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Structure and reactivity of As(III)- and As(V)-rich schwertmannites and amorphous ferric arsenate sulfate from the Carnoulès acid mine drainage, France: Comparison with biotic and abiotic model compounds and implications for As remediation

Fabien Maillot^a, Guillaume Morin^{a,*}, Farid Juillot^a, Odile Bruneel^b,
Corinne Casiot^b, Georges Ona-Nguema^a, Yuheng Wang^a, Sophie Lebrun^a,
Emmanuel Aubry^c, Gilberto Vlaic^d, Gordon E. Brown Jr.^{e,f}

^a *Institut de Minéralogie et de Physique des Milieux Condensés (IMPMC), UMR CNRS 7590, UPMC, 4 Place Jussieu, 75252 Paris Cedex 05, France*

^b *Laboratoire HydroSciences Montpellier (HSM), UMR 5569 CNRS-IRD-UMI&2, Université Montpellier 2, Place E. Bataillon, CC MSE, 34095 Montpellier Cedex 05, France*

^c *Biogéochimie et Ecologie des Milieux Continentaux (Bioemco), UMR 7618 Université Paris 6, INRA, INAPG, CNRS, ENS, ENSCP, Case 120, Tour 56, couloir 56-66, 4^{ème} étage, 4 place Jussieu, 75252 Paris cedex 05, France*

^d *Department of Chemical Science, University of Trieste, Trieste I-34127, Italy*

^e *Surface & Aqueous Geochemistry Group, Department of Geological & Environmental Sciences, Stanford University, Stanford, CA 94305-2115, USA*

^f *Department of Photon Science and Stanford Synchrotron Radiation Lightsource, SLAC National Accelerator Laboratory, 2575 Sand Hill Road, Menlo Park, CA 94025, USA*

Abstract

Poorly ordered nanocrystalline hydroxysulfate minerals of microbial origin, such as schwertmannite, $\text{Fe}_8\text{O}_8(\text{OH})_6\text{SO}_4$, are important arsenic scavengers in sulfate-rich acid mine drainage (AMD) environments. However, despite the fact that As(III) and As(V) have been shown to sorb on schwertmannite, little is known about the actual mechanism of arsenic scavenging processes after microbial Fe(II) oxidation in AMD environments. The major focus of the present study is to determine the molecular-level structure of poorly ordered As(III) and As(V) bearing Fe oxyhydroxysulfate minerals from the Carnoulès AMD, France, which exhibits exceptional As(III) concentrations. Powder X-ray diffraction (XRD) and extended X-ray absorption fine structure (EXAFS) spectroscopy were used to compare field samples with a large set of synthetic analogs prepared via biotic or abiotic pathways, with As/Fe ratios typical of minerals and mineraloids ranging from nanocrystalline schwertmannite to amorphous hydroxysulfate compounds. Our results yield further evidence for the poisoning effect of As(V) in limiting the nucleation of schwertmannite. For initial dissolved As(V)/Fe(III) molar ratios ≥ 0.2 , amorphous Fe(III)–As(V) hydroxysulfate forms, with a local structure consistent with that of amorphous ferric arsenate. EXAFS data for this amorphous material are consistent with corner-sharing FeO_6 octahedra to which AsO_4 tetrahedra attach via double-corner ^2C linkages. For As(V)/Fe(III) molar ratios lower than 0.2, As(V) binds to schwertmannite via ^2C surface complexes. In contrast with the As(V)-containing samples, As(III) has a lower affinity for schwertmannite following its nucleation, as this mineral phase persists up to an initial As(III)/Fe(III) molar ratio of 0.6. EXAFS data indicate that during

* Corresponding author.

E-mail address: guillaume.morin@impmc.upmc.fr (G. Morin).

the precipitation process, As(III) forms dominantly 2C surface complexes on schwertmannite surfaces, likely on the sides of double-chains of Fe(III)(O,OH)_6 octahedra, with a smaller proportion of edge-sharing 2E surface complexes at the apexes of these chains. Importantly, dissolved As(V) concentrations in contact with As(V)–schwertmannite or ferric arsenate were found to be ~ 10 times lower than dissolved As(III) concentrations in contact with As(III)–schwertmannite for similar As/Fe ratios in the solid phase. Consequently, remediation of As-rich AMD environments is greatly improved by oxidation of As(III) to As(V).

1. INTRODUCTION

Arsenic is toxic to humans and is naturally released at Earth's surface from volcanic activity and from the chemical weathering of rocks, as well as from mining and industrial activities (Vaughan, 2006). Thanks to detailed investigations of such As-impacted environments as well as extensive laboratory studies, much progress has recently been made in understanding the relationship between arsenic crystal chemistry and its speciation and distribution at Earth's surface. Among other major processes affecting the fate of arsenic in the environment, arsenic scavenging by iron-bearing minerals has been shown to play an important role in delaying the long-term transfer of this element to surface- and groundwaters. Arsenic trapping associated with iron-bearing minerals has been observed in natural and polluted soils as well as in acid mine drainage (AMD) systems and hot springs (Morin and Calas, 2006; and references therein). Among these systems, AMDs have received much attention because they can locally produce high levels of dissolved arsenic in surface waters. Indeed, in AMDs, acidic Fe^{2+} - and SO_4^{2-} -rich waters originating from the oxidation of sulfide-bearing rocks can often carry high concentrations of dissolved As(III) resulting from dissolution of arsenopyrite or As-rich pyrite. In such sulfate- and iron-rich acidic waters, arsenic is mainly trapped by sorption and/or coprecipitation, with ferric oxyhydroxysulfates and oxyhydroxides formed via the oxidation of dissolved Fe^{2+} (e.g., Jambor and Blowes, 1994; Nordstrom and Alpers, 1999). This iron oxidation reaction has been demonstrated to often be catalyzed by the metabolic activity of acidophilic microorganisms such as *Acidithiobacillus ferrooxidans* (e.g., Leblanc et al., 1996; Duquesne et al., 2003; Morin et al., 2003). Moreover, oxidation of As(III) to As(V), which is less toxic than the reduced form and less soluble in low pH waters, can be catalyzed by arsenite-oxidizing bacteria such as *Thiomonas* sp. (Battaglia-Brunet et al., 2002; Morin et al., 2003; Arsène-Ploetze et al., 2010; Bruneel et al., 2011). Such microbial oxidation has recently been suggested to play a key role in the natural attenuation of arsenic in the heavily contaminated Carnoulès AMD (Egal et al., 2010; Bruneel et al., 2011). Oxidation of soluble Fe(II) into relatively insoluble Fe(III) leads to over-saturation of water with respect to Fe(III)-bearing minerals. In acidic sulfate-rich waters, iron sulfates and oxyhydroxides form, following the sequence jarosite $[\text{KFe}_3(\text{SO}_4)_2(\text{OH})_6]$, schwertmannite $[\text{Fe}_8\text{O}_8(\text{OH})_6\text{SO}_4]$, ferrihydrite $[\text{Fe}_3(\text{OH})_8 \cdot 4\text{H}_2\text{O}]$, and goethite $[\alpha\text{FeOOH}]$ with increasing pH (Bigham et al., 1996a,b; Foster et al., 1998; Savage et al., 2000; Burgos et al., 2012). These phases

may be crystalline (jarosite, goethite) or poorly ordered (schwertmannite, ferrihydrite). Indeed, nanocrystalline or amorphous iron oxyhydroxides and hydroxysulfates are kinetically favored with respect to crystalline phases upon rapid Fe(II) oxidation because high oversaturation with respect to Fe(III)-bearing minerals leads to high nucleation rates. Therefore, these poorly ordered ferric minerals frequently occur as transient phases in AMD systems. They exhibit very high surface areas (several $100 \text{ m}^2/\text{g}$) and thus play a key role in the geochemical cycling of minor and trace elements, which sorb on their surfaces or are incorporated in their structures.

For instance, a number of laboratory studies have shown that As(V) sorbs onto the surface of minerals such as ferrihydrite (Pierce and Moore, 1982; Waychunas et al., 1993, 1995a; Raven et al., 1998) and goethite (Fendorf et al., 1997; Manning et al., 1998). Such sorption processes have also been found in EXAFS studies of mine tailings and AMD environments by Foster et al. (1998) and Savage et al. (2000). This latter study has shown that As(V) may be incorporated at low concentrations in jarosite $[\text{KFe}_3(\text{SO}_4)_2(\text{OH})_6]$ where it substitutes for $(\text{SO}_4)^{2-}$ oxoanions. Importantly, As(V) sorbs at high loadings onto schwertmannite $[\text{Fe}_8\text{O}_8(\text{OH})_6\text{SO}_4]$ (Burton et al., 2009), and this mineral has been identified as an As(V) scavenger in AMD systems (Fukushi et al., 2003) and has further been proposed as a water treatment sorbent in these environments (Carlson et al., 2002; Fukushi et al., 2004). In a comparison of AMD sediments with synthetic analogs using XRD, Carlson et al. (2002) showed that As(V) can also co-precipitate with ferric iron to form amorphous ferric arsenate with As/Fe ratios as high as 0.8, whereas schwertmannite forms at lower As/Fe ratios. In a previous EXAFS study of synthetic samples, Paktunc et al. (2008) proposed that the local structure of amorphous ferric arsenate consists of single chains of corner-sharing FeO_6 octahedra with bridging AsO_4 tetrahedra alternating along the chains, as in the structure of sulfate minerals such as butlerite, parabutlerite, and fibroferrite. An investigation of arsenate-rich Fe-precipitates from continental acidic hot-springs (Inskeep et al., 2004) also illustrated the strong ability of As(V) to coprecipitate with Fe(III) in acidic sulfate waters to form hydrated amorphous ferric arsenate phases. These studies have shown that $\text{H}_2\text{As(V)O}_4^-$ oxoanions retard or inhibit the formation of schwertmannite (Waychunas et al., 1995b; Carlson et al., 2002) or ferrihydrite (Paktunc et al., 2008) when the As(V)/Fe(III) molar ratio exceeds ~ 0.25 in the starting solution. However, the uniqueness of the local structure of amorphous ferric arsenate, especially the possibility of the existence of various local structures

depending on formation conditions of the ferric arsenate is still under debate (Paktunc, 2010; Jiang et al., 2010).

In addition, the As(V) to schwertmannite binding modes are still poorly documented from direct structural analysis. XRD (Carlson et al., 2002) and EXAFS (Waychunas et al., 1995b) studies have suggested that $\text{H}_2\text{As(V)}\text{O}_4^-$ oxoanions strongly bind to FeO_6 octahedra via inner sphere complexes upon precipitation of schwertmannite, which is thought to inhibit crystal growth and may explain the formation of an amorphous ferric arsenate at high As/Fe ratio (Waychunas et al., 1995a; Paktunc et al., 2008). On the other hand, the significant displacement of sulfate ions upon sorption of arsenate oxoanions on schwertmannite has led to the suggestion that arsenate could substitute for sulfate, assuming an akaganeite-like structure for schwertmannite (Bigham et al., 1990; Burton et al., 2009; Fernandez-Martinez et al., 2010). In such a structure, sulfate ions are proposed to be mainly hydrogen bonded to the iron-oxyhydroxide network within the channels and at the particle surface (Majzlan and Myneni, 2005). There is less sulfate exchange for $\text{H}_3\text{As(III)}\text{O}_3$, which is known to sorb less efficiently than $\text{H}_2\text{As(V)}\text{O}_4^-$ on schwertmannite at acidic pH (Burton et al., 2009; Paikaray et al., 2011). Despite this existing knowledge concerning the reactivity of As(III) with respect to schwertmannite in the laboratory, the mechanisms of As(III) scavenging by schwertmannite in AMD systems is still poorly documented, although the more toxic $\text{H}_3\text{As(III)}\text{O}_3$ species may be the dominant dissolved arsenic species in acidic waters of AMD systems (Casiot et al., 2003; Morin et al., 2003; Morin and Calas, 2006), and hot springs (Inskip et al., 2004). In addition, the possible influence of microbial activity and exopolymeric substances on the structure of As-bearing Fe minerals has been rarely investigated in AMD samples (Benzerara et al., 2008). For instance, a study by Egal et al. (2009) showed that the kinetics of bacterial Fe(II) oxidation can influence the nature of minerals forming in high As(III) AMD systems, with slow or fast oxidation rates leading to the formation of tooeleite [$(\text{Fe}_6(-\text{AsO}_3)_4\text{SO}_4(\text{OH})_4 \cdot 4\text{H}_2\text{O})$] or As(III)-rich schwertmannite, respectively.

The main objective of the present study is to directly determine the molecular-level structure of poorly crystalline As(III)- and As(V)-bearing Fe oxyhydroxysulfate minerals from the Carnoulès AMD system, France, which exhibits exceptionally high As(III) concentrations. Powder X-ray diffraction (XRD) and extended X-ray absorption fine structure (EXAFS) spectroscopy were used to compare the local As structural environments in these field samples with those of synthetic analogs prepared via biotic or abiotic pathways, with As/Fe ratios ranging from those in nanocrystalline schwertmannite to those in amorphous ferric arsenate compounds.

2. MATERIALS AND METHODS

2.1. AMD samples

Hydrated and loosely packed sediments were sampled in the upstream section of the acidic Reigous creek (pH 3–4.5) (Carnoulès AMD, France), which exhibits exceptionally

high concentrations of dissolved As(III) and As(V) (up to 250 mg/L dissolved total As) and of dissolved iron and sulfate (~ 0.5 –1 g/L and ~ 1 –3 g/L, respectively) (Casiot et al., 2003, 2005; Egal et al., 2010). Samples were air dried for 24 h and stored in a dry atmosphere before spectroscopic analysis. Sample ‘Sed S32F00’ was sampled 32 m downstream from the acidic spring and consists of amorphous Fe(III)–As(V) oxyhydroxysulfate with an As/Fe molar ratio of 0.63 and with an As(III)/AsTotal ratio of 0.05 as determined by electron microprobe analysis (EMPA) and X-ray absorption near edge structure (XANES) spectroscopy, respectively (Fig. S1; Morin et al., 2003). Sample ‘Sed arch0503’ was sampled ~ 200 m downstream from the acidic spring and consists of poorly ordered oxyhydroxysulfate with an As/Fe molar ratio of 0.45 and an As(III)/AsTotal ratio of 0.23, as determined using the same methods (Lebrun, 2004). Sample ‘Sed foyer0303’ was sampled ~ 1000 m downstream from the acidic spring and consists of poorly ordered schwertmannite with an As/Fe molar ratio of 0.25 and an As(III)/AsTotal ratio of 0.46, as determined using the same methods (Lebrun, 2004).

2.2. Biogenic laboratory samples

Model compound samples were prepared via microbial incubation of sterilized water from the Carnoulès AMD water with autochthonous bacterial strains: *Thiomonas* sp. (*Tm* B2 and *Tm* B3 strains; Morin et al., 2003) and *A. ferrooxidans* (*Af* CC1 strain; Duquesne et al., 2003; Egal et al., 2009). Incubations with strain *Tm* B2 or *Tm* B3 yielded amorphous As(V)–Fe(III) oxyhydroxysulfate as the single solid product, owing to the ability of these strains to catalyze the oxidation of As(III) to As(V) at acidic pH, with Fe(II) being slowly oxidized abiotically by dissolved O_2 over the 1 week experiment (Morin et al., 2003; Casiot et al., 2003). Samples obtained after incubation by strains *Tm* B2 or *Tm* B3 are characterized by As/Fe molar ratios of 0.63 and 0.8, respectively, and do not contain As(III) as determined by EMPA and XANES spectroscopy (Fig. S1; Morin et al., 2003). In contrast, incubation with strain *Af* CC1 for 5 days yields pure As(III)-rich schwertmannite due to the ability of this strain to catalyze Fe(II) to Fe(III) oxidation without significant abiotic oxidation of As(III) to As(V) (Duquesne et al., 2003; Egal et al., 2009). This sample exhibits an As/Fe molar ratio of 0.19 and does not contain As(V) as determined by EMPA and XANES spectroscopy (Egal et al., 2009).

2.3. Synthetic abiotic analogs

Abiotic model compounds were prepared according to the schwertmannite synthesis protocol from Loan et al. (2004), which was modified by adding various amounts of dissolved As(III) or As(V) in the starting solutions. This protocol relies on hydrolysis of ferric iron in acidic sulfate water at pH 3, at a temperature of 80 °C. This protocol was chosen for As-coprecipitation samples instead of the one we used for pure schwertmannite and schwertmannite sorption samples (see next paragraph) because of the absence of an H_2O_2 oxidation step, in order to avoid As(III)

oxidation. The protocol was modified by mixing appropriate volumes of 100 mM NaAsO₂ or Na₂HAsO₄ stock solutions with 100 mL of a 16 mM ferric sulfate stock solution, in order to obtain As(III)/Fe(III) or As(V)/Fe(III) dissolved molar ratios of 0, 0.025, 0.05, 0.1, 0.2, 0.4, 0.6, and 0.8. The volume of the mixture was increased to 250 mL by adding deionized water to polypropylene flasks containing the mixture, and the pH was adjusted to 3 using H₂SO₄, thus giving a starting dissolved Fe(III) concentration of 6.4 mM. Initial dissolved As concentrations were thus 0, 0.16, 0.32, 0.64, 1.28, 2.56, 3.85, and 5.12 mM. Each aliquot was then heated to 80 °C for one hour in closed flasks under magnetic stirring. Suspensions were then cooled in ice-water, centrifuged at 10,000g, and washed two times in deionized water. In order to preserve the As(III) oxidation state, all experiments with As(III) were conducted with degassed deionized water. The solids were then dried under vacuum and stored in a dry atmosphere before spectroscopic analysis. Supernatants were analyzed for dissolved As and Fe concentrations after 0.2 µm filtration using flame Atomic Absorption Spectroscopy (Solaar S, Thermo electron corporation). Chemical compositions of the solids were determined by averaging 15 EDXS analyses per sample. X-ray emission spectra were collected on thin powder samples deposited on carbon tape using a Bruker Si-drift detector mounted in a Zeiss Scanning Electron Microscope operating at 15 kV with a 60 µm diaphragm. Hematite and scorodite were used as standards for semi-quantitative analyses with the Esprit software (http://www.bruker-axs.com/esprit_software.html). The solid As/Fe molar ratio values estimated with this method were consistent with those determined from the supernatant analyses. Solid As/Fe values and corresponding standard deviations were then estimated by averaging these two sets of values.

In addition, model compounds consisting of arsenic sorbed on schwertmannite were prepared at pH 3 by reacting for 24 h 550 µM solutions of Na₂HAsO₄ or NaAsO₂ with 1 g/L of schwertmannite prepared according to the protocol of Barham (1997). For this synthesis protocol, 100 mL of a 0.1 M ferrous sulfate solution adjusted to pH 2.5 using H₂SO₄ was oxidized by an excess of H₂O₂. The schwertmannite precipitate was washed and air-dried before resuspension in deionized water for the sorption experiment. After 24 h of interaction of the schwertmannite precipitate with aqueous arsenic, the sorption samples were centrifuged and vacuum dried, and then stored under a dry atmosphere before spectroscopic analyses. Final As/Fe ratios in the solids were 0.04 and 0.09 for the As(III)- and As(V)-schwertmannite sorption samples, respectively. XANES analysis indicated that the oxidation state of arsenic was fully preserved during the sorption process (Lebrun, 2004).

2.4. XRD and EXAFS analyses

X-ray powder diffraction patterns of AMD samples and of biogenic model compounds were recorded using a PW1710 Philips diffractometer equipped with a back-mounted graphite monochromator, and using Co K α radiation. Data were recorded in step-scan mode using 0.04° 2 θ

steps and a counting time of 6–12 h per sample. XRD patterns of the synthetic model compounds were recorded using an Xpert pro PanAnalytical diffractometer equipped with an XCellerator linear detector and filtering the Co K β radiation using a Fe filter. Data were recorded in continuous mode with a resulting 2 θ step of 0.05° and counting for 3 to 6 h per sample over the 10–100° 2 θ range. Bragg peaks from the aluminum sample holder were removed from the data for samples that were too thin due to a low amount of powder.

EXAFS spectroscopy data were recorded at a temperature below 20 K in a liquid helium cryostat in order to maximize second-neighbor contributions in the EXAFS of the poorly crystalline samples studied. All Fe K-edge data were recorded in transmission detection mode using a Si(111) double-crystal monochromator at the BM29 beamline (ESRF), except for samples Sed S32F00, *Tm* B2, and *Tm* B3, that were analyzed at the D44 beamline (LURE) using a similar setup. Most of the As K-edge EXAFS data were recorded in transmission mode at the XAFS beamline (ELETTRA) using a Si(111) double-crystal monochromator, except for the *Af* CCl sample that was analyzed at the 11–2 beamline (SSRL) using a Si(220) double-crystal monochromator. Iron K-edge EXAFS data of sample Sed arch0503 were recorded in transmission mode at the FAME beamline (ESRF) using a Si(220) double-crystal monochromator. Arsenic K-edge EXAFS data of the two arsenic sorption samples and of sample Sed arch0503 were recorded in fluorescence detection mode using a Canberra Ge 30 elements detector at the FAME beamline (ESRF), using a Si(220) double-crystal monochromator.

EXAFS spectra were extracted from the raw fluorescence data using the XAFS program (Winterer, 1996) after averaging and normalization using the Athena Program (Ravel and Newville, 2005) or the SixPack Program (Webb, 2005) for SSRL data. Radial distribution functions (RDF) were obtained by Fourier transforming the $k^3\chi(k)$ EXAFS functions using a Kaiser-Bessel window, with a Bessel weight of 2.5. Least-squares fitting of the unfiltered $k^3\chi(k)$ functions was performed with the plane-wave formalism, with a S_0^2 value set to 1, using a Levenberg–Marquard minimization algorithm. Theoretical phase-shift and amplitude functions employed in this fitting procedure were calculated with the curved-wave formalism using the *ab initio* FEFF8 code (Ankudinov et al., 1998). As–O and As–Fe phase-shift and amplitude functions were extracted from the scorodite (Hawthorne, 1976) and tooeleite (Morin et al., 2007) crystal structures. For faster calculations, the threshold energy was calculated without using the self-consistent potential option in FEFF8, which explains the high ΔE_0 values observed in our fits of As K-edge EXAFS data. The fit quality was estimated using the classical reduced χ^2 (see Table 1).

3. RESULTS AND DISCUSSION

3.1. Chemistry and mineralogy of the laboratory and field samples

The chemical compositions of the solids and supernatants for the synthetic samples obtained from precipitation

Table 1

Results of shell-by-shell fitting of unfiltered EXAFS data at the Fe K-edge for the As(III) samples studied. $R(\text{\AA})$: interatomic distances; N : number of neighbors; $\sigma(\text{\AA})$: Debye Waller factor; ΔE_0 (eV): difference between the experimentally determined threshold energy and the FEFF8-calculated threshold energy, without using the self-consistent potential calculation; Standard deviations of these fit parameters are 0.02 $\text{\AA}$, $\pm 30\%$, 0.01 $\text{\AA}$, and 2 eV, respectively χ^2_{FT} is a goodness of fit estimate $\chi^2_{FT} = N_{ind}/[(N_{ind} - p)n]\sum_1^n (||FT||_{exp_i} - ||FT||_{calc_i})^2$ with $N_{ind} = (2\Delta k\Delta R)/\pi$, the number of independent parameters, p the number of free fit parameters, n the number of data points fitted, and $||FT||_{exp}$ and $||FT||_{calc}$ the experimental and theoretical Fourier transform magnitudes within the 0–8 $\text{\AA}$ R -range. Note: V As(V)/AsTotal if greater than 0.00; * fixed N value. As/Fe and S/Fe molar ratio of the solid samples are reported. n.d.: not determined.

Sample	Origin	As/Fe mol/mol	S/Fe mol/mol	$R(\text{\AA})$	N	$\sigma(\text{\AA})$	$R(\text{\AA})$	N	$R(\text{\AA})$	N	$\sigma(\text{\AA})$	χ^2_{FT}	ΔE_0 (eV)
As free	Abiotic lab sample	0.00	n.d.	1.94	3.15 FeO	0.08	3.03	1.2 FeFe	5.53	0.8 FeFe	0.06	0.02	−2
				2.07	1.93 FeO		3.31	0.7 FeFe	6.03	0.6 FeFeFeFe			
							3.40	24.0 FeOO *					
							3.42	1.1 FeFe					
As3_0.1	Abiotic lab sample	0.06(1)	0.19(4)	1.93	2.55 FeO	0.08	3.03	1.2 FeFe	5.50	0.8 FeFe	0.07	0.01	−2
				2.05	2.25 FeO		3.30	0.6 FeFe	6.02	0.6 FeFeFeFe			
							3.42	1.2 FeFe					
							3.39	24.0 FeOO *					
As3_0.2	Abiotic lab sample	0.10(1)	0.22(1)	1.91	2.15 FeO	0.07	3.02	1.1 FeFe	5.51	0.7 FeFe	0.07	0.02	−3
				2.04	2.45 FeO		3.33	0.7 FeFe	6.02	0.5 FeFeFeFe			
							3.41	24.0 FeOO *					
							3.44	0.9 FeFe					
As3_0.4	Abiotic lab sample	0.16(2)	0.24(1)	1.93	2.30 FeO	0.07	3.04	1.2 FeFe	5.55	1.0 FeFe	0.07	0.03	−2
				2.06	2.25 FeO		3.31	0.8 FeFe	6.03	0.7 FeFeFeFe			
							3.37	24.0 FeOO *					
							3.43	1.3 FeFe					
As3_0.6	Abiotic lab sample	0.21(2)	0.22(1)	1.93	2.55 FeO	0.08	3.03	1.3 FeFe	5.52	0.6 FeFe	0.08	0.02	−2
				2.05	2.45 FeO		3.41	1.2 FeFe	6.01	0.6 FeFeFeFe			
							3.46	24.0 FeOO *					
							3.04	2.0 FeFe	5.48	0.6 FeFe			
Af CC1	Biogenic lab sample	0.19(1)	n.d.	1.92	2.30 FeO	0.07	3.04	2.0 FeFe	5.48	0.6 FeFe	0.09	0.03	−2
		0.05 V		2.04	2.80 FeO		3.26	1.5 FeFe	6.03	0.7 FeFeFeFe			
							3.34	24.0 FeOO *					
							3.42	1.9 FeFe					
Sed foyer0303	AMD sample	0.25(1)	0.20(1)	1.94	2.55 FeO	0.08	3.03	1.0 FeFe	5.51	0.5 FeFe	0.08	0.01	−2
		0.47 V		2.05	2.35 FeO		3.43	24.0 FeOO *	6.03	0.4 FeFeFeFe			
							3.39	0.8 FeFe

Fig. 1. Chemical composition of the solids (centrifuged) and supernatants (filtered through $<0.2 \mu\text{m}$ cellulose filters) for the series of abiotic synthetic samples studied. (a) As/Fe molar ratio in the solid phase as a function of the initial As/Fe ratio in the aqueous phase. (b) As concentration in the solid phase as a function of the final As concentration in solution. Sample labels are recalled on the right side of each point.

of schwertmannite at pH 3 in the presence of increasing concentrations of As(III) or As(V) in the starting solution are plotted in Figs. 1 and S2. These results indicate a high uptake of As(V) in or on the solid phase (89–98% of the initial dissolved As), whereas only 20–40% uptake was observed for As(III) (Fig. S2a). The As/Fe ratio in the solid phase is systematically lower than the initial As/Fe in the case of As(III), whereas it is close to the initial ratio in the case of As(V) (Fig. 1a). This lower affinity of As(III) for the solid phase is also well illustrated by the ~ 10 times higher final concentrations of dissolved As(III) compared to As(V), for similar As concentration in the solid (Fig. 1b).

XRD patterns of the As(III) and As(V)-rich samples are displayed in Figs. 2 and 3, respectively, and their corresponding As/Fe ratios in the solid phases are reported in Tables 1 and 2 for As(III) samples, and in Tables 3 and 4 for As(V) samples. Fig. 2 shows that all As(III) samples studied, including the field sample Sed foyer0303 and the biotic and abiotic analogs, consist of schwertmannite, based on comparison with a synthetic As-free schwertmannite, which matches the pattern indexed using the crystal structure recently proposed by Fernandez-Martinez et al. (2010). However, line broadening is observed when increasing the As/Fe ratio in the solid, which may be related either to a decrease in size of the schwertmannite crystallites and/or to the occurrence of structural defects. Assuming that the broadening would be only due to the decrease of the mean coherent domain size (MCD), application of the Scherrer formula to the observed line width indicates that the MCD size along the [221] direction (sharpest Bragg peak) would remain about constant at $\sim 7.5 \pm 0.5 \text{ nm}$ for all samples, whereas the MCD [301] value decreases from 5.2 ± 0.5 for the As-free sample to $4.1 \pm 0.5 \text{ nm}$ for the As-rich Sed foyer0303 AMD sample. Assuming that part

of the broadening might be due to structural defects, XRD results suggest that the MCD values of schwertmannite vary less than 25% within the range of As(III)/Fe ratios investigated. The As(V) samples consist of schwertmannite up to an initial As/Fe molar ratio of 0.1, without significant line broadening. However, for higher As/Fe ratios, the XRD patterns indicate a mixture with an amorphous phase, which becomes dominant at an initial As/Fe ratio of 0.2, and fully replaces schwertmannite above this ratio. This trend is observed for both AMD samples and model compounds, both biotic and abiotic. The XRD pattern of this amorphous phase as well as its chemical composition are consistent with those of amorphous As(V)–Fe(III) oxyhydroxysulfate, also referred to as ferric arsenate (Paktunc et al., 2008). EXAFS spectroscopy at the As and Fe K-edges were used to determine the mode of As associations with schwertmannite and ferric arsenate mineral phases.

3.2. Molecular-level structure of the As(III) samples

EXAFS data at the Fe K-edge of the As(III) samples are displayed in Fig. 4. Results of shell-by-shell fits of these data are reported in Table 1. The field sediment sample ‘Sed foyer0303’ is compared with a biogenic As(III)-rich schwertmannite ‘Af CCI’, prepared by inoculating sterilized AMD water with the model *A. ferrooxidans* strain CCI (Figs. S3 and S4), and with synthetic schwertmannite samples prepared by aqueous precipitation in the presence of dissolved As(III) (Figs. S5 and S6). The similarity of the EXAFS spectra at the Fe K-edge of all As(III)-rich samples with the spectrum of As-free schwertmannite indicates that they all have a schwertmannite-like local structure (Table 1), in agreement with XRD data (Fig. 2). The local environment around Fe in schwertmannite is

Fig. 2. Powder XRD patterns of the As(III) samples studied. All samples exhibit a schwertmannite-like pattern whatever the As/Fe molar ratio is in the solid phase. However, progressive broadening of the schwertmannite Bragg peaks with increasing As loading indicate either a decrease in crystallite size or a defective structure, or both. Peaks from the Al sample holder have been subtracted from the pattern of sample Af CC1. Peak indexes refer to the monoclinic cell proposed by [Fernandez-Martinez et al. \(2010\)](#). Corresponding As/Fe ratios in the solid phases are reported in [Tables 1 and 2](#).

characterized by three types of Fe–Fe distances at 3.0, 3.3, and 3.45 Å, in agreement with the crystal structure of goethite ([Forsyth et al., 1968](#)), or akaganeite ([Post et al., 2003](#)), both of which are characterized by double chains of edge-sharing $\text{Fe}(\text{O},\text{OH})_6$ octahedra (Fe–Fe distances of 3.0 and 3.3 Å) connected by double corner sharing linkage (Fe–Fe distance of 3.4–3.5 Å) (e.g., [Maillot et al., 2011](#) and references therein). In addition, longer distance Fe–Fe paths (5.5 Å) and multiple scattering Fe–Fe–Fe–Fe paths (6.0 Å) observed in our EXAFS data for schwertmannite are also consistent with these structures ([Fig. S7](#)), the latter MS paths being consistent with the alignment of $\text{Fe}(\text{O},\text{OH})_6$ octahedra forming double-chains along the *b*-axis in the akaganeite and goethite structures. Although the observed Fe–Fe distances are consistent with the goethite–akaganeite local structure, the number of observed Fe second neighbors ([Table 1](#)) is about equal to or less than half that expected from the bulk crystal structure of akaganeite, i.e., $N = 2, 2$, and 4 for the Fe neighbors at distances of 3.0, 3.3, and 3.45 Å, respectively. This discrepancy could be explained by destructive interferences between the EXAFS signals from the Fe–Fe pairs at 3.3 and 3.45 Å ([Maillot et al., 2011](#)). It could also be related to the nanometer size of the schwertmannite crystallites (4–8 nm), for which the surface Fe atoms are expected to have 10–20% less

neighbors than in the bulk structure ([Maillot et al., 2011](#)). Alternatively, this discrepancy could be due to the occurrence of a significant amount of structural defects, in agreement with the XRD-PDF study of the schwertmannite structure performed by [Fernandez-Martinez et al. \(2010\)](#). Comparison of the Fe K-edge EXAFS results obtained for the series of As(III)-bearing schwertmannite samples in the present study indicate that the presence of As(III) during the precipitation of schwertmannite does not lead to major changes in the schwertmannite local structure ([Table 1](#), [Fig. 4](#)). However, the intensity of the second-neighbor peak in the FT of the Fe K-EXAFS spectra decreases significantly with increasing the As(III)/Fe ratio in the solid ([Fig. 4](#)). This decrease is, however, better modeled by an increase of the Debye–Waller parameter (from 0.06 to 0.08–0.09 Å) than by a net decrease in the number of Fe second neighbors ([Table 1](#)). Accordingly, the 3.3 and 3.4 Å Fe–Fe paths could not be resolved in the EXAFS spectra of samples As3_0.6 and Sed foyer0303, with As(III)/Fe > 0.16 in the solid ([Table 1](#)). These results suggest that the presence of As(III) during schwertmannite precipitation leads to an increase in structural disorder rather than to a significant decrease in crystallite size, which would be consistent with the moderate change in crystallite size estimated from XRD, as discussed above.

Fig. 3. Powder XRD patterns of the As(V) samples studied. Field samples from the Carnoulès AMD are compared with biogenic samples obtained after inoculating *Thiomonas* sp. strains B2 or B3 in sterilized AMD water, and with synthetic schwertmannite precipitated from ferric sulfate solution at pH 3 in the presence of various concentrations of dissolved arsenate. Below As/Fe molar ratio of 0.2 in the solid phase, the products consist essentially of schwertmannite. The gradual broadening of Bragg peaks with increasing As/Fe ratio may be related to a decrease in crystallite size or to a defective structure, or both. Above As/Fe ratio of 0.2 in the solid phase, an amorphous ferric arsenate phase starts to form in mixture with schwertmannite and further dominates at high As/Fe ratio. The field sample contains minor amounts of quartz and mica. Peak indexes refer to the monoclinic cell proposed by Fernandez-Martinez et al. (2010). Corresponding As/Fe ratios in the solid phases are reported in Tables 3 and 4.

EXAFS data at the As K-edge for the As(III)-samples are displayed in Fig. 5. The first coordination shell around arsenic was fit by 3.0 ± 0.5 oxygen atoms at a distance of 1.79 ± 0.02 Å (Table 2), which is consistent with the known geometry of the As(III)O₃ pyramid (e.g., Morin et al. 2007 and references therein). EXAFS data for the AMD sample Sed foyer0303 were not analyzed using the present shell-by-shell procedure because this sample contains a high amount (47%) of As(V). For all the other As(III)-containing samples, the second-neighbor contribution to the EXAFS is weak, suggesting that As(III) is mainly adsorbed at the surface of the schwertmannite particles rather than substituted

for Fe(III) in the structure. These second-neighbor contributions are well fit by a combination of two distinct As–Fe paths at distances of 2.9 and 3.4 Å in variable proportions, depending both on the sample origin and the As(III)/Fe ratio (Table 2; Figs. S8 and S9). Multiple-scattering As–O–O paths within the AsO₃ pyramid with a fixed multiplicity were systematically included in the fit in order to improve the reliability of the results (Table 2; Ona-Nguema et al., 2005). The contribution from the As–Fe path at 3.4 Å is systematically larger than that at 2.9 Å, with a ratio $N_{3.4}/N_{2.9}$ ranging between 3 and 6 (Table 2). Interestingly, this ratio is maximum for the sorption sample ($N_{3.4}/$

Table 2

Results of shell-by-shell fitting of unfiltered EXAFS data at the As K-edge for the As(III) samples studied. $R(\text{\AA})$: interatomic distance; N : number of neighbors; $\sigma(\text{\AA})$: Debye Waller factor; ΔE_0 (eV): difference between the experimentally determined threshold energy and the FEFF8-calculated threshold energy, without using the self-consistent potential calculation; Standard deviations of these fit parameters are 0.02 $\text{\AA}$, $\pm 15\%$ and $\pm 30\%$ for first and second neighbors respectively, 0.01 $\text{\AA}$, and 2 eV, respectively χ^2_{FT} is a goodness of fit estimate (Table 1). Note: *fixed N value. As/Fe and S/Fe molar ratio of the solid samples are reported.

Sample	Origin	As/Fe mol/mol	S/Fe mol/mol	$R(\text{\AA})$	N	$\sigma(\text{\AA})$	$R(\text{\AA})$	N	$\sigma(\text{\AA})$	χ^2_{FT}	ΔE_0 (eV)
As3_0.04_ads	Abiotic lab sample	0.04(1)	0.18(6)	1.77	3.00 AsO	0.07	2.94 3.21 3.41	0.3 AsFe 6.0 AsOO*	0.08	0.06	15
As3_0.05	Abiotic lab sample	0.03(1)	0.21(1)	1.79	3.05 AsO	0.06	2.95 3.17 3.41	1.8 AsFe 0.7 AsFe 6.0 AsOO*	0.11	0.04	16
As3_0.1	Abiotic lab sample	0.06(1)	0.22(1)	1.79	2.65 AsO	0.06	2.94 3.18 3.41	2.6 AsFe 0.7 AsFe 6.0 AsOO*	0.11	0.04	16
As3_0.2	Abiotic lab sample	0.10(1)	0.19(4)	1.79	2.80 AsO	0.06	2.97 3.21 3.41	2.7 AsFe 0.7 AsFe 6.0 AsOO*	0.11	0.06	16
As3_0.4	Abiotic lab sample	0.16(2)	0.24(1)	1.79	2.85 AsO	0.06	2.95 3.22 3.41	0.5 AsFe 6.0 AsOO*	0.10	0.06	15
As3_0.6	Abiotic lab sample	0.21(2)	0.22(1)	1.79	2.80 AsO	0.06	2.94 3.27 3.41	2.5 AsFe 0.8 AsFe 6.0 AsOO*	0.10	0.03	17
AsfCC1	Biogenic lab sample	0.19(1)	n.d.	1.79	3.45 AsO	0.07	2.91 3.28 3.41	2.3 AsFe 0.8 AsFe 6.0 AsOO*	0.10	0.07	17
								2.1 AsFe			

Table 3

Results of shell-by-shell fitting of unfiltered EXAFS data at the Fe K-edge for the As(V) samples studied. $R(\text{\AA})$: interatomic distance; N : number of neighbors; $\sigma(\text{\AA})$: Debye Waller factor; ΔE_0 (eV): difference between the experimentally determined threshold energy and the FEFF8-calculated threshold energy, without using the self-consistent potential calculation; Standard deviations of these fit parameters are 0.02 $\text{\AA}$, $\pm 30\%$, 0.01 $\text{\AA}$, and 2 eV, respectively. χ^2_{FT} is a goodness of fit estimate. (Table. 1).

Sample	Origin	As/Fe mol/mol	S/Fe mol/mol	$R(\text{\AA})$	N	$\sigma(\text{\AA})$	$R(\text{\AA})$	N	$R(\text{\AA})$	N	$\sigma(\text{\AA})$	χ^2_{FT}	ΔE_0 (eV)
As free	Abiotic lab sample	0.00	n.d.	1.94	3.15 FeO	0.08	3.03	1.2 FeFe	5.53	0.8 FeFe	0.06	0.02	-2
				2.07	1.93 FeO	—	3.31	0.7 FeFe	6.03	0.6 FeFeFeFe	—		
							3.40	24.0 FeOO*			—		
							3.42	1.1 FeFe			—		
As5_0.05	Abiotic lab sample	0.05(1)	n.d.	1.94	2.85 FeO	0.08	3.03	1.2 FeFe	5.53	0.9 FeFe	0.07	0.02	-2
				2.06	2.00 FeO	—	3.29	0.7 FeFe	6.02	0.6 FeFeFeFe	—		
							3.37	24.0 FeOO*			—		
							3.41	1.3 FeFe			—		
As5_0.1	Abiotic lab sample	0.13(1)	0.20(2)	1.91	2.50 FeO	0.07	3.01	0.7 FeFe	5.46	0.4 FeFe	0.06	0.02	-4
				2.04	2.45 FeO	—	3.32	0.8 FeFe	6.00	0.3 FeFeFeFe	—		
							3.34	24.0 FeOO*			—		
							3.43	0.9 FeFe			—		
As5_0.2	Abiotic lab sample	0.24(1) 0.85(2) ^v	0.16(6)	1.93	2.40 FeO	0.07	3.03	0.9 FeFe	5.50	0.6 FeFe	0.07	0.01	-2
				2.05	2.40 FeO	—	3.31	0.4 FeFe	6.02	0.4 FeFeFeFe	—		
							3.39	24.0 FeOO*			—		
							3.41	0.9 FeFe			—		
Sed arch0503	AMD sample	0.45(1) 0.77(2) ^v	0.18(1)	1.98	4.95 FeO	0.09	3.04	0.5 FeFe	3.92	6.0 FeOFeO*	0.08	0.01	-2
							3.30	0.4 FeAs	6.02	0.2 FeFeFeFe	—		
							3.39	24.0 FeOO*			—		
							3.42	0.7 FeFe			—		
As5_0.4	Abiotic lab sample	0.48(2)	0.18(2)	1.98	5.20 FeO	0.09	3.32	1.8 FeAs	3.91	6.0 FeOFeO*	0.10	0.01	-2
							3.43	24.0 FeOO*			—		
							3.60	1.5 FeFe			—		
							3.31	2.8 FeAs	3.89	6.0 FeOFeO*	0.10		
Sed S32F00	AMD sample	0.62(2) 0.93(2) ^v	0.09(1)	1.99	5.25 FeO	0.08	3.31	2.8 FeAs			0.10	0.02	-2
							3.44	24.0 FeOO*			—		
							3.60	2.2 FeFe			—		
							3.32	2.5 FeAs	3.90	6.0 FeOFeO*	0.10		
Tm B2	Biogenic lab sample	0.63(2)	n.d.	1.98	5.30 FeO	0.09	3.32	2.5 FeAs			0.10	0.01	-3
							3.49	24.0 FeOO*			—		
							3.60	2.0 FeFe			—		
							3.32	3.0 FeAs	3.90	6.0 FeOFeO*	0.10		
Tm B3	Biogenic lab sample	0.80(2)	0.12(1)	1.99	5.20 FeO	0.09	3.32	3.0 FeAs			0.10	0.02	-2
							3.50	24.0 FeOO*			—		
							3.60	2.3 FeFe			—		
							3.32	2.2 FeAs	3.93	6.0 FeOFeO*	0.10		
As5_0.8	Abiotic lab sample	0.78(2)	0.10(3)	1.98	5.15 FeO	0.09	3.32	2.2 FeAs			0.10	0.01	-3
							3.46	24.0 FeOO*			—		
							3.61	1.5 FeFe			—		
											—		

Note: vAs(V)/AsTotal when lower than 1.00; *fixed N value.

Table 4

Results of shell-by-shell fitting of unfiltered EXAFS data at the As K-edge for the As(V) samples studied. $R(\text{\AA})$: interatomic distance; N : number of neighbors; $\sigma(\text{\AA})$: Debye Waller factor; ΔE_0 (eV): difference between the experimentally determined threshold energy and the FEFF8-calculated threshold energy, without using the self-consistent potential calculation; Standard deviations of these fit parameters are 0.02 $\text{\AA}$, $\pm 15\%$ and $\pm 30\%$ for first and second neighbors respectively, 0.01 $\text{\AA}$, and 2 eV, respectively χ^2_{FT} is a goodness of fit estimate (Table 1). Note: $^v\text{As(V)}/\text{AsTotal}$ when lower than 1.00; *fixed N value.

Sample	As/Fe solid	S/Fe mol/mol	$R(\text{\AA})$	N	$\sigma(\text{\AA})$	χ^2_{FT}	ΔE_0 (eV)
As5_ads	0.09(2)	0.17(1)	1.68	4.1 AsO	0.05	0.07	3
			3.11	12.0 AsOO*	—		
			3.20	4.0 AsOAsO*	—		
			3.27	1.5 AsFe	0.07		
			4.35	4.9 AsOO	—		
As5_0.025	0.03(1)	n.d.	1.69	4.6 AsO	0.05	0.07	5
			3.12	12.0 AsOO*	—		
			3.30	4.0 AsOAsO*	—		
			3.30	2.0 AsFe	0.08		
			4.34	8.0 AsOO	—		
As5_0.05	0.05(1)	n.d.	1.69	4.4 AsO	0.05	0.08	4
			3.13	12.0 AsOO*	—		
			3.31	4.0 AsOAsO*	—		
			3.30	1.9 AsFe	0.08		
			4.34	7.2 AsOO	—		
As5_0.1	0.13(1)	0.20(2)	1.69	4.4 AsO	0.05	0.08	5
			3.12	12.0 AsOO*	—		
			3.30	4.0 AsOAsO*	—		
			3.31	1.9 AsFe	0.08		
			4.36	7.1 AsOO	—		
As5_0.2	0.24(1) 0.80(2) ^v	0.16(6)	1.68	4.0 AsO	0.05	0.06	3
			3.12	12.0 AsOO*	—		
			3.24	4.0 AsOAsO*	—		
			3.30	1.9 AsFe	0.08		
			4.33	6.0 AsOO	—		
As5_0.4	0.48(2)	0.18(2)	1.68	4.5 AsO	0.05	0.05	5
			3.12	12.0 AsOO*	—		
			3.26	4.0 AsOAsO*	—		
			3.31	1.7 AsFe	0.07		
			4.39	6.2 AsOO	—		
As5_0.8	0.78(2)	0.10(3)	1.68	4.2 AsO	0.05	0.05	4
			3.16	12.0 AsOO*	—		
			3.21	4.0 AsOAsO*	—		
			3.32	2.0 AsFe	0.08		
			4.41	7.3 AsOO	—		
Sed arch0503	0.45(1) 0.77(2) ^v	0.18(1)	1.68	3.9 AsO	0.05	0.04	3
			3.11	12.0 AsOO*	—		
			3.26	4.0 AsOAsO*	—		
			3.30	1.9 AsFe	0.08		
			4.36	6.1 AsOO	—		
Sed S32F00	0.62(2) 0.93(2) ^v	0.09(1)	1.68	4.3 AsO	0.05	0.04	2
			3.12	12.0 AsOO*	—		
			3.25	4.0 AsOAsO*	—		
			3.31	1.9 AsFe	0.07		
			4.38	6.1 AsOO	—		
Tm B2	0.63(2)	n.d.	1.68	4.3 AsO	0.05	0.04	2
			3.14	12.0 AsOO*	—		
			3.21	4.0 AsOAsO*	—		
			3.31	2.0 AsFe	0.08		
			4.40	6.6 AsOO	—		
Tm B3	0.80(2)	0.12(1)	1.69	4.6 AsO	0.06	0.05	3
			3.16	12.0 AsOO*	—		
			3.25	4.0 AsOAsO*	—		
			3.32	1.5 AsFe	0.07		
			4.45	5.7 AsOO	—

Fig. 4. Unfiltered k^3 -weighted EXAFS data at the Fe K-edge for the As(III) samples studied, and corresponding amplitude and imaginary parts of the Fourier transform. Experimental and fit curves are displayed in black and red color respectively. Fit parameters are reported in Table 1. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

$N_{2,9} = 6$) and larger than that observed for the AMD samples and for those prepared by precipitating schwertmannite in the presence of dissolved As(III) ($N_{3,4}/N_{2,9} = 2.7$ – 5.4). This trend, although less marked for the As_{0.4} sample, indicates that the As(III) adsorption process observed when As(III) interacts with already formed schwertmannite differs slightly from that obtained after schwertmannite precipitation in the presence of As(III). Because As–Fe pairs at distances of 2.9 and 3.4 Å are generally assigned to bidentate edge sharing 2E and corner sharing 2C As(III) surface complexes, respectively (Ona-Nguema et al., 2005; Hohmann et al., 2011), our EXAFS results suggest that the As(III) adsorption mode is dominated by 2C surface complexes for all samples, and that the contribution of 2E complexes, although minor, tends to increase when As(III) interacts with schwertmannite during the nucleation and crystal growth process. Fig. 6 displays tentative structural models for such As(III) surface complexes on schwertmannite. It shows that FeO₆ octahedral edges with reactive singly coordinated oxygen atoms only occur at the extremities of the double chains along the b -axis, and which would cor-

respond to (010) facets in the monoclinic akaganeite structure (Post et al., 2003) or the monoclinic schwertmannite structure (Fernandez-Martinez et al., 2010). According to the generally observed needle shape of schwertmannite crystallites, which are elongated along the b -axis (Cornell and Schwertmann, 2003), (010) facets likely correspond to crystal growth faces that are significantly underrepresented after the needle particles have grown. Such (010) facets would, however, be available for As(III) surface complexation during the nucleation and growth stages of the schwertmannite particles (Fig. 6, right). Further interaction of As(III) with the needle-shaped particles is then expected to be dominated by 2C As(III) complexes forming on the sides of the needles and which involve two adjacent FeO₆ octahedra from the b -axis double chains (Fig. 6, left).

3.3. Molecular-level structure for the As(V) samples

Fe K-edge EXAFS data for the As(V)-samples are displayed in Fig. 7. Filtered first- and second-neighbor contributions to the EXAFS are analyzed in Figures S10 and S11.

Fig. 5. Unfiltered k^3 -weighted EXAFS data at the As K-edge for the As(III) samples studied, and corresponding amplitude and imaginary parts of the Fourier transform. Experimental and fit curves are displayed in black and red color respectively. Fit parameters are reported in Table 2. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

Examination of the Fe K-edge spectra indicate a progressive change in the spectrum shape with increasing As(V)/Fe ratio in the samples, whatever their provenance, including AMD samples as well as biotic and abiotic model compounds. Up to an initial As(V)/Fe ratio of 0.2, the samples exhibit a schwertmannite-like local structure, whereas for higher As/Fe ratios, the spectrum progressively changes toward that of amorphous ferric arsenate, which was identified by XRD as the main component in these latter samples (Fig. 3). According to our fit results for the Fe K-edge EXAFS data (Table 3), the local structure of this XRD-amorphous phase is characterized by Fe–As pairs at 3.3 Å and Fe–Fe pairs at 3.6 Å (Fig. S12). The Fe–Fe pairs at shorter distances (3.0, 3.3, and 3.4 Å), as well as the Fe–Fe paths and MS Fe–Fe–Fe–Fe paths at longer distance, which are indicative of the schwertmannite–akaganite–goethite local structure, are absent in this phase. Although the Fe–As pairs at 3.3 Å cannot be easily distin-

guished from Fe–Fe pairs because backscattering amplitude functions of Fe and As do not differ significantly, the goodness of fit indicator χ^2_{FT} was significantly better when assuming As instead of Fe second neighbors at this distance (data not shown). More importantly, the dominant second-neighbor contribution around arsenic to the As K-edge EXAFS data of the same samples is an As–Fe pair at the same distance of 3.3 Å (Fig. 8, Table 4). The first coordination shell around Fe observed by EXAFS at the Fe K-edge in the amorphous ferric arsenate phase (Fig. 7) is fit with ~ 5 oxygen atoms at a distance of 1.985 ± 0.02 Å, which indicates that Fe(III) is likely octahedrally coordinated to oxygen in this phase (note that the S_0 factor was set to 1; see Section 2). This first coordination shell can be fit with only one oxygen shell (Figs. S10 and S12), which differs from ferric oxyhydroxides for which two shells are necessary to account for the FeO_6 octahedral distortion (Maillot et al., 2011). Although the Fe–O distance measured by EXAFS

Fig. 6. Tentative structural models for As(III) surface complexes on schwertmannite that are consistent with our EXAFS data at the As K-edge. The 3D view is drawn according to the local structure of akaganeite in monoclinic (SG $I2/m$) unit cell (Post et al. 2003). 2C As(III) surface complexes, forming on the sides of the FeO_6 octahedra double-chains, dominate over 2E complexes in the case of adsorption on already formed schwertmannite (left), while the contribution of 2E complexes forming on the extremities of the FeO_6 octahedra double-chains, i.e., (010) facets, increases in the case of precipitation of schwertmannite in the presence of dissolved As(III) (left). Nevertheless, both type of processes only lead to surface adsorption of As(III) and the As/Fe molar ratio of the solid phase does not exceeds 0.25 in our experiments and in the AMD samples studied (Table 2).

in amorphous ferric arsenate ($1.985 \pm 0.02 \text{ \AA}$) is slightly lower than that measured by EXAFS in ferric (oxyhydr)oxides in which Fe(III) is octahedrally coordinated ($2.01 \pm 0.01 \text{ \AA}$ in goethite, $2.00 \pm 0.01 \text{ \AA}$ in akaganeite; Maillot et al., 2011), it is, however, larger than the $\langle Fe-O \rangle$ measured by EXAFS in compounds containing a mixture of octahedrally and tetrahedrally coordinated Fe(III) ($1.97 \pm 0.01 \text{ \AA}$ in ferrihydrite, $1.96 \pm 0.01 \text{ \AA}$ in maghemite; Maillot et al., 2011). Moreover, the absence of tetrahedrally coordinated Fe(III) in amorphous ferric arsenate is fully confirmed by the smaller intensity of the Fe K-pre-edge intensity for this compound compared to other mineral phases such as goethite and akaganeite, in which Fe(III) is solely octahedral (Fig. S13). This smaller intensity of the pre-edge feature, as well as the fact that the first-neighbor contribution to the EXAFS can be fit using only one oxygen shell, might be related to a more regular FeO_6 octahedron than in Fe(III) oxyhydroxides. Such lack of octahedral distortion could be explained by the absence of long-range order in this amorphous phase, which may favor relaxation of the local structure, or by the absence of edge- or face-sharing between $Fe(III)O_6$ octahedra in these amorphous ferric arsenate phases. Indeed, these types of polyhedral linkages create local displacement of the iron atoms from the center of their octahedra, due to cation-cation repulsion, as illustrated by the distorted octahedral in the structure of (α - $FeOOH$) goethite or hematite (α - Fe_2O_3). The As-Fe and Fe-Fe linkages observed in our EXAFS data for all amorphous ferric arsenate samples investigated here, either from AMD or from the biotic and abiotic syntheses, are similar to those reported by Paktunc et al. (2008) for synthetic amorphous ferric arsenate. The Fe-Fe distance at $3.6 \text{ \AA}$ corresponds to corner linkage (2C) between

tilted $Fe(O,OH)_6$ octahedra, and the As-Fe distance at $3.3 \text{ \AA}$ corresponds to arsenate corner linkage (2C) to adjacent $Fe(O,OH)_6$ octahedra, which is consistent with the local chain structure found in the ferric sulfate minerals butlerite, parabutlerite, and fibroferrite (Paktunc et al., 2008). It is also interesting to note that such linkages are consistent with the ring local structure of jarosite, and its lead-arsenate isotype beudantite (Fig. 9). However, the number of Fe(III) second neighbors around As(V) in jarosite and beudantite is 3 instead of 2 for a single-chain model as present in butlerite, parabutlerite, and fibroferrite. According to our As K-edge EXAFS results (Table 4), the observed number of Fe neighbors around As(V) at $3.3 \text{ \AA}$ ranges between 1.5 and 2.0 ± 0.5 Fe atoms, which supports a single-chain model rather than a jarosite/beudantite model with 3 Fe neighbors. Besides, as shown in Fig. 9, a single-ring model, based on a modified jarosite/beudantite structure, could match the EXAFS results as well as the butlerite, parabutlerite, and fibroferrite single-chains models.

Arsenic K-edge EXAFS data for the As(V)-containing samples are displayed in Fig. 8. Detailed analysis of the second-neighbor atoms around the absorbing arsenate oxoanion is reported in Fig. S14. It is noteworthy that the shape of the spectrum as well as the fit results remain similar for the whole set of samples, irrespective of their As/Fe ratio, including adsorption experiments, abiotic or biotic precipitation pathways, as well as field AMD samples. The first coordination shell around arsenic was fit with 4.0 ± 0.5 oxygen atoms at a distance of $1.69 \pm 0.02 \text{ \AA}$, which is characteristic of the AsO_4 tetrahedron. The second-neighbor contribution to the EXAFS was fit with $1.5\text{--}2.0 \pm 0.5$ Fe atoms at a distance of $3.3 \pm 0.05 \text{ \AA}$, which is consistent with the binding of an AsO_4 tetrahedron with two FeO_6 octahedra via double corner linkage (2C complex geometry). This surface complex binding mode is the most commonly observed one for arsenate surface complexes on ferric oxyhydroxides such as ferrihydrite, goethite (Hohmann et al., 2011 and references therein), and maghemite (Morin et al., 2008). This geometry is also consistent with the local structure of amorphous ferric arsenate (Fig. 9) observed in our samples with As/Fe ratios above 0.2, as discussed above. It is also noted that the same As-Fe distance occurs in scorodite (Fig. S15). However, EXAFS data for amorphous ferric arsenate indicate a Fe-Fe distance at $3.6 \text{ \AA}$ in amorphous ferric arsenate (Fig. 9), which is absent in the scorodite structure (Fig. S15; Paktunc et al., 2008, 2010). Altogether, in the present study, the similarity between the local structure of As(V) O_4 2C surface complexes and that of amorphous ferric arsenate makes it difficult to determine unambiguously the molecular environment of As(V) in our arsenate samples with As/Fe ratios below 0.2. Indeed, AsO_4 tetrahedra may be adsorbed at the surface of schwertmannite particles attached to the sides of the FeO_6 octahedra double chains (Fig. 9a), or they may form minute amounts of amorphous ferric arsenate (Fig. 9b and c) that would not be detected by XRD in a mixture with schwertmannite. Both adsorbed and amorphous species could also occur simultaneously in these low As/Fe samples.

Fig. 7. Unfiltered k^3 -weighted EXAFS data at the Fe K-edge for the As(V) samples studied, and corresponding amplitude and imaginary parts of the Fourier transform. Experimental and fit curves are displayed in black and red color respectively. Fit parameters are reported in Table 3. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

3.4. Implications for As(V) fate in AMDs

The similarity in compositions and molecular level structures of the AMD field samples and of their biotic and abiotic synthetic analogs investigated here allows us to derive important implications about the structure and reactivity of As(III)- and As(V)-rich hydroxysulfate minerals forming in AMD systems. First, in agreement with previous studies by Carlson et al. (2002) and Waychunas et al. (1995b), our results yield further evidence for the poisoning effect of As(V)O₄ that inhibits the nucleation of schwertmannite. This effect can be directly related to the strong affinity of the AsO₄ tetrahedron for binding to FeO₆ octahedra, which may hinder the formation of edge-sharing linkages between FeO₆ octahedra during the nucleation of schwertmannite. It is noteworthy that similar retardation of the nucleation of Fe oxides and oxyhydroxides have been reported in the presence of dissolved silica (Pokrovski et al., 2003), phosphate (Rose et al., 1996) or germanate (Pokrovsky et al., 2006). As a result of this strong affinity between dissolved arsenate an ferric species, an amorphous Fe(III)–As(V)

hydroxysulfate forms for initial As(V)/Fe ratios higher than 0.2, with a local structure similar to that of amorphous ferric arsenate described by Paktunc et al. (2008). EXAFS data of amorphous ferric arsenate at the As and Fe K-edge are consistent with single chains or rings of corner-sharing FeO₆ octahedra on which AsO₄ tetrahedra attach via ²C linkages. In the study by Paktunc et al. (2008), this phase formed at the expense of ferrihydrite, whereas in the present study it formed at the expense of schwertmannite. This difference can likely be attributed to the fact that, even though both studies were conducted in sulfate-rich systems, the study by Paktunc et al. (2008) was conducted at pH 4.5, whereas the present study was conducted at pH 3. Such a difference in pH could explain the formation of ferrihydrite instead of schwertmannite in the experiments by Paktunc et al. (2008), owing to the known stability fields of these phases (Bigham et al., 1996b). The 30 times higher starting concentration of both dissolved Fe(III) and As(V) (0.2 M and 0.02–0.2 M, respectively) used by Paktunc et al. (2008) relative to those used in the present study (6.4 mM Fe(III) and 0.16–5.12 mM As(V)) is another difference that

Fig. 8. Unfiltered k^3 -weighted EXAFS data at the As K-edge for the As(V) samples studied, and corresponding amplitude and imaginary parts of the Fourier transform. Experimental and fit curves are displayed in black and red color respectively. Fit parameters are reported in Table 4. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

also could have influenced the mineral phases formed. The starting dissolved concentrations used to prepare the synthetic abiotic model compounds in the present study were chosen to better match the concentration levels of the dissolved Fe and As in the Carnoulès AMD (10–40 mM and 1–5 As mM, respectively; Casiot et al., 2003). To the best of our knowledge, the Carnoulès AMD is one of the most As-rich AMDs reported to date (Morin and Calas, 2006). Our results show that the As(V)-bearing mineral phases synthesized under abiotic conditions in our laboratory experiments are similar to those formed after microbial incubation of sterilized AMD water from Carnoulès with As-oxidizing bacteria from the *Thiomonas* sp., as well as

to those formed in the field. It is especially noteworthy that the evolution of As(V) and Fe mineralogy with increasing As/Fe ratio is similar for the biotic and abiotic laboratory samples as well as for the field samples in the present study, as previously suggested by XRD (Carlson et al., 2002; Morin et al., 2003) and XANES (Morin et al., 2003) studies. Moreover, the present results yield direct evidence for the molecular-scale breakdown of the schwertmannite structure with increasing concentration of As(V) in the AMD system. A comparison of the EXAFS-derived local structure of amorphous ferric arsenate sulfate in the present study and that of Paktunc et al. (2008) with the EXAFS results of Inskeep et al. (2004) for the Beowulf acidic hot springs in

Fig. 9. Tentative structural models for As(V) speciation obtained after sorption on schwertmannite or coprecipitation with SO_4^{2-} and Fe(III) at pH 3, and which are consistent with our EXAFS data at the As and Fe K-edges. (a) Possible geometry for 2C complexes adsorbed at the surface of schwertmannite. (b) Local structure of beudantite/jarosite modified to match the number of Fe neighbors ($N = 1.5\text{--}2$) measured by EXAFS around arsenic in our amorphous ferric arsenate samples. (c) Local structure of the sulfate mineral butlerite, which also matches EXAFS data of amorphous ferric arsenate when replacing sulfate tetrahedra by arsenate ones, as proposed by Paktunc et al. (2008). Other sulfate minerals as parabutlerite and fibroferrite (not shown) are also reliable candidates, as proposed by the same authors.

Yellowstone National Park, USA, shows that the structure of biogenic ferric arsenate phases forming in the acidic hot springs is similar to that of the amorphous ferric arsenate sulfate forming in AMD systems. Interestingly, the formation of this phase for dissolved Fe and As(V) concentrations as low as $30\text{--}50\text{ }\mu\text{M}$ and $24\text{ }\mu\text{M}$, respectively, indicates that the high As(V)/Fe(III) ratio in the aqueous phase ($0.48\text{--}0.80\text{ mol/mol}$ in the Beowulf hot spring) is the driving parameter for the formation of amorphous ferric arsenate sulfate instead of schwertmannite in acidic sulfate-rich waters.

3.5. Implications for As(III) behavior in AMDs

The present study also shows that dissolved As(III) does not play the same role as As(V) when it is present during the precipitation of schwertmannite. These different behaviors of As(III) and As(V) are likely explained by the respective aqueous speciation of As(III) and As(V) during the nucleation stage of schwertmannite at pH 3, with the neutral $\text{H}_3\text{As(III)O}_3$ molecule having a lower affinity than the $\text{H}_2\text{As(V)O}_4^-$ oxoanion for $[\text{Fe}(\text{OH})(\text{H}_2\text{O})_5]^{2+}$ cations. Such differences in As(III) and As(V) speciation as a function of pH have been directly demonstrated by EXAFS (Perfetti et al., 2008; Testemale et al., 2011 and references therein). This difference in As(III) speciation compared to that of As(V) can explain the persistence of the schwertmannite structure for biotic and abiotic precipitation experiments in the presence of high concentrations of As(III), as well as in AMD samples (up to initial As/Fe ratio of 0.6). Indeed, our EXAFS data indicate that As(III) is adsorbed on the schwertmannite surface in all samples studied. More precisely, it forms 2C and 2E inner-sphere surface complexes on schwertmannite, similar to those observed on other Fe-minerals such as ferrihydrite and hematite (Ona-Nguema et al., 2005; Hohmann et al., 2011 and references therein). In addition, although 2C As(III) surface complexes dominate in all AMD and synthetic schwertmannite samples studied here, the presence of dissolved As(III) during the precipitation of schwertmannite slightly increases the proportion of 2E complexes, with respect to the sorption

experiments. This difference likely reflects the higher proportion of available FeO_6 octahedral edges during the schwertmannite nucleation stage rather than on the surfaces of well-developed schwertmannite particles. The formation of As(III) 2C and 2E surface complexes is found to favor the occurrence of structural defects in schwertmannite and to moderately decrease the size of the schwertmannite crystallites; however, it does not lead to a breakdown of the structure in the AMD samples or in the synthetic As(III)-rich samples (up to an initial As/Fe ratio of 0.6).

3.6. As(III)–schwertmannite vs. tooeleite

It is important to note that we did not observe the formation of the ferric arsenite hydroxysulfate mineral tooeleite ($\text{Fe}_6(\text{AsO}_3)_4\text{SO}_4(\text{OH})_4 \cdot 4\text{H}_2\text{O}$; As(III)/Fe(III) = 0.63; Morin et al., 2007) in the abiotic and biotic synthesis experiments, although this mineral was found to form in As(III) and Fe(III) rich acidic sulfate waters, including the Carnoulès AMD (Morin et al., 2003, 2006), other microbial incubation experiments of sterilized Carnoulès-Reigous water (Morin et al., 2003; Egal et al., 2009), and abiotic precipitation experiments at room temperature (Nishimura and Robins, 2008). Detailed analysis of these previous studies indicates that the formation of tooeleite requires a high initial As(III)/Fe(III) ratio (≥ 0.8) and a slow nucleation process that can be achieved at room temperature in abiotic experiments at pH 1.8–3 (Nishimura and Robins, 2008). These requirements can likely be obtained by slow microbial Fe(II) oxidation at room temperature, even at low As(III)/Fe(II) starting ratios, because the concentration of Fe(III) progressively produced by microbial oxidation may be lower than the initial As(III) concentration, thus generating high As(III)/Fe(III) ratios in the solution (Morin et al., 2003; Egal et al., 2009). In the present study rapid precipitation of Fe(III) and As(III) at $80\text{ }^\circ\text{C}$ for 1 h with starting As/Fe ratios not exceeding 0.6 in a pH 3 solution led to the formation of As(III)–schwertmannite. Our EXAFS analysis shows that As(III) is mainly adsorbed at the surface of the schwertmannite particles in these samples, as well as in the biotic and AMD ones. Conversely,

formation of tooeite may improve the efficiency of As removal from the solution compared to As(III)-sorbed schwertmannite in AMD systems. Indeed, the final dissolved concentration of As(III) after tooeite formation was found to be ~ 2 to 4 times lower than that measured after precipitation of As(III)-schwertmannite (Egal et al., 2009, and Fig. 1, this study, respectively).

3.7. Implications for arsenic remediation in AMD

Regarding the much higher residual dissolved As(III) concentrations relative to dissolved As(V) after precipitation of schwertmannite in acidic waters (Fig. 1), remediation of As-rich AMD systems is greatly improved by oxidizing As(III) to As(V), which allows the formation of As(V) hydroxysulfates exhibiting As/Fe solid ratios similar to those of the starting solution. Importantly, dissolved As(V) concentrations in contact with these phases are found to be ~ 10 times lower than dissolved As(III) concentrations in contact with these phases for a similar As/Fe ratio in the solid phase (Fig. 1). These phases are As(V)-schwertmannite for low starting As/Fe ratio (≤ 0.2 mol/mol) and amorphous ferric arsenate sulfate for high starting As/Fe ratio (≥ 0.2 mol/mol). In AMD systems, formation of such phases is accelerated by microorganisms such as *Thiomonas* sp., which are able to catalyze As(III) to As(V) oxidation by dissolved oxygen (Morin et al., 2003, 2006). In the Carnoulès AMD system, seasonal variations of the microbial activity of As(III)-oxidizing microorganisms likely influence the efficiency of the natural remediation process and lead to larger production of As(V) in the summer season (Morin et al., 2003, 2006). Nevertheless, aggregation and sedimentation of tooeite and amorphous ferric arsenate sulfate in thick biofilms rapidly lower the dissolved arsenic concentration in the upstream section (first 50–100 m downstream the spring) of the AMD system (Casiot et al., 2003; Egal et al., 2010). Because the dissolved As concentration decreases, schwertmannite can, therefore, precipitate downstream, and it becomes the dominant solid phase in the Reigous creek sediments. The remaining dissolved arsenic thus progressively immobilizes by forming surface complexes on the surfaces of this mineral, as shown for sample Sed foyer0303 sampled in Reigous creek, 1000 m downstream of the acidic spring (Lebrun, 2004; Casiot et al., 2005). The present study improves our understanding of the relationships between the structure and reactivity of As-bearing nanocrystalline and amorphous phases involved in natural and (bio)technological As remediation processes.

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APPENDIX A. SUPPLEMENTARY DATA

Supplementary data associated with this article can be found, in the online version, at <http://dx.doi.org/10.1016/j.gca.2012.11.016>.

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