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BiOCl nano/microstructures on substrates: synthesis and photocatalytic properties

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Abstract

Bismuth oxychloride (BiOCl) nano/microstructures, including flake and nanowire arrays, were successfully synthesized on Anodic Aluminum Oxide (AAO) templates via sol-gel combined with the vacuum air-extraction method. The flakes are almost vertically aligned on the surface, but nanowire arrays at a lower sol concentration are aligned along the channels. A possible formation mechanism is proposed. Furthermore, the photocatalytic activity of the BiOCl nano/microstructures is investigated by photocatalytic decomposition of Rhodamine B (Rh B) dye under UV-Visible light irradiation. Compared with the BiOCl flake film on the glass substrate, where the flakes are horizontally oriented on the surface, the vertically aligned flake and nanowire arrays on AAO templates, have a higher photocatalytic efficiency.

Keywords:

BiOCl; flake film; microstructures; sol-gel preparation.

Introduction

Recently, Bi-based photocatalytic materials have aroused great interest due to the unique electron structure of Bi, which is beneficial for promoting the mobility of photo-generated carriers [1, 2]. In particular BiOCl, which has exhibited wide applications as a pigment in the cosmetic industry and as a candidate material for optoelectronic and photovoltaic devices, e.g.

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light-emitting diodes, lasers and solar cells, could perform as an potential excellent photocatalytic material for purifying textile dye polluted wastewater [3]. In 2007, Huang and co-workers firstly reported that BiOCl exhibited better performance than TiO₂ (P25, Degussa) on the photocatalytic degradation of methyl orange (MO) dye [1]. An [4] and Zhang [5] found that BiOCl exhibited excellent photocatalytic activity on degradation of Rh B and MO dye under both UV and visible light irradiation. Therefore, it is necessary to further investigate BiOCl growth and morphology to achieve optimum photocatalytic activity and explore the possible applications.

There are two major factors, i.e. high photocatalytic activity and immobilization, that should be considered for the application of photocatalysts [6]. The following strategies can be adopted: phase and morphological control, doping, surface sensitization, composite materials etc. Morphological control of nanostructured materials is one of the efficient and available ways to improve the photocatalytic efficiency due to the high surface-to-volume (S/V) reachable ratio. Until now, nano-structure materials with different morphologies, such as nanotube, nanowires, flakes, spheres, v-shaped structure, have been fabricated and used to promote the efficiency [7-9]. The second important factor is immobilization. The photocatalyst powders are generally suspended in water, which limits their practical use because of the difficulty in collecting and recycling photocatalyst powders from a suspension. One of the methods to solve this problem is to immobilize the photocatalyst on a substrate, e.g. to form thin films, coatings, or directly control growth of nanostructures on the substrate, to meet the requirements of the future applications.

Although different preparation methods have been reported, the AAO template assisted method is the most widely used and also turns out to be quite efficient [10]. AAO templates can not only control the growth of nanostructure, but also can behave as substrate. Recently, we have prepared the BiOCl nanowire arrays successfully by employing AAO templates as substrate [11]. Herein, we report the control growth of different morphological BiOCl nano/microstructures on AAO templates and the BiOCl flake film on glass substrates. Furthermore, their photocatalytic activities were also investigated by photocatalytic decomposition of Rhodamine B (Rh B) dye.

Experiment

BiOCl powder was dissolved in 100 ml alcohol to form 0.3 M, 0.5 M sol, in which 5 M HCl

was added to adjust the pH=1 under constant stirring at 60 °C for 2 h. Then, the commercial AAO template (Whatman company) with a pore diameter of 100 nm and depth of 60 μm was placed in a vessel that contained 2 ml of the prepared BiOCl sol. After, they were treated under the vacuum air-extraction[12]. The condition of vacuum pressure was 1 Pa at room temperature for 30 minutes. The resulting samples were collected and dried at 100 °C for 10 h. For comparison, the BiOCl flake film on a glass substrate was prepared via the sol-gel dipping method. The glass substrate was dipped slowly into the 0.5 M sol and a raising velocity at 25mm/min was maintained for deposition.

The crystallinity and morphology of the samples was examined by X-ray diffraction (XRD), and field effect-scanning electron microscopy (FE-SEM). Photocatalytic activity of the samples was evaluated by the degradation of Rh B dye solution. The samples with the average area of 4 cm² were put into 20 ml 0.01mmol Rh B solution, respectively, and kept in darkness for 24 h to reach the adsorption-desorption equilibrium. The light source was a 35W Xe lamp and the distance between the light source and samples was 20 cm. The UV-Visible absorption spectra and diffuse reflectance spectrum were carried out on a Hitachi U-3010 spectrometer.

Results and discussion

X-ray diffraction (XRD) was used to investigate the phase structures of the samples. All of the sharp diffraction peaks in the XRD patterns were perfectly indexed as pure phase of BiOCl, in good agreement with the standard JCPDS file (No. 06-0249) of BiOCl. A tetragonal lattice with $a=3.890 \text{ \AA}$ and $c=7.369 \text{ \AA}$ was obtained. The three strongest peaks were indexed as (001), (002) and (003) for the BiOCl flake film on the glass substrate (Figure 1c), whereas (101), (110) and (211) for the 0.5 M sol (Figure 1a) and 0.3 M sol (Figure 1b) on the AAO template.

Figure 2 displays the SEM images of the BiOCl morphologies on AAO templates and glass substrates. The flakes, with a diameter of $\sim 3\ \mu\text{m}$, thickness of $\sim 300\ \text{nm}$, are densely aligned on the surface of the AAO template (Figure 2a). They are almost perpendicular to the surface. A detailed cross-section image (inset of Figure 2a) shows that the flakes are co-existent with nanowires. By decreasing the concentration of the sol to 0.3 M, the flakes disappear and nanowires with the diameter around 100 nm and length around 3 μm are formed (Figure 2b). It can be seen that all nanowires are roughly parallel to each other and vertically oriented on the AAO template to form an array [11]. The morphology of the BiOCl on glass substrate is different from that on the AAO template. The flakes are parallel to the surface and are stacked layer by layer on the glass substrate. The thickness of these flakes ranges from 50 to 100 nm with corresponding diameters between 0.5 to 1.5 μm (Figure 2c). It can be clearly seen that the BiOCl on the AAO template possesses larger surface-volume area than that on glass.

As a consequence, a possible formation mechanism of different BiOCl morphologies is proposed (Figure 3). The formation of the flakes can be attributed to the tetragonal matlockite structure of BiOCl expressed by $[\text{M}_2\text{O}_2][\text{Cl}_m]$ or $[\text{M}_3\text{O}_{4+n}][\text{Cl}_m]$ ($m=1-3$), where bismuth oxide based fluorite-like layers, $[\text{M}_2\text{O}_2]$ or $[\text{M}_3\text{O}_{4+n}]$, are intergrown with double chlorine layers [5,13]. Hence, the BiOCl flakes prefer to be formed on the surface of glass and AAO template. During the process of the vacuum air-extraction treatment, these flakes on the template start to stand up and densely packed together. Meanwhile, part of the sol is sucked into the channels and becomes coagulated and solidified in the channels. In the other hand, the concentration of the sol determines its viscosity, which will affect its diffusing ability. The viscosity will decrease correspondingly as the concentration decreases. Therefore, the sol with lower concentration is

easier to impenetrate into the channels and fully fill them to form nanowire arrays.

The photocatalytic activities of the samples were evaluated for decomposition of Rh B dye. The removal of Rh B versus irradiation time is plotted in Figure 4a. The initial data shows the normalized concentration after 24 h equilibration of adsorption-desorption. Remarkably, ~39% and ~42% of the initial dye is removed from the aqueous solution by adsorption on the BiOCl flakes and nanowire arrays on AAO templates, respectively. The BiOCl on glass substrate removes only ~ 3% of the dye. The photocatalytic reaction is simply described by

$$-d[c]/dt = k[c],$$

where $[c]$ is the Rh B concentration, and k being the degradation rate constant. The calculated degradation rate constants in decomposing Rh B with photocatalytic samples is shown in Figure 4b., and is found to dramatically increase from 0.0038 min^{-1} for the horizontally oriented flakes film, to 0.0067 min^{-1} for the nanowire arrays and 0.0242 min^{-1} for the vertically aligned flakes. The BiOCl nanowire arrays and the vertically aligned flakes on the AAO template exhibit high photocatalytic activities, whereas the horizontally oriented BiOCl flakes on the glass substrate show a lower efficiency. As indicated in the Figure 4a, the bare AAO template can not cause any adsorption and degradation of the dye. Hence, the higher adsorption and efficiency of the BiOCl on the AAO templates can be attributed to the special morphology of BiOCl structures.

Conclusion

The vertically aligned flakes, nanowire arrays on AAO templates, and horizontally oriented flakes on glass substrates have been successfully formed. It is found that the concentration of the sol has a strong impact on the morphologies of the BiOCl nanostructures on AAO templates. Specifically, nanowire arrays are prepared at a concentration of the sol is 0.3 M, whereas vertically aligned

flakes are formed on surface at a concentration of 0.5 M. The photocatalytic action has been evaluated by degradation of Rh B dye under UV-visible light. We find that BiOCl nano/micro structures on AAO templates possesses higher photocatalytic efficiency than the BiOCl on glass substrates. This efficiency can be dramatically improved by controlling the morphology.

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Figure captions

Fig 1. XRD patterns of the 0.5 M sol on AAO template (a); 0.3 M sol on AAO templates (b); and the BiOCl film on glass substrates (c).

Fig 2. The SEM images of BiOCl flakes structure, from 0.5 M sol (a) and nanowire array structures, from 0.3 M sol (b) on AAO templates, and BiOCl film on glass substrates (c).

Fig 3. Schematic illustration of the formation mechanism of BiOCl nano/micro structures on AAO templates.

Fig 4. First order plots for the photocatalytic degradation of Rh B (a) and the corresponding degradation rate constants (b) for flakes, nanowire arrays, and film on glass substrate.

Fig 1

Fig2

Fig 3

Fig 4