

Room temperature triggered single-photon source in the near infrared

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Abstract. We report the realization of a solid-state triggered single-photon source with narrow emission in the near infrared at room temperature. It is based on the photoluminescence of a single nickel-nitrogen NE8 colour centre in a chemical vapour deposited diamond nanocrystal. Stable single-photon emission has been observed in the photoluminescence under both continuous-wave and pulsed excitations. The realization of this source represents a step forward in the application of diamond-based single-photon sources to Quantum Key Distribution (QKD) under practical operating conditions.

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1. Introduction

Practical and reliable triggered single-photon sources have attracted a lot of attention in recent years [1], owing to their numerous applications in quantum information processing. Among them, faithful implementation of the BB84 quantum key distribution protocol [2] relies on the use of single-photon pulses which ensure perfect security, over long distances, against eavesdropper attack [3]. As production of single-photon pulses at a sufficiently high rate became practical enough, quantum key distribution experiments was demonstrated with triggered single-photons [5, 6, 7] and with heralded single-photons from parametric down-conversion [8, 9].

Alternate protocols using weak coherent pulses have been proposed [10, 11, 12]. These protocols, e.g. using “decoy-state” are promising alternatives to single-photon QKD in the prospect of increasing the distance of perfectly secure communication compared to the “classical” limit that was determined in Ref.[3]. It was also recently shown that a modified decoy-state protocol using single-photon pulses can improve practical QKD [13, 14]. This result is another motivation of developing QKD dedicated practical single-photon sources.

In this context, optically-active defects in diamond provide numerous possibilities. Thanks to a near perfect room temperature photostability, the Nitrogen-Vacancy (N-V) colour centre in diamond has been demonstrated as a remarkable source for emitting single-photon pulses on demand [4].

This system successfully led to the realization of the single-photon QKD testbeds under realistic operating conditions [6, 7] and to the observation of single-photon interference in the delayed-choice regime [15]. The NV colour center has nevertheless a broad spectral emission at room temperature (100 nm FWHM) due to phonon broadening. Spectral filtering is a possible solution but it greatly decreases the effective single-photon transmission rate.

More recently, nickel-nitrogen-related point defects in diamond generated strong interest. These defects can be found in some natural type II-a diamonds and more generally in high-pressure high-temperature (HPHT) grown diamond samples where nickel is used as a solvent/catalyst for the HPHT crystal growth. Nickel then incorporates in the lattice as a substitutional or an interstitial impurity [16]. In the presence of nitrogen impurities, it can form a variety of nickel-nitrogen colour centres where a single nickel atom is surrounded by different number of nitrogen atoms located at adjacent lattice sites [17].

The photoluminescence of individual nickel-nitrogen-related colour centres has several striking features [18, 19]: a narrow emission band around 800 nm almost entirely concentrated in the zero phonon line (ZPL), with a Debye-Waller factor of approximately 0.7 at room temperature and a spectral width of the order of 1 nm; an excited-level lifetime around 2 ns; and a linearly-polarized light emission. These centres can be isolated at the single-emitter level in bulk diamond samples [19] but the high refractive index of diamond leads to a low collection efficiency due to total

internal reflection and spherical aberrations. However, if the size of the diamond sample is much smaller than the radiated light wavelength, refraction effects can be neglected and the colour centre can be approximated to a point source radiating at the air-glass interface [20]. It was demonstrated that nickel-nitrogen-related centres can be fabricated by chemical vapour deposition (CVD) of diamond [21]. The control of the CVD process parameters enables one to grow well isolated nanodiamonds with a specific size and with controlled defect concentration [22], contrary to natural diamond in which the defects are randomly distributed. This capability opens numerous possibilities for the development of highly efficient diamond-based single-photon sources, e.g. by growing photoluminescing diamond nanocrystals on an optical fibre endface [23] or by coupling them to microcavity resonance modes [24].

We report the observation of individual nickel-nitrogen-related colour centres in well isolated CVD diamond nanocrystals with photoluminescence centered at 793 nm which corresponds to the NE8 colour centre emission [25]. Using a tunable continuous-wave titanium-doped sapphire (Ti:Sa) laser, we measure the photoluminescence excitation (PLE) spectrum of these defects. We then find out the optimal excitation wavelength which leads to the highest signal-to-background ratio with the detection setup of the experiment. We then switch to a pulsed excitation regime, using a ps-pulsed laser tuned at this optimal wavelength. Contrary to previous observations on NV colour centers [26], we observe that cw photoluminescence properties of NE8 colour centers, e.g. its photostability, remain unaffected under this pulsed excitation. Finally, we demonstrate a room-temperature triggered single-photon source based on the time control of the optical excitation adapted to individual NE8 colour centres. This source has all the appropriate features for operating an open-air QKD testbed under daylight.

2. Material preparation

The process parameters of diamond synthesis using the CVD technique enables control over nanodiamond density and size and also dopant concentration [27, 28]. In a previous work [21], continuous films of polycrystallite diamond containing single nickel-nitrogen-related colour centres were fabricated by CVD technique. Using the same procedure while adjusting the growth time, we have prepared well isolated nickel-doped diamond crystallites which have nanometer scale.

These diamond nanocrystals were grown on a 170-micron-thick quartz glass substrate. After ultrasonic nucleation in a suspension containing fine nickel powder and diamond powder with size smaller than 100 nm, the substrates were washed with acetone and deionized water and then loaded into a 1.2 kW microwave plasma CVD reactor (ASTeX) for a 30 minutes growth time. The chamber gas flow (500 sccm) consisted of 0.7% CH₄ in H₂ gas mixture and the chamber pressure was maintained at 40 mbar during the growth process, with a substrate temperature of 700 °C. Nitrogen was not deliberately added to the gas feedstock; however, it was known to be present at a background level of approximately 0.1% corresponding to an N/C atomic ratio

≈ 0.15 . Due to growth rate estimated to be less than $1 \mu\text{m}/\text{h}$ and a deliberately low nucleation density, isolated nanodiamonds were formed on the substrate surface instead of a polycrystalline thin film.

The sample topography was characterized using an atomic force microscope (AFM) in order to evaluate the size and distribution of the diamond nanocrystals. The surface density was of the order of $0.1 \text{ particle}/(\mu\text{m})^2$ and the typical nanodiamond size was ranging between 30 nm to 100 nm (figure 1).

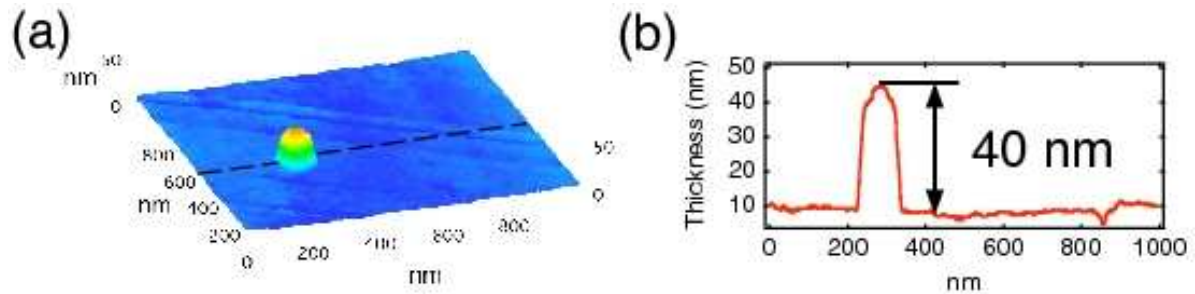


Figure 1. (a) $1 \times 1 \mu\text{m}^2$ AFM raster scan displaying a single diamond nanoparticle grown on the quartz substrate. (b) Cross-section of the crystallite indicating a height of approximately 40 nm . The corresponding in-plane dimensions are larger due to the convolution with the radius of curvature of the AFM tip.

3. Experimental setup

The experimental setup for the optical measurements of the single nickel-nitrogen-related colour centres was based on a home-made confocal microscope (figure 2). The excitation laser, being either a tunable cw Ti:Sa laser or a picosecond pulsed laser emitting at 765 nm , was focused on the sample with a metallographic microscope objective (O) of high numerical aperture ($\text{NA} = 0.95$, $\times 100$) which yielded a spot size of about $1 \mu\text{m}$. By driving a piezo-electric transducer-mounted mirror (MM), the objective allowed for an $x - y$ raster scan of the sample. Focused $z - \text{scan}$ was obtained by translating the microscope objective with another piezoelectric transducer.

The same microscope objective also collected the photoluminescence from the colour centres which was then passed through a dichroic mirror (DM) and optical filters (F1 and F2) that removed any residual excitation light. Proper spatial filtering was obtained by focusing in a pinhole (PH). The transmitted photoluminescent light was sent to the detection arms for spectral characterization and assessment of the emitter unicity by the observation of antibunching using a standard Hanbury Brown and Twiss (HBT) setup for photon correlation measurements [29, 30].

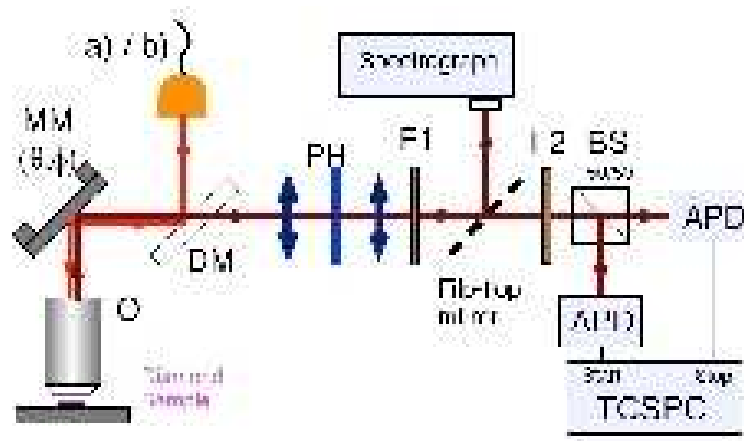


Figure 2. Experimental set-up. Excitation source: a) cw Ti:Sa laser tunable from 700 to 800 nm; b) picosecond pulsed laser emitting at 765 nm with 6 ps pulse duration and 20 MHz repetition rate; MM: Mobile mirror mounted on a scanning piezoelectric transducer; O: Microscope objective; DM: Dichroic mirror; PH: Pinhole with 100 μm diameter; F1: Long-pass interference filter transmitting wavelengths above 780 nm; F2: Band-pass filter centered at 800 nm and with a bandwidth of 10 nm FWHM; BS: non-polarized beamsplitter; APD: Silicon avalanche photodiode operated in photon counting regime; TCSPC: Time-correlated single-photon counting system (PicoHarp 300, PicoQuant, Germany); Spectrograph: Micro HR (Horiba Jobin Yvon, France) equipped with a CCD detector cooled at 200 K and a 1200 lines/mm grating blazed at 630 nm. The spectral resolution of the spectrograph is approximately 0.25 nm.

4. Continuous-wave excitation

A raster scan over a photoluminescent nanodiamond is shown in figure 3(a). The bright spot gave rise to a sharp spectral emission line centered at 793.7 nm with a width of 2-nm FWHM, as displayed in figure 3(b). This feature is characteristic of the NE8 nickel-nitrogen colour centre [25]. The atomic structure of this point defect was suggested to consist of a nickel atom located at equal distance of two carbon vacancies and associated to four nitrogen impurity atoms in its first coordination shell [31].

Photoluminescence of this NE8 colour centre was then recorded as a function of the cw excitation wavelength, tuned from 700 nm to 770 nm. Both the position and the shape of the emission line at 793.7 nm remained identical. As shown in figure 4, the photoluminescence intensity increased with the excitation wavelength while the background level remained constant except for a small increase when the excitation wavelength reached the cutoff edge of the long-pass filter F1. The highest signal-to-background ratio of ≈ 80 was achieved for a 765 nm excitation wavelength, which corresponds to a second-harmonic frequency of the 1.55 μm telecom band.

The HBT setup records the histogram of the time separation between successive photodetection events. For our operating conditions and at the limit of short time

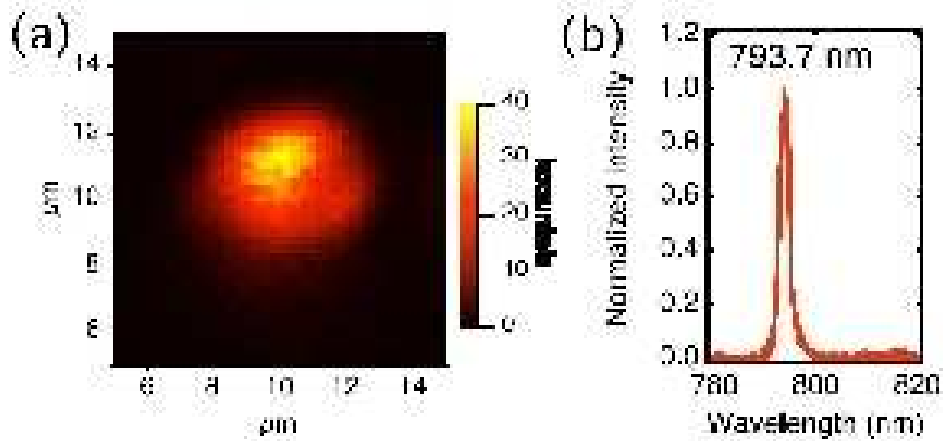


Figure 3. (a) Confocal microscopy raster scan over a photoluminescent nanodiamond under cw excitation at 765 nm. The colour scale indicates the total photoluminescence intensity at the output of the two detectors of the HBT setup. The peak counting rate value is approximately 70 kcounts/s while the background level is of the order of 0.8 kcounts/s. (b) Corresponding photoluminescence spectrum of the single emitter displaying the characteristic emission line of the NE8 colour centre.

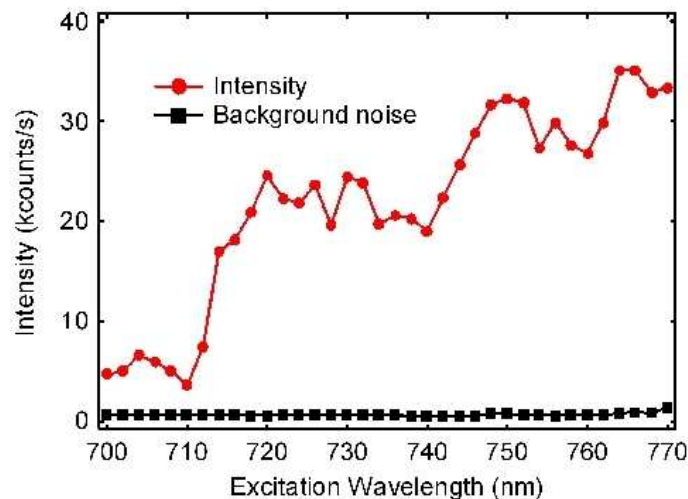


Figure 4. Photoluminescent excitation spectrum of the single NE8 colour centre associated to figure 3. The counting rate corresponds to the sum of the output of the two detectors of the HBT setup. Red dots and black squares respectively represent the evolution of the total counting rate and of the background as a function of the excitation wavelength, while maintaining a constant excitation power of 1 mW focused on the sample.

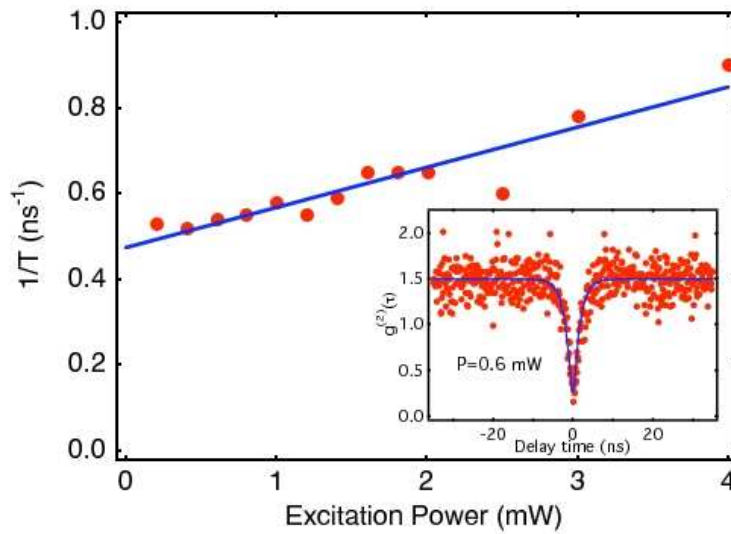


Figure 5. Evolution of the emission rate $1/T$ (red dots) as a function of the excitation power and associated linear fit (blue line). Inset: Normalized normally-ordered intensity autocorrelation function $g^{(2)}(\tau)$ associated to a time bin of 0.128 ns. The histogram of time intervals between successive detection events (red dots) has been recorded for a 0.6 mW excitation power and a 800 s total integration time. The corresponding detectors counting rates are 15.8 and 18.2 kcounts/s. The solid blue line is the intensity autocorrelation function computed for a single emitter with random on/off emission, convoluted with the instrumental response function of the photon correlation measurement setup (see Ref.[19]).

scale, this histogram is equivalent to a measurement of the normally-ordered intensity autocorrelation function $g^{(2)}(\tau) = \langle :I(t)I(t + \tau): \rangle / \langle :I(t): \rangle^2$. As shown in the inset of figure 5, a distinct minimum at zero delay, corresponding to $g^{(2)}(0) \approx 0.2$, is observed. This indicates that the addressed colour centre was indeed a single emitter. As explained in Ref. [19], fits of $g^{(2)}(\tau)$ recorded with increasing excitation powers yield the evolution of the exponential decay time T of the emitter as it gets more and more saturated (figure 5). The extrapolated value for a null excitation intensity allows one to infer an emission lifetime of approximately 2 ns. This value is similar to the one that has been previously measured for individual nickel-nitrogen-related colour centres in a type II-a natural diamond sample [19].

5. Pulsed excitation

The pulsed excitation starts from a mode-locked Erbium-doped fibre laser (PriTel, USA). The laser output, at a wavelength of 1530 nm and emitting a train of a 6.1 ps pulses with a 20 MHz repetition rate, is then amplified in a fiber amplifier and frequency doubled in a periodically-poled lithium niobate crystal. We then obtained picosecond pulses of energy 0.75 nJ at the optimized wavelength of 765 nm.

The TCSPC unit previously used to record the intensity autocorrelation function in the cw excitation regime was employed to measure the histogram of time delays between the laser pulses that trigger the colour centre photoluminescence and the corresponding photon detection events. An exponential decay fit yields an emission lifetime of 2.1 ns, in agreement with the value inferred from the antibunching curves recorded under cw excitation for the same wavelength (see figure 5).

The intensity autocorrelation function $g^{(2)}(\tau)$ is then measured under this pulsed excitation regime and compared to equivalent classical light pulses with Poissonian photon number statistics. Figure 6 shows the corresponding data for the triggered emission of the single NE8 defect and for the attenuated laser pulses, with peaks separated by the excitation repetition period of $\theta = 50$ ns [32]. In the case of the single NE8 defect, the peak at $\tau = 0$ has almost disappeared which implies that the probability of having two photons in the same output pulse is greatly reduced compared to the equivalent Poissonian statistics.

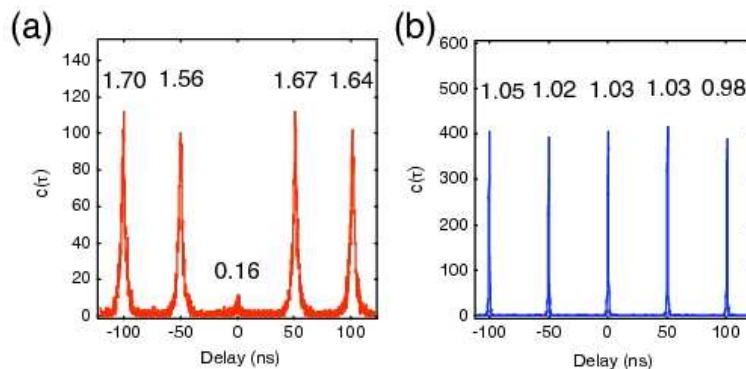


Figure 6. Normally-ordered intensity autocorrelation functions of (a) a single NE8 colour centre under a pulsed excitation, corresponding to an average power of $50 \mu\text{W}$ focused on the sample, and of (b) attenuated pulses from the excitation laser. The normalized total number of coincidence counts associated to each peak is indicated on its top.

To estimate the quality of the single-photon source, we then quantified this reduction following a procedure described in Ref. [4]. The total number of coincidence counts $c(m)$ associated to peak m is normalized as:

$$C_N(m) = \frac{c(m)}{N_1 N_2 \theta T_{\text{acq}}} \quad (1)$$

where N_1 and N_2 are the counting rates at the output of the two detectors of the HBT setup and T_{acq} is the acquisition time during which the temporal correlation histogram has been recorded.

The normalized C_N value for each peak is indicated in figures 6(a) and 6(b). For attenuated pulses from the excitation laser, C_N is equal to unity within the measurement uncertainty, as it is expected for Poissonian photon number fluctuations. For the single NE8 colour centre emission, the peak at $\tau = 0$ corresponds to $C_N(0) = 0.16$, well below

unity. The difference between the null value that would correspond to an ideal single-photon source is attributed to a lower signal-to-background ratio observed under pulsed excitation compared to the one measured under cw excitation (see figure 3).

Figure 6(a) also shows that the peaks corresponding to $m \neq 0$ are associated to a C_N value larger than unity. This is attributed to the existence of a “dark” metastable state which is also responsible for the bunching effect observed in the photon correlation measurement under cw excitation. This bunching corresponds to a limit value of $g^{(2)}(\tau)$ higher than unity when $|\tau|$ exceeds the characteristic decay time T (see the inset of figure 5) [19]. Using a model that accounts for random ON/OFF emission [33], the normalized total number of coincidence counts for peak $m \neq 0$ is given by:

$$C_N(m \neq 0) = 1 + \frac{T_{\text{off}}}{T_{\text{on}}} \exp \left[- \left(\frac{1}{T_{\text{on}}} + \frac{1}{T_{\text{off}}} \right) |m|\theta \right] \quad (2)$$

where T_{on} is the mean duration during which the emitter is active and T_{off} the mean duration during which it is trapped in the dark state and its emission turned off. Fits of the C_N values give $T_{\text{on}} = 9.1 \mu\text{s}$ and $T_{\text{off}} = 7.0 \mu\text{s}$ for an average excitation power of $50 \mu\text{W}$ focused on the sample.

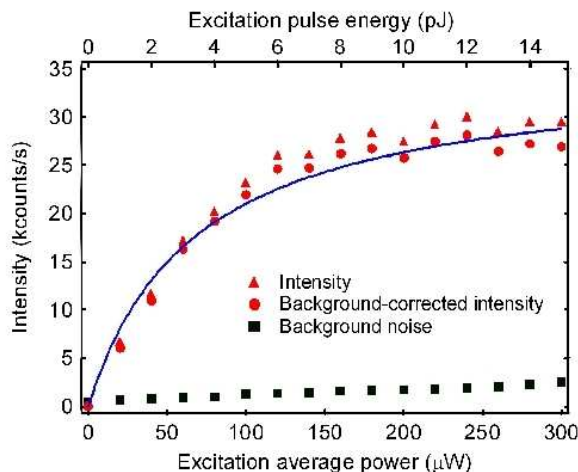


Figure 7. Detection counts rate as a function of the average excitation power focused on the sample, in the pulsed excitation regime. Experimental data are presented with points (triangle for the detection counting rate, dots for the background-corrected detected counting rate, and squares for the background level). The solid line is a fit of the background-corrected signal using equation (3).

These values can be used to estimate the overall efficiency of the single-photon source. The rate R of total detection counts as a function of the average excitation power P is shown in figure 7 and can be fitted by

$$R = R_{\infty} \frac{P}{P + P_{\text{sat}}} \quad (3)$$

where P_{sat} is the saturation average excitation power and R_{∞} is the rate of detection counts at the limit of saturated emission. Fit of the excitation saturation curve shown in

figure 7 yields $R_\infty = 35$ kcounts/s and $P_{\text{sat}} = 66 \mu\text{W}$. Taking into account the quantum efficiency of the emitter resulting from parameters T_{on} and T_{off} and the overall detection efficiency η_{det} of the setup, the saturated counting rate R_∞ can also be estimated by:

$$R_\infty = \eta_{\text{det}} \frac{T_{\text{on}}}{T_{\text{on}} + T_{\text{off}}} \frac{1}{\theta}. \quad (4)$$

The inferred value of this parameter allows us to estimate the detection efficiency $\eta_{\text{det}} \approx 0.35\%$ for the single NE8 colour centre in a CVD diamond nanocrystal. This value is three times larger than the one previously estimated for the emission of nickel-nitrogen-related colour centres in a bulk diamond sample [19]. The improvement in the collection efficiency is attributed to the sub-wavelength size of the nanocrystal structure.

Conclusion

NE8 nickel-nitrogen colour centers were observed in CVD-grown diamond nanocrystals. Optimization of the CVD growth technique lead to well separated nanocrystals, in which we could observe single colour centers with a high signal-to-background ratio. The photoluminescence dynamics of NE8 colour centers, previously studied in bulk samples, was demonstrated to scale down to these nanosize structures. Moreover, a higher single-photon detection efficiency compared to the previously reported value in bulk diamond [19] was achieved by avoiding the influence of the high index of refraction of diamond.

Both cw and pulsed excitations were used for single-photon emission by these NE8 colour centers. A triggered single-photon source was realized emitting in the 793 ± 1 nm transmission window. We determined the optimal pulsed excitation wavelength according to the excitation spectrum recorded in the cw regime for a single emitter.

We now plan to increase the triggered single-photon emission rate by growing nanodiamonds on optimized photonic structures, and to implement this single-photon source in an open-air QKD testbed.

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