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Enhancement and Inhibition of Spontaneous Photon Emission by Resonant Silicon Nanoantennas

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Substituting noble metals for high-index dielectrics has recently been proposed as an alternative strategy in nanophotonics to design broadband optical resonators and circumvent the Ohmic losses of plasmonic materials. In this paper, we demonstrate that subwavelength silicon nanoantennas can manipulate the photon emission dynamics of fluorescent molecules. In practice, we show that dielectric nanoantennas can both increase and decrease the local density of optical states at room temperature, a process that is inaccessible with noble metals at the nanoscale. Using scanning probe microscopy, we analyze quantitatively, in three dimensions, the near-field interaction between a 100-nm fluorescent nanosphere and silicon nanoantennas with diameters ranging between 170 and 250 nm. Associated with numerical simulations, these measurements indicate increased or decreased total spontaneous decay rates by up to 15% and a gain in the collection efficiency of emitted photons by up to 85%. Our study demonstrates the potential of silicon-based nanoantennas for the low-loss manipulation of solid-state emitters at the nanoscale and at room temperature.

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I. INTRODUCTION

Broadband subwavelength optical resonators, or nanoantennas, have the ability to enhance the luminescence decay rate and brightness of solid-state emitters at room temperature by increasing the local density of optical states (LDOS) [1–8]. Generally based on noble metal nanoparticles exhibiting surface-plasmon polaritons, these resonators feature Ohmic losses that locally quench spontaneous photon emission and reduce fluorescence lifetimes [2,3,8]. Importantly, these nonradiative processes prevent plasmonic antennas from inhibiting spontaneous emission in their vicinity, as the LDOS can never be lower than the one found in vacuum. Even when plasmonic nanoantennas feature a dark mode weakly coupled to the far field, the nonradiative decay channels dominate and increase the decay rate of a coupled emitter [9,10]. In contrast, high-quality-factor dielectric cavities can exhibit a reduced LDOS to lengthen the excited-state lifetime of narrowband emitters such as isolated atoms [11] or solid-state light sources at cryogenic temperatures [12–15].

High-index dielectric nanoparticles featuring broadband low-order Mie resonances have been recently proposed as alternatives for the design of optical nanoantennas with weak Ohmic losses and strong magnetic modes [16–24], as well as enhanced local fields [25–29]. In particular, several theoretical studies have proposed using subwavelength-sized silicon resonators to accelerate spontaneous decay processes from isolated emitters [30–33]. In this work, we experimentally demonstrate that silicon nanoantennas can be used to resonantly enhance but also to inhibit spontaneous photon emission in fluorescent molecules at room temperature because of their low nonradiative losses. Furthermore, we show that these resonators increase the measured fluorescence signal by enhancing the collection efficiency of emitted photons through a transparent substrate, in excellent agreement with numerical simulations.

In practice, the interaction between a silicon nanoantenna and solid-state emitters is fully characterized by tuning the position of a fluorescent nanosphere with nanometer precision in the vicinity of an e-beam-fabricated silicon disk using scanning probe microscopy. The continuous
monitoring of the emission intensity and excited-state lifetime of the molecules then provides a three-dimensional map of the LDOS [34–36]. By experimentally showing how subwavelength-sized Si nanoantennas manipulate and engineer spontaneous light emission with low Ohmic losses, this study highlights the potential of dielectric resonators for nanoscale photon management.

II. EXPERIMENTAL RESULTS AND DISCUSSION

A. Principle of the experiment

To control the relative axial and lateral position of solid-state emitters with respect to a silicon nanoantenna, a 100-nm-diameter fluorescent sphere (Invitrogen Red FluoSpheres) is grafted at the extremity of a tapered optical fiber, which is mounted on a tuning fork to control its height using a shear-force feedback loop. A Si nanoantenna is then scanned in three dimensions using a piezoelectric stage [Fig. 1(a)]. Nanometer-scale silicon disks are fabricated by electron-beam lithography and reactive ion etching in a 105-nm-thick amorphous Si film deposited on a 1-mm-thick silica substrate (see Appendix A). Figures 1(b) and 1(c) show scanning-electron-microscopy (SEM) images of two nanoantennas featuring base diameters of 170 ± 5 and 250 ± 5 nm, respectively. The etching process introduces a typical 10°–15° clearance angle between the base and the top of the fabricated silicon disks. In practice, the sample consists of 20 arrays of 100 Si disks with diameters ranging from 110 to 300 nm (in increments of 10 nm).

Choosing a film thickness close to 100 nm and tuning the nanoantenna diameters between 100 and 300 nm allows us to redshift the resonance wavelength of the first-order Mie modes (induced electric and magnetic dipoles) over the entire visible to near-infrared range [37–42]. This spectral evolution is evidenced in Figs. 1(d) and 1(e), which present a dark-field optical microscopy image of the entire sample and scattering spectra of selected Si nanoantennas, respectively. When varying the diameter from 110 nm (upper left) to 200 nm (second row right), the scattered light increases in intensity and shifts from the blue to the red part of the spectrum. As shown in Fig. 1(e), larger resonators (lowest two rows) feature quadrupolar resonances in addition to their dipolar modes, leading to broader scattering spectra and larger cross sections.

The absorption and emission spectra of the fluorescent nanosphere are centered on 580 and 610 nm, respectively [see Fig. 2(a)]. In order to analyze Si nanoantennas in and out of resonance with the emitters, we consider two disk diameters of 170 and 250 nm. As shown in Fig. 2(a), the smaller nanoantenna features a maximum scattering cross section corresponding to induced electric and magnetic dipoles [43] that overlap with the maximum of the fluorescence spectrum of the nanosphere. On the other hand, the dipolar and quadrupolar resonances of the 250-nm-diameter Si nanoantenna are centered on 550 and 710 nm, respectively, with a local minimum of the scattering cross section around 610 nm. These spectral properties are in very good agreement with the theoretical scattering cross sections shown in Fig. 2(b) obtained using finite-difference time-domain simulations (FDTDs) with the disk dimensions measured with SEM, the dielectric constant of the amorphous Si film estimated in ellipsometry, and considering a 10-nm-thick silica layer on the surface of the resonator due to the etching process [44], as well as an infinite silica substrate.
\[ \tau \]

The lifetime and emission count rate of the nanosphere are measured while scanning the Si nanoantenna in three orientations of the molecular transition dipoles and the spatial positions of approximately 10^3 emitters in the 100-nm-diameter sphere [34].

Figure 2(c) shows the evolution of the excited-state decay rate as a function of the axial distance between the fluorescent nanosphere and the 170-nm-diameter (red data points and solid line) or 250-nm-diameter (blue data points and solid line) Si nanoantennas.

\[ \Gamma \]

To confirm the distance dependence of the emitter-resonator interaction, we simulate in Fig. 2(d) the orientation-averaged decay-rate enhancement for an emitter in the center of the polystyrene sphere as a function of the axial distance (z axis) to the resonator. FDTD simulations are performed for three orthogonal electric dipole sources emitting at 610 nm, at the center of a 100-nm-diameter sphere with a refractive index of 1.57 and with the silicon disk and substrate properties used for the theoretical spectra of Fig. 2(b) (see Appendix B). In particular, the shear-force feedback control of the tapered fiber and the grafting process of the polystyrene sphere produces a slight offset of the minimum axial separation between the fluorescent nanosphere and the nanoantenna surface, which can be estimated from the simulations. The \( z \) dependence of the lifetime oscillations simulated in Fig. 2(d) is in good agreement with the experimental data for both resonator diameters when considering a minimum of 35 \( \pm \) 5 nm for the axial separation between the polystyrene sphere and the resonator. Furthermore, compared to the experimental conditions, the FDTD simulations neglect spatial averaging over the dimensions of the polystyrene sphere. As further illustrated in the following paragraphs, the simulations considering a single-orientation-averaged emitter provide a good agreement with the experimentally measured spatial
dependences of the fluorescence decay rate and intensity, although with higher amplitudes.

C. Enhanced fluorescence collection efficiency

The nanoscale dimensions of broadband silicon nanoantennas allow for a local modulation of fluorescence emission in three dimensions. To illustrate this, we plot in Fig. 3 the measured and simulated evolution of the fluorescence count rate when scanning the fluorescent nanosphere laterally over the 170- or 250-nm Si nanoantenna, with an estimated fixed axial separation of 120 nm with respect to the top of the resonator. We observe in Fig. 3(a) that the collected fluorescence intensity increases by 12% when the sphere is above the on-resonance silicon nanoantenna, compared to the count rate of the sphere without the antenna (at the same axial separation from the silica substrate). In contrast, the fluorescence intensity reduces by 40% on top of the 250-nm-diameter disk. When increasing the lateral separation between the nanosphere and the Si nanoantenna, the collected intensity reduces by up to 38% and 60% for the smaller and larger disk, respectively, before recovering the intensity of the fluorescent sphere without the resonator at distances larger than 700 nm.

In order to understand these results, we simulate two-dimensional fluorescence intensity maps. The simulations require the computation of the excitation intensity at 560 nm as well as the emitter quantum yield and photon-collection efficiency at 610 nm. The collection efficiency is estimated by taking into account the silica substrate and the limited NA of the microscope objective while the initial quantum yield of the fluorescent emitter is set to 1. The simulated fluorescence intensities in the sample plane given in Figs. 3(c) and 3(d) are in very good agreement with the experimental maps of Figs. 3(a) and 3(b). As mentioned above, the simulations do not consider the spatial distribution of fluorescent molecules in the sphere, which can account for the overestimated count-rate increase on the 170-nm-diameter nanoantenna (the reduction with the off-resonance antenna is in good agreement with the experimental data). However, the numerical simulations allow the separate analysis of the excitation intensity, the luminescence quantum yield, and the photon-collection efficiency [Figs. 4(a)–4(c)]. As shown in Fig. 4(a), a 170-nm-diameter disk provides a 20% enhancement of the excitation intensity but not the off-resonance Si antenna. More importantly, the interaction between the emitter and a resonant Si nanoantenna leads to a 40% increase in the collection efficiency (from 10% to 14%) compared to the case of the fluorescent sphere without a resonator [Fig. 4(b)]. Inversely, an off-resonance antenna reduces the collected photon flux by 20%. As shown in Fig. 4(d), the influence of the resonator size on the relative phase of the induced dipoles in the Si nanoantenna leads to changes in the angular radiation pattern in the silica substrate and, thus, to a reduced collection efficiency for the off-resonance case. On the other hand, the quantum yield of the emitter is similar in the vicinity of both antennas [Fig. 4(c)] with a 10% reduction because of the imaginary part of the dielectric constant of amorphous silicon. Finally, the reduced collected fluorescence intensity at distances of 200–300 nm from the resonators seen in Fig. 3 is related both to a reduction of the excitation intensity and to a lower percentage of photons emitted in the silica substrate.

D. Nanoscale enhancement of spontaneous emission

While Fig. 3 shows how silicon nanoantennas can modulate the collected fluorescence intensity, it does not demonstrate their ability to engineer the LDOS in three dimensions at the nanoscale. This is because with a fluorescent sphere located at $z = 120$ nm, the lifetime of the emitters is similar for both resonator sizes [see Figs. 2(c) and 2(d)] and its fluctuations remain within the experimental error when scanning the Si nanoantenna laterally. We, thus, reduce the axial separation between the fluorescent nanosphere and the 170-nm-diameter Si antenna down to 35 nm, which is the lowest value allowed by the shear-force feedback loop applied to the tapered optical fiber. At these short distances, a strong nanoscale modulation of the excited-state decay rate is observed [Figs. 5(a) and 5(b)]. In particular, both experimental and simulated values for the 2D spatial fluctuations of $\Gamma$ reveal an enhancement region centered on the nanoantenna with a typical size of 100 nm [Figs. 5(a) and 5(b)]. Furthermore, by radially averaging the spatial modulations of the
fluorescence decay rate in Fig. 5(a), the line scan of Fig. 5(c) shows a 15% increase of $\Gamma$ at the center of the resonator ($x = 0$), which is in excellent agreement with the simulations provided in Fig. 5(d).

As a reference, Fig. 5(c) also shows the spatial evolution of the total decay rate when the axial separation between the sphere and the 170-nm nanoantenna is $z = 120$ nm. As already mentioned, the lifetime fluctuations are within the experimental error. Nevertheless, the fluorescence lifetime changes measured in Fig. 5(c) feature significantly weaker fluctuations as compared to the simulations of Fig. 5(d). This is again due to the spatial averaging of the LDOS changes over the fluorescent sphere diameter, which is neglected in the simulations. Importantly, both experiments and simulations for $z = 35$ nm [Figs. 5(c) and 5(d)] indicate a reduced LDOS at a lateral distance of $x \approx 120$ nm, demonstrating that the interference between the source dipoles and the dipoles induced in the Si nanoantenna lead to enhanced and inhibited spontaneous emission in the sample plane similar to what is observed along the $z$ axis in Fig. 2.

While changes in the axial separation between the sphere and the Si antenna strongly modify the total decay rate of the emitters, its influence on the measured fluorescence count rates is much weaker, as shown on Figs. 5(e) and 5(f). In particular, the solid red lines in Figs. 5(e) and 5(f) correspond to the radially averaged data of Figs. 3(a)–3(c). Both experimental data and simulations indicate similar damped spatial oscillations: the measured intensity is enhanced above the resonator but decreases at a lateral distance of about 200 nm before increasing again at larger spacings. As in Fig. 3, the simulated fluorescence count rate exhibits spatial fluctuations that are in good agreement with the experimental data, albeit with weaker amplitudes. Although lowering the position of the fluorescent sphere along the $z$ axis does not significantly influence the collected fluorescence signals shown in Figs. 5(e) and 5(f), numerical simulations indicate that it increases the photon-collection efficiency but reduces the excitation enhancement and the fluorescence quantum yield [Figs. 6(a)–6(c)]. In particular, reducing the axial distance of the fluorescent sphere above the resonator from 120 to 35 nm leads to a twofold increase in the collection efficiency enhancement [from 40% to 80%; Fig. 6(b)] but decreases the excitation enhancement [from a 20% gain to a 10% reduction; Fig. 6(a)] and reduces the quantum yield [from 90% to 80%; Fig. 6(c)]. The enhanced collection efficiency and decreased quantum yield are expected when increasing the coupling between emitters and a resonator that radiates efficiently into the silica substrate but features some Ohmic losses. However, the reduced excitation intensity is more surprising. As shown
Furthermore, good agreement between electrodynamics simulations and experimental data proves that the resonant behavior of nanometer-scale silicon particles allows them to modulate the emission and far-field collection of fluorescence photons and makes them excellent candidates for the design of optical nanoantennas with low Ohmic losses. In particular, the simulated fluorescence intensity enhancements clearly show that the gain in the measured photon count rates for the resonant 170-nm-diameter Si nanoantenna is dominated by enhanced collection efficiency (by up to 85%) and not by changes in excitation probability or by reciprocity of the local density of optical states (up to a 15% increase in the total spontaneous decay rate). This result highlights not only the interest of dielectric antennas to control the directivity of emitted photons but also the need to improve the near-field coupling between emitter and resonator. The larger field confinements expected in coupled silicon particles [26,29,30,32] associated with the top-down nanofabrication [52] or self-assembly strategies [6–8] developed for plasmonic nanoantennas will allow the optimized interaction between solid-state light sources and dielectric-based resonators.

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D. B. and M. M. contributed equally to this work.

**APPENDIX A: EXPERIMENTAL DETAILS**

1. **Sample fabrication**

A thin film of amorphous silicon is obtained by e-beam evaporation of a solid source on a 1-mm-thick UV-grade fused silica substrate in high vacuum. After chemical cleaning, the substrate is introduced in the growth chamber and undergoes a smooth plasma cleaning before deposition of the silicon layer at room temperature. The substrates are then transferred in a tubular furnace for a 1-h annealing step at 600°C in vacuum to increase the density of the silicon.
thin film. The thickness (105 nm) and optical constants of the annealed films are determined \textit{ex situ} by ellipsometry ($n = 4.55 + 0.34i$ at 610 nm).

After oxygen plasma cleaning to remove organic residues and increase the wettability of the silicon surface, the annealed amorphous film is covered with a 60-nm-thick layer of $e$-beam resist (PMMA diluted at 2\% in ethyl-lactate, AR-P 679.04 from Allresist, Germany) by spin coating at 6000 rpm. After a soft baking process (10 min at 170°C on a hot plate), a 30-nm-thick conductive resist is spin coated on the $e$-beam resist (SX AR-PC 5000/90.1 from Allresist) and baked 2 min at 85°C. The $e$-beam lithography is then performed by a field electron gun SEM system (PIONEER from RAITH). After exposure, the conductive resist is removed with deionized water, and the sample is dried under nitrogen. The development is performed for 60 s in a commercial solution [AR 600-55 from Allresist containing methyl isobutyl ketone and propan-2-ol (IPA)]. The development is stopped in IPA, and the sample dries under nitrogen. A 15-nm-thick nickel mask is then evaporated under vacuum (Auto 306 tool from Edwards). Nickel is used as the mask material because it is resistant to fluorine attacks [54]. A lift-off process in acetone using an ultrasonic bath removes the $e$-beam resist and the excess of nickel in areas of the sample that are not irradiated with the electron beam.

The nickel mask pattern is transferred into the amorphous thin silicon film by reactive ion etching (MG200, Plassys, France). The unprotected areas are etched by the plasma of a gas mixture containing SF$_6$, O$_2$, and CHF$_3$ (respective fluxes of 20, 8, and 5 sccm) alternated with a pure O$_2$ plasma. Excited SF$_6$ efficiently etches silicon, and the two key roles of the CHF$_3$ gas are to passivate the vertical feature walls [54] and to etch the silicon oxide that forms during the process on the very reactive amorphous silicon surface. This configuration allows a very good etching anisotropy and nearly vertical walls for the fabricated structures. The rf power used to excite the plasma is 120 W for the gas mixture and 15 W for pure O$_2$. This sequence is repeated as many times as required for the total etching of the thin silicon film. The nickel mask is removed by immersion in an aqueous FeCl$_3$ solution (3 min). Then the sample is rinsed by deionized water and dried under nitrogen flow.

FIG. 6. Theoretical evolution of the excitation intensity enhancement in the center of the fluorescent sphere (a), of the fluorescence collection efficiency enhancement (b), and of the orientation-averaged quantum yield (c), as a function of the lateral distance between the nanosphere and the 170-nm-diameter Si nanoantenna, with an axial separation of 35 nm (green solid line) or 120 nm (red solid line). (d) Spatial distribution of the electric field intensity, in a longitudinal plane containing the 170-nm-diameter silicon antenna and the fluorescent sphere, for a 560-nm excitation wavelength (axial separation of 35 nm). (e) Simulated decay rate enhancement for a longitudinal (solid green line), a transverse (dashed green line), or three orientation-averaged dipoles (solid black line) in the center of the sphere as a function of the lateral distance between the nanosphere and the 170-nm-diameter Si antenna, with an axial separation of 35 nm.
2. Scanning electron microscopy and dark-field microscopy

After all optical experiments are carried out, the sample featuring silicon resonators is covered by a 5-nm-thick titanium layer and imaged with the GEMINI scanning electron microscope (1-nm resolution) of a Zeiss Auriga 60 FIB-SEM. Dark-field images and spectra are measured in an inverted microscope (IX71, Olympus) coupled to a color CCD (Quicam, Roper) and to a fiber-coupled (50-μm core diameter) imaging spectrometer (Acton SP300 with PIXIS 100 CCD detector, Princeton Instruments). White light from a 100-W halogen lamp is focused on the sample using a (0.8–0.92)-NA dark-field condenser, and the scattered light is collected through the 1-mm-thick silica substrate with a (60 × 0.7)-NA objective. The dark-field spectra of individual silicon nanoantennas are measured by aligning a single disk in the confocal volume corresponding to the 50-μm fiber core of the imaging spectrometer with a 1-s acquisition time. A background spectrum is measured with the same acquisition time on an empty area of the sample. The background is subtracted to the measured spectra before correction according to the wavelength-dependent illumination and detection.

3. Near-field optical measurements

The sample is mounted on a sample-scanning inverted confocal microscope (Olympus, IX71). A fluorescent polystyrene sphere (Invitrogen Red FluoSpheres, 100 nm diameter) is grafted at the extremity of a tapered optical fiber (100-nm tip curvature) that is mounted on a tuning fork to meter) is grafted at the extremity of a tapered optical fiber (100-nm tip curvature) that is mounted on a tuning fork to control its height using a shear-force feedback loop [35,36]. Pulsed laser excitation at 560 nm (Fianium SC450, 40-ps pulses, 10-MHz rate) is focused on the sample through the silica substrate with a 0.7-NA 60× objective. Fluorescence photons are collected through the same objective and are separated from the excitation with a dichroic mirror and a high-pass filter (λ > 580 nm). Time-resolved photon detection is performed with a time-correlated single-photon counting system (PDMR detector by Micro Photon Devices combined with a PicQuant HydraHarp 400 acquisition board), which allows one to simultaneously map the fluorescence intensity and decay rate (instrument response function with a (190 ± 30)-ps full width at half maximum). The tapered optical fiber supporting the fluorescent sphere is held by shear-force feedback at a constant distance of approximately 20 nm to the surface of a nanostructured sample that is scanned in the (x, y) plane (control electronics by RHK). The pixel size in the 2D intensity and decay rate maps is 30 × 30 nm² with a 1-s acquisition time. The nanoscale drift of the experimental setup during the total measurement time of a 2D map (1000 s) explains the observed astigmatism in Figs. 3 and 5.

4. Data analysis

To estimate the data and error bars of Fig. 2(c), the evolution of the decay rate when changing the height of the fluorescent nanosphere is measured four (resp. three) consecutive times in the vicinity of the 170-nm-diameter (resp. 250 nm) Si nanodisk. Each measurement consists in a series of ten successive acquisitions (10-s integration time) ranging from 35 to 300 nm. Figure 7 shows typical fluorescence lifetime measurements close to the on-resonance and off-resonance nanoantennas or with a large axial distance of z = 300 nm. The experimental results presented in Fig. 2(c) are the estimated mean decay rates when fitting these data with a single-exponential decay, while the error bars correspond to the 95% confidence interval (plus or minus twice the standard error). The mean decay rates are normalized with respect to the mean value estimated at z = 300 nm.

The data of Figs. 5(c) and 5(e) are estimated by radially averaging the decay rate and intensity maps of Figs. 3(a) and 5(a). The center of the disk is first identified using the corresponding intensity map. All possible line scans going from the center of the disk to the edge of the map are then plotted, resulting in a set of measurements of the emission decay rate and intensity as a function of the distance from the resonator center in the sample plane. The experimental values presented in Fig. 5 are, for each distance, the mean decay rates and intensities, which are normalized with respect to the value estimated at x = 650 nm. The error bars in Fig. 5(c) for each estimated decay rate correspond to the 95% confidence interval.

**APPENDIX B: NUMERICAL SIMULATIONS**

Numerical simulations are performed using an in-house FDTD code [55]. The model takes into account the
nanoantenna dimensions estimated in SEM (105 nm thickness and 170 or 250 nm diameters), the dielectric constants of the amorphous silicon thin film measured by ellipsometry, and an infinite silica substrate. A 10-nm-thick layer of silicon dioxide \( (n = 1.5) \) is added around the silicon resonator to account for the oxidation of the semiconductor matrix after the RIE treatment. The fluorescent molecules in the polystyrene sphere are modeled by three dipoles oriented along the \( x \), \( y \), and \( z \) axes, in the center of a 100-nm sphere with a dielectric index of 1.57. For simplicity, the tapered optical fiber is not taken into account. The simulations consider a volume spanning \( \pm 2 \mu \text{m} \) in \( x \), \( y \), and \( z \) around the nanostructures. The antennas are surrounded by air (optical dielectric index of 1). All six boundaries of the computational volume are terminated with convolution-periodic matching layers to avoid parasitic unphysical reflections around the nanostructures. The nonuniform grid resolution varies from 25 nm, for areas at the periphery of the simulations, to 5 nm for the region in the immediate vicinity of the antennas.

Scattering spectra are simulated by impinging an unpolarized plane wave on the silicon nanoantenna from the upper air side of the sample. The scattered light is then collected in the silica lower half-space to account for the transmission configuration of the dark-field measurement.

Local excitation enhancement in the center of the sphere is estimated by impinging a plane wave propagating perpendicularly to the sample plane from the silica substrate on the sample. Changes in the total decay rate are obtained by computing the orientation-averaged changes in power dissipated by the three orthogonal dipoles in the center of the polystyrene sphere when changing its \( x \), \( y \), and \( z \) position with respect to the silicon antenna. The orientation-averaged radiative-decay-rate evolution due to the resonator is estimated by projecting the Poynting vector produced by each of the three source dipoles on the surface of the computation volume. Finally, the collection efficiency is calculated by integrating the projected Poynting vectors on the limited numerical aperture of the microscope objective (solid angle with a 56° width in the infinite silica substrate). For all calculations, we consider that the source electric dipoles have an intrinsic quantum yield of 1 and normalize the changes in local field distribution, as well as dissipated and radiated powers, with respect to a 100-nm polystyrene sphere with a given \( (z + 105 \text{ nm}) \) height above the silica substrate, without the silicon antenna. Simulations are performed for each axial \( (z \text{ axis}) \) and lateral \( (x \text{ axis}) \) position of the fluorescent nanosphere with respect to the center of the Si nanoantenna. The minimum considered axial distance (35 nm) is limited by the position of the fluorescent nanosphere on the tapered optical fiber and by the applied shear-force feedback loop during optical measurements. For shorter axial distances, strong variations of the excitation field and of the LDOS over the dimension of the nanosphere require simulating the emission properties of source dipoles at different positions of the nanosphere and not simply in its center.


[29] R. Regmi et al., All-dielectric silicon nanogap antennas to enhance the fluorescence of single molecules, Nano Lett. 16, 5143 (2016).


