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Statistical analyses of repolarisation current of a PZT film deposited on ITO electrode with different thermal treatments.

Salvatore Marino¹, Andrei Th. Ionescu², Anca-Luiza Alexe-Ionescu³, Gaetano Nicastro¹, Giuseppe Strangi¹ and Nicola Scaramuzza^{1,*}

1) INFM-CNR-LICRYL Laboratory-CEMIF.CAL, Department of Physics University of Calabria, Via P. Bucci, Cubo 33B, 87036 Rende (CS) Italy

2) Facultatea de Fizica, Universitatea din Bucuresti, P.O.B MG-11, Ro-077125 Bucharest, Romania

3) Facultatea de Stiinte Aplicate, Universitatea ‘‘Politehnica’’ din Bucuresti, Splaiul Independentei 313, Ro-060042 Bucharest, Romania

*) Corresponding author. Tel.: (+39) 0984-496113 Fax. (+39) 0984-494401 E-mail: scaramuzza@fis.unical.it

Abstract

In the vast application fields of PZT (lead zirconium titanate) thin films, of particular interest are the interaction effects occurring at the ferroelectric - substrate interface [1,2]. Relevant for this purpose are polarity sensitive liquid crystals (LC) cells, micro and nano-electronic applications [3]. In the present work the polarization current of a PZT film ($\text{PbZr}_{0.47}\text{Ti}_{0.53}\text{O}_3$), obtained by sol-gel synthesis and deposited by spin coating on ITO (indium tin oxide) electrode, has been studied. The different behaviour showed by such a system when the support electrode is previously submitted to a thermal treatment could be attributed to the change of the electrical properties of the ITO layer. In particular, a higher negative charge in the conductive band of the ITO electrode seems to be responsible of a higher order in the ferroelectric film.

Keywords: PZT film, repolarisation current, thermal treatments, Ferroelectric properties

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1. Introduction

PZT is a well known material for its ferroelectric properties that are suitable for a large number of applications as non-volatile ferroelectric memories, sensors [4,5] and so on [6-11]. Recently, thin films of PZT have been used to create asymmetric nematic liquid crystal cells (ANLC) in order to obtain a polarity sensitive electro-optical response. The advantage of such cells consists in controlling both ON and OFF states by an external field, consequently reducing the switching time [1]. The electro-optical response obtained with ANLCs containing PZT thin films was very interesting in terms of response time, contrast and rectifying effect. Such behaviour had been ascribed to the ferroelectric nature of the PZT film and to the interlayer phenomena occurring at the PZT-ITO interface that are responsible for a favoured orientation of the PZT nanodomains and, consequently, for the polar sensitive response of the liquid crystal layer. In this work the repolarisation current of PZT thin film obtained by sol-gel synthesis has been studied in two different cases. In the first case the PZT film has been deposited on an ITO electrode deprived of whatever thermal treatment and subsequently submitted to a heating of 600°C for 1h in order to ensure the transition to the ferroelectric perovskite phase. In the second case the ITO electrode has been previously submitted to a thermal treatment of 700°C in order to induce a change of its electric properties, then covered by the sol-gel process and eventually heated at 600°C for 1h. Interesting conclusions have been drawn from the

comparison between the behaviours of the repolarisation current showed by these samples.

2. Experimental setup

The repolarisation currents of PZT film were investigated using the setup described in Fig.1. The ITO substrate is connected to the output of a voltage source (Wavetek Universal Waveform Generator model 195). The signal applied to the electrode was a triangular wave with a 1s period. The upper electrode of cylindrical shape and 1 cm in diameter was placed with one of its optically polished bases directly in contact to the PZT film (the contact pressure was provided by the very weight of the copper cube). This electrode is connected to the ground by a rather small and well known resistor (100Ω). The voltages drop on this resistor, measured and recorded by a high input impedance oscilloscope (Agilent Infiniium 54832D MSO), is directly proportional to the electric current through the PZT film. To minimize the noise, the measuring setup (substrate, PZT film, upper electrode, and the measuring resistor) was surrounded by a grounded Faraday cage (not shown in the figure). Another channel of the oscilloscope was used to visualize and record the applied triangular wave, this last one also serving to trigger the oscilloscope. For each frequency two sets of data have been recorded, allowing representing currents and voltages versus time. The digitized sets of data are amenable to several kind of mathematical processing (for instance: peak area integration, smoothing, curve fitting, etc.). All the measurements have been performed at room temperature.

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We have used a rather large upper electrode because we are concerned of repolarization measurements. The repolarization is controlled by the electric field in the PZT and, therefore, we tried to create a thin planar condenser between the ITO substrate and the polished copper surface. We are not interested in having good ohmic contacts on top of PZT free surface, on the contrary. Actually, the ohmic current is very small as one can easily see from Fig.2, analyzing the slopes of the currents for the two regimes of the triangular signal (increasing and decreasing). On the other hand, the dielectric displacement current change is quite evident at the time instants where the two regimes of the triangular signal change the slopes. On top of this two currents there appear the ferroelectric repolarization currents manifested as two narrow peaks.

. In Fig.2 is reported the voltage-current vs time plot. The two peaks of the current flowing through the PZT (open circle) that appear in correspondence of around +40V and -29V of the applied voltage (black line) are due to the switch of the ferroelectric dipole. Since the applied signal includes 200 cycles we have 400 repolarisation switches (200 for positive polarity and 200 for negative one) that are enough to do statistical analyses. From such measurement has been possible to carry out information about the intensity of the repolarisation current (peak height), the voltage value at which the repolarisation switches happens and the repolarisation switching time (peak width measured at the base).

3. Results and discussion

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3 The peak height variation as function of the time, measured for both polarities of the
4 applied voltage, is shown in Fig.3, while in Fig.4 are shown the same measurements
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6 for the ITO electrode submitted to a previous thermal treatment at 700°C. Comparing
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8 these results it is evident that in the first case a higher correlation between the
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10 experimental data is present, in particular the peak height occurring during the
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12 negative polarity of the applied voltage. Such difference could be attributed to a
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14 higher order of the ferroelectric nanodomain orientation in the first case with respect
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16 to the second case.
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25 In Fig.5 are showed the histograms related to the value of the applied voltage at
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27 which the repolarisation switches appear both for positive and negative polarity when
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29 the ITO electrode has not been thermally treated. It is possible to notice that for
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31 negative polarity (ITO negatively charged) the distribution of voltage values at which
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33 the polar switch happens, within a quite narrow range, centres around 33V. On the
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35 contrary, for positive polarity the distribution is rather wide and centres on 18V.
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42 In Fig. 6 the switching time histograms of the ferroelectric repolarisation are reported
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44 both for positive and negative polarization of the ITO not thermally treated. In this
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46 case also it is possible to notice a remarkable difference between the opposite
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48 polarities. In fact, for negative polarity the switching time distribution has a narrow
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50 range cantered around 0.02s, while for positive polarity the switching time
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52 distribution is wider and cantered on 0.04s. What can be deduced by these data is that
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54 when the ITO electrode is negatively charged the ferroelectric repolarisation appears
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56 in a faster and cooperative way that when the ITO electrode is positively charged.
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The same histograms have been drawn for the sample in which the ITO electrode has been previously annealed at 700°C. The histograms related to the value of the applied voltage at which the repolarisation switch happens for both positive and negative polarity are reported in Fig.7. A remarkable difference from the precedent case is that now the required voltage to command the polar switch is higher. In fact, for positive polarity, the distribution is cantered on 60V while, for the negative one, around 43V, moreover, the two distributions are not so different in width as the preceding ones. As the ferroelectric switching time is concerned (see fig.8), in the last case we still observe narrowed distribution for negative polarity respect to the positive one, but the average switching time is similar, in fact both positive and negative polarity distributions are cantered around 0.01s.

4. Conclusions

In this work the switching properties of $\text{PbZr}_{0.47}\text{Ti}_{0.53}\text{O}_3$ (PZT) ferroelectric films were studied. Such films was obtained by sol-gel synthesis and deposited on ITO covered glass substrates by spin coating deposition and submitted to a thermal treatment at 600°C for one hour to ensure the transition to the perovskite phase. The analysis of the repolarisation current was effectuate on two different kinds of samples: in one of them the bottom electrode was previously submitted to a thermal treatment at 700°C in order to induce a structural modification of the ITO layer, while in the other one the substrate was thermally untreated. From the statistical analysis can be deduced that, in the case of the sample having the ITO electrode untreated, the

polar switch occurring when the ITO electrode is negatively charged appears in a faster and cooperative way than the ITO electrode is positively charged. This difference was less evident for the other sample, moreover, even in this last case, an increased value of the voltage at which the ferroelectric switch happen was observed for both polarities. Such information, together with the distribution shapes, suggests that the structure modification induced on the ITO film by the thermal treatment is responsible of a higher disorder of the nanodomain polarization. A possible explanation of this behaviour should take into account the influence of the annealing on the ITO charge carriers. In particular, we can imagine the fact that ITO/PZT interface is not just a geometric dividing plane but it allows injection of electrons into the PZT film. For instance, with negative polarity applied to ITO side of the sample not only the ITO film has got an influx of electrons but also the PZT. As both zirconium and titanium are transition metals there are d-type orbitals that can accept extra electrons. The twin potential walls, typical for ferroelectric materials, go deeper and the potential barrier increase (see fig.9). On the contrary, the positive potential extracts electrons also from PZT and the barrier lowers permitting the electric dipoles to flip over at a smaller voltage. This synergic bond mechanism [12] is consistent to the nefelauxetic effect seen in transition metal complexes as an auxiliary contribution to the usual ionic ligand field.

When the ITO substrate was previously treated up, its electrical properties worsen, the permeability of electrons to PZT diminishes and the histograms cannot show clearly the effect.

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Figure captions

Fig.1 Repolarization current setup . r is the measuring resistor.

Fig.2 1 s part out of the 200 s plot. The black line is the applied voltage, the open circles are the current flowing through the PZT film.

Fig.3 Peaks height as function of time (sample with the substrate previously untreated). The black squares are referred to the positive polarity, while the open circles to the negative one.

Fig.4 Peaks height as function of time (sample with the substrate previously treated). The black squares are referred to the positive polarity, while the open circles to the negative one.

Fig.5 Voltage value histograms at which the repolarization happens. Positive polarity (a), negative polarity (b). (Sample with the substrate previously untreated).

Fig.6 Switching time histograms. Positive polarity (a), negative polarity (b). (Sample with the substrate previously untreated).

Fig.7 Voltage value histograms at which the repolarization happen. Positive polarity (a), negative polarity (b). (Sample with the substrate previously treated).

Fig.8 Switching time histograms. Positive polarity (a), negative polarity (b). (Sample with the substrate previously treated).

Fig.9 Twin potential walls for a ferroelectric material in an ideal symmetric condition (a) and with the electrons contribution (b).

Figure 1

fig. 1
163x110mm (500 x 500 DPI)

FIGURE 2

fig 2
124x93mm (500 x 500 DPI)

FIGURE 3

fig 3
112x96mm (500 x 500 DPI)

FIGURE 4

fig 4
113x96mm (500 x 500 DPI)

FIGURES 5

fig 5
99x171mm (500 x 500 DPI)

FIGURES 6

fig 6
99x171mm (500 x 500 DPI)

FIGURES 7

fig 7
99x171mm (500 x 500 DPI)

FIGURES 8

fig 8
99x172mm (500 x 500 DPI)

a) Without electron contribution

b) With electron contribution

FIGURE 9

fig 9
161x134mm (500 x 500 DPI)