybrid potential was the generalized gradient approximation with the Hubbard U parameter GGA + U form [53,54], with $U = 4.61 \text{ eV}$ for $3d$ iron states. Partial majority and minority electronic density of states (DOS)

FIG. 3. (a) Partial density of minority-spin $3d$ states at Fe(1) and Fe(2) sites of the HP phase. Inset depicts minority-spin electron hopping between neighboring sites. (b) Minority-spin band structure contribution from $3d$ states near the Fermi level of each Fe site (weighting contribution to a subband represented by line thickness). The Fe(1)-Fe(2) pair contributes higher weightings in upper subbands, while Fe(3)-Fe(4) weightings dominate lower subbands. Filled minority-spin states occur at *all* Fe sites (dashed line highlights). A higher weight of filled states occurs at one site in both pairs, indicating asymmetric dimer formation.

profiles for the $3d$ states are shown in Fig. 3(a) and in Fig. S2 of the Supplemental Material [28]. To smooth the DOS plots, we used a Gaussian broadening scheme with a width of 0.04 eV.

Calculations indicate that all Fe sites in LuFe_2O_4 -HP are high spin; i.e., $3d$ majority-spin states are filled. Based on the smoothing parameter used, the gap in minority spin states is deduced to be at least ~ 100 meV, which is relatively close to that determined in electrical-transport measurements. Minority-spin $3d$ states are mainly empty, but for *all* Fe sites, there is some fraction of minority-spin states below the Fermi level as shown for Fe(1) and Fe(2) sites; see Fig. 3(a). Integration of minority-spin DOS up to the Fermi energy gives the fractional number of electrons at these sites as 0.3 and 0.7, implying asymmetric dimer formation $[\text{Fe}^{(2.3)+} : \text{Fe}^{(2.7)+}]$, compatible with our experimental findings. The electronic structure of both Fe(1)-Fe(2) and Fe(3)-Fe(4) pairs is essentially the same, although the four sites are structurally inequivalent.

To further verify dimer formation by Fe(1)-Fe(2) and Fe(3)-Fe(4) couples, we calculated the band structure in the vicinity of the Fermi level; see Fig. 3(b). Two flat subbands

above and below the Fermi level, mainly from Fe $3d$ states, are evident. Panels in Fig. 3(b) show contributions to minority-spin subbands from each Fe site (represented by subband line thickness). There are dominant upper subband contributions from Fe(1) and Fe(2) and dominant lower subband contributions from Fe(3) and Fe(4). This implies $3d$ states of iron are strongly hybridized within Fe(1)-Fe(2) and Fe(3)-Fe(4) couples. Each pair contains a common $3d$ minority-spin electron (filled states below the Fermi level), confined preferentially to one site in the pair. The sizable asymmetry of the dimers arises from the crystallographically inequivalent sites.

Asymmetric dimer formation in recovered LuFe_2O_4 -HP deduced from IS values and electronic structure calculations in Figs. 2(c) and 3, respectively, consistently confirms the admixture of site-centered ($\Delta \sim 0$) and bond-centered ($\Delta \sim 0.5$) charge confinement. Recovered LuFe_2O_4 -HP also evidences electron diffraction satellites in addition to Bragg reflections from the Pm parent unit cell [18,19]. If these Bragg satellites emanate from long-range charge modulation, then this is intermediate-state CO, because site-centered CO is precluded by Fig. 2(c). An asymmetric dimer phase is predicted to render ferroelectricity because it breaks spatial inversion symmetry [3,6].

In LuFe_2O_4 -HP, ferromagnetic spins occur in the a direction. There are also alternating ~ 3.00 Å and ~ 2.80 Å Fe-Fe distances, reminiscent of a Peierls distortion [19]; see Fig. 2(d). Along the b direction Fe-Fe distances are longer (3.29 Å) for neighboring antiferromagnetic spins [19]. The a direction has optimum Fe-O-Fe orbital overlap for Zener double-exchange involving ferromagnetic spins; see Figs. 1(b) and 2(d) [11]. It is the geometrically and energetically favored pathway for σ -bonded minority-spin electron deconfinement and consequent dimer formation [55,56].

Corroborative temperature-dependent resistivity-pressure data in Fig. 2(e) show that the pressure dependence of ρ in LuFe_2O_4 -HP is much lower than in the LP phase. Thus, charge carrier characteristics change at the LP $\rightarrow$ HP structural transition. The spectral analysis in Figs. 2(b) and 2(c) suggests minority-spin electron hopping $\text{Fe}^{2+} \leftrightarrow \text{Fe}^{3+}$ involving nearest neighbors in LuFe_2O_4 -HP, as shown in Fig. 2(d). This hopping is prone to a Jahn-Teller induced lattice polarization barrier [4,13]. Consequently, LuFe_2O_4 -HP resistivity data were analyzed as thermally activated small polaron motion $\rho/T \propto \exp(W_H/k_B T)$, valid for $T > \theta_D/2$, where W_H is the thermal activation barrier for hopping, and $\theta_D/2$ is half the Debye temperature 150–200 K [57]. Best fits to linearized data from 180 to 320 K [Fig. 2(e)] yield $W_H = 230$ meV at ~ 25 GPa and ~ 270 meV in decompressed (recovered) LuFe_2O_4 -HP at ~ 1 GPa.

IV. CONCLUDING REMARKS

$[\text{Fe}_2\text{O}_4]_\infty$ bilayer substructures persist in LuFe_2O_4 -HP, and a rectangular network of Fe atoms occurs [19]; see Fig. 1(b). Nearest-neighbor Fe-Fe interatomic distances both within and between sheets of a bilayer are smaller than respective distances in the triangular networked LP phase. Evidently, site-centered CO is incompatible with this rectangular network and its reduced interatomic spacings. This triggers back-and-forth minority-spin electron migration between neighboring Fe sites

and the formation of asymmetric dimers $[\text{Fe}^{(2.4)+} : \text{Fe}^{(2.6)+}]$. This is a lower repulsive energy situation than in statically configured Fe^{3+} and Fe^{2+} valences of site-centered CO [47]. A superstructure of asymmetric dimers rationalizes the additional charge modulation (Bragg satellites) discerned in electron diffraction experiments of recovered HP phase at 300 K. Asymmetric dimers in the metastable HP magnetic phase of LuFe_2O_4 potentially manifest as intermediate-state CO at ambient conditions. This yields the requisite inversion-symmetry breaking for CO electronically-invoked ferroelectricity in the metastable HP polymorph.

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 - [27] The magnetic hyperfine field is a result of the nuclear Zeeman splitting of ^{57}Fe nuclear levels due to an internal magnetic field at the nucleus. The primary contribution to this internal field, B_{hf} , is from the contact term. This is the effect of d -electron spin polarization on the s -electrons. Other contributions to B_{hf} are from comparatively small dipolar and orbital terms. In ferric iron, the orbital and dipolar terms are negligibly small. The orbital term in ferrous iron may be appreciable and of opposite sign to the contact term. Hence, B_{hf} in ferrous compounds tends to be much smaller than in ferric compounds. At well below the magnetic ordering temperatures, B_{hf} in ferric oxides is typically clustered around ~ 50 T, whereas in ferrous compounds, it may range from near zero to 44 T, depending on the orbital configuration.
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